

1 **Trace Level Doping of Lithium-Rich Cathode Materials**

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16 REVIEW OF SYNTHESIS METHODS FOR PRODUCING LITHIUM-MANGANESE RICH MATERIALS

17 - Various synthesis methods exist for producing layered cathode materials including co-
18 precipitation, solid-state synthesis, polymer-assisted synthesis, ball milling, combustion synthesis
19 and spray pyrolysis¹⁻⁵. Among these methods, spray pyrolysis allows for a level of control of
20 product purity that is unlikely to be reproduced by any other synthesis method. In spray pyrolysis
21 each droplet acts as a microreactor, thus preserving the composition of the precursor solution. No
22 additives are required, which allows accurate control of the dopant profile to trace levels in the
23 product.

24 DOPANT EFFECTS - Substituting Li⁺ with an alkali metal may improve the stability of the
25 structure due to steric effects caused by the larger size dopant, as they also carry a single positive
26 charge. Kang and Ceder using computational techniques⁶ have shown that Li⁺ ion hopping is
27 affected by the size of the tetrahedral vacancy site, and the electrostatic interaction between Li⁺
28 in that site and the cation in the adjacent face sharing octahedron. In general, the screening effect
29 of the original transitional metal ions is very important for the diffusion of Li⁺ ions. Therefore,
30 doped samples may have lower diffusivities, where the bonds with oxygen become more
31 polarized due to an increase in electronegativity difference. This is a complicated effect,
32 especially when it is coupled to a change in oxidation state and the dopant does not carry the
33 same theoretical charge (e.g. Mn⁴⁺↔Al³⁺).

34 Replacing Li⁺ ions in the interstitial sites with larger alkali metals leads to an increase in
35 the amount of residual ions in the interstitial host sites after charging, because often the dopants
36 do not participate in the intercalation reaction, as reported for Na doping⁷. Kim *et al.* found that
37 Li_xMO₂ based structures have strong driving forces and low energy kinetic paths to transform
38 into spinel structures and that these driving forces do not exist for Na-based counterparts of the

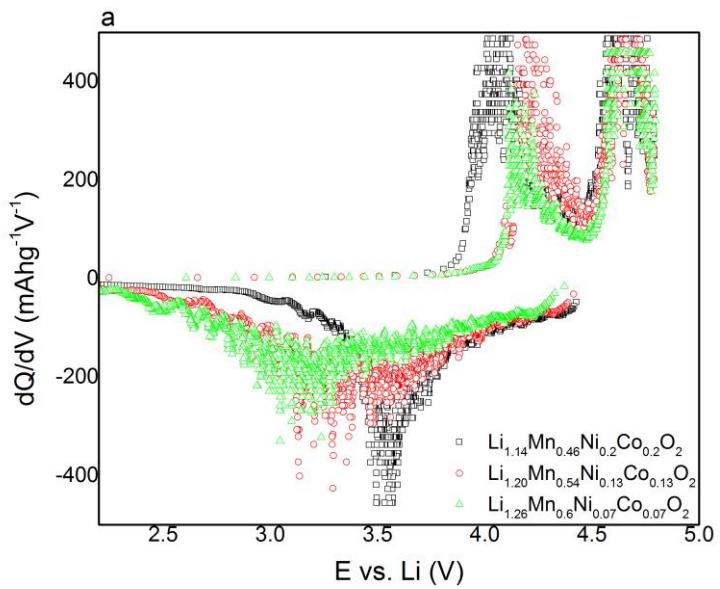
39 materials⁸. These spinel-transformation reactions are speculated to proceed via transitional metal
40 hopping. The migration of the TM (e.g., Mn) ions has to be coupled with the migration of Li⁺
41 ions, which may become hindered due to the larger size of the Na⁺ ions.

42 Substituting Mn with Al may lead to a slight reduction in the electrochemically active
43 cathode material content of the sample with aluminum having only a single ⁺³ oxidation state. Al
44 is similar to Mn in size, and the single oxidation state may be the reason behind the improved
45 stability of Al doped compounds. Several studies demonstrated the beneficial effect of Al on
46 stabilizing the cathode material primarily by creating an Al-coated surface to prevent surface
47 dissolution of the Mn-based active cathode material ⁹⁻¹¹. Additionally, replacing Mn⁺⁴ with Al⁺³
48 ions should lead to an increase in the average oxidation state of the Mn ions. To maintain charge
49 neutrality this beneficial effect can mitigate the potentially arising Jahn-Teller distortion effect ¹².

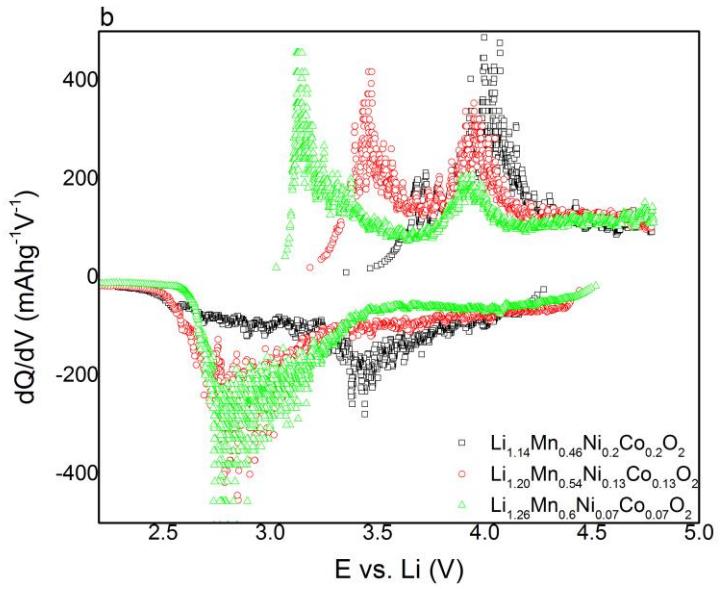
50 Alkali earth metals (Mg, Ca, Sr, Ba) have larger ionic radii than Co³⁺ or Ni²⁺ atoms.
51 Their introduction to the structure has been shown to improve the stability similar to Al and may
52 reduce the capacity reducing the available intercalation sites due to their single +2 oxidation state
53 ¹³⁻¹⁷. Substituting Co⁺³ for M⁺² ions is also expected to increase the average oxidation state of the
54 transitional metal ions originally residing in the structure thereby reducing TM dissolution.

55 REVIEW OF REVERSIBLE OXYGEN CHEMISTRY – In a conventional description of the
56 oxidation states in a solid compound alkali atoms are anticipated to carry a single, positive
57 charge, alkali earth atoms two positive charges, while oxygen two negative charges. To maintain
58 electroneutrality for LiCoO₂, the previously mentioned would require Co to be in a Co³⁺
59 oxidation state. Earlier several groups ¹⁸⁻²¹ have shown based upon both experimental
60 observations and computational results that the oxidation state of cobalt remains the same during
61 the intercalation process in LiCoO₂, and that the Fermi-level electrons of the material are located

62 at the oxygen atoms. They concluded that the only possible way to maintain charge neutrality
63 would be to require a reversible oxygen chemistry to occur in the LiCoO₂ cathode. Furthermore,
64 similar results were obtained for LiNiO₂ and its derivatives as well^{22, 23}. Recently, this was
65 demonstrated by Sathiya et al.²⁴ for a Ru and Sn based chemistry, and more recent results by
66 Oishi et al. have indicated the phenomenon to occur in the lithium-rich NMC chemistry^{25, 26}.



