

**Supporting information**

# Conductive surfaces responsive to temperature, salt and electrical potential

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## Electrochemical synthesis of PBrEDOT films

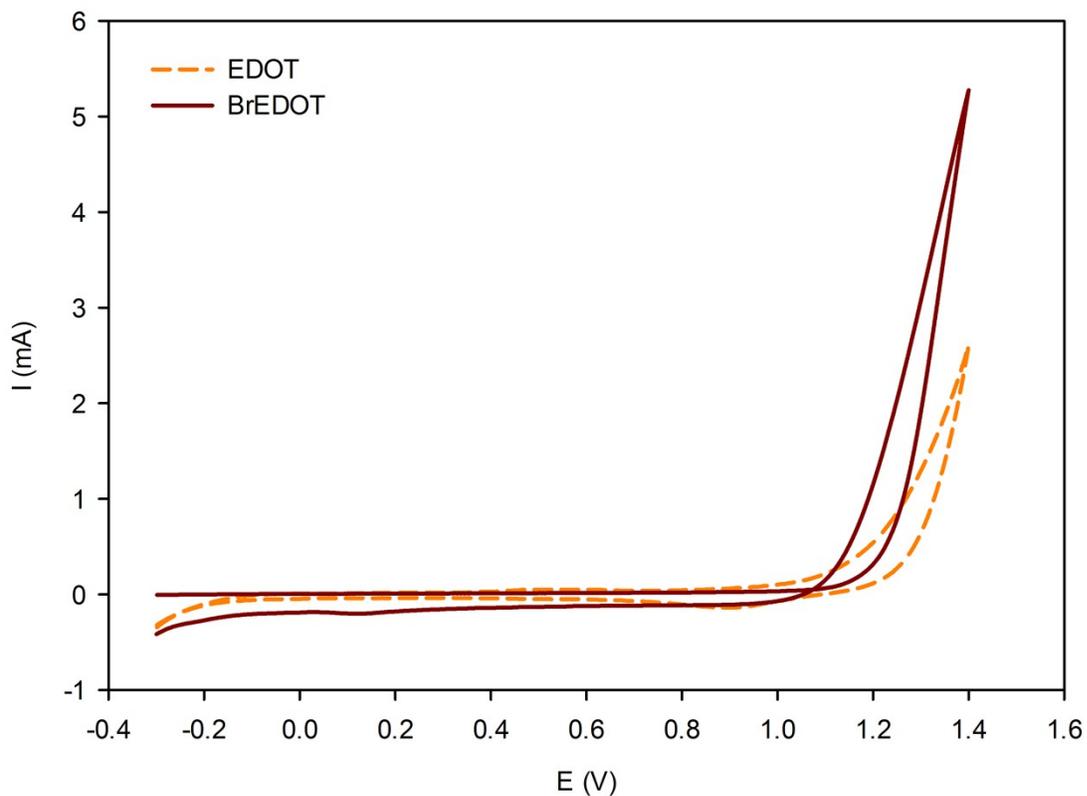


Figure S1. Subsequent electropolymerization of PEDOT adhesion layer (0.1 M sodium *p*-toluenesulfonate in 3:1 acetonitrile/water) and PBrEDOT surface layer (0.1 M LiClO<sub>4</sub>/acetonitrile) by cyclic voltammetry (-0.3 to 1.4 V vs Ag/AgCl (3 M KCl), 100 mV/s, 1 cycle each).

### Additional details of characterization methods

#### *Ellipsometry data modelling*

Simple models of polymer brushes on silicon substrates were generated using a Cauchy film model on top of a silicon substrate with a native oxide layer. To generate the gold-PBrEDOT-

brush models, the optical constant of the gold substrate (~230 nm grown on top of a silicon wafer with a thin titanium adhesion layer) were first measured on the bare gold substrate using a b-spline model (model of choice from the software package and used for materials absorbing in the optical range probed and for which the optical constants are not known). The PBrEDOT layer was also modelled using a b-spline model. The refractive index  $n$  obtained from this model for the dry PBrEDOT at 633 nm was 1.491 and its extinction coefficient  $k$  was 0.129, in good agreement with the values obtained by Baba et al.<sup>1</sup> The brush layer was generated on top of these two surfaces using a Cauchy film model.

The optical constants of the salt solutions used were obtained by fitting ellipsometry data obtained for a silicon substrate with a 25 nm silicon oxide layer. These were used, for each concentration and salt type, for modelling the ellipsometry data of gold-PBrEDOT-brush and silicon-brush samples. Given the complexity of the multilayer model presently used and the heterogeneity of the materials characterized, a graded layer was not introduced to model the brush layer, to limit the number of parameters to be fit. In addition, to distinguish PEDOT-driven swelling from brush swelling, the optical constants and swelling (50%  $\pm$ 10%) of the PEDOT initiator coatings were measured in aqueous solutions and used to build the model. Measurements were carried out in triplicate for each salt concentration at room temperature (23 °C).

#### *EIS model circuit parameters*

The EIS model circuit (Figure S1) consists of a resistor ( $R_2$ ) in series with a constant phase element (CPE) representing a non-ideal capacitance  $Q_2$  with a fractional exponent  $0 < \alpha < 1$ , both of which are in parallel to a capacitor  $C_1$ . Finally, a series resistance  $R_1$  is included to represent

the solution resistance. The total impedance can be calculated using the following equation, where  $\omega$  is the angular frequency:

$$Z = R_1 + \left[ j\omega C_1 + \left( \frac{1}{Q_2(j\omega)^\alpha} + R_2 \right)^{-1} \right]^{-1}$$

$R_1$  was calculated as the limit of  $\log(|Z|)$  vs  $\log(\omega)$  at high frequencies and fixed during modelling.  $\alpha$  can also be estimated from the low frequency gradient of  $\log(|Z|)$  vs  $\log(\omega)$  and fixed during modelling.

