

Electronic Supplementary Information

Tuning SEI formation on nanoporous carbon-titania composite sodium ion batteries anodes and performance with subtle processing changes

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Figure S1 illustrates the morphology of the porous carbon/TiO₂ composite fabricated using 0.35 g PHEMA/0.3 g FA/0.3 g TiO₂ as a function of molecular weight of PHEMA (20K, 300K, and 1000K). The pores of the carbon/TiO₂ composite are significantly larger for the composite produced using 20K PHEMA (Figure S1A) than the carbon/TiO₂ composites from 300K (Figure S1B) and 1000K PHEMA (Figure S1C). The morphology of porous carbon/TiO₂ composites pyrolyzed from PHEMA/FA/TiO₂ appears to be controlled by the mobility of the precursor suspension that decreases as the molecular weight of PHEMA increases. Additionally, agglomeration of the TiO₂ nanoparticles increases as the molecular weight of PHEMA increases.

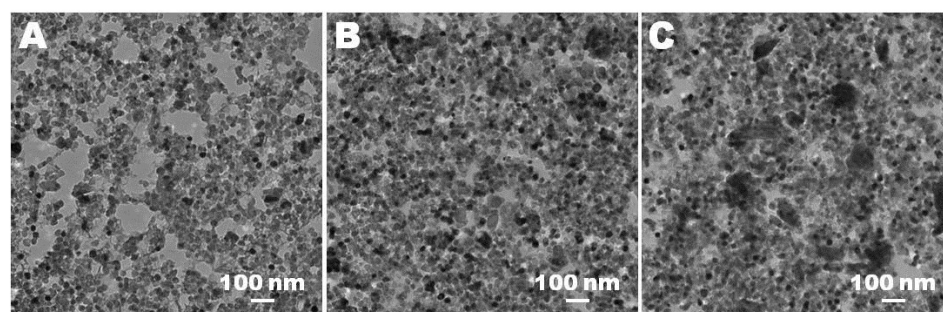


Figure S1. TEM micrographs of porous carbon/TiO₂ composites fabricated with (A) 20K, (B) 300K, and (C) 1000K PHEMA using 0.35 g PHEMA /0.3 g FA/0.3 g TiO₂ for the precursor.

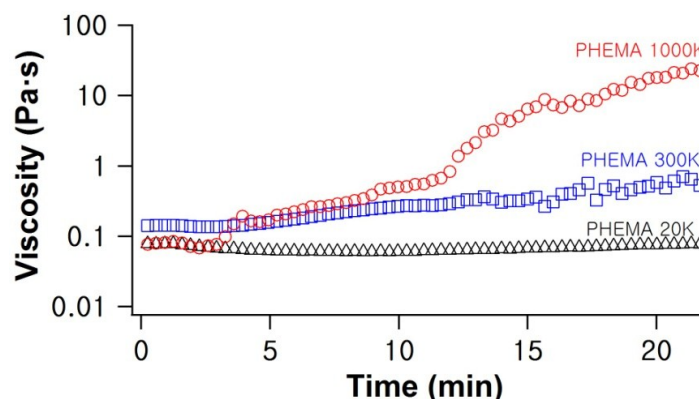


Figure S2. Viscosity variation of FA/PHEMA solutions with different molecular weight of PHEMA during FA polymerization.

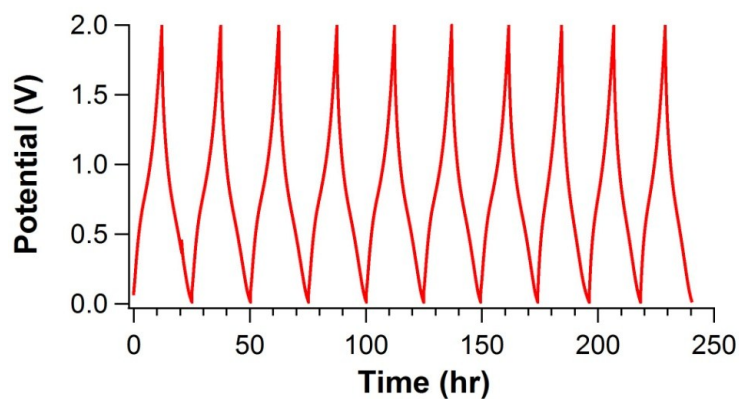


Figure S3. Galvanostatic charge/discharge behavior of carbon/TiO₂ composites from 20K pf PHEMA in a potential range of 0.01–2.0 V at a current density of 10 mA/g.