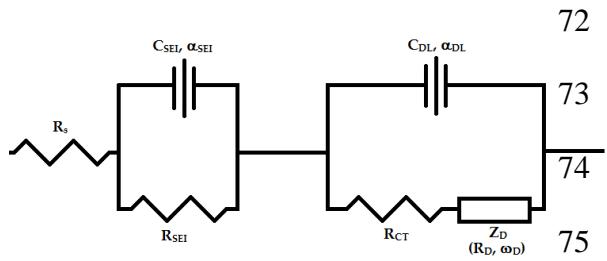
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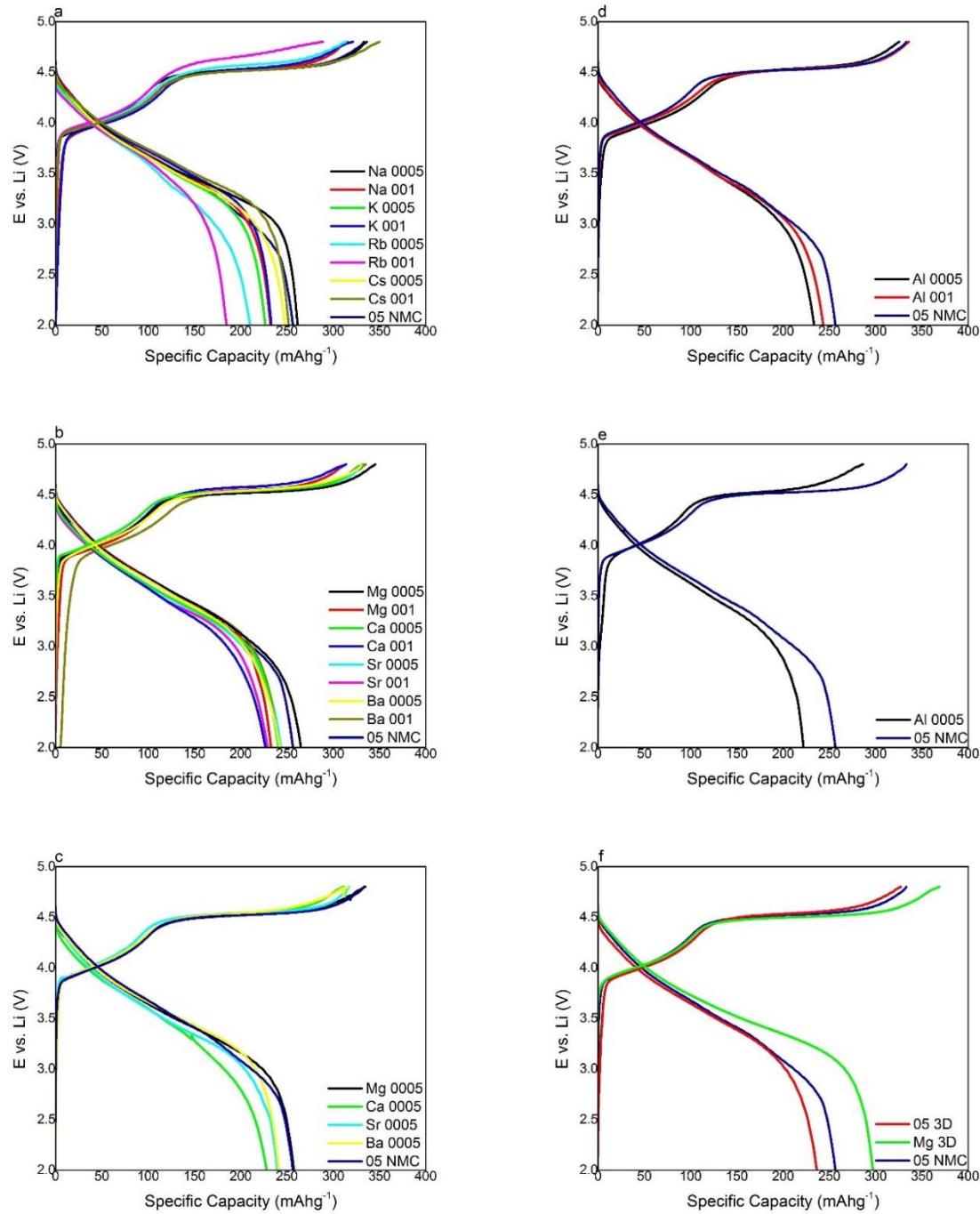
69 Fig. S1 - dQ/dV curves of $\text{Li}_{1.14}\text{Mn}_{0.46}\text{Ni}_{0.2}\text{Co}_{0.2}\text{O}_2$, $\text{Li}_{1.20}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$ and
70 $\text{Li}_{1.26}\text{Mn}_{0.6}\text{Ni}_{0.07}\text{Co}_{0.07}\text{O}_2$ for (a) cycle 1 and (b) cycle 50.

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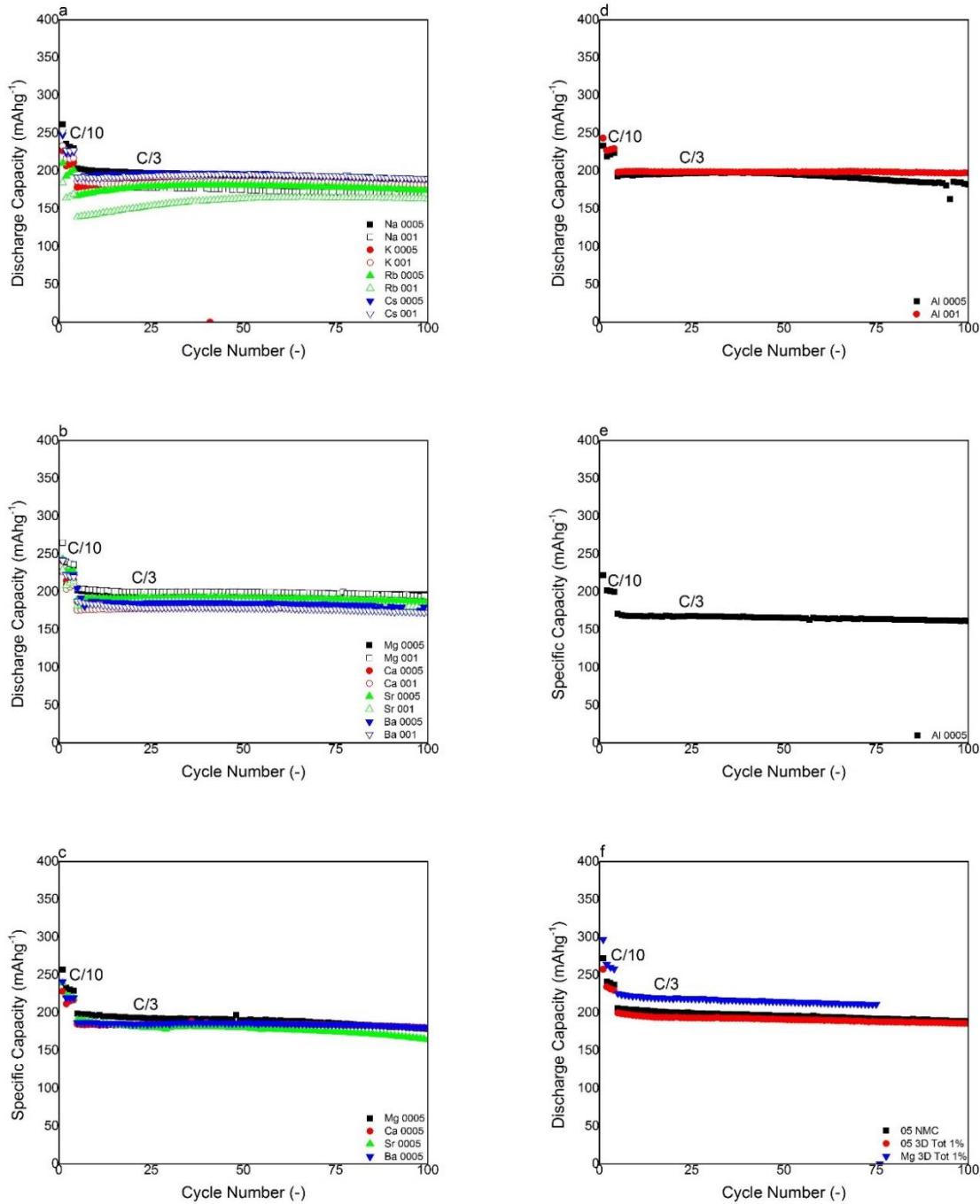
76 Fig. S2 - The equivalent circuit model used for the interpretation of the EIS spectra.