The two capacitive elements  $C_1$  and  $Q_2$  can be interpreted as the capacitance of an ionic double layer associated with the solution-CP interface, and a surface-state capacitance charged across the CP/brush layer respectively. The CPE includes an exponent  $\alpha$  to account for deviations from perfect capacitance, which are thought to arise from a variety of factors such as surface roughness, variable film thickness, and other inhomogeneities leading to distributed reaction rates and uneven current distributions.<sup>2-3</sup> At high frequencies, the impedance spectrum will be dominated by the capacitance of the  $C_1$  component; as the applied frequency decreases, the  $R_2 + Q_2$  branch becomes more prominent.<sup>4</sup>

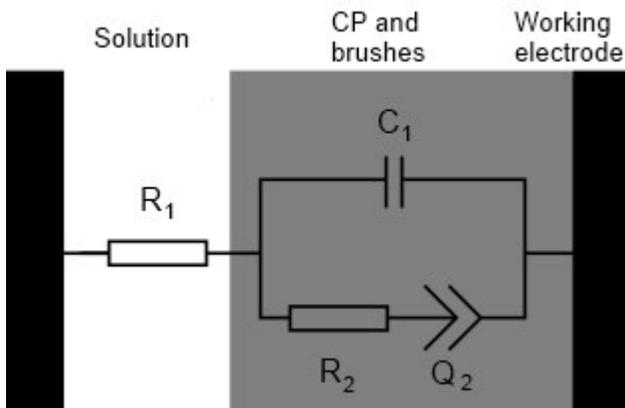


Figure S2. Simplified Bisquert model for a transmission line used to model EIS data.

### UV-vis turbidity measurements of free polymers

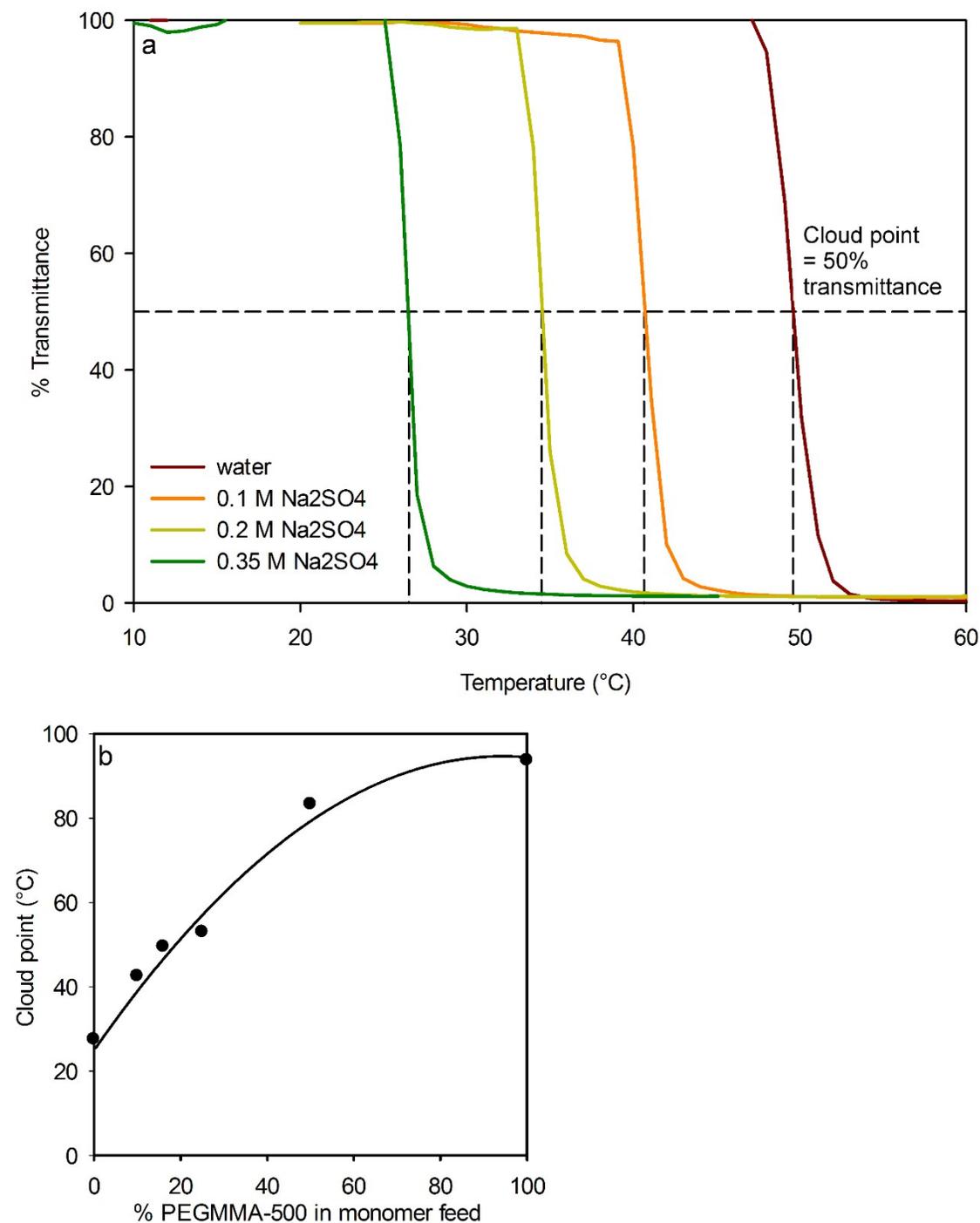


Figure S3. a) UV-vis turbidity measurements of P(PEGMMA-co-DEGMMA, 1:5) with increasing Na<sub>2</sub>SO<sub>4</sub> concentrations. The cloud point is determined as the temperature at which

transmittance drops to 50% of its original value. b) Effect of PEGMMA/DEGMMA monomer feed ratio on the cloud point temperature of the polymer in water.