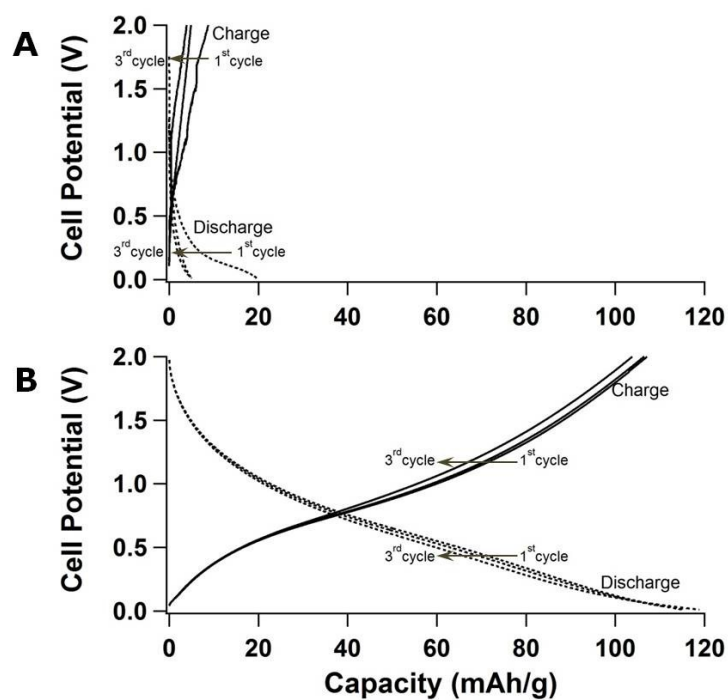
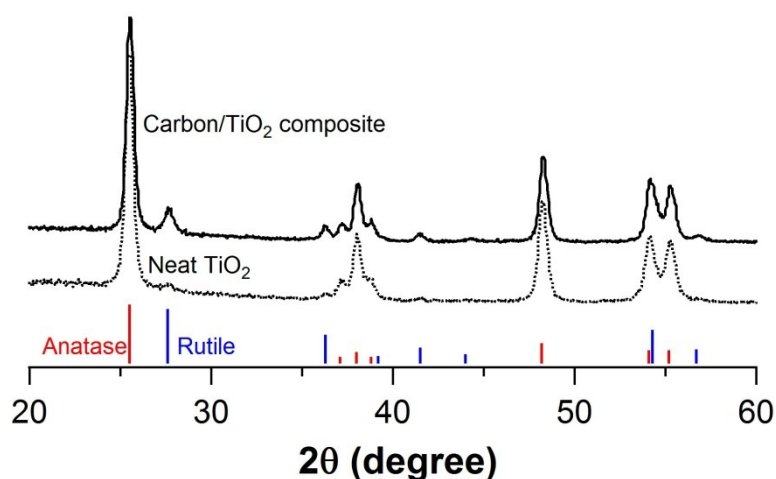


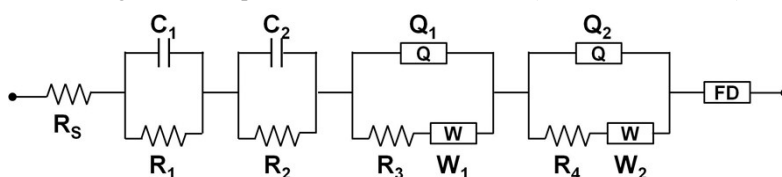
Figure S4. Galvanostatic charge/discharge behavior of (A) TiO₂ and (B) carbon/TiO₂ composites from 20K of PHEMA and 0.3 g of FA in a potential range of 0.01–2.0 V at a current density of 10 mA/g.



	Anatase	Rutile
Neat TiO ₂	98.3 %	1.7 %
Carbon/TiO ₂ composite	89.6 %	10.4 %

Figure S5. XRD profiles for the neat TiO₂ and carbon/TiO₂ composite produced by carbonization of PHBMA/PFA polymer blend with TiO₂ and the standard spectrum of anatase TiO₂ (JCPDS 84–1286, red lines) and rutile TiO₂ (JCPDS 75-1753, blue lines).

