

# Supplementary Information

## Highly Efficient Electrochemical Reduction of CO<sub>2</sub> to CH<sub>4</sub> in Ionic Liquid using Metal-Organic Framework Cathode

Xinchen Kang, Qinggong Zhu,\* Xiaofu Sun, Jiayin Hu, Jianling Zhang, Zhimin Liu, Buxing Han\*

Beijing National Laboratory for Molecular Sciences, Institute of Chemistry, Chinese Academy of Sciences, Beijing 100190, China

### Contents

Results and discussion-----	2
Supplementary Figures-----	4
Supplementary Tables-----	12
References -----	15

## Results and discussion

### Zn-MOF synthesis

Ionic liquids (ILs) are excellent solvents to dissolve the precursors of MOFs.<sup>S1</sup> It has been claimed that the phase behavior, intermolecular interaction and microstructure are the most critical aspects in the application of IL-containing systems. The microstructures of the synthetic media influence the morphology of the Zn-MOFs significantly.<sup>S2</sup> In this work, the solvent consisted of 75 wt% C<sub>12</sub>mimCl and 25 wt% glycerol. The mass fractions of ZnCl<sub>2</sub> (x) in the C<sub>12</sub>mimCl + glycerol + ZnCl<sub>2</sub> system were 0.17, 0.29, 0.38, 0.44, 0.50, respectively.

The microstructure of the C<sub>12</sub>mimCl + glycerol + ZnCl<sub>2</sub> system with different ZnCl<sub>2</sub> contents were characterized by SAXS method, and the results are given in Fig. S1, together with the SEM images of the Zn-MOFs synthesized. The domains in the solution were rod-like at lower x, and the rod-like Zn-MOFs were formed. With the increase of the x, the length-width ratio  $L/D_c$  of the IL/glycerol/ZnCl<sub>2</sub> domains decreased, which resulted in podgy rods. When the x reached 0.38, the domains in the system became sheet-like, and the Zn-MOF was sheet-like material synthesized at this condition. The domains became spherical when x was higher than 0.44, which can be known because the ordered structure was completely disappeared (Fig. S2). The Zn-MOFs synthesized at the conditions had spherical morphology, which was consistent with poorer crystallinity of the Zn-MOFs prepared at this condition (Fig. 1). With x increasing continuously, the domains in the system became larger, which resulted mainly from the high viscosity of the solution, and larger Zn-MOFs spheres were formed.

The results above indicate that the shape of the Zn-MOFs is similar to that of the domains in the C<sub>12</sub>mimCl + glycerol + ZnCl<sub>2</sub> system. This is easy to understand, and is discussed taking the formation of the sheet-like Zn-MOFs as example, which is shown schematically in Fig. S3. The ligands existed in the C<sub>12</sub>mimCl + glycerol + ZnCl<sub>2</sub> system, and the sheet-like Zn-MOFs nuclei were formed by coordination reaction of the Zn<sup>2+</sup> and BTC in the domains. The sheet-like Zn-MOF nuclei were formed because the domains acted as the templates. The nuclei behaved as seeds of the Zn-MOFs and grew gradually to form larger sheet-like particles.

Higher ZnCl<sub>2</sub> concentration in the synthesis solution resulted in higher Zn content in the as-synthesized Zn-MOFs. The diffraction peaks moved to lower 2 $\theta$  degree, indicating the enlarged interplanar spacing d (Fig. 1), which illustrating that excess Zn<sup>2+</sup> ions incorporated interstitially instead of segregate out of the MOFs.