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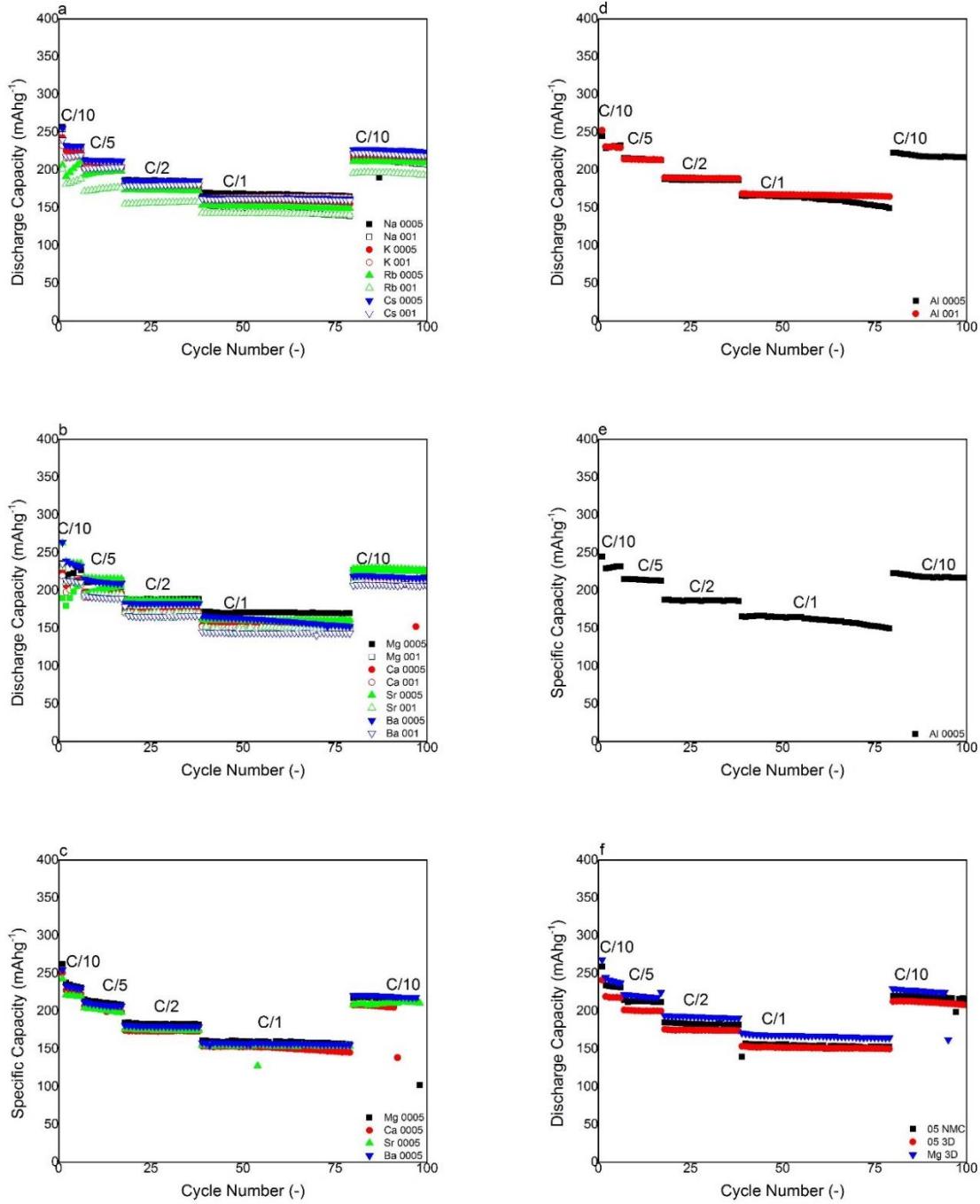
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79 Fig. S3 - Initial charge and discharge profiles at a constant current density of 20 mAg^{-1} between
80 2.0 and 4.8 V for $\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$ samples doped with typically 0.005 or 0.01 level of (a)
81 Na, K, Rb or Cs substituting Li; (b) Mg, Ca, Sr or Ba substituting Co; (c) Mg, Ca, Sr or Ba
82 substituting Ni; (d) Al substituting Mn; (e) Al substituting Ni; (f) $\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$ triply
83 substituted with Na, Ba, Al indicated as “05 3D” or Na, Mg, Al indicated as “Mg 3D” at a total
84 dopant level of 0.01 with equal amount of dopant. All materials display two activation-plateaus
85 during the initial charge cycle characteristic of the layered compounds.



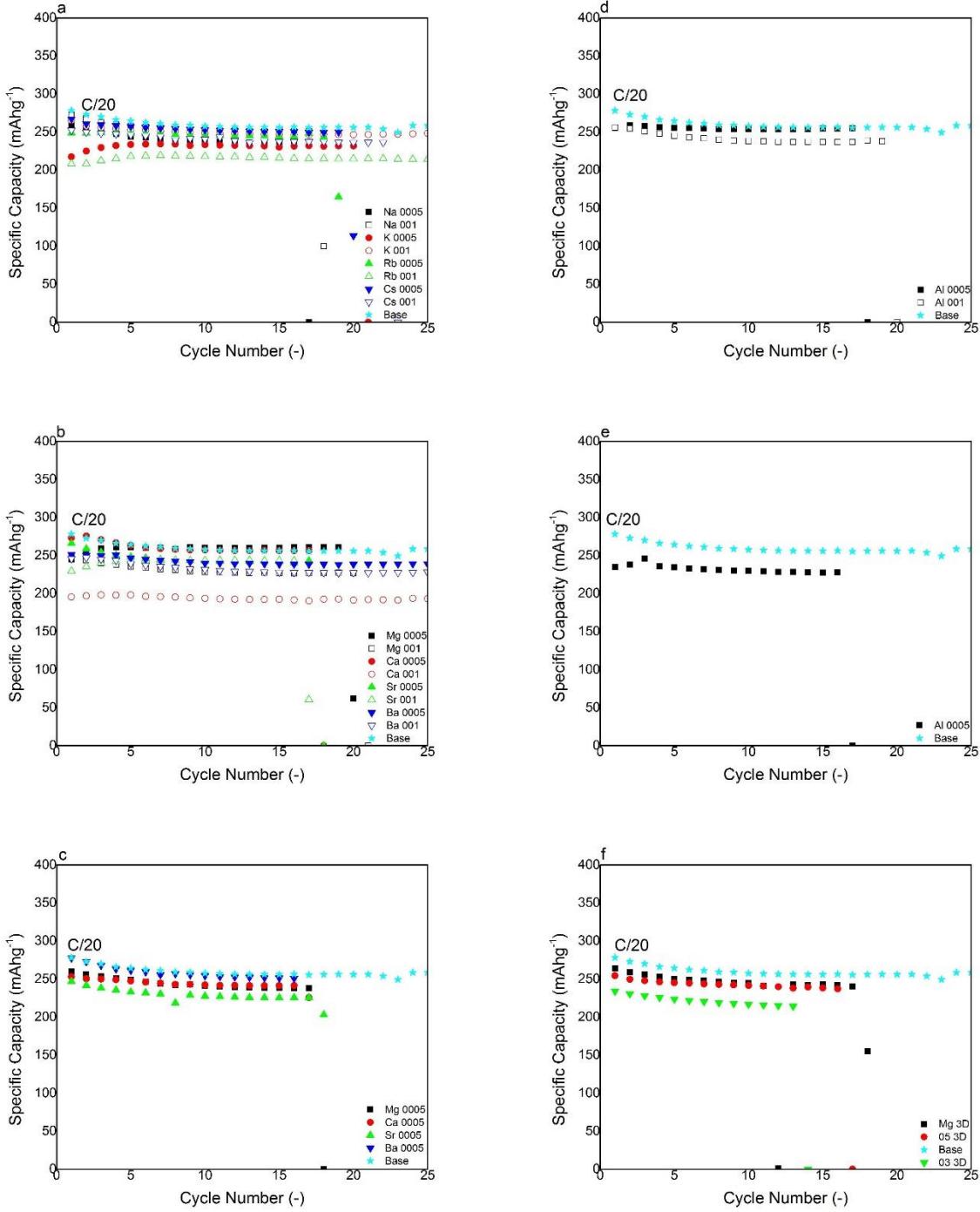
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87 Fig. S4 - Cycling performance of cells doped with (a) 0.005 or 0.01 levels of Na, K, Rb or Cs
88 substituting Li; (b) 0.005 or 0.01 levels of Mg, Ca, Sr or Ba substituting Co; (c) 0.005 Mg, Ca, Sr
89 or Ba substituting Ni; (d) 0.005 or 0.01 level of Al substituting Mn; (e) 0.005 level of Al
90 substituting Ni; (f) Li_{1.2}Mn_{0.54}Ni_{0.13}Co_{0.13}O₂ triply substituted with Na, Ba, Al indicated as “05
91 3D” or Na, Mg, Al indicated as “Mg 3D” at a total dopant level of 0.01 with equal amount of
92 dopant. C/10 equals a current density of 20 mA⁻¹, while C/3 equals 66.67 mA⁻¹.