The XRD in Figure S5 illustrates that the carbonized composite still contains anatase TiO₂ nanoparticles with strong diffraction peaks at ~26° and ~48°. These peaks are in good agreement with the standard spectrum for anatase (JCPDS no.: 84–1286). Although the content of rutile TiO₂ increases from 1.7 % to 10.4 % after carbonization of PFA/PHBMA/TiO₂ composite, over 91 % of anatase-TiO₂ remains without transformation. The Rutile TiO₂ has strong diffraction peaks at ~28°, ~36° and ~54° (JCPDS no.: 76-1940).



	R_s (Ω)	R_1 (Ω)	R_2 (Ω)	R_3 (Ω)	R_4 (Ω)	C_1 (F)	C_2 (F)	Q_1 ($S \cdot s^n$)	Q_2 ($S \cdot s^n$)	W_1 ($S \cdot s^{1/2}$)	W_2 ($S \cdot s^{1/2}$)
PHBMA 20K	3 ± 0.5	174 ± 44	266 ± 40	86 ± 5	21 ± 3	$(1.8 \pm 1.8) \times 10^{-1}$	$(9.4 \pm 2.4) \times 10^{-6}$	$(3.0 \pm 2.9) \times 10^{-4}$	$(2.1 \pm 2.1) \times 10^{-3}$	$(3.8 \pm 0.5) \times 10^{-2}$	$(2.1 \pm 2.1) \times 10^{-4}$
PHBMA 300K	15 ± 0.2	339 ± 9	439 ± 16	465 ± 8	12 ± 3	$(3.2 \pm 0.3) \times 10^{-5}$	$(8.8 \pm 0.4) \times 10^{-6}$	$(2.7 \pm 0.1) \times 10^{-4}$	$(1.8 \pm 0.1) \times 10^{-2}$	$(2.6 \pm 0.8) \times 10^5$	$(2.7 \pm 1.7) \times 10^{-12}$
PHBMA 1000K	17 ± 7	336 ± 50	202 ± 30	394 ± 105	61 ± 27	$(2.7 \pm 1.3) \times 10^{-5}$	$(3.2 \pm 3.2) \times 10^{-6}$	$(2.9 \pm 2.0) \times 10^{-3}$	$(1.9 \pm 0.9) \times 10^{-4}$	$(2.9 \pm 0.8) \times 10^{-2}$	$(5.0 \pm 3.8) \times 10^4$
FA 0.2 g	10 ± 6	190 ± 49	229 ± 30	0.1 ± 0.01	21 ± 9	$(6.4 \pm 0.6) \times 10^{-4}$	$(2.8 \pm 1.2) \times 10^{-5}$	$(5.2 \pm 4.0) \times 10^{-12}$	$(3.0 \pm 2.0) \times 10^{-3}$	$(1.8 \pm 0.2) \times 10^{-2}$	$(3.3 \pm 3.1) \times 10^{-2}$
FA 0.3 g	3 ± 0.5	174 ± 44	266 ± 40	86 ± 5	21 ± 3	$(1.8 \pm 1.8) \times 10^{-1}$	$(9.4 \pm 2.4) \times 10^{-6}$	$(3.0 \pm 2.9) \times 10^{-4}$	$(2.1 \pm 2.1) \times 10^{-3}$	$(3.8 \pm 0.5) \times 10^{-2}$	$(2.1 \pm 2.1) \times 10^{-4}$
FA 0.5 g	10 ± 0.2	180 ± 3	205 ± 7	134 ± 7	573 ± 16	$(2.4 \pm 0.3) \times 10^{-5}$	$(7.5 \pm 0.3) \times 10^{-6}$	$(3.9 \pm 0.4) \times 10^{-4}$	$(4.6 \pm 0.3) \times 10^{-3}$	$(2.3 \pm 0.2) \times 10^{-2}$	$(9.8 \pm 1.5) \times 10^4$
FA 0.7 g	9 ± 3	190 ± 20	27 ± 23	154 ± 36	2 ± 0.2	$(8.5 \pm 0.4) \times 10^{-6}$	$(1.5 \pm 0.2) \times 10^{-5}$	$(4.7 \pm 1.3) \times 10^{-4}$	$(6.1 \pm 1.0) \times 10^{-3}$	$(5.4 \pm 4.6) \times 10^4$	$(2.1 \pm 2.1) \times 10^9$

Figure S6. Equivalent circuit to fit the EIS data of porous carbon/TiO₂ composite electrodes and numerical values of the equivalent circuit components obtained for the impedance data.

