Supporting Information

Mixed Multilayered Vertical Heterostructures Utilizing Strained Monolayer WS₂

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S1. AFM Characterization of WS₂ residues



Figure S1. (a) AFM topological mapping of the WS_2 residues on SiO_2/Si after being in contact with DI water. Scale bar: 2µm. (b) Corresponding line profile for the blue dashed line in (a).

Figure S1(a) presents an AFM topological map of the WS₂ residues on SiO₂/Si after transferring a layer of h-BN using the standard wet transfer method. The line profile (Figure S1(b)) for the blue dashed line in Figure S1(a) shows that the thickness of the surviving WS₂ is \sim 3nm after standard wet transfer, indicating the material left on the surface is multilayer WS₂.



S2. Detection of WO₃ or W-oxide formation

Figure S2. Probing WO₃ or W-oxide formation after transferring a layer of h-BN onto as-grown WS₂ using the standard wet transfer method. Optical images of WS₂ with a layer of h-BN on top before (a) and after (b) PMMA removal. Scale bar: 100μ m. (c) SEM image of the trace of the WS₂ crystals on SiO₂/Si with and without h-BN on top. Scale bar: 0.5mm. (d) Zoomed-in SEM image of the trace of surviving WS₂ and surviving WS₂ under h-BN. Scale bar: 50μ m. (e) Raman spectra of washed off WS₂, pristine WS₂ and WO₃ powder measured using 532nm excitation. (f) Comparison of Raman spectra among pristine WS₂ on SiO₂/Si, surviving WS₂ with and without h-BN on top.

Figure S2(a) shows the optical image of WS₂ with a PMMA/h-BN stack on top. After soaking in acetone at 45°C for 2 hours to remove PMMA, as shown in Figure S2(b), it was found that PMMA, which sits on top of h-BN, can be well removed during this process and the WS₂ will not be removed from the substrate by the acetone. The small optical contrast from the h-BN film on SiO₂ is attributed to its negligible opacity in the visible spectrum arising from its large band gap (>5 eV), making it hard to detect. The high transparency of the h-BN makes it easy to find WS₂ survivors and the traces of WS₂ crystals using an optical microscope even when it is covered with a layer of h-BN film. According to the optical images, it seems that there is still some residue of the material lying at locations where the WS_2 triangles or hexagons used to be before exposure to DI water. If WS₂ has been changed to some other material, which has a large band gap, it might be transparent under optical microscope. So it is necessary to examine the samples under SEM to see whether there is anything left on the surface. From the SEM images as shown in Figures $S_2(c)$ and (d), it was confirmed that just a very small number of WS₂ domains were left on the surface after transferring a layer of h-BN using the standard wet transfer method. The trace of WS₂ under h-BN on top (darker in contrast) and without h-BN (brighter in contrast) can be easily found in Figure S2(c). The water might be oxidizing the WS₂ and changing it to WO₃ or other W-oxides during the transfer process. So it is necessary to check the darker and brighter areas using Raman spectroscopy in order to confirm whether the disappearance of WS₂ leads to a rise of 2D W-oxides.

It was reported by Xu et al.¹ that Raman peaks centered at 272, 326, 717 and 807 cm⁻¹ can be found from WO₃, the first two are induced by W-O-W bending mode vibration and the other two correspond to W-O-W stretching vibration mode. As can be seen in Figure S2(e), peaks in the Raman spectrum at similar positions (270, 326, 716 and 804 cm⁻¹) were found for

pure WO₃ powder deposited onto the sample for the measurement, which is the same as the precursor used to grow monolayer WS₂ by CVD. There is no difference between the spectra of washed off WS₂ without h-BN on top and bare Si, indicating that the WS₂ was completely washed off and no W-oxides were left on the surface after exposure to water. The Raman spectra of pristine WS₂ on SiO₂/Si, surviving WS₂ with and without h-BN on top after exposure to DI water were compared and show almost the same peak positions, but with slightly different peak intensity. Although the contrast for washed off WS₂ under SEM is brighter on SiO₂/Si and darker under h-BN, as can be seen from Figure S2(c), that their Raman spectra are almost the same (Figure S2(f)), showing only one peak centered at 520 cm⁻¹, which corresponds to bare Si. The contrast in both SEM and optical microscopy can be attributed to the concave surface left by the detachment of WS₂, most likely formed during the initial growth of WS₂ on the SiO₂ surface by CVD at high temperatures.

S3. Characterization of CVD-grown multilayer h-BN



Figure S3. (a) A typical SEM image of the CVD-grown multilayer h-BN film on Cu substrate. Scale bar: 10 μ m. (b) Raman spectrum of the transferred h-BN film on SiO₂/Si substrate. (c) AFM topological mapping of the transferred h-BN film. Scale bar: 5 μ m. (d) Corresponding height profile taken across the black dashed line in (c).

Figure S3(a) presents a typical SEM image of the CVD-grown continuous h-BN film on Cu. The Raman spectrum, as can be seen from Figure S3(b), shows a characteristic peak centered at ~1368.7 cm⁻¹, which indicates the film thickness to be multilayer. Figures S3(c) and (d) gives a typical AFM measurement, showing the thickness of the h-BN film is \sim 2.5nm following transfer onto a SiO₂/Si substrate.



S4. Characterization of CVD graphene

Figure S4. (a) SEM image of graphene/h-BN/WS₂ (left part) and graphene/WS₂ (right part) on SiO₂/Si. Scale bar: 50 μ m. (b) Optical image of the graphene/h-BN/WS₂ stack on SiO₂/Si. The blue dashed line and the red dashed line indicate the boundary of graphene and h-BN film, respectively. Scale bar: 200 μ m. (c) Optical image of the graphene/WS₂ stack. Scale bar: 30 μ m. (d) Raman spectra of the marked spots with corresponding colored crosses showing the existence of bi- and multi-layer graphene. Blue line: multilayer graphene. Red line: bilayer graphene.

Figures S4(a) and (b) present typical areas of the graphene/h-BN/WS₂ stack on SiO₂/Si examined by SEM and optical microscopy, respectively. The non-uniform contrast indicates the variation of the film thickness.² The graphene film is continuous with several thicker graphene strips. Figure S4(d) shows the Raman spectra of two marked spots with different color in Figure S4(c). The 2D/G intensity ratio of the spectra in red is ~1.3, which indicates bilayer graphene.^{3–6} The blue spectrum with a ratio of ~0.6 indicates that the graphene is about 2-4 layers thick.⁷ The D peak (~1350 cm⁻¹) used to identify the level of defects⁸ in graphene was found to be negligible for both spectra, indicating the high quality of the transferred CVD graphene.

Reference

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