### Electrochemical impedance spectroscopy (EIS)

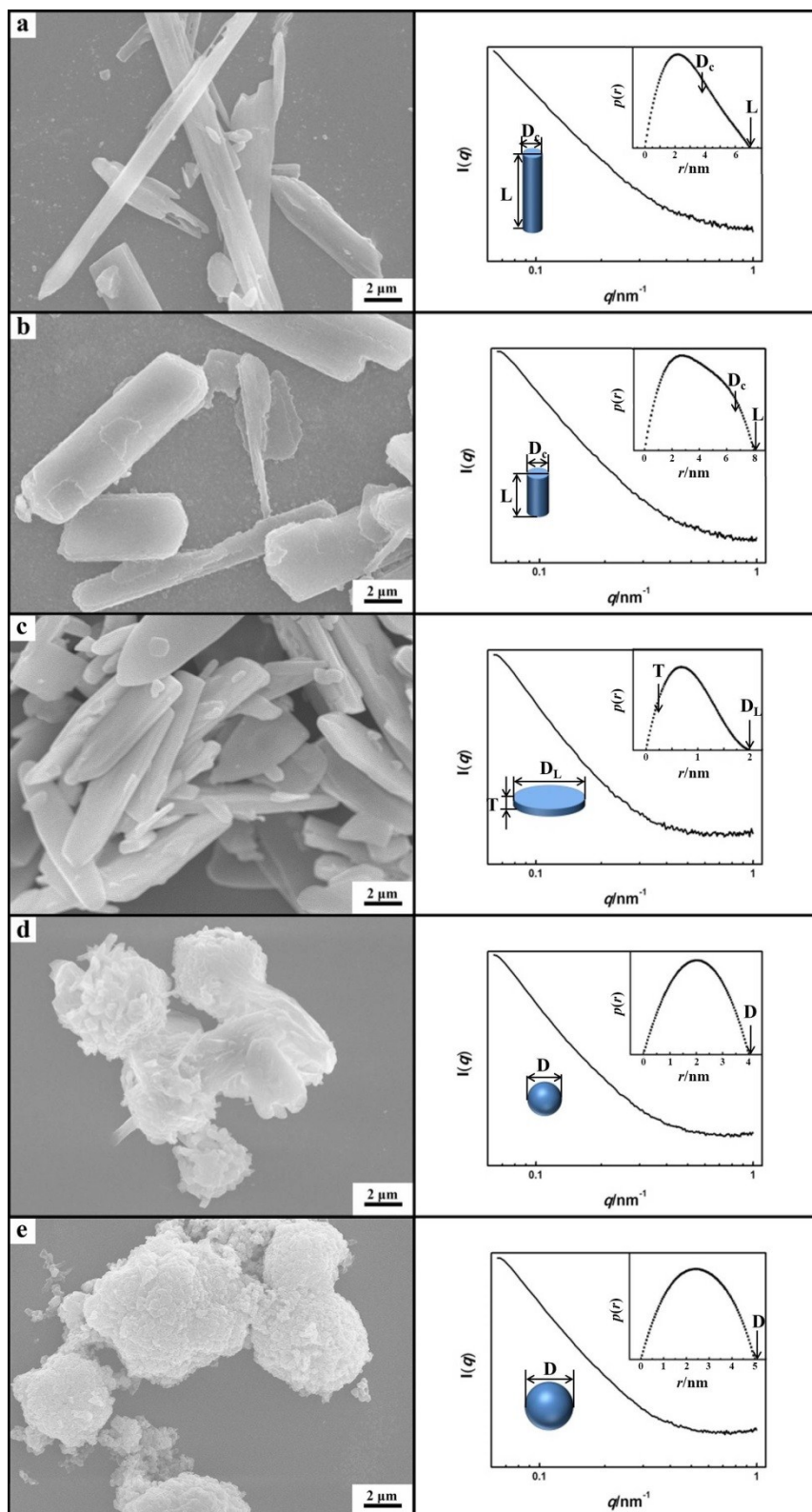
The interfacial effect between Zn-MOF/CP electrode and ionic liquid on the influence of CO<sub>2</sub> reduction was characterized by EIS. Fig. S6 shows the EIS of each Zn-MOF/CP electrode at an open circuit potential (OCP) and -2.0 V closer to the CO<sub>2</sub> reduction potential. A simple equivalent circuit (Fig. S7) is used for simulating the experimental impedance data. The resulting experimental and simulated EIS spectra

of various Zn-MOF/CP cathodes with Randles' equivalent circuit  $R(C(R(Q(RW))))$  are illustrated in Fig. S8, and values of the main parameters of Randles equivalent circuit elements obtained by fitting the EIS spectra are listed in Supplementary Tables 1 and 2. Relevant data extracted from Table S1 represents a pure capacitive behavior of Zn-MOF/CP electrode in BmimBF<sub>4</sub>,<sup>S3</sup> indicating the Zn-MOFs have partial charge on the surface. The capacitive behavior could also be explained the presence of a new molecule in the double layer for Zn-MOF electrode when a highly ordered ionic layer is generated on the electrode surface. Since the charge transfer resistance ( $R_{ct}$ ) which is related to electronic transfer occurring at the interface will strongly depend on the double layer properties,<sup>S4</sup> and the  $R_{ct}$  values obtained at OCP only reflect the electronic transfer ability of the electrode. It is concluded that electrode fabricated from the sheet-like Zn-MOF synthesized at  $x=0.38$  had lowest  $R_{ct}$  value of  $16.83 \Omega \cdot \text{cm}^2$ . In contrast, the contact surface between the rod-like or between the spherical Zn-MOFs particles is small, and thus the charge transfer is blocked significantly in Zn-MOFs electrodes synthesized. On the other hand, there is a decrease in the value of  $R_{ct}$  when the formal potential was set at  $-2.0\text{V}$  closer to the onset potential of CO<sub>2</sub> reduction. The decreasing  $R_{ct}$  and increasing CPE values are related to deformation of the double layer due to the inclusion of CO<sub>2</sub> molecules, and the migration of species to the electrode allows an easier electron transfer on the electrode surface.