In Figure S6, R_s indicates the bulk resistance of electrolyte, separator, and electrode. These composite anode electrodes consist of carbon and TiO₂ particles. Therefore, it leads to two different SEIs (C_1/R_1 and C_2/R_2) on carbon and TiO₂ surfaces. The interfacial charge transfer, Na ion diffusion impedance, and double layer capacitance for carbon and TiO₂ are also considered individually, and thus they are presented by a combination of $Q_1/(R_3-W_1)$ and $Q_2/(R_4-W_2)$ circuits in series. FD is the finite length diffusion impedance.

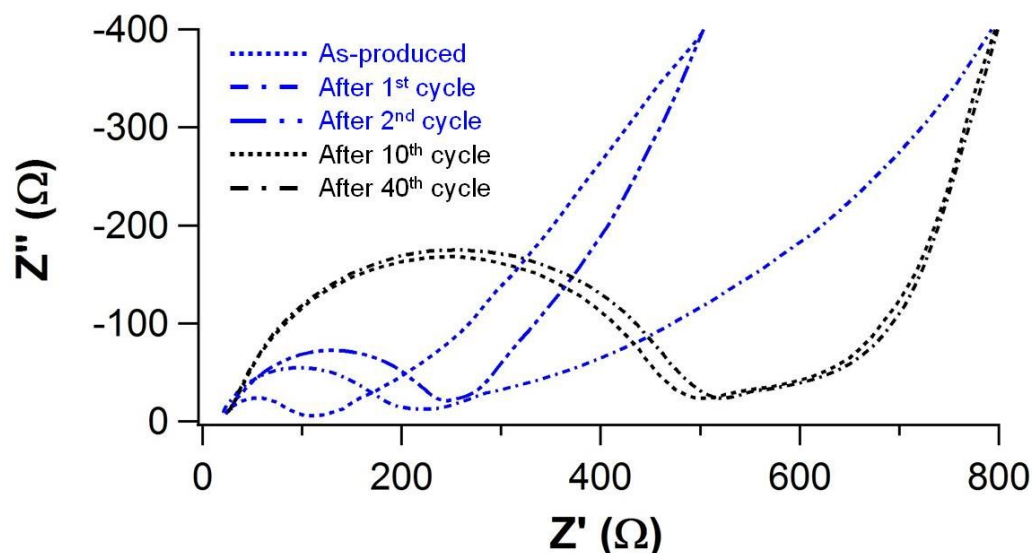


Figure S7. Nyquist plot from EIS data associated with cycling of carbon/TiO₂ composites from 20K of PHEMA and 0.3 g of FA. The impedance changes significantly in the first 10 charge-discharge cycles.

Cyclic voltammetry was carried out by cycling a carbon/TiO₂ composite from 20K of PHEMA for 40 cycles between 0.01 and 2.0 V at a scan rate of 0.01 mV/s. Figure S8 shows the corresponding CV response, with oxidation currents shown as positive and reduction currents shown as negative.

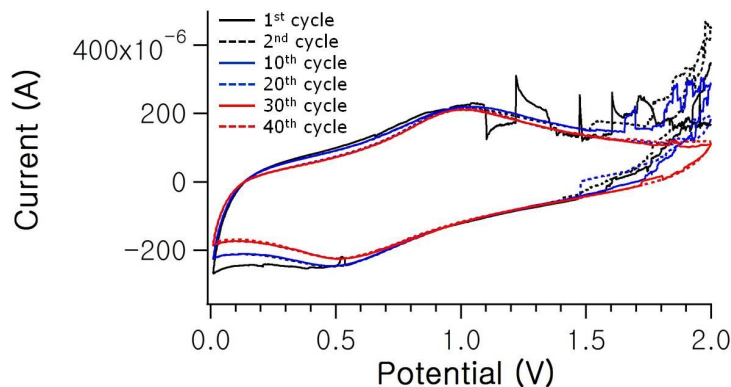


Figure S8. Cyclic voltammetry of the carbon/TiO₂ composite from 20K of PHEMA and 0.3 g of FA at a scan rate of 0.01 mV/s with sodium metal as both counter and reference electrodes.

The distribution of pore size of carbon/TiO₂ composites is shown in Figure S9. The small size pores (<6 nm) are more formed in carbon/TiO₂ composites with increasing the molecular weight of PHEMA in the precursor from 20 kg/mol to 1000 kg/mol. This indicates that the morphology of porous carbon/TiO₂ composites pyrolyzed from PHEMA/FA/TiO₂ can be controlled by the molecular weight of PHEMA used as a thermal decomposition matrix.

