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Electronic Supplementary Information

Novel Al₂Mo₃O₁₂-based temperature-stable microwave dielectric ceramics for LTCC applications

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Figure S1 XRD patterns of the obtained Al₂Mo₃O₁₂ powders calcined at 550-750 °C for 4 h.

From Figure S1, it is seen that although for the sample prepared at the calcination temperature of 700 °C, Al₂Mo₃O₁₂ phase has been detected from the XRD patterns, Al₂O₃ and MoO₃ phases could still be identified from the patterns of the samples calcined at 550-700 °C, which are originating from the applied source powders. At the lowest synthesis temperature of 750 °C, pure Al₂Mo₃O₁₂ with monoclinic phase (JCPDS card no. 84-1652) could be obtained.



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Figure S2 (a) XRD patterns of the obtained $Al_2Mo_3O_{12}$ ceramics sintered at 750-830 °C for 12 h, and (b) an elaborated observation on the diffraction peak around 37.7°.

It is seen in **Figure S2a** that, the main phase of all the samples is a monoclinic structure of $Al_2Mo_3O_{12}$ (JCPDS card no. 84-1652). However, through an elaborated observation on the characteristic diffraction peak at the 2 θ angle around 37.7° (see **Figure S2b**), Al_2O_3 phase could be identified from the samples sintered at a temperature higher than 810 °C, because the volatilization of MoO₃ happens at such high sintering temperature, thus leaving Al_2O_3 in the samples alone.



Figure S3 Typical SEM images of the obtained Al₂Mo₃O₁₂ ceramics sintered at different temperatures for 12 h: (a) 750, (b) 770, (c) 790, (d) 810 and (e) 830 °C. (f) EDX data on the marked grain in (e).

As is seen in **Figure S3**, when sintering at 750-770 °C, there exist inhomogeneously sized $Al_2Mo_3O_{12}$ grains and many pores in the samples, indicating that the sintering is not complete. At 790-830 °C, denser samples with more homogeneous grain size could be obtained. However, when sintering at 830 °C, some dark impurities consisting of Al_2O_3 appears in the samples together with the main $Al_2Mo_3O_{12}$ phase (see **Figure S3f**).





Figure S4 Apparent density (a), dielectric constant (b), quality factor (c) and temperature coefficient of resonant

frequency (d) of the obtained Al₂Mo₃O₁₂ ceramics sintered at 750-830 °C for 12 h.

From **Figure S4**, it can be seen that the apparent density reaches the maximum when the sample is sintered at 790 °C. Meanwhile, the dielectric constant and quality factor of the obtained $Al_2Mo_3O_{12}$ ceramics reach their maximum value, respectively. The temperature coefficient of resonant frequency of the obtained ceramics decreased with increasing sintering temperature.



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Figure S5 XRD patterns of the obtained Al₂Mo₃O₁₂ ceramics sintered at 790 °C for 4-20 h.

As is seen in **Figure S5**, a single phase of monoclinic $Al_2Mo_3O_{12}$ (JCPDS card no. 84-1652) could be detected in all the samples, and no obvious second phases could be identified, indicating that $Al_2Mo_3O_{12}$ phase in the samples was stable at 790 °C.



Figure S6 Typical SEM images of the obtained Al₂Mo₃O₁₂ ceramics sintered at 790 °C for different times: (a) 4, (b) 8, (c) 12, (d) 16 and (e) 20 h. (f) EDX data on the marked grain in (e).

As is seen in **Figure S6**, when sintering at 790 °C for 4-8 h, there exist inhomogeneously sized $Al_2Mo_3O_{12}$ grains and many pores in the samples, indicating that the sintering is not complete. For 12-16 h, denser samples with more homogeneous grain size could be obtained. However, when sintering for 20 h, some dark impurities consisting of Al_2O_3 appears in the samples together with the main $Al_2Mo_3O_{12}$ phase (see **Figure S6f**).





Figure S7 Apparent density (a), dielectric constant (b), quality factor (c) and temperature coefficient of resonant frequency (d) of the obtained Al₂Mo₃O₁₂ ceramics sintered at 790 °C for 4-20 h.

From **Figure S7**, it can be seen that the apparent density reaches the maximum when the sample is sintered for 12 h at 790 °C. Meanwhile, the dielectric constant and quality factor of the obtained $Al_2Mo_3O_{12}$ ceramics reach their maximum value, respectively. The temperature coefficient of resonant frequency of the obtained ceramics decreased with extending sintering time.



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Figure S8 (a) XRD patterns of the obtained 0.6Al₂Mo₃O₁₂-0.4TiO₂ ceramics sintered at 750-830 °C for 12 h, and (b) an elaborated observation on the diffraction peak around 37.7°.

From **Figure S8a**, it can be seen that monoclinic $Al_2Mo_3O_{12}$ (JCPDS card no. 84-1652) and rutile TiO₂ (JCPDS card no. 21-1276) phases could be detected in all the samples, indicating that TiO₂ could coexist with $Al_2Mo_3O_{12}$ phase. However, through an elaborated observation on the characteristic diffraction peaks at the 20 angles around 37.7° (see **Figure S8b**), Al_2O_3 phase could be identified from the samples sintered at a temperature higher than 790 °C. Thus, from the point of phase composition, 790 °C was the optimized sintering temperature for $0.6Al_2Mo_3O_{12}$ - $0.4TiO_2$ ceramics.