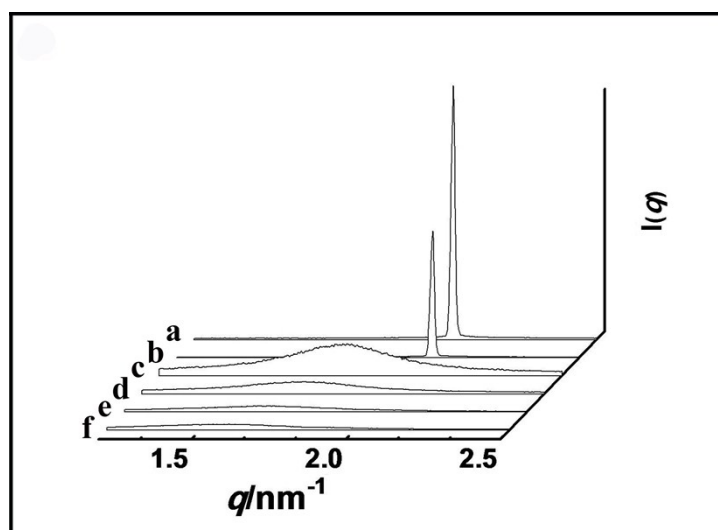
## Reference Electrode Calibration

The calibration was based on the reported literatures.<sup>S5-S6</sup> In the experiment, a ferrocene/ferrocenium (2.5mM) redox couple in acetonitrile and BmimBF<sub>4</sub> is used to calibrate the reference electrode with scan rate of 10 mV/s, which is shown in Fig. S13. We found the potential difference of the redox couple in acetonitrile and in BmimBF<sub>4</sub> was 99 mV. Therefore, we conclude that there is a  $537+99=636\text{mV}$  difference between the potential of the designed electrode in BmimBF<sub>4</sub> and SHE at 25 °C, indicating that the influence of the addition resistance is negligible in IL medium.

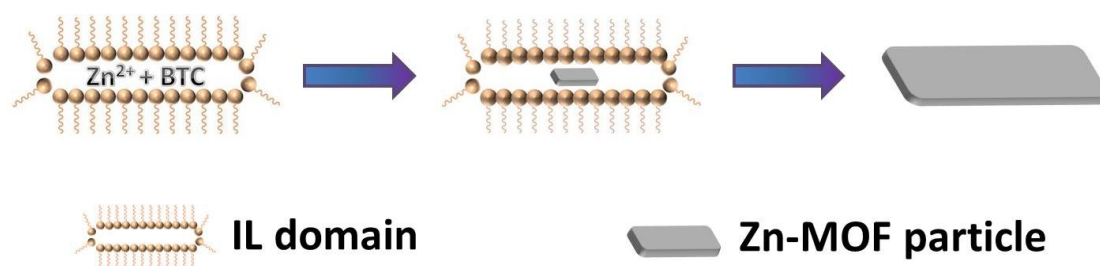
## Supplementary Figures



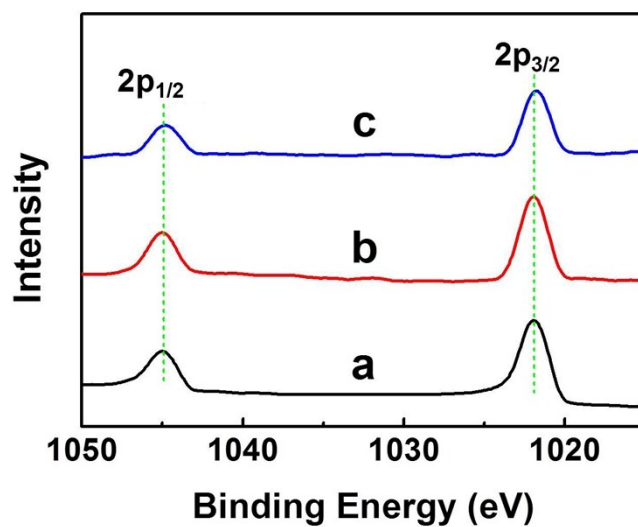
**Fig. S1** SEM images of the Zn-MOFs (left) and SAXS curves (right). The solvent consisted of 75 wt%  $C_{12}mimCl$  and 25 wt% glycerol. The mass fractions of  $ZnCl_2$  (x) in the  $C_{12}mimCl$  + glycerol +  $ZnCl_2$  system were 0.17 (a), 0.29 (b), 0.38 (c), 0.44 (d), 0.50 (e).



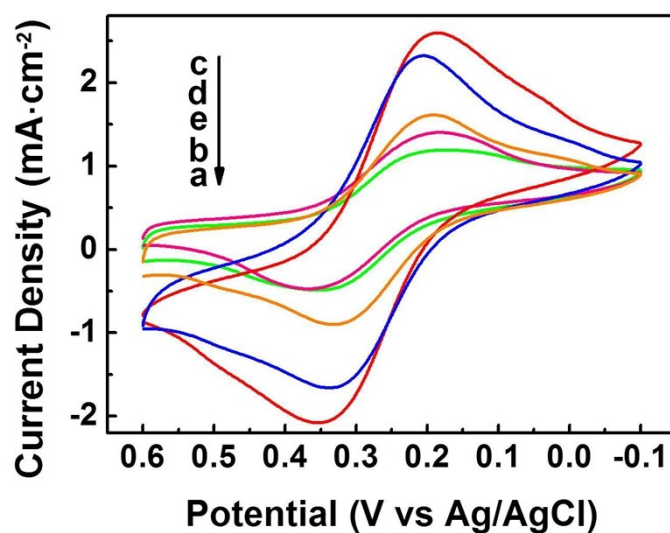
**Fig. S2** The SAXS curves of  $C_{12}mimCl$  + glycerol +  $ZnCl_2$  system. The solvent consisted of 75 wt%  $C_{12}mimCl$  and 25 wt% glycerol. The mass fractions of  $ZnCl_2$  (x) in the  $C_{12}mimCl$  + glycerol +  $ZnCl_2$  system were 0 (a), 0.17 (b), 0.29 (c), 0.38 (d), 0.44 (e) and 0.50 (f). The sharp peaks indicate the existence of the ordered structure in the mixture.