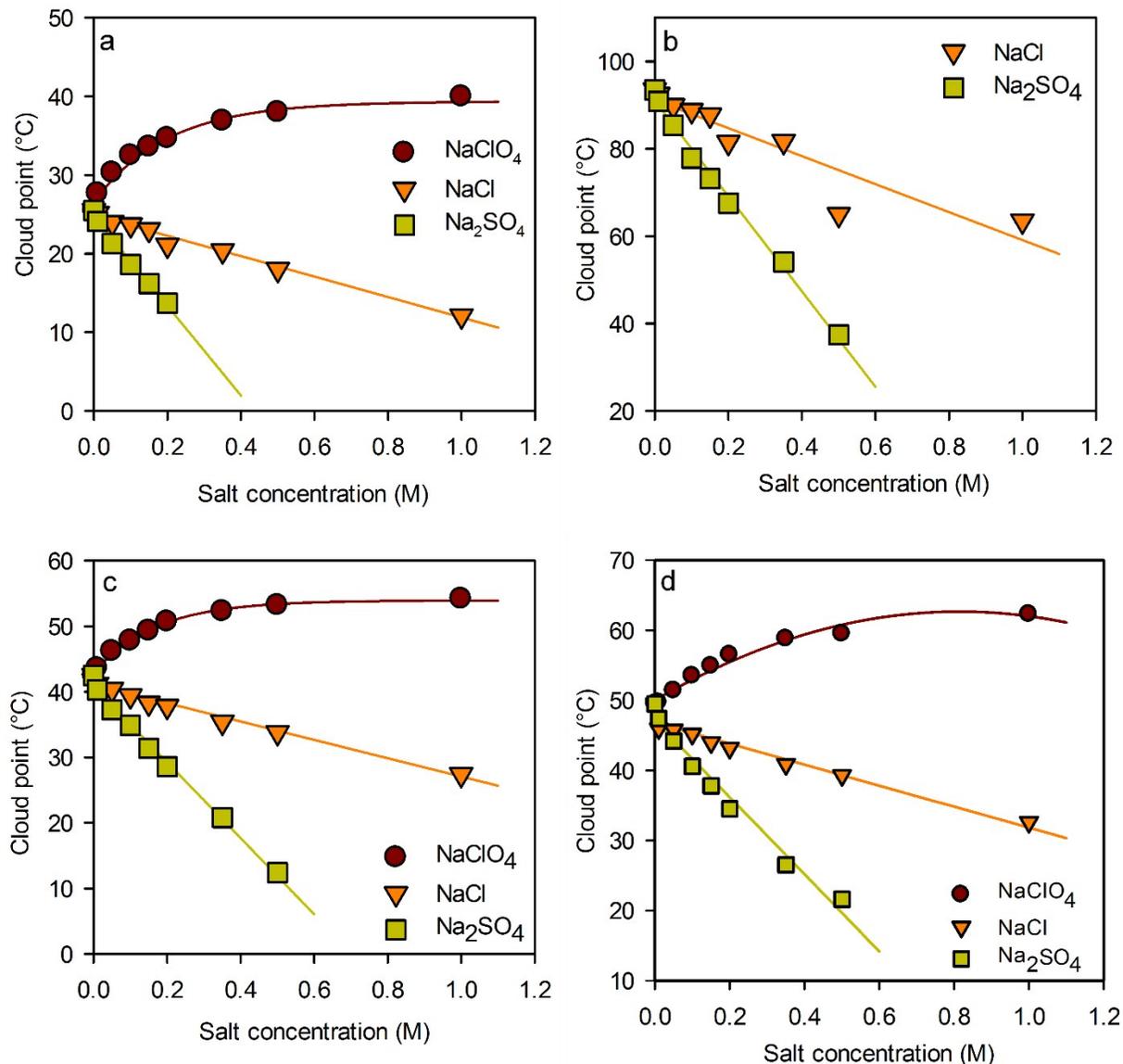


Figure S4. Effect of salt concentration on cloud point temperature as determined by UV-vis turbidity measurements in NaClO<sub>4</sub>, NaCl, and Na<sub>2</sub>SO<sub>4</sub>: a) PDEGMMA, b) P(PEGMMA-500), c) P(PEGMMA-co-DEGMMA, 1:9), and d) P(PEGMMA-co-DEGMMA, 1:5).

## **FTIR spectra of grafted brushes**

FT-IR spectra of films grafted with P(PEGMMA-*co*-DEGMMA, 1:5) for various times, as well as the ungrafted control was analyzed and shows the appearance of the characteristic C-H stretch peaks at 2800–3100  $\text{cm}^{-1}$ , corresponding to C-H groups on the PEDOT backbone and ethylene glycol units of the grafted brushes, and the growth of the ester C=O stretch at around 1700  $\text{cm}^{-1}$  verify successful grafting (Figure S5b). Growth of the brushes is tracked by calculating the area under the C-H peak at 2800–3100  $\text{cm}^{-1}$ , as shown in Figure S5c.