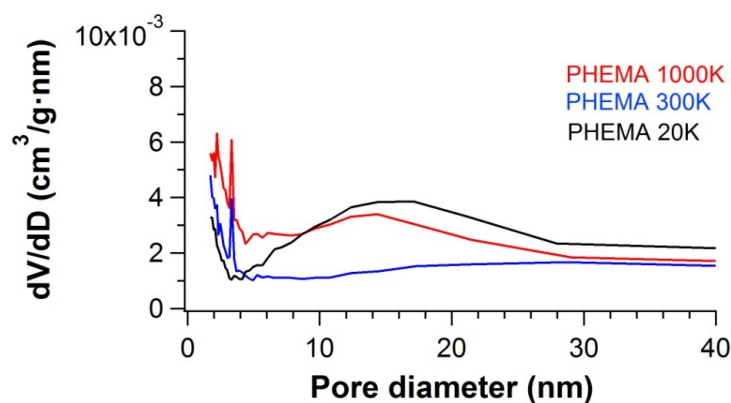


Figure S9. Pore size distribution of carbon/TiO₂ composites fabricated with 20K, 300K, and 1000K PHEMA using 0.35 g PHEMA/0.3 g FA/0.3 g TiO₂ for the precursor.

Figure S10 shows the XPS spectra of the neat and 40 charge-discharge cycled carbon/TiO₂ composite anodes fabricated with 20K, 300K, and 1000K PHEMA using 0.35 g PHEMA/0.3 g FA/0.3g TiO₂ for the precursor. All the cycled samples show that the Na 1s peak appears and the intensity of O 1s peak increase compared to the neat anodes. It indicates that the sodium- included inorganic compounds and oxygen-rich substances form on the surface of anodes as the SEI layer.

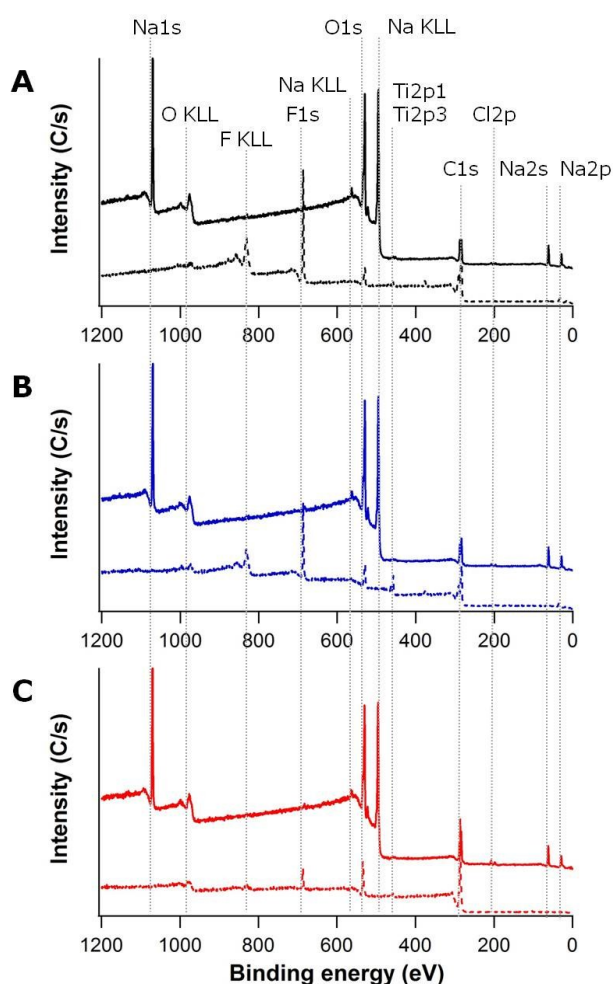


Figure S10. XPS spectra of neat (dash line) and 40 charge-discharge cycled (solid line) carbon/TiO₂ composite anodes fabricated with (A) 20K, (B) 300K, and (C) 1000K PHEMA using 0.35 g PHEMA/0.3 g FA/0.3g TiO₂ for the precursor.