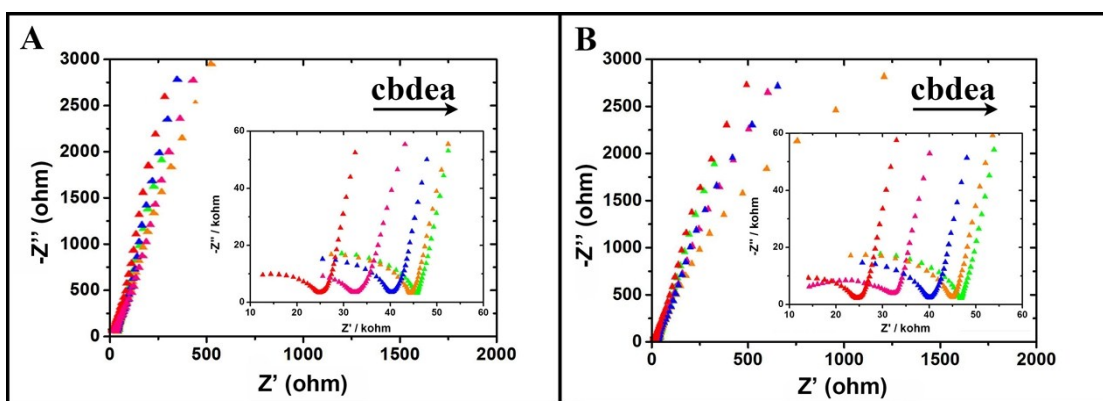
**Fig. S3** Schematic diagram to show the formation mechanism of the Zn-MOF sheets.



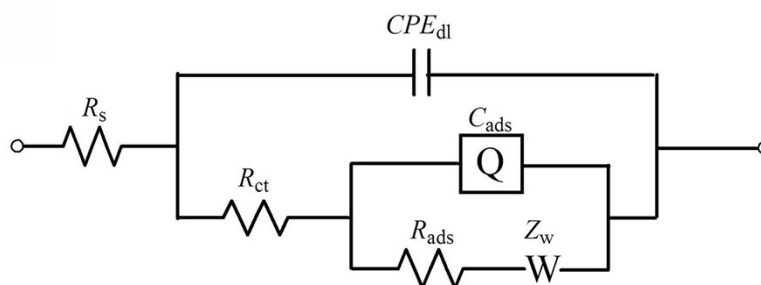
**Fig. S4** Zn 2p XPS spectra of Zn-MOF (a), Zn-MOF/CP cathode (b), and Zn-MOF/CP cathode after electrolysis (c). The Zn-MOF cathode was prepared using Zn-MOF synthesized at  $x=0.38$ .