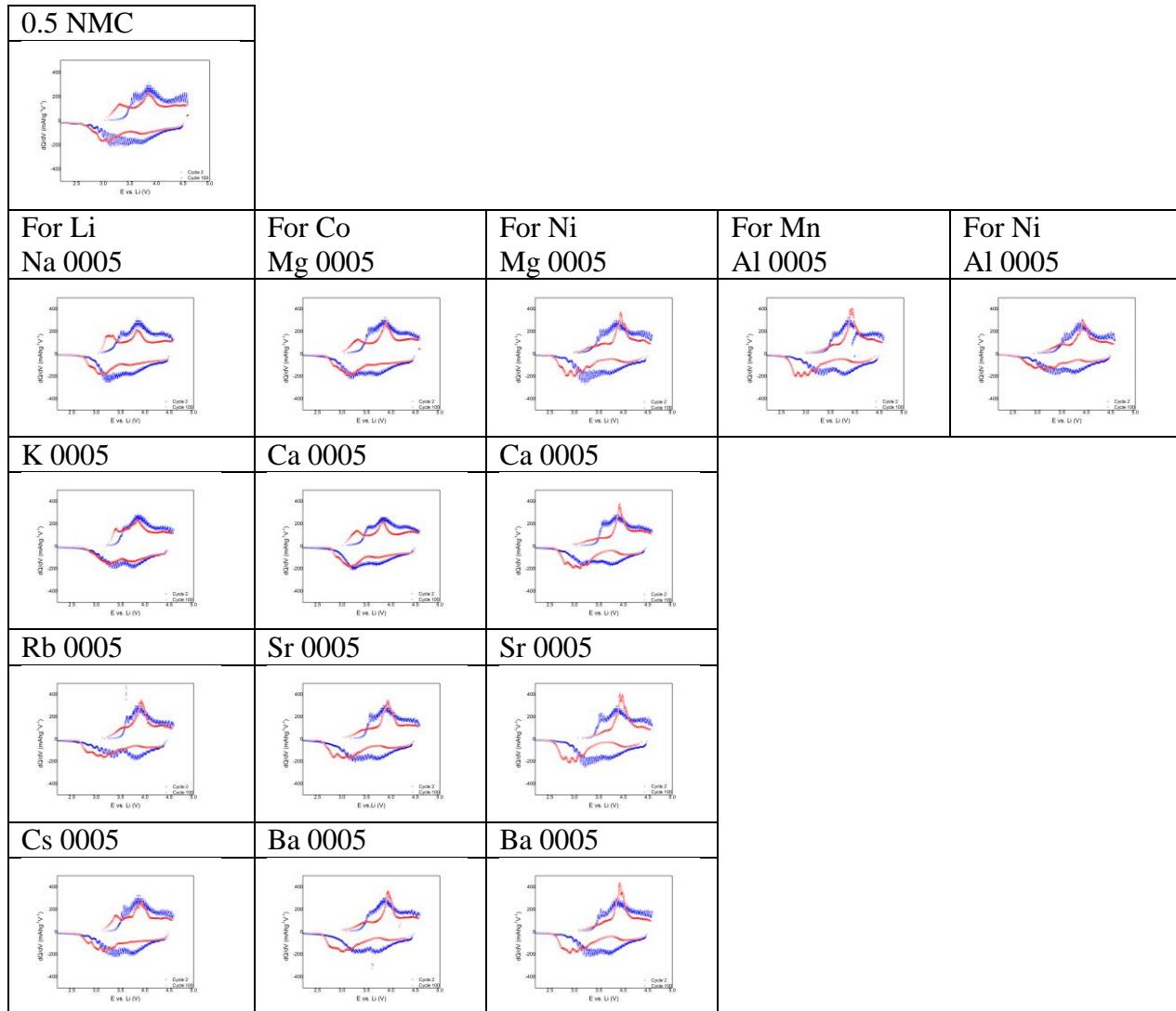


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94 Fig. S5 – Rate capability test performance of cells doped with (a) 0.005 or 0.01 levels of Na, K,
95 Rb or Cs substituting Li; (b) 0.005 or 0.01 levels of Mg, Ca, Sr or Ba substituting Co; (c) 0.005
96 Mg, Ca, Sr or Ba substituting Ni; (d) 0.005 or 0.01 level of Al substituting Mn; (e) 0.005 level of
97 Al substituting Ni; (f) Li_{1.2}Mn_{0.54}Ni_{0.13}Co_{0.13}O₂ triply substituted with Na, Ba, Al indicated as “05
98 3D” or Na, Mg, Al indicated as “Mg 3D” at a total dopant level of 0.01 with equal amount of
99 dopant. C/10, C/5, C/2 and C/1 correspond to a current density of 20, 40, 100, 200 mAg⁻¹,
100 respectively.



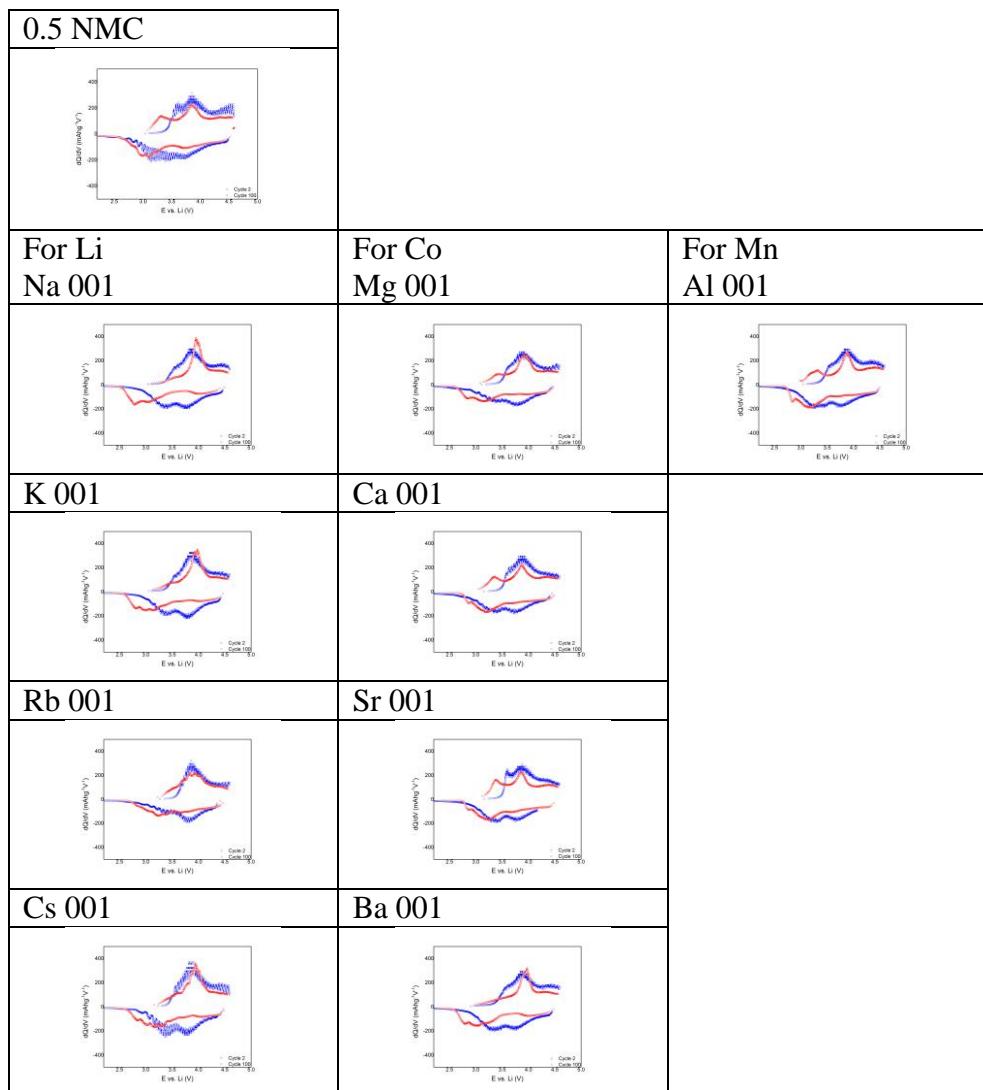
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102 Fig. S6 - Cycling performance of cells doped with (a) 0.005 or 0.01 levels of Na, K, Rb or Cs
103 substituting Li; (b) 0.005 or 0.01 levels of Mg, Ca, Sr or Ba substituting Co; (c) 0.005 Mg, Ca, Sr
104 or Ba substituting Ni; (d) 0.005 or 0.01 level of Al substituting Mn; (e) 0.005 level of Al
105 substituting Ni; (f) $\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$ triply substituted with Na, Ba, Al indicated as “05
106 3D” or Na, Mg, Al indicated as “Mg 3D” at a total dopant level of 0.01 with equal amount of
107 dopant $\text{Li}_{1.14}\text{Mn}_{0.46}\text{Ni}_{0.2}\text{Co}_{0.2}\text{O}_2$ triply substituted with Na, Ba, Al indicated as “03 3D”. The
108 dopant free sample is added to all curves as a teal star for reference.