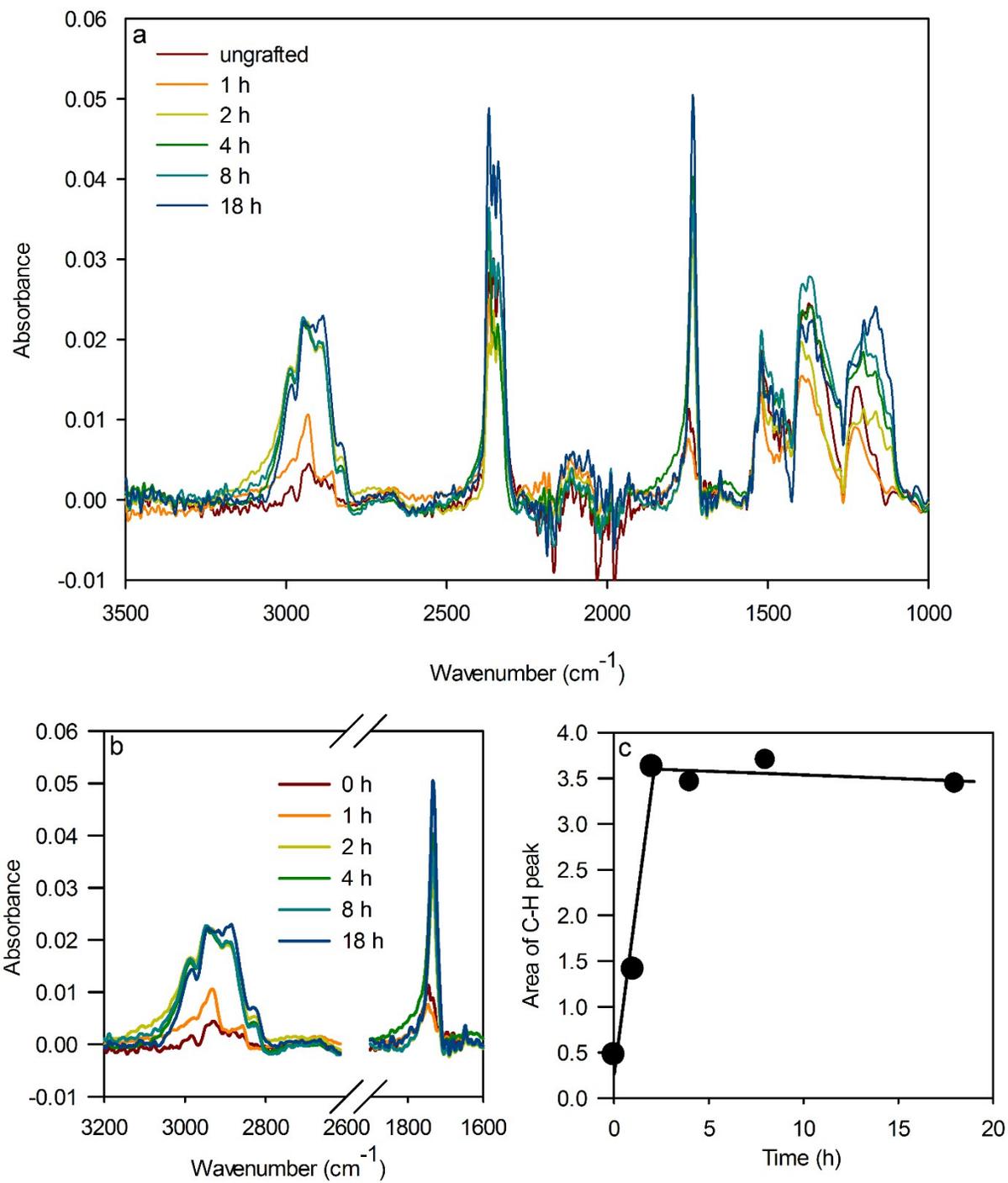


Figure S5. a) FTIR spectra of ungrafted PBrEDOT and grafted PBrEDOT-*g*-P(PEGMMA-*co*-DEGMMA, 1:5) grown for various periods of time and normalized on 1540  $\text{cm}^{-1}$ . b) Increase in

C-H stretch ( $3100\text{-}2800\text{ cm}^{-1}$ ) and C=O stretch ( $1800\text{-}1650\text{ cm}^{-1}$ ) with grafting time. c) Area under the C-H peak vs grafting time.

### Ellipsometry from Si substrates

Brushes were grown directly from Si substrates as controls to aid with modelling, due to the improved fit (as demonstrated by the lower mean square errors provided in Table 1 in the main paper). For the brushes grafted from silicon wafers, the brushes all exhibited a gradual collapse starting between 0.2 – 0.4 M. The higher mean square errors observed for the PBrEDOT-grafted brushes (MSE  $\sim 30$  vs. below 5 for the Si-grafted brushes) suggest some level of interpenetration of the brush within the PEDOT layer, leading to reduced fitting.

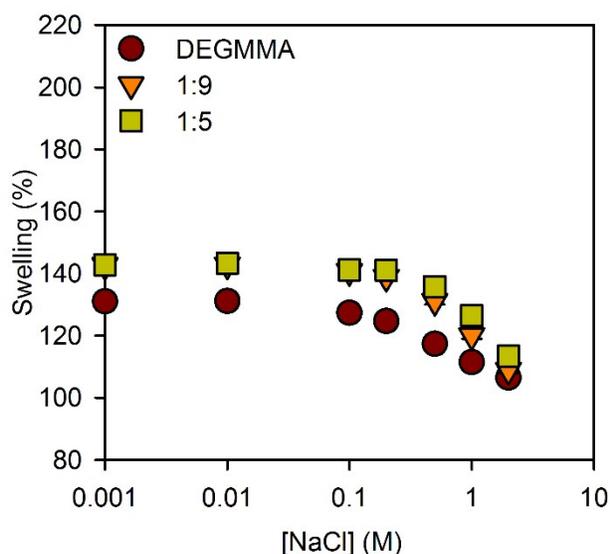


Figure S6. Ellipsometry measurements showing degree of swelling of brush compositions grafted from Si substrates as a function of concentration of NaCl solution.