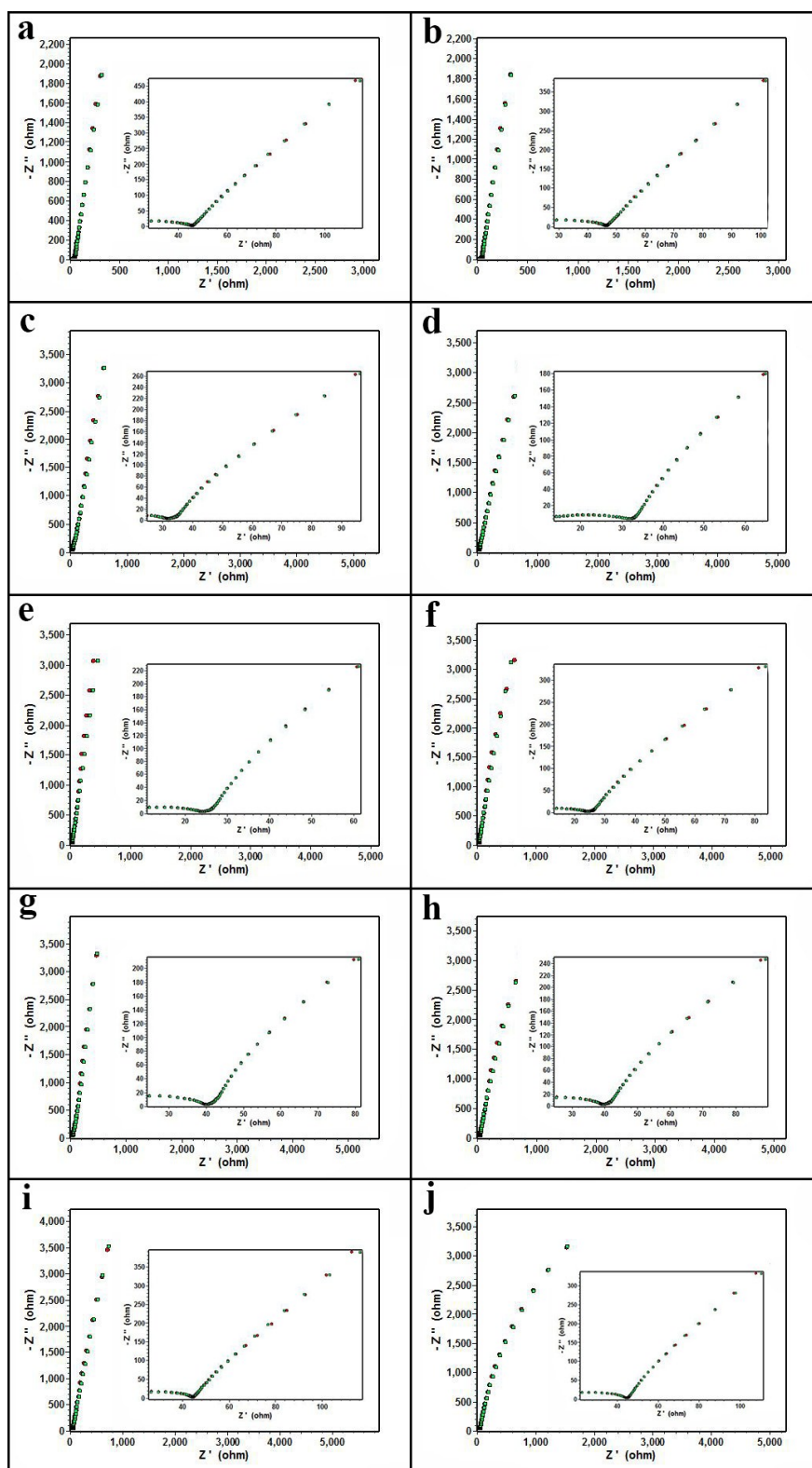
**Fig. S5** CVs of 10 mM  $\text{Fe(CN)}_6^{3-/4-}$  in 1 M KCl using different Zn-MOF/CP cathodes prepared using Zn-MOFs synthesized at  $x=0.17$  (a),  $0.29$  (b),  $0.38$  (c)  $0.44$  (d) and  $0.5$  (e).



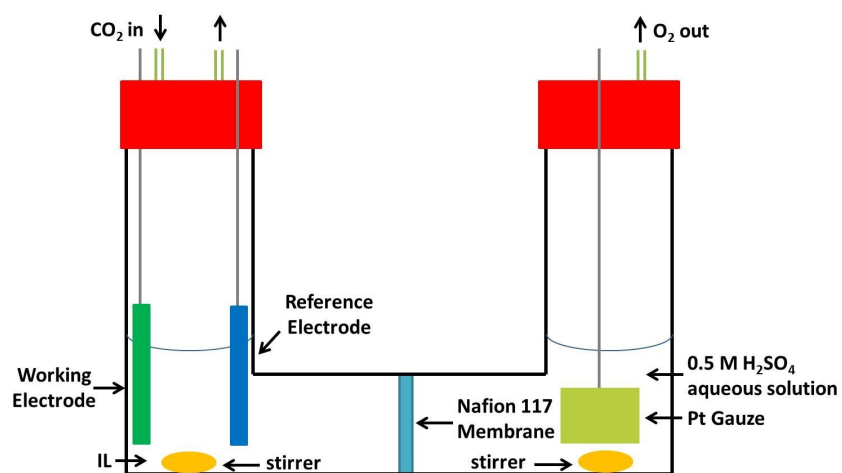
**Fig. S6** Nyquist plots for various Zn-MOF/CP cathodes prepared using Zn-MOFs synthesized at  $x=0.17$  (a),  $0.29$  (b),  $0.38$  (c),  $0.44$  (d),  $0.50$  (e) in BmimBF<sub>4</sub> at OCP and  $-2.0$  V under CO<sub>2</sub> atmospheres.



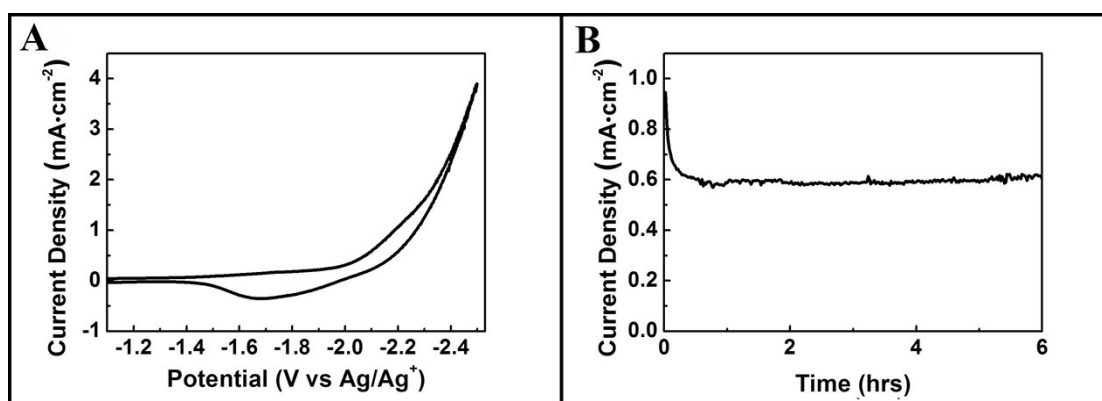
**Fig. S7** Electrical equivalent circuit used for simulating the experimental impedance data. The components contain solution resistance ( $R_s$ ), electron transfer resistance ( $R_{ct}$ ), double layer capacitance ( $CPE_{dl}$ ), surface adsorption capacitance ( $C_{ads}$ ), surface adsorption resistance ( $R_{ads}$ ) and Warburg-type impedance ( $Z_w$ ).