109 Fig. S7 – dQ/dV curves of cycle 2 and cycle 100 of the doped samples when cycled at C/3 rate
110 (66.67 mAg⁻¹) between 2.0-4.6 V. The curves are organized according to the position of the
111 dopant element in the periodic table and above each curve the dopant is indicated. All dopant
112 levels in the current table are fixed at 0.005.

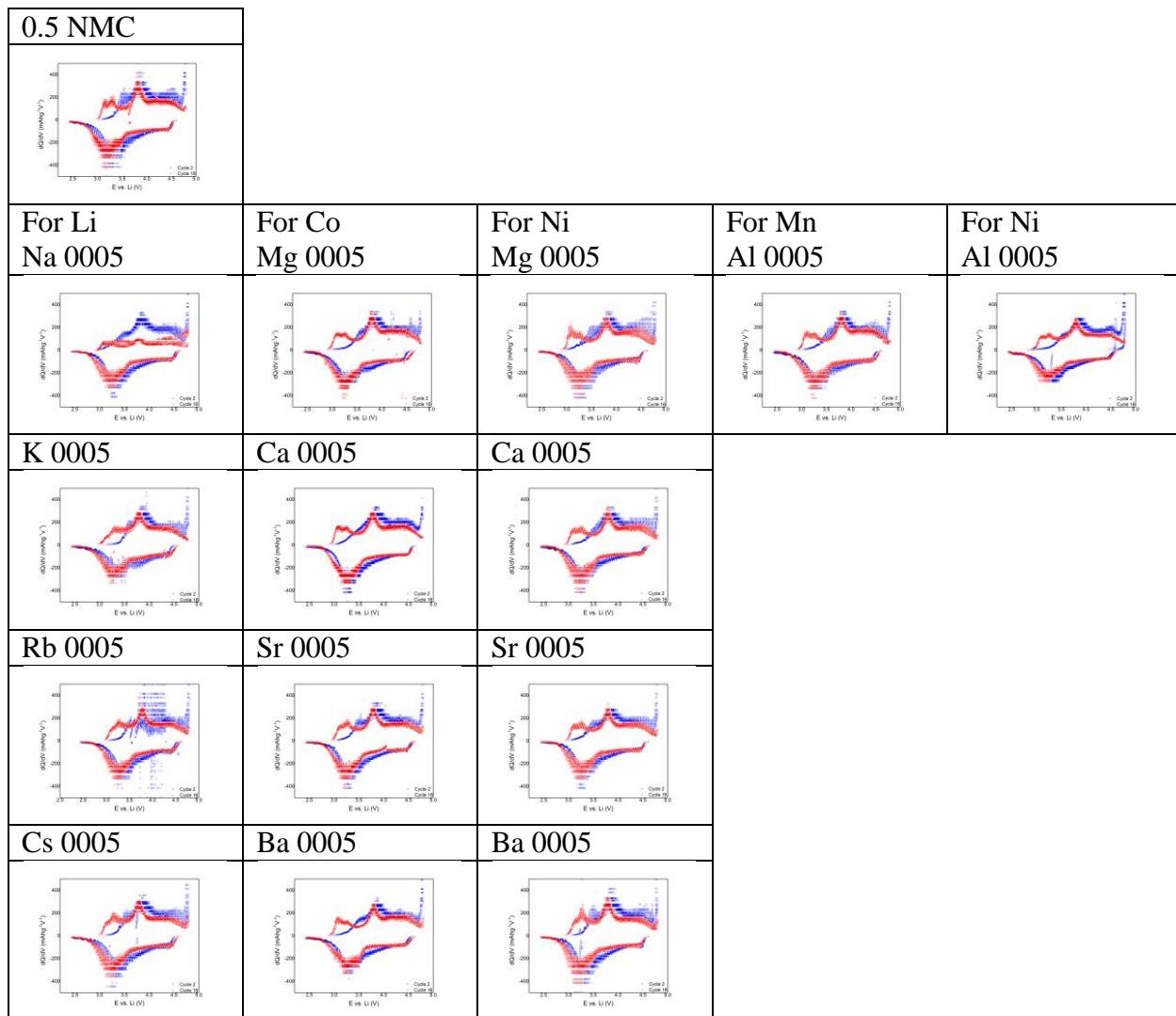
113 Despite the ability of Al to improve capacity retention when substituted for Mn, Al does not
114 appear to reduce the growth of the spinel charge peak that indicates voltage fade. This implies
115 that Al doping improves capacity retention via a different mechanism, for example by reducing
116 the strains along the *c* axis during cycling, as reported for several LiMn_xO_y based spinel
117 materials^{27, 28}.

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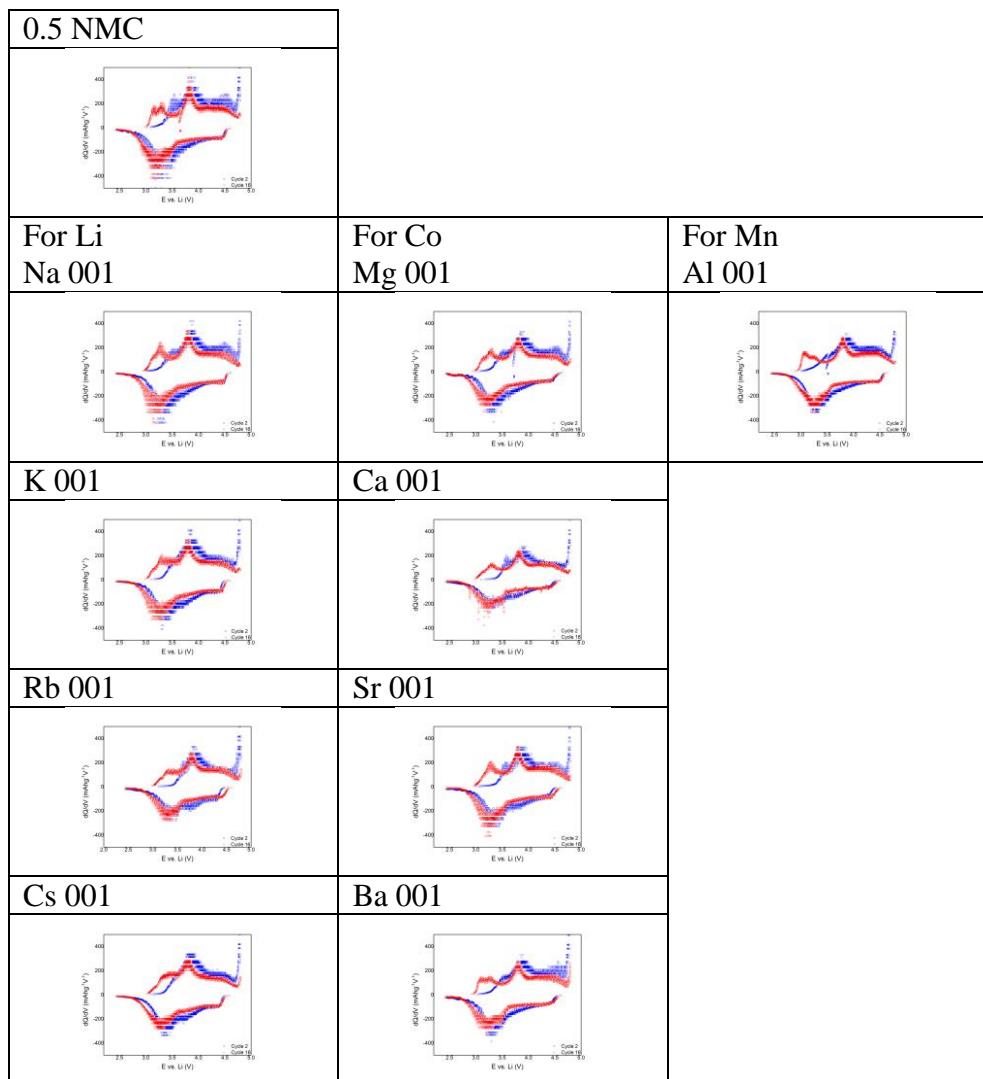
119 Fig. S8 - dQ/dV curves of cycle 2 and cycle 100 of the doped samples when cycled at C/3 rate
120 (66.67 mA g^{-1}) between 2.0-4.6 V. The curves are organized according to the position of the
121 dopant element in the periodic table and above each curve the dopant is indicated. All dopant
122 levels in the current table are fixed at 0.01. Ni-doping was omitted from the plot. When
123 compared to the results in Figure S7 it is clear that doubling the dopant levels leads to more
124 significant suppression of the spinel peaks in the charge curves for the alkali and alkali earth
125 metals. Increasing the Al dopant's level for Mn however seems to lead to additional peaks
126 forming in the structure.