Figure S11 illustrates the morphology of the porous carbon/TiO₂ composite fabricated using 0.35 g PHEMA (20K)/FA/0.3 g TiO₂ as a function of molecular weight of FA (0.2, 0.3, 0.5 and 0.7 g). The pores of the carbon/TiO₂ composite are significantly larger for the composite produced using 0.3 g (Figure S11B) than the carbon/TiO₂ composites from 0.5 g (Figure S11C) and 0.7 g FA (Figure S11D). Figure S11A illustrates the carbon/TiO₂ composite fabricated with 0.2 g FA, which has larger apertures and exposes more TiO₂ particles than other composites because it has small carbon content.

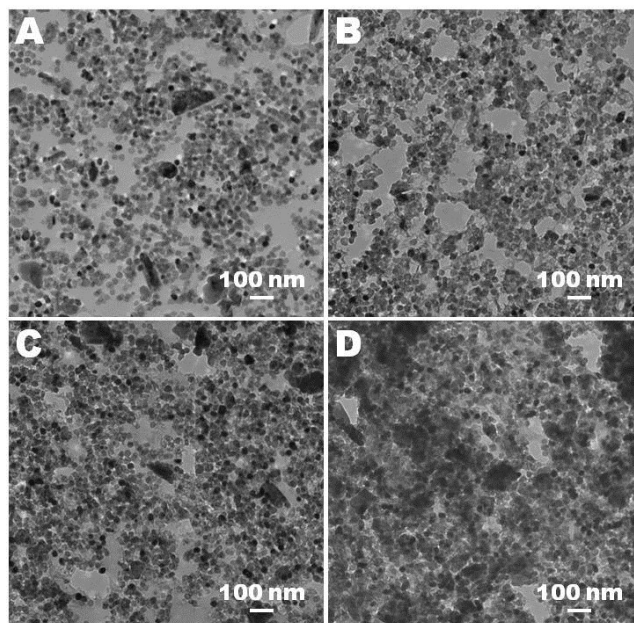


Figure S11. TEM micrographs of porous carbon/TiO₂ composites fabricated with 0.35 g PHEMA (20K)/0.3 g TiO₂/FA of (A) 0.2, (B) 0.3, (C) 0.5, and (D) 0.7 g for the precursor.

The TGA under air atmosphere is hired to confirm the carbon content of the porous carbon/TiO₂ composite fabricated using 0.35 g PHEMA (20K)/FA/0.3 g TiO₂ as a function of amount of FA (0.2, 0.3, 0.5 and 0.7 g). The TGA curves are shown in Figure S12, and the values are 12, 15, 16, and 18 wt% for the composites using 0.2, 0.3, 0.5, and 0.7 g FA, respectively.

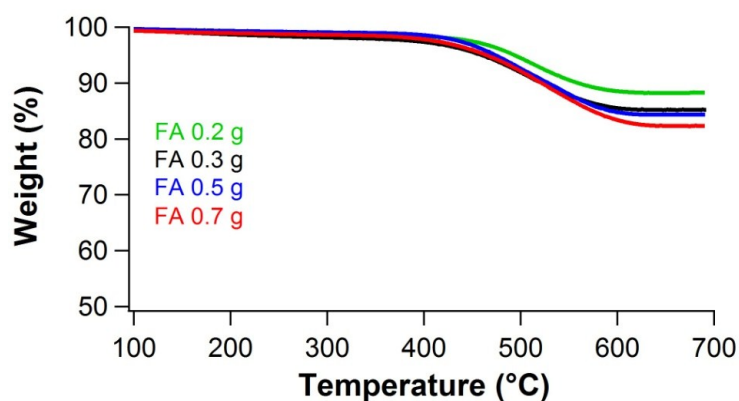


Figure S12. TGA curves of carbon/TiO₂ composites with different amount of FA to determine the carbon content by oxidation in air.

The distribution of pore size of carbon/TiO₂ composites is shown in Figure S13. The small size pores (<6 nm) are more formed in carbon/TiO₂ composites with increasing the amount of FA in the precursor from 0.3 g to 0.7 g. The carbon/TiO₂ composite fabricated with 0.2 g FA has the largest apertures, but also exposes a significant fraction of the TiO₂ nanoparticles. This indicates that the morphology of porous carbon/TiO₂ composites pyrolyzed from PHEMA/FA/TiO₂ can be controlled by the amount of FA used in the synthesis.