## QCM-D data

Raw data

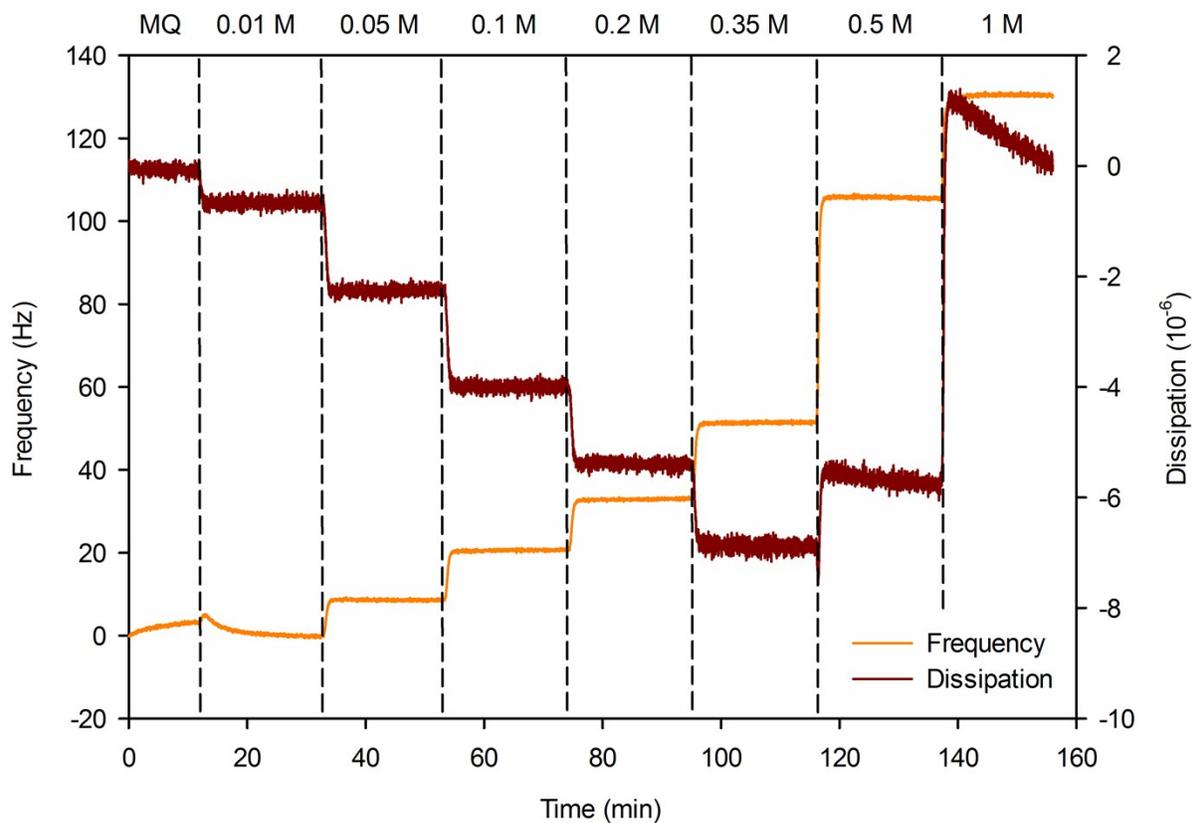
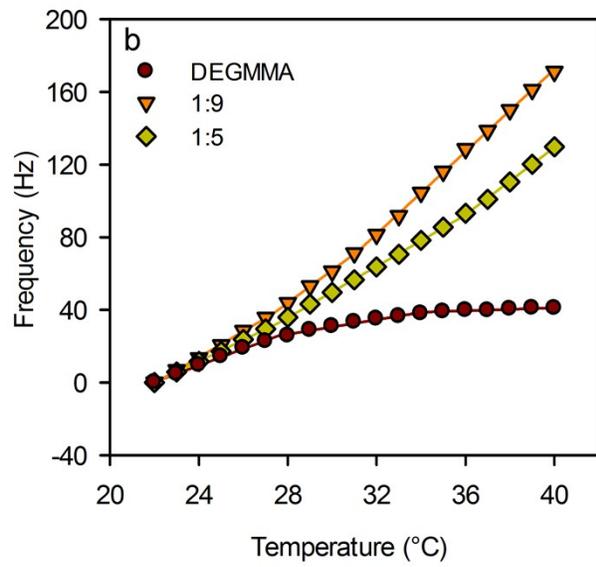
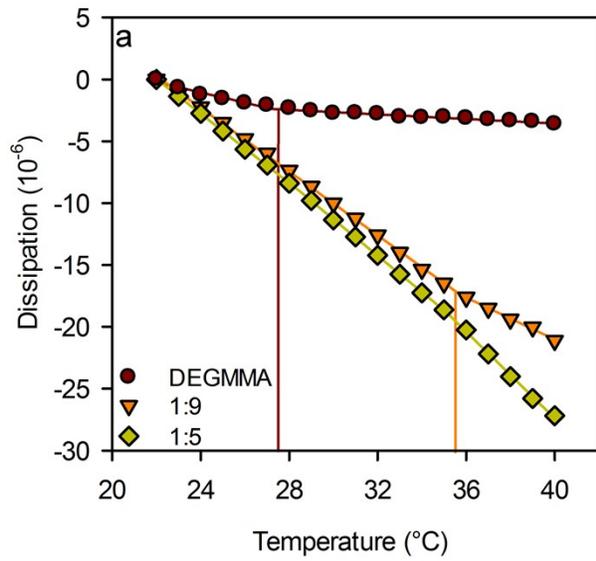


Figure S7. An example of recorded QCM-D data showing the change in frequency and dissipation of PBrEDOT-*g*-P(PEGMMA-*co*-DEGMMA, 1:9) with increasing Na<sub>2</sub>SO<sub>4</sub> concentration (5<sup>th</sup> overtone, 22 °C, data not normalized to ungrafted PBrEDOT).

*QCM-D response with temperature*



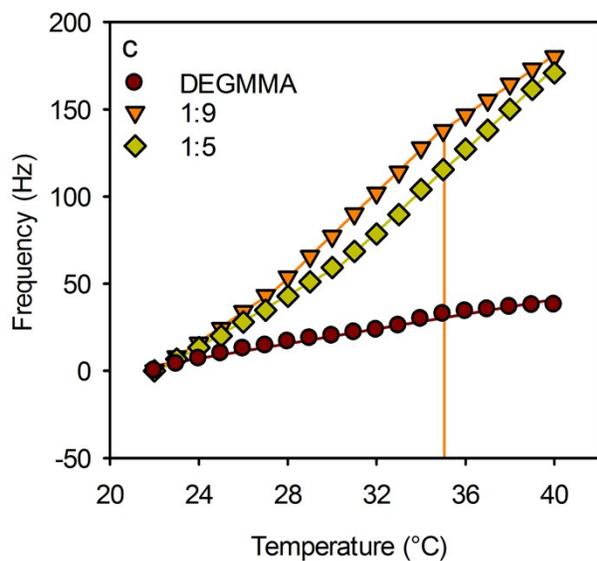
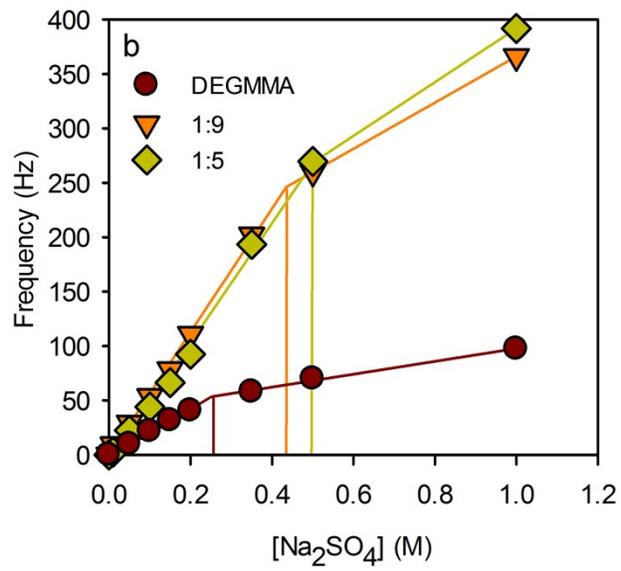
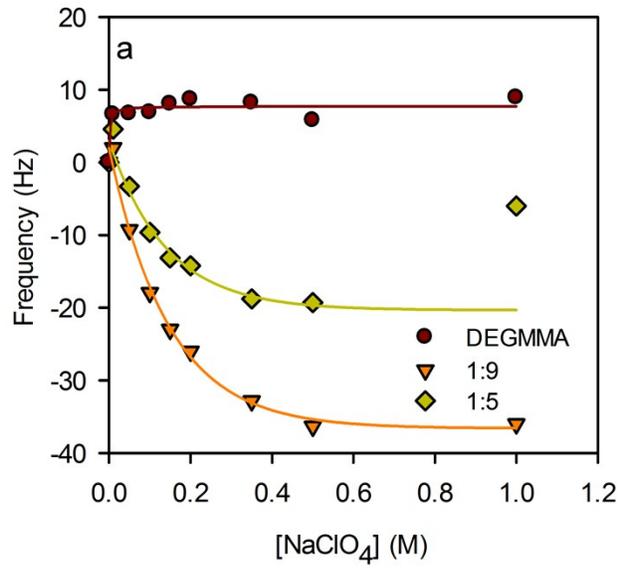