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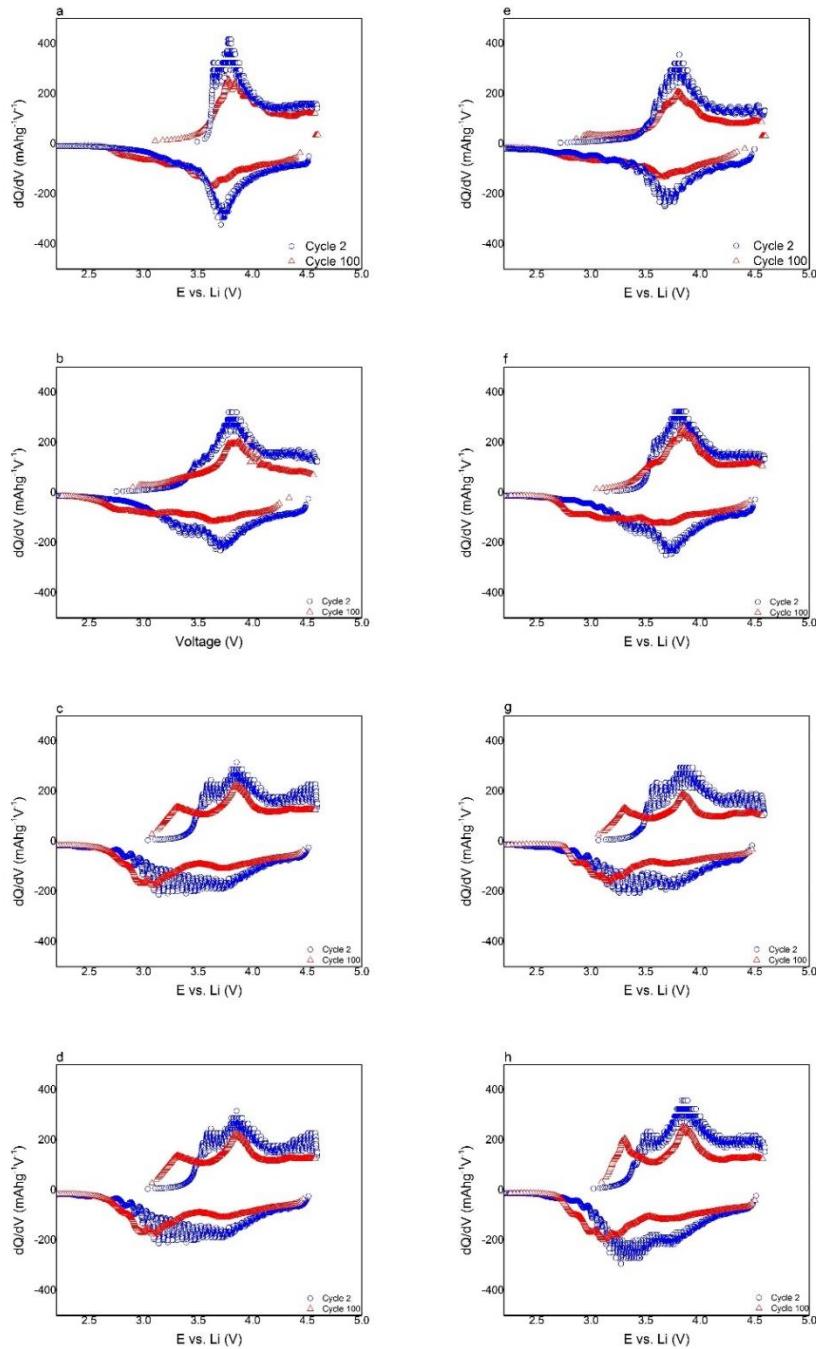
129 Fig. S9 - dQ/dV curves of cycle 2 and cycle 16 of the doped samples when cycled at C/20 rate
130 (10 mAg^{-1}) between 2.0-4.8 V. All dopant levels in the current table are fixed at 0.005. While
131 improvements can be observed when cycled between 2.0-4.6 V, the dopants have minor effect on
132 the spinel peak growth and peak shift. Additionally, it is clear that all samples still display the
133 original 4.6 V activation peak in the charge profiles at these low currents in cycle 2, which
134 disappears gradually by cycle 16. Similar trends were reported before by Yabuuchi et al.²⁹ for 0.5
135 NMC, but it is a surprising result for the doped samples.

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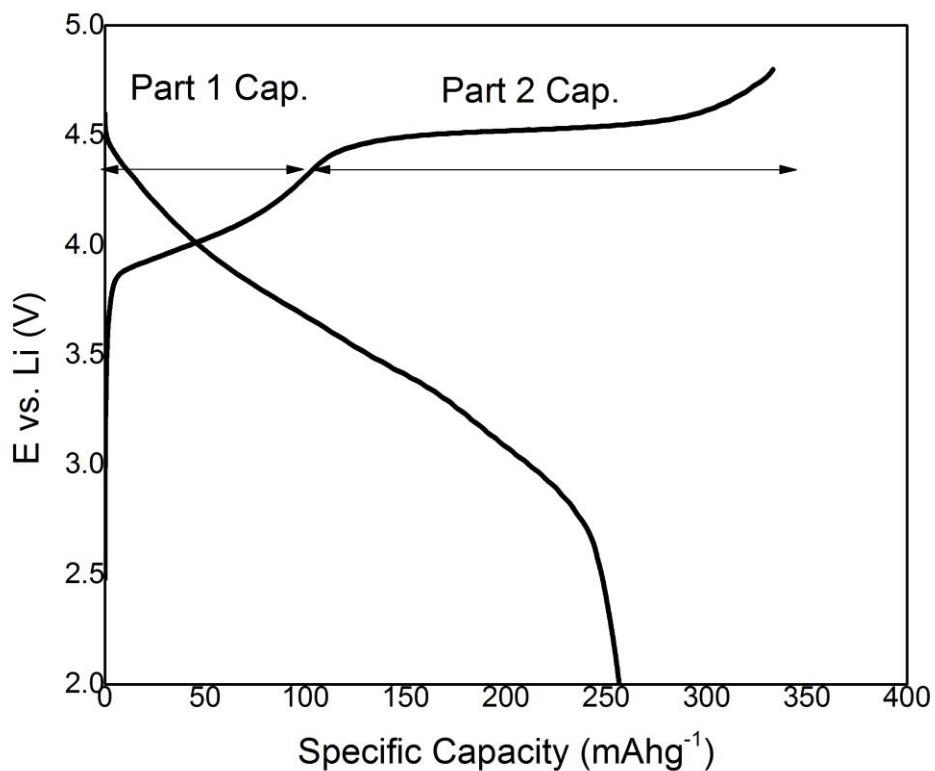
137 Fig. S10 - dQ/dV curves of cycle 2 and cycle 16 of the doped samples when cycled at C/20 rate
138 (10 mAg^{-1}) between 2.0-4.8 V. The curves are organized according to the position of the dopant
139 element in the periodic table and above each curve the dopant is included. All dopant levels in
140 the current table are fixed at 0.01.