Figure S9 Typical SEM images of the obtained 0.6Al₂Mo₃O₁₂-0.4TiO₂ ceramics sintered at different temperatures for 12 h: (a) 750, (b) 770, (c) 790 C, (d) 810 °C and (e) 830 °C. (f) EDX data on the marked grain in (e).

As is seen in **Figure S9(a-c)**, after adding TiO₂, the average grain size of $Al_2Mo_3O_{12}$ decreased. When sintering at a temperature below 790 °C, there exist inhomogeneously sized $Al_2Mo_3O_{12}$ grains and many pores in the samples, indicating that the sintering is not complete. At 790 °C, denser samples with more homogeneous grain size could be obtained. However, when sintering at a temperature higher than 790 °C, some dark impurities consisting of Al_2O_3 appears in the samples together with the main $Al_2Mo_3O_{12}$ phase (see **Figure S9e**).



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Figure S10 Apparent density (a), dielectric constant (b), quality factor (c) and temperature coefficient of resonant

frequency (d) of the obtained 0.6Al₂Mo₃O₁₂-0.4TiO₂ ceramics sintered at 750-830 °C for 12 h.

From **Figure S10**, it is seen that the apparent density reaches the maximum of 95.5% when the sample is sintered at 790 °C. Meanwhile, the dielectric constant and quality factor of the obtained $Al_2Mo_3O_{12}$ ceramics reach their maximum value, respectively. The temperature coefficient of resonant frequency of the obtained ceramics increased with increasing sintering temperature. Thus, from the point of microwave dielectric properties, 790 °C was the optimized sintering temperature for $0.6Al_2Mo_3O_{12}$ - $0.4TiO_2$ ceramics.



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Figure S11 XRD patterns of the obtained 0.6Al₂Mo₃O₁₂-0.4TiO₂ ceramics sintered at 790 °C for 4-20 h. As is seen in Figure S11, monoclinic Al₂Mo₃O₁₂ (JCPDS card no. 84-1652) and rutile TiO₂ (JCPDS card no. 21-1276) phases could be detected in all the samples, indicating that TiO₂ could coexist with Al₂Mo₃O₁₂ phase. No other phases could be identified, when the samples were sintered at 790 °C.

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Figure S12 Typical SEM images of the obtained 0.6Al₂Mo₃O₁₂-0.4TiO₂ ceramics sintered at 790 °C for different

times: (a) 4, (b) 8, (c) 12, (d) 16, and (e) 20 h. (f) EDX data on the marked grain in (e).

As is seen in **Figure S12**, with the extension of sintering time, the average grain size of $Al_2Mo_3O_{12}$ increased. When sintering at 790 °C for a time shorter than 12 h, there exist inhomogeneously sized $Al_2Mo_3O_{12}$ grains and many pores in the samples. At 790 °C for 12 h, denser samples with more homogeneous grain size could be obtained. However, when sintering at 790 °C for a time longer than 12 h, some dark impurities consisting of Al_2O_3 appears in the samples together with the main $Al_2Mo_3O_{12}$ phase (see **Figure S12e**).



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Figure S13 Apparent density (a), dielectric constant (b), quality factor (c) and temperature coefficient of resonant frequency (d) of the obtained 0.6Al₂Mo₃O₁₂-0.4TiO₂ ceramics sintered at 790 °C for 4-20 h.

From **Figure S13**, it is seen that the apparent density reaches the maximum when the sample is sintered for 12 h at 790 °C. Meanwhile, the dielectric constant and quality factor of the obtained $Al_2Mo_3O_{12}$ ceramics reach their maximum value, respectively. The temperature coefficient of resonant frequency of the obtained ceramics increased with the time of sintering time. Thus, from the point of microwave dielectric properties, 12 h was the optimized sintering temperature for $0.6Al_2Mo_3O_{12}$ - $0.4TiO_2$ ceramics.

In summary, $Al_2Mo_3O_{12}$ and $0.6Al_2Mo_3O_{12}$ -0.4TiO₂ ceramics with the optimum microwave dielectric properties and highest density could be obtained at 790 °C for 12 h.



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Figure S14 The grain size distribution of the obtained $(1-x)Al_2Mo_3O_{12}-xTiO_2$ ceramics sintered at 790 °C for 12 h with x=: (a) 0, (b) 0.1, (c) 0.2, (d) 0.3 and (e) 0.4. And (f) the average grain size of the $(1-x)Al_2Mo_3O_{12}-xTiO_2$ ceramics as a function of x value.

As is seen in **Figure S14**, with x from 0 to 0.4, the average size of $Al_2Mo_3O_{12}$ grain decreased from 6.09 to 4.9, indicating that the addition of TiO₂ could block the growth of $Al_2Mo_3O_{12}$ grains. Moreover, with increasing addition amount of TiO₂, the grain size distribution became more homogeneous.

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Figure S15 (a) Typical SEM image of the obtained 0.6Al₂Mo₃O₁₂-0.4TiO₂ ceramics sintered at 790 °C for 12 h,

and EDX data on the marked Points A (b) and B (c).

From Figure S15, it is seen that there are two kinds of color in the sample. The smaller white ones represent

 TiO_2 grains and the bigger grey ones shows $Al_2Mo_3O_{12}$ grains.