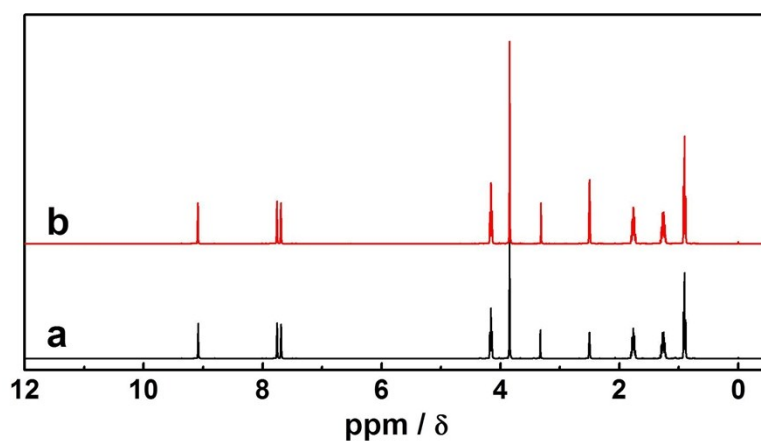
**Fig. S8** The experimental and simulated EIS spectra of Zn-MOF/CP cathodes prepared using Zn-MOFs synthesized at  $x=0.17$  (a-b),  $0.29$  (c-d),  $0.38$  (e-f),  $0.44$  (g-h),  $0.50$  (i-j) with Randles' equivalent circuit  $R(C(R(QRW)))$  at OCP (left column) and  $-2.0$  V (right column).



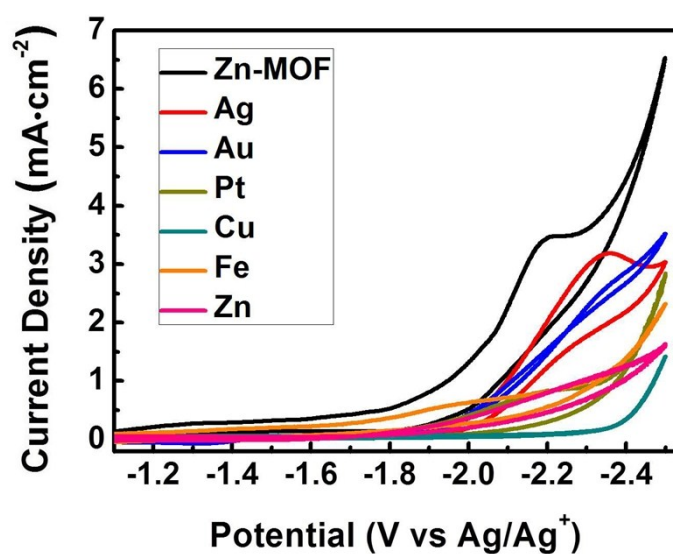
**Fig. S9** The schematic diagram of the electrolysis device.



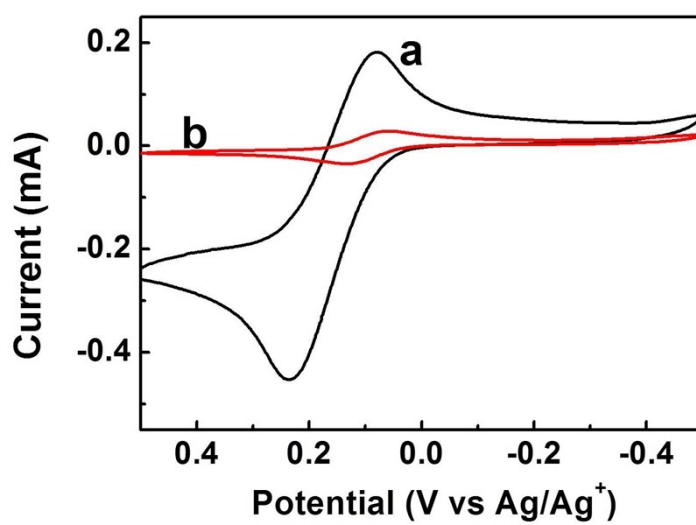
**Fig. S10** CV and Current density profile using CP as cathode.



**Fig. S11**  $^1\text{H}$ -NMR spectra of the electrolyte BmimBF<sub>4</sub> before (a) and after (b) electrolysis in DMSO- $d_6$  with TMS as an internal standard.



