Engineering Oxygen Vacancies towards Self-Activated BaLuAl_xZn_{4-x}O_{7-(1-x)/2} Photoluminescent Materials: An Experimental and Theoretical Analysis

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Supporting Information

Details of DFT calculation and level of theory:

The full potential linearized augmented plane wave (FPLAPW) method [1] is a procedure for solving the Kohn-Sham equations for the ground state density, total energy, and (Kohn-Sham) eigenvalues of a many-electron system by using a basis set. In FPLAPW method we divide the unit cell into (I) non-overlapping atomic spheres (centered at the atomic sites) and (II)

an interstitial region. In the interstitial region a plane wave expansion is used $\Phi_{k_n} = \frac{1}{\sqrt{\Omega}} e^{i k_n \cdot \vec{r}}$

; with
$$k_n = \kappa + K_n$$
 (1)

where κ is the wave vector in the Brillouin zone (BZ), K_n is a reciprocal lattice vectors and Ω is the volume of the unit cell. Each plane wave is augmented by an atomic-like function in every atomic sphere. Inside each atomic sphere, the solutions for a spherically symmetric potential are the atomic basis functions which consist of radial functions, $u_l(r, E)$, multiplied by spherical harmonics.

For typical valence states with azimuthal quantum number l, the radial function is regular at the origin and varies with energy. This energy dependence is linearized by selecting an expansion energy E_l , near the center of the band and writing the atomic function as:

$$\Phi_{k_n} = \sum_{lm} \left[A_{lm,k_n} u_l(r, E_l) + B_{lm,k_n} \mathcal{U}_l(r, E_l) \right] Y_{lm}(\hat{r})$$
⁽²⁾

where u_l is solution of the radial Schrödinger equation and $u_{lm}^{\mathbf{k}}$ is its energy derivative. For each atom and lm there are two free coefficients, A_{lm} and B_{lm} , which are chosen such that (at the sphere boundary) the plane wave with wave vector, \mathbf{k}_n , joins continuously on to the atomic function value and slope. These linearized augmented plane waves (LAPW's) form the basis for expanding the crystalline orbitals (Bloch states) and provide a rapidly converging series.

$$\Psi_{\kappa} = \sum_{n} C_{n} \Phi_{k_{n}} \tag{3}$$

All low lying core states are included as frozen core states (k independent but but calculated self-consistently by solving an atomic Dirac equation.

Sometimes it is necessary to extend the LAPW basis set with so called local orbitals (LO's)[2]. With such an extension, higher-level semi-core states can be treated, together with the valence states, in one LAPW calculation instead of two (valence and semi-core states) energy windows. This LO scheme avoids the problem of non-orthogonality that can occur in calculations in which the semi-core states are either frozen or treated in separate energy window. A LO is chosen such that it vanishes in value and slope at the muffin-tin radius and includes an additional radial function, corresponding to a different principle quantum number:

$$\Phi_{lm}^{LO} = \left[A_{lm} u_l(r, E_{1,l}) + B_{lm} u_{\ell}(r, E_{1,l}) + C_{lm} u_l(r, E_{2,l}) \right] Y_{lm}(\hat{r}).$$
(4)

The LO is then added to usual LAPW basis set in equation (3). The inclusion of LO's makes the computation of matrix elements somewhat more difficult but one has a better model and saves computational time, since no semi-core states need to be calculated in separate LAPW run.

In FPLAPW method the potential and charge density are expanded (a) into lattice harmonics (inside each atomic sphere) and (b) as a Fourier series (in interstitial region) and thus they are completely general, so that the scheme is termed as full-potential calculation

$$\sum_{LM} V_{LM} \left(\stackrel{f}{r} \right) Y_{LM} \left(\stackrel{f}{r} \right)$$
 Inside sphere
$$V \left(\stackrel{f}{r} \right) =$$

(5)

 $\sum_{K} V_{K} \exp\left(i\frac{1}{K}r\right) \qquad \text{Outside sphere}$

In order to have few LM values in the lattice harmonics equation (5), a local coordinate system for each atom sphere is defined, according to the point symmetry of the corresponding atom. This specifies a rotation matrix that relates the local to the global coordinate system of the unit cell.



Fig. S1. XRD patterns of nominal BaLuAl_{0.91}Zn_{3.09}O₇ phase prepared at different atmosphere, and the standard pattern (PDF# 85-0252) of BaLuAl_{0.91}Zn_{3.09}O₇ is also given as a comparison.



Fig. S2. Difference Rietveld plot $BaLuAl_{0.91}Zn_{3.09}O_7$ prepared in air (a), Ar (b), and N₂-H₂ (c) atmosphere.



Fig. S3. Photoluminescence excitation (left) and emission (right) spectra of nominal $BaLuAl_{0.91}Zn_{3.09}O_7$ phase prepared at different atmosphere.



Fig. S4. XRD patterns of the BaLuAlZn₃O₇, BaLuAl_{0.91}Zn_{3.09}O₇ and BaLuAl_{0.8}Zn_{3.2}O₇ samples depending on different Al/Zn ratio.



Fig. S5. Photoluminescence excitation (left) and emission (right) spectra of BaLu(Al,Zn)O₇ depending on different Al/Zn ratio, (a) Al/Zn = 1:3, (b) Al/Zn = 0.91/3.09, (c) Al/Zn = 0.8/3.2.



Fig. S6. XRD patterns of $BaLuAl_{0.91}Zn_{3.09}O_7$ phosphors prepared at different annealing temperatures.



Fig. S7. Photoluminescence excitation (left) and emission (right) spectra of BaLuAl_{0.91}Zn_{3.09}O₇ phosphors prepared at different annealing temperatures.



Fig. S8. Relaxed unit cells of (a) $BaLuAl_{0.91}Zn_{3.09}O_7$ and (b) that with two oxygen vacancies introduced at Q1 and O3 sites.



Fig. S9. Total DOS and partial DOS of constituted atoms for stoichiometric and defected $BaLuAl_{0.91}Zn_{3.09}O_7$ phosphor for different cases.

References

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