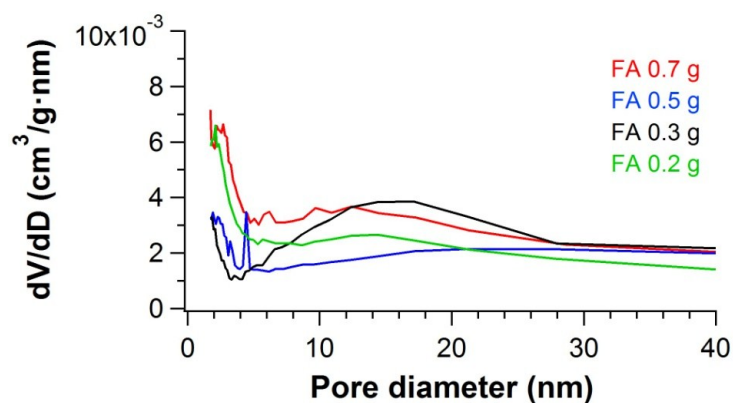


Figure S13. Pore size distribution of carbon/TiO₂ composites fabricated with 0.35 g PHEMA (20K)/0.3 g TiO₂/different amount of FA for the precursor.

Figure S14 shows the XPS spectra of the neat and 30 charge-discharge cycled carbon/TiO₂ composite anodes fabricated with different carbon contents. It indicates that the sodium- included inorganic compounds and oxygen-rich substances form on the surface of anodes as the SEI layer.

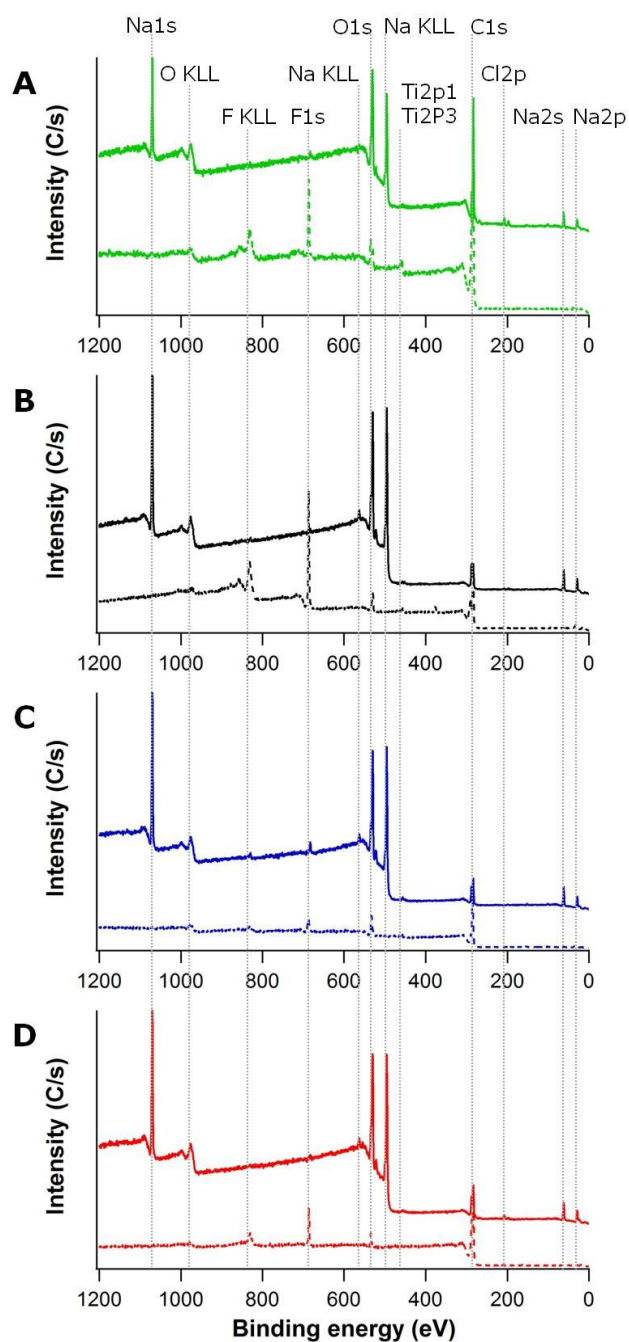


Figure S14. XPS spectra of the neat (dash line) and 30 charge-discharge cycled (solid line) composite anodes fabricated with different carbon contents of carbon/TiO₂ composite: (A) 12 wt%, (B) 15 wt%, (C) 16 wt%, and (D) 18 wt%.