141



142

143 Fig. S11 - dQ/dV curves of cycle 2 and cycle 100 of (a) $\text{Li}_{1.09}\text{Mn}_{0.43}\text{Ni}_{0.24}\text{Co}_{0.24}\text{O}_2$ (850 °C); (b)
 144 $\text{Li}_{1.14}\text{Mn}_{0.46}\text{Ni}_{0.2}\text{Co}_{0.2}\text{O}_2$ (850 °C); (c) $\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$ (900 °C); (d)
 145 $\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$ (900 °C); (e) $\text{Li}_{1.08}\text{Na}_{0.01}\text{Mn}_{0.42}\text{Al}_{0.01}\text{Ni}_{0.24}\text{Co}_{0.23}\text{Ba}_{0.01}\text{O}_2$ (800 °C); (f)
 146 $\text{Li}_{1.13}\text{Na}_{0.01}\text{Mn}_{0.45}\text{Al}_{0.01}\text{Ni}_{0.2}\text{Co}_{0.19}\text{Ba}_{0.01}\text{O}_2$ (800 °C); (g)
 147 $\text{Li}_{1.197}\text{Na}_{0.003}\text{Mn}_{0.541}\text{Al}_{0.003}\text{Ni}_{0.13}\text{Co}_{0.12}\text{Ba}_{0.003}\text{O}_2$ (900 °C); (h)
 148 $\text{Li}_{1.197}\text{Na}_{0.003}\text{Mn}_{0.541}\text{Al}_{0.003}\text{Ni}_{0.13}\text{Co}_{0.12}\text{Mg}_{0.003}\text{O}_2$ (900 °C);. The temperatures after the
 149 compositions indicate the annealing temperature of the samples. Effects of the annealing
 150 temperature on these compositions via spray pyrolysis were reported earlier³⁰.



151
152 Fig. S12 – A representative first cycle curve displaying the method for estimating capacities in
153 Table S2.
154
155

156 Table S1 - List of doped samples and dopants synthesized in the current study

	Chemical formula	Individual dopant levels
Alkali	$\text{Li}_{1.2-x}\text{Na}_x\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$	0.005, 0.01, 0.025, 0.05, 0.1
	$\text{Li}_{1.2-x}\text{K}_x\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$	0.005, 0.01
	$\text{Li}_{1.2-x}\text{Rb}_x\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$	0.005, 0.01
	$\text{Li}_{1.2-x}\text{Cs}_x\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$	0.005, 0.01
Alkali earth	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13-z}\text{Mg}_z\text{O}_2$	0.005, 0.01, 0.02
	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13-z}\text{Ca}_z\text{O}_2$	0.005, 0.01
	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13-z}\text{Sr}_z\text{O}_2$	0.005, 0.01
	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13-z}\text{Ba}_z\text{O}_2$	0.005, 0.01
	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13-z}\text{Co}_{0.13}\text{Mg}_z\text{O}_2$	0.005
	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13-z}\text{Co}_{0.13}\text{Ca}_z\text{O}_2$	0.005
	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13-z}\text{Co}_{0.13}\text{Sr}_z\text{O}_2$	0.005
	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13-z}\text{Co}_{0.13}\text{Ba}_z\text{O}_2$	0.005
Al	$\text{Li}_{1.2}\text{Mn}_{0.54-y}\text{Al}_y\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$	0.005, 0.01
	$\text{Li}_{1.2}\text{Mn}_{0.54}\text{Al}_y\text{Ni}_{0.13-y}\text{Co}_{0.13}\text{O}_2$	0.005
Multi-doped samples	$\text{Li}_{1.19}\text{Na}_{0.01}\text{Mn}_{0.53}\text{Al}_{0.01}\text{Ni}_{0.13}\text{Co}_{0.12}\text{Ba}_{0.01}\text{O}_2$	0.01
	$\text{Li}_{1.197}\text{Na}_{0.003}\text{Mn}_{0.541}\text{Al}_{0.003}\text{Ni}_{0.13}\text{Co}_{0.12}\text{Ba}_{0.003}\text{O}_2$	0.0033
	$\text{Li}_{1.197}\text{Na}_{0.003}\text{Mn}_{0.541}\text{Al}_{0.003}\text{Ni}_{0.13}\text{Co}_{0.12}\text{Mg}_{0.003}\text{O}_2$	0.0033
	$\text{Li}_{1.13}\text{Na}_{0.01}\text{Mn}_{0.45}\text{Al}_{0.01}\text{Ni}_{0.2}\text{Co}_{0.19}\text{Ba}_{0.01}\text{O}_2$	0.01
	$\text{Li}_{1.13}\text{Na}_{0.01}\text{Mn}_{0.45}\text{Al}_{0.01}\text{Ni}_{0.2}\text{Co}_{0.2}\text{O}_2$	0.01
	$\text{Li}_{1.08}\text{Na}_{0.01}\text{Mn}_{0.42}\text{Al}_{0.01}\text{Ni}_{0.24}\text{Co}_{0.23}\text{Ba}_{0.01}\text{O}_2$	0.01
	$\text{Li}_{1.08}\text{Na}_{0.01}\text{Mn}_{0.42}\text{Al}_{0.01}\text{Ni}_{0.24}\text{Co}_{0.24}\text{O}_2$	0.01

158 Table S2 – Capacity of selected doped samples of $\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$ with a dopant level of
 159 (a) 0.005 and (b) 0.01. 05 3D and Mg 3D stand for a triple doped sample of
 160 $\text{Li}_{1.2}\text{Mn}_{0.54}\text{Ni}_{0.13}\text{Co}_{0.13}\text{O}_2$ with Na, Ba and Al and Na, Mg and Al substituted in a total
 161 concentration of 0.01, respectively. The capacities were estimated by measuring the different
 162 capacities until the inflection point in the initial charge and discharge curve as indicated in Fig.
 163 S12.

164 a,

	Part 1 Capacity mAhg ⁻¹	Part 2 Capacity mAhg ⁻¹	Total Capacity mAhg ⁻¹
Theoretical 05 NMC	138.9	229	367.9
Base	132	225	357
Na	130	207	337
K	130	194	324
Rb	130	185	315
Cs	130	203	333
Al	132	195	327
Mg	150	196	346
Ca	130	202	332
Sr	125	209	334
Ba	132	198	330

165 b,

	Part 1 Capacity mAhg ⁻¹	Part 2 Capacity mAhg ⁻¹	Total Capacity mAhg ⁻¹
Theoretical 05 NMC	138.9	229	367.9
Base	132	225	357
Na	130	186	316
K	130	193	323
Rb	130	170	300
Cs	130	222	352
Al	128	209	337
Mg	110	211	321
Ca	100	216	316
Sr	130	202	332
Ba	120	217	337
05 3D	128	220	348
Mg 3D	128	221	349

166

167 Table S3 – Comparison of the effect of different dopants on the rate capability of the doped
 168 samples at identical dopant levels and current densities. When Ni is substituted with aluminum
 169 vs. Mn substitution the rate capability is considerably inferior at any rate. For alkali earth
 170 dopants replacing Ni instead of Co lowers the rate capability of the material, particularly at
 171 higher C-rates.

	Cycle 1 @ C/20	Cycle 1 @ C/10	Cycle 1 @ C/5	Cycle 1 @ C/2	Cycle 1 @ C/1
Al 0005	255.4	244.6	215.4	187.8	165.7
Ni-Al	235.1	217.0	180.9	156.1	133.3
% Change For Mn-Ni	108.6	112.7	119.1	120.3	124.3
Mg 0005	244.6	235.7	210.6	188.3	171.9
Ni-Mg	259.9	262.6	215.2	184.4	161.2
% Change For Co-Ni	94.1	89.8	97.8	102.1	106.6
Ca 0005	273.2	224.6	197.7	172.9	153.9
Ni-Ca	252.9	250.6	206.5	174.5	153.6
% Change For Co-Ni	108.0	89.6	95.8	99.1	100.2
Sr 0005	266.0	262.8	217.3	187.7	161.5
Ni-Sr	246.8	242.6	204.2	176.0	154.6
% Change For Co-Ni	107.8	108.3	106.4	106.7	104.5
Ba 0005	251.4	263.7	214.9	183.3	166.2
Ni-Ba	277.5	254.6	212.0	181.2	158.1
% Change For Co-Ni	90.6	103.5	101.4	101.2	105.1

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173

174 Table S4 – (a) Observed first cycle charge and discharge capacity and; (b) % capacity lost
 175 compared to dopant-free sample.