Figure S8. QCM-D measurements recorded as a function of temperature: a) dissipation and b) frequency measurements of PDEGMMA and P(PEGMMA-co-DEGMMA, 1:9 and 1:5) brushes in 0.1 M NaCl, and c) frequency measurements in 0.1 M Na<sub>2</sub>SO<sub>4</sub>. Dissipation in Na<sub>2</sub>SO<sub>4</sub> is provided in the main paper (Figure 4b).

*QCM-D response to salt concentration*



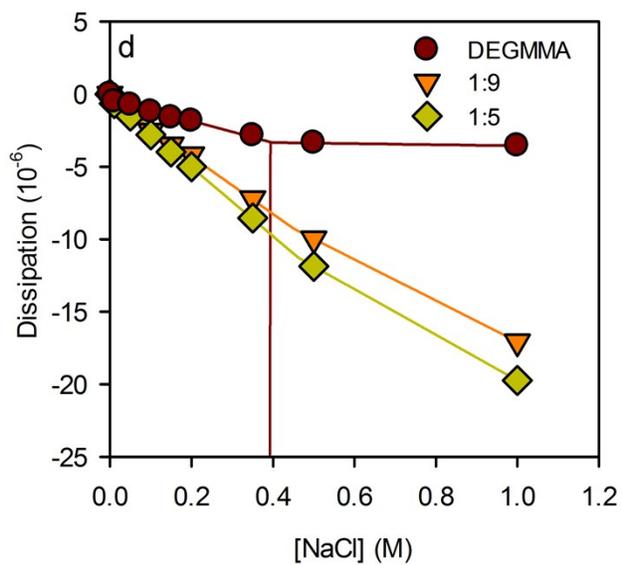
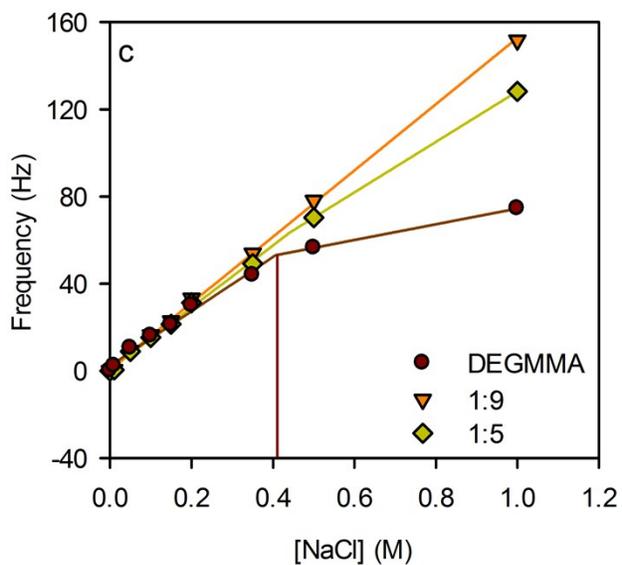
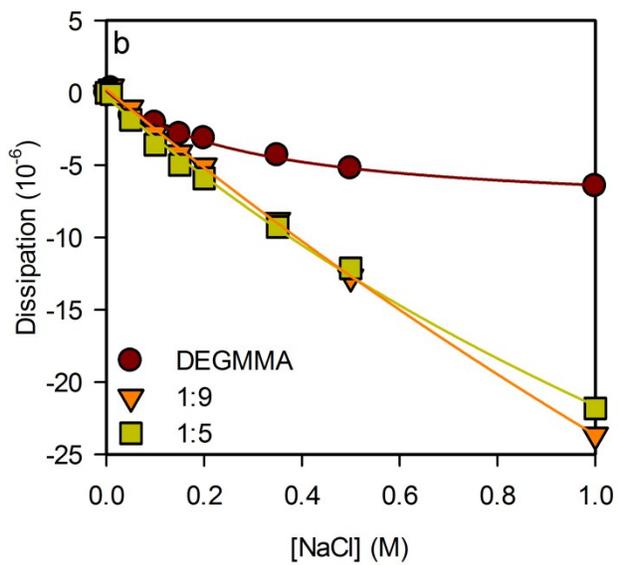
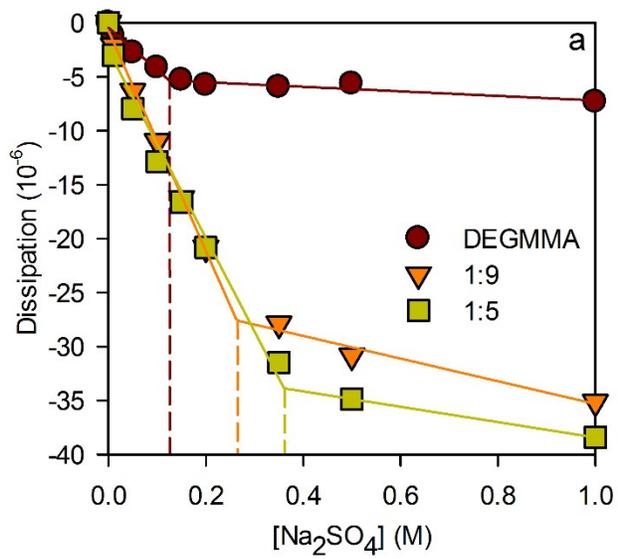


Figure S9. Frequency of PEGMMA/DEGMMA copolymers (1:9 and 1:5) and DEGMMMA homopolymer as a function of increasing salt concentration: a) NaClO<sub>4</sub>, b) Na<sub>2</sub>SO<sub>4</sub>, and c) NaCl. d) Dissipation in increasing NaCl concentration (NaClO<sub>4</sub> and Na<sub>2</sub>SO<sub>4</sub> provided in main paper, Figure 4c-d).



