Strong localized surface plasmon resonance effects of Ag/TiO2 core-shell nanowire arrays in visible light for photocatalytic activity

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Deposition of TiO2 thin film:
The TiO2 thin film was deposited by widely used cathodic electrodeposition in aqueous electrolyte1 that contains 0.1M NH4NO3 and 0.05M (NH4)2TiF6.2 Some studies on the mechanism of TiO2 cathodic deposition in (NH4)2TiF6 aqueous solution explain that NO3− reduction works as the oxidant for Ti(IV) formation3 and significantly affects the formation of TiO2 film.4 The electrodeposition of TiO2 thin film via the reduction of NO3− and accumulation of OH− facilitates the TiO2 formation on the electrode by the combination of OH− and Ti(IV) species. More details of TiO2 cathodic electrodeposition can be found in the literature.5 Electrodeposition was performed at -1.1V (vs. Ag/AgCl reference electrode) as an optimal potential for reasonably high deposition rate while suppressing the evolution of hydrogen and nitrogen. Current-time (I-t) curve of TiO2 film electrodeposition is represented on inset of Fig. S1a. The current was stabilized at ~0.4mA after 80sec. Gel-phase TiO2 film was also fabricated by sol-gel spin coating method. The solution consisted of titanium isopropoxide (Ti(OC3H7)4) as the source for formation of gel TiO2 film.6 After deposition process, films were annealed at 80°C for 30 min to remove the contaminant. The surface morphology of Ag/gel TiO2 films by electrodeposition process is shown on Fig. S1a and gel-TiO2 by sol-gel spin coating method is shown on Fig. S1b. The thickness of both gel-TiO2 films which are fabricated by electrodeposition and sol-gel spin coating methods were measured as ~50nm by spectroscopic ellipsometer (Woollam, M2000D)).

XRD analysis of TiO2 layer:
For the X-ray diffraction (XRD) analysis of TiO2 thin film, the films with thickness over 200 nm were deposited due to the detection limit of the XRD equipment. Fig.S2a shows the XRD pattern of the TiO2 thin film that was electrodeposited on Si (100) substrate with thermally evaporated Ag thin film and thermally annealed at 300 °C for the formation of anatase-phase TiO2. First, diffraction peaks of Ag and shoulder at low angle range are observed. Also, the peaks for anatase phase ((101) and (200)) can be observed. The peaks of silver oxide (AgO) are not observed from the XRD data.

XPS analysis of Ag NW and Ag/TiO2 core-shell NW arrays:
The full range XPS survey spectra of A0 NW, A0/gel NW and A0/anatase NW samples are shown in Fig.S2b. The XPS results of Ti 2p binding energies for as-deposited A0/gel NW and A0/anatase NW are shown in Fig.S2c. The peak position of the Ti(2p3/2) shifted from 459.4 eV to 458.1 eV by the transformation of A0/gel NW to A0/anatase NW. The higher binding energy of A0/gel NW samples was caused by fluorine (F) intermediate on the surface because of the incomplete reaction.2,7 The peak of F in A0/gel NW at 687eV represents the substitutional F atom on oxygen site in the TiO2 (Fig.S2b)8. After annealing, gel film was changed to anatase film (Fig.S2a) and F was removed as shown in Fig.S2b. The peaks of Ti 2p binding energies of Ag/anatase NW at 458.1eV were in accord with anatase-phase TiO2 in the literature.9,10
Fig. S2. Surface characterization of fabricated samples: (a) X-ray diffraction pattern of Ag/electrodeposited-TiO$_2$ dual layer film heated at 300°C. (b) XPS spectra of A0 NW, A0/TiO$_2$ NW and A0/anatase NW, and (c) Ti 2p of A0/TiO$_2$ NW array and A0/anatase NW array.

Fig. S3. Measured absorption for transverse electric (TE) mode of (a) A0 NW array, A45 NW array, A60 NW array, (b) A0/TiO$_2$ NW array, A0/anatase NW array, A45/TiO$_2$ NW array and A60/TiO$_2$ NW array; (c) A0 NW array, A45 NW array, A60 NW array, A0/anatase NW array, A45/TiO$_2$ NW array and A60/TiO$_2$ NW array. Black dotted lines for each sub-figure represent non-polarized absorption of a) Ag thin film and b) Ag/gel-TiO$_2$ dual layer thin film, respectively.

FDTD Numerical Simulation:

The calculations of light absorption were performed by using commercially available 2D finite difference time domain (FDTD) software (Lumerical®, Lumerical Solutions, Inc.). The geometrical parameters for modeling were obtained from the SEM images of the fabricated samples. The structures were illuminated by a linearly polarized broadband incident plane wave. An infinite 2D array of unit cells was simulated using periodic boundary conditions at the left and right boundaries, while perfectly matched layer (PML) absorbing boundary conditions were set at the top and bottom boundaries. The size of mesh cell in the calculations was set as 1 x 1 nm. The Lumerical’s multi-coefficient model was used to fit the empirical dielectric constants of Ag, which were taken from Palik’s Handbook of Optical Constants. The dielectric constants of electrodeposited TiO$_2$ and anatase TiO$_2$ were measured by using a Spectroscopic Ellipsometer (M2000D, Woollam) and used in simulation. Measured refractive index (n) and extinction coefficient (k) of gel and anatase TiO$_2$ are presented on Fig. S5.
Fig. S5. Optical properties of the gel phase TiO$_2$ film and anatase TiO$_2$ film: (a) Refractive index ($n$) and (b) extinction coefficient ($k$)

References