176 a,

For Li		Na 0005	K 0005	Rb 0005	Cs 0005	Na 001	K 001	Rb 001	Cs 001
Cycles 1 (mAhg ⁻¹)	Charge	337.2	324.5	316.0	333.0	316.7	323.5	300.9	352.7
	Discharge	261.5	226.5	210.2	247.8	232.2	233.2	184.6	252.0

177

For Co		Mg 0005	Ca 0005	Sr 0005	Ba 0005	Mg 001	Ca 001	Sr 001	Ba 001
Cycles 1 (mAhg ⁻¹)	Charge	346.8	332.4	334.3	330.2	321.3	316.2	332.8	337.3
	Discharge	264.6	240.8	243.2	242.2	226.4	226.8	228.9	241.3

178

		For Mn Al 0005	For Mn Al 001	05 3D	Mg 3D	03 NMC	03 3D NMC	0.5 NMC
Cycles 1 (mAhg ⁻¹)	Charge	327.1	338.0	340.4	370.1	328.9	309.0	356.5
	Discharge	233.4	243.3	257.4	296.9	233.9	228.8	271.5

179

Ni-sub		Mg 0005	Ca 0005	Sr 0005	Ba 0005	Al 0005
Cycles 1 (mAhg ⁻¹)	Charge	337.6	312.9	318.1	319.0	290.7
	Discharge	257.4	228.0	239.4	241.2	221.9

180 b,

For Li		Na 0005	K 0005	Rb 0005	Cs 0005	Na 001	K 001	Rb 001	Cs 001
	Charge	5.4	9.0	11.4	6.6	11.2	9.3	15.6	1.1
Cycles 1	Discharge	3.7	16.6	22.6	8.7	14.5	14.1	32.0	7.2

181

For Co		Mg 0005	Ca 0005	Sr 0005	Ba 0005	Mg 001	Ca 001	Sr 001	Ba 001
	Charge	2.7	6.8	6.2	7.4	9.9	11.3	6.7	5.4
Cycles 1	Discharge	2.5	11.3	10.4	10.8	16.6	16.5	15.7	11.1

182

		For Mn Al 0005	For Mn Al 001	05 3D	Mg 3D	03 3D NMC	0.5 NMC
	Charge	8.3	5.2	4.5	0.0	6.0	0
Cycles 1	Discharge	14.0	10.4	5.2	0.0	2.2	0

183

Ni-sub		Mg 0005	Ca 0005	Sr 0005	Ba 0005	Al 0005
	Charge	5.3	12.2	10.8	10.5	18.5
Cycles 1	Discharge	5.2	16.0	11.8	11.1	18.3

184 **References:**

- 185 1. M. M. Thackeray, S.-H. Kang, C. S. Johnson, J. T. Vaughey, R. Benedek and S. A.
186 Hackney, *J. Mater. Chem.*, 2007, **17**, 3112-3125.
- 187 2. W. C. West, J. Soler and B. V. Ratnakumar, *J. Power Sources*, 2012, **204**, 200-204.
- 188 3. F. Amalraj, D. Kovacheva, M. Talianker, L. Zeiri, J. Grinblat, N. Leifer, G. Goobes, B.
189 Markovsky and D. Aurbach, *J. Electrochem. Soc.*, 2010, **157**, A1121-A1130.
- 190 4. Y. J. Hong, J. H. Kim, M. H. Kim and Y. C. Kang, *Mat. Res. Bull.*, 2012, **47**, 2022-2026.
- 191 5. X. Zhang, M. Lengyel and R. L. Axelbaum, *AIChE J.*, 2014, **60**, 443-450.
- 192 6. K. Kang and G. Ceder, *Phys. Rev. B*, 2006, **74**, 094105.
- 193 7. W. He, D. Yuan, J. Qian, X. Ai, H. Yang and Y. Cao, *J. Mater. Chem. A*, 2013, **1**, 11397.
- 194 8. S. Kim, X. Ma, S. P. Ong and G. Ceder, *Phys. Chem. Chem. Phys.*, 2012, **14**, 15571-
195 15578.
- 196 9. Y.-K. Sun, S.-W. Cho, S.-W. Lee, C. S. Yoon and K. Amine, *J. Electrochem. Soc.*, 2007,
197 **154**, A168-A172.
- 198 10. W. C. West, J. Soler, M. C. Smart, B. V. Ratnakumar, S. Firdosy, V. Ravi, M. S.
199 Anderson, J. Hrbacek, E. S. Lee and A. Manthiram, *J. Electrochem. Soc.*, 2011, **158**,
200 A883-A889.
- 201 11. F. Amalraj, M. Talianker, B. Markovsky, L. Burlaka, N. Leifer, G. Goobes, E. M.
202 Erickson, O. Haik, J. Grinblat, E. Zinigrad, D. Aurbach, J. K. Lampert, J.-Y. Shin, M.
203 Schulz-Dobrick and A. Gasuch, *J. Electrochem. Soc.*, 2013, **160**, A2220-A2233.
- 204 12. J. Cabana, S.-H. Kang, C. S. Johnson, M. M. Thackeray and C. P. Grey, *J. Electrochem.
205 Soc.*, 2009, **156**, A730-A736.
- 206 13. B. V. R. Chowdari, G. V. S. Rao and S. Y. Chow, *Solid State Ionics*, 2001, **140**, 55-62.
- 207 14. R. Pelosato, C. Cristiani, G. Dotelli, S. Latorrata, R. Ruffo and L. Zampori, *J. Power
208 Sources*, 2010, **195**, 8116-8123.
- 209 15. W. Luo, F. Zhou, X. Zhao, Z. Lu, X. Li and J. R. Dahn, *Chem. Mater.*, 2010, **22**, 1164-
210 1172.
- 211 16. W. Chung, H. Jung, C. H. Lee and S. H. Kim, *Thin Solid Films*, 2013, **546**, 98-103.
- 212 17. A. H. Tavakoli, H. Kondo, Y. Ukyo and A. Navrotsky, *J. Electrochem. Soc.*, 2013, **160**,
213 A302-A305.
- 214 18. P. Kuiper, G. Kruizinga, J. Ghijsen, G. A. Sawatzky and H. Verweij, *Phys. Rev. Lett.*,
215 1989, **62**, 221.
- 216 19. M. Abbate, F. M. F. d. Groot, J. C. Fuggle, A. Fujimori, Y. Tokura, Y. Fujishima, O.
217 Strelbel, M. Domke, G. Kaindl, J. v. Elp, B. T. Thole, G. A. Sawatzky, M. Sacchi and N.
218 Tsuda, *Phys. Rev. B*, 1991, **44**, 5419.
- 219 20. L. A. Montoro, M. Abbate and J. M. Rosolen, *J. Electrochem. Soc.*, 2000, **147**, 1651-
220 1657.
- 221 21. J. Graetz, A. Hightower, C. C. Ahn, R. Yazami, P. Rez and B. Fultz, *J. Phys. Chem. B*,
222 2001, **106**, 1286-1289.
- 223 22. Y. Uchimoto, H. Sawada and T. Yao, *J. Power Sources*, 2001, **97-98**, 326-327.
- 224 23. Y. Koyama, Y.-S. Kim, I. Tanaka and H. Adachi, *J. Appl. Phys.*, 1999, **38**, 2024-2027.
- 225 24. M. Sathiya, G. Rousse, K. Ramesha, C. P. Laisa, H. Vezin, M. T. Sougrati, M.-L.
226 Doublet, D. Foix, D. Gonbeau, W. Walker, A. S. Prakash, M. B. Hassine, L. Dupont and
227 J.-M. Tarascon, *Nat. Mater.*, 2013, **12**, 827.

- 228 25. M. Oishi, C. Yogi, I. Watanabe, T. Ohta, Y. Orikasa, Y. Uchimoto and Z. Ogumi, *J.*
229 *Power Sources*, 2015, **276**, 89-94.
230 26. T. Ohzuku, M. Nagayama, K. Tsuji and K. Ariyoshi, *J. Mater. Chem.*, 2011, **21**, 10179.
231 27. T. E. Conry, A. Mehta, J. Cabana and M. M. Doeff, *Chem. Mater.*, 2012, **24**, 3307-3317.
232 28. T. E. Conry, A. Mehta, J. Cabana and M. M. Doeff, *J. Electrochem. Soc.*, 2012, **159**,
233 A1562-A1571.
234 29. N. Yabuuchi, K. Yoshii, S.-T. Myung, I. Nakai and S. Komaba, *J. Am. Chem. Soc.*, 2011,
235 **133**, 4404-4419.
236 30. M. Lengyel, X. Zhang, G. Atlas, H. L. Bretscher, I. Belharouak and R. L. Axelbaum, *J.*
237 *Electrochem. Soc.*, 2014, **161**, A1338-A1349.
238 31. D. Mohanty, S. Kalnaus, R. A. Meisner, K. J. Rhodes, J. Li, E. A. Payzant, D. L. W. III
239 and C. Daniel, *J. Power Sources*, 2013, **229**, 239-248.
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