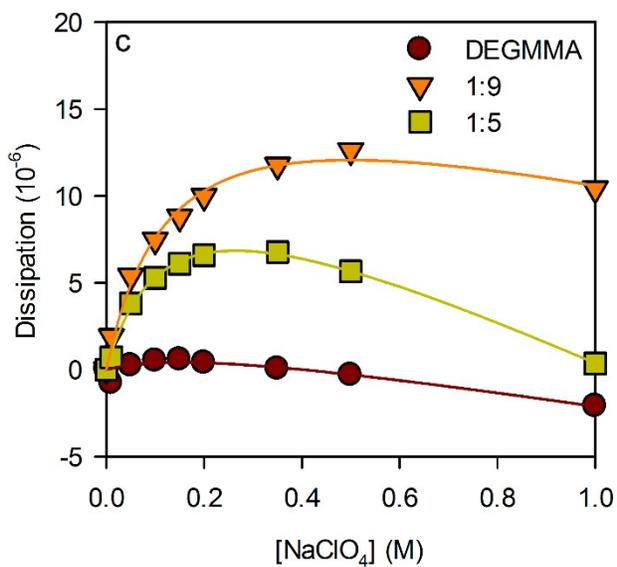


Figure S10. Repeated QCM-D dissipation measurements in response to increasing salt concentration, in a) Na<sub>2</sub>SO<sub>4</sub>, b) NaCl and c) NaClO<sub>4</sub>, repeated on the same samples as Figures 4c-d (main paper) and S8. These measurements follow the same trends as the previous measurements, although the recorded values differ too much for statistical analysis.

## Cyclic voltammetry characterization

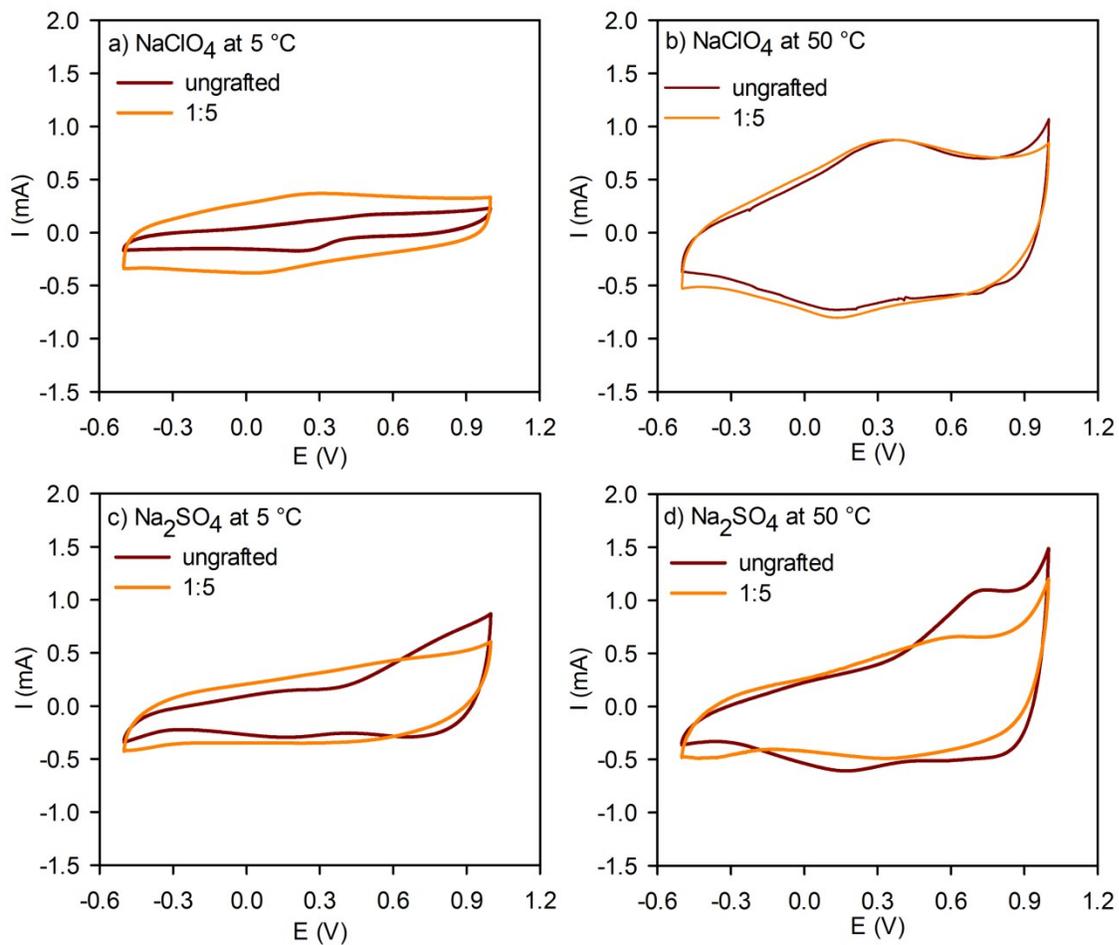


Figure S11. Cyclic voltammograms of ungrafted PBrEDOT and grafted PBrEDOT-g-P(PEGMMA-co-DEGMMA, 1:5) in 0.1 M salt solutions: a) NaClO<sub>4</sub>/5 °C; b) NaClO<sub>4</sub>/50 °C; c) Na<sub>2</sub>SO<sub>4</sub>/5 °C; d) Na<sub>2</sub>SO<sub>4</sub>/50 °C.

## Electrochemical impedance spectroscopy

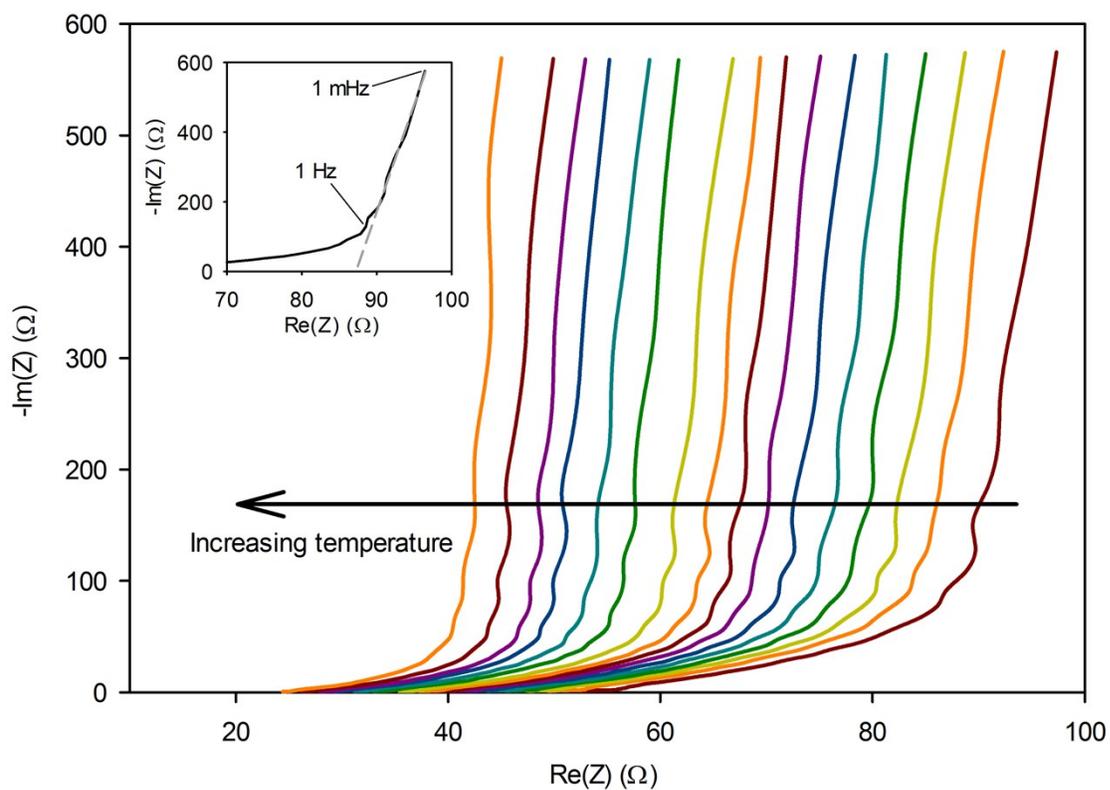
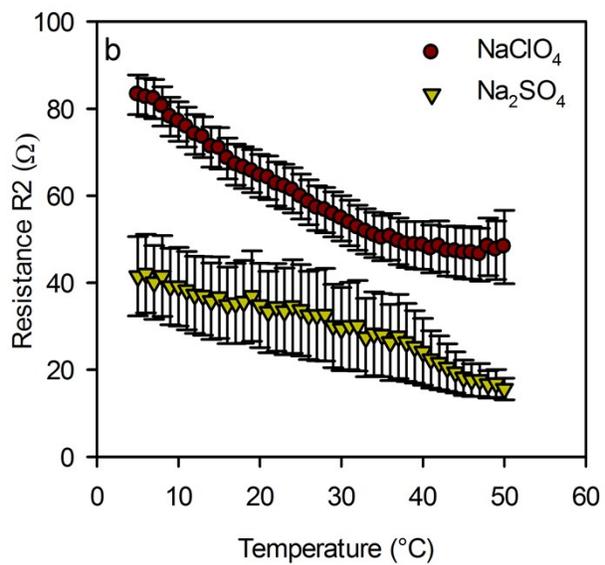
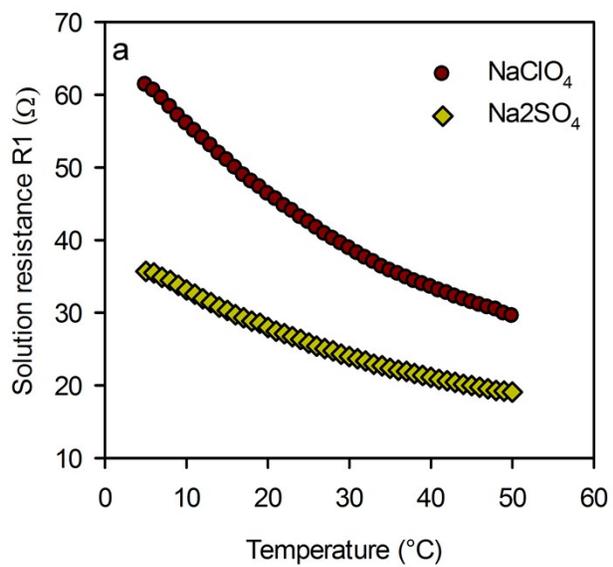


Figure S12. Overlaid Nyquist plots of grafted PEGMMA/DEGMMA sample in 0.1 M Na<sub>2</sub>SO<sub>4</sub>, recorded every 1 °C between 5-50 °C. (Inset: The slope of the impedance plot was calculated from 1 mHz to 1 Hz.)



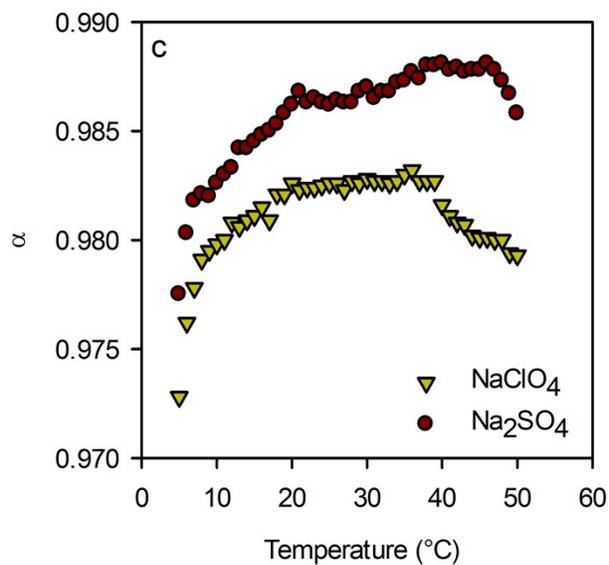


Figure S13. Solution resistance  $R_1$  (a), polymer film charge transfer resistance  $R_2$  (b) and fractional exponent  $\alpha$  (c) for grafted PEGMMA/DEGMMA sample in 0.1 M  $\text{NaClO}_4$  and 0.1 M  $\text{Na}_2\text{SO}_4$ .  $R_1$  and  $\alpha$  were approximated and fixed prior to modelling, and therefore do not have associated error bars.

### Anti-fouling behavior (QCM-D)