**Fig. S12** CV traces obtained in BmimBF<sub>4</sub> using different kinds of electrodes. The Zn-MOF/CP cathode was prepared using Zn-MOF synthesized at  $x=0.38$ .



**Fig. S13** Anodic and cathodic wave for ferrocene/ferrocenium (2.5mM) redox couple in Acetonitrile (a) and BmimBF<sub>4</sub> (b) using the  $\text{Ag}/\text{Ag}^+$  reference.

## Supplementary Tables

**Table S1** Values of the main parameters of Randles equivalent circuit elements obtained by fitting the EIS spectra with Randles' equivalent circuit  $R(C(R(Q(RW))))$  at OCP.

Entry	$x^a$	$R_s (\Omega \cdot \text{cm}^2)^b$	$CPE (\Omega^{-1} \cdot \text{cm}^{-2} \cdot \text{S}^n)^c$	$n^d$	$R_{ct} (\Omega \cdot \text{cm}^2)^e$
1	0.17	11.37	$9.3 \times 10^{-5}$	0.80	34.31
2	0.29	10.61	$8.4 \times 10^{-5}$	0.87	19.28
3	0.38	5.39	$10.2 \times 10^{-5}$	0.80	16.83
4	0.44	10.34	$10.8 \times 10^{-5}$	0.80	29.93
5	0.50	15.71	$11.7 \times 10^{-5}$	0.91	33.68

<sup>a</sup>The mass fraction of  $\text{ZnCl}_2$  ( $x$ ) in the  $\text{C}_{12}\text{mimCl} + \text{glycerol} + \text{ZnCl}_2$  systems. <sup>b</sup> $R_s$  is solution resistance. <sup>c</sup>CPE is double layer capacitance. <sup>d</sup> $n$  is dimensionless parameter. <sup>e</sup> $R_{ct}$  is electron transfer resistance.

**Table S2** Values of the main parameters of Randles equivalent circuit elements obtained by fitting the EIS spectra with Randles' equivalent circuit  $R(C(R(Q(RW))))$  at -2.0 V.

Entry	$x^a$	$R_s (\Omega \cdot \text{cm}^2)^b$	$CPE (\Omega^{-1} \cdot \text{cm}^{-2} \cdot \text{S}^n)^c$	$n^d$	$R_{ct} (\Omega \cdot \text{cm}^2)^e$
1	0.17	11.16	$9.9 \times 10^{-5}$	0.91	33.2
2	0.29	12.57	$11.5 \times 10^{-5}$	0.89	19.16
3	0.38	5.98	$11.8 \times 10^{-5}$	0.89	16.10
4	0.44	10.03	$12.2 \times 10^{-5}$	0.90	18.69
5	0.50	10.87	$13.0 \times 10^{-5}$	0.87	19.86

<sup>a</sup>The mass fraction of  $\text{ZnCl}_2$  ( $x$ ) in the  $\text{C}_{12}\text{mimCl} + \text{glycerol} + \text{ZnCl}_2$  systems. <sup>b</sup> $R_s$  is solution resistance. <sup>c</sup>CPE is double layer capacitance. <sup>d</sup> $n$  is dimensionless parameter. <sup>e</sup> $R_{ct}$  is electron transfer resistance.

**Table S3** CO<sub>2</sub> reduction performance of Zn-MOF/CP cathode in various electrolytes at -2.2 V vs Ag/Ag<sup>+</sup> after 2 hrs.<sup>a</sup>

Entry	Electrolytes	$j_{\text{tot}}$ [mA·cm <sup>-2</sup> ]	FE <sub>CH<sub>4</sub></sub> [%]	FE <sub>CO</sub> [%]	FE <sub>HCOOH</sub> [%]	FE <sub>H<sub>2</sub></sub> [%]
1	0.01M TBABF <sub>4</sub> /DMF	1.0±0.06	26.3	-	6.8	59.6
2	0.1M TBAPF <sub>6</sub> /MeCN	3.1±0.5	23.2	15.4	19.0	48.6
3	0.1M BmimBF <sub>4</sub> /MeCN	5.5±0.9	10.3	8.3	56.4	29.0

<sup>a</sup>The Zn-MOF/CP cathode was prepared using Zn-MOF synthesized at x=0.38 for reduction of CO<sub>2</sub>.

**Table S4** The overpotentials at -2.2 V vs Ag/Ag<sup>+</sup>, linear range in Tafel plots, and Tafel slopes for the main product of different kinds of cathodes.

entry	Cathode	Main product	Linear range [V]	Tafel slope [mV·dec <sup>-1</sup> ]
1	Zn-MOF/CP <sup>a</sup>	CH <sub>4</sub>	0.19-0.37	146
2	Au	CO	0.05-0.27	186
3	Ag	CO	0.23-0.38	152
4	Pt	H <sub>2</sub>	0.3-0.54	197
5	Fe	CH <sub>4</sub>	0.38-0.6	253
6	Zn	CH <sub>4</sub>	0.2-0.43	209
7	Cu	CH <sub>4</sub>	0.45-0.65	200

<sup>a</sup>The cathode was prepared using Zn-MOF synthesized at x=0.38.

**Table S5** CH<sub>4</sub> selectivity using various cathodes reported in literature.

entry	Electrode	Electrolytes	Potential [V]	FE <sub>CH<sub>4</sub></sub> [%]	Ref.
1	Cu/C (7 nm)	0.1 M NaHCO <sub>3</sub>	-1.35V vs Ag/AgCl	76	S7
2	Cu (210)	0.1 M KHCO <sub>3</sub>	-1.52V vs Ag/AgCl	60.5	S8
3	Cu	0.1 M KHCO <sub>3</sub>	-1.05V vs RHE	24.4	S9
4	Cu	0.1 M KHCO <sub>3</sub>	-2.2V vs SCE	60	S10
5	Ag	0.1 M KHCO <sub>3</sub>	-1.35±0.025 vs RHE	0.07	S11
6	Zn	0.1 M KHCO <sub>3</sub>	-1.31±0.009 vs RHE	0.16	S11
7	Cu	0.1 M KHCO <sub>3</sub>	-1.05±0.008 vs RHE	24.4	S11
8	Ni	0.1 M KHCO <sub>3</sub>	-1.04±0.018 vs RHE	0.4	S11
9	Pt	0.1 M KHCO <sub>3</sub>	-0.75±0.006 vs RHE	0.02	S11
10	Fe	0.1 M KHCO <sub>3</sub>	-0.54±0.006 vs RHE	0.01	S11
11	Polycrystalline Cu	0.1 M KHCO <sub>3</sub>	-1.0V vs RHE	43	S12
12	Annealed Cu	0.1 M KHCO <sub>3</sub>	-1.0V vs RHE	22	S12
13	Cu <sub>2</sub> O (200 nm)	0.1 M KHCO <sub>3</sub>	-0.99V vs RHE	9.85	S13
14	Cu (210)	0.1 M KHCO <sub>3</sub>	-1.52V vs SHE	64.0	S14
15	Cu (755)	0.1 M KHCO <sub>3</sub>	-1.43V vs SHE	62.9	S14
16	Cu nanoparticles (1.9 nm)	0.1 M KHCO <sub>3</sub>	-1.1V vs RHE	15	S15

## References

- S1. Q. B. Liu, M. H. A. Janssen, F. van Rantwijk and R. A. Sheldon, *Green Chem.*, 2005, **7**, 39.
- S2. W. T. Shang, X. C. Kang, H. Ning, J. L. Zhang, X. G. Zhang, Z. H. Wu, G. Mo, X. Q. Xing and B. X. Han, *Langmuir*, 2013, **29**, 13168.
- S3. S. C. Canobre, D. A. L. Almeida, C. P. Fonseca, S. Neves, *Electrochim. Acta* 2009, **54**, 6383-6388; b) S. A. Mamuru, K. I. Ozoemena, T. Fukuda, N. Kobayashi and T. Nyokong, *Electrochim. Acta*, 2010, **55**, 6367.
- S4. P. Acevedo-Peña, M. Haro, M. E. Rincón, J. Bisquert and G. Garcia-Belmonte, *J. Power Sources*, 2014, **268**, 397.
- S5. V. V. Pavlishchuk, A. W. Addison, *Inorg. Chim. Acta*, 2000, **298**, 97.
- S6. B. A. Rosen, A. Salehi-Khojin, M. R. Thorson, W. Zhu, D. T. Whipple, P. J. A. Kenis and R. I. Masel, *Science*, 2011, **334**, 643.
- S7. K. Manthiram, B. J. Beberwyck and A. P. Aivisatos, *J. Am. Chem. Soc.*, 2014, **136**, 13319.
- S8. Y. Hori, I. Takahashi, O. Koga and N. Hoshi, *J. Phys. Chem. B*, 2002, **106**, 15.
- S9. K. P. Kuhl, E. R. Cave, D. N. Abram and T. F. Jaramillo, *Energy Environ. Sci.*, 2012, **5**, 7050.
- S10. Y. Momose, K. Sato and O. Ohno, *Surf. Interface Anal.*, 2002, **34**, 615.
- S11. K. P. Kuhl, T. Hatsukade, E. R. Cave, D. N. Abram, J. Kibsgaard and T. F. Jaramillo, *J. Am. Chem. Soc.*, 2014, **136**, 14107.
- S12. C. W. Li and M. W. Kanan, *J. Am. Chem. Soc.*, 2012, **134**, 7231.
- S13. D. Ren, Y. L. Deng, A. D. Handoko, C. S. Chen, S. Malkhandi and B. S. Yeo, *Acs Catal.*, 2015, **5**, 2814.
- S14. *J. Mol. Catal. A-Chem.* 199, 39-47 (2003). Y. Hori, I. Takahashi, O. Koga and N. Hoshi, *J. Mol. Catal. A-Chem.*, 2003, **199**, 39.
- S15. R. Reske, H. Mistry, F. Behafarid, B. R. Cuenya and P. Strasser, *J. Am. Chem. Soc.*, 2014, **136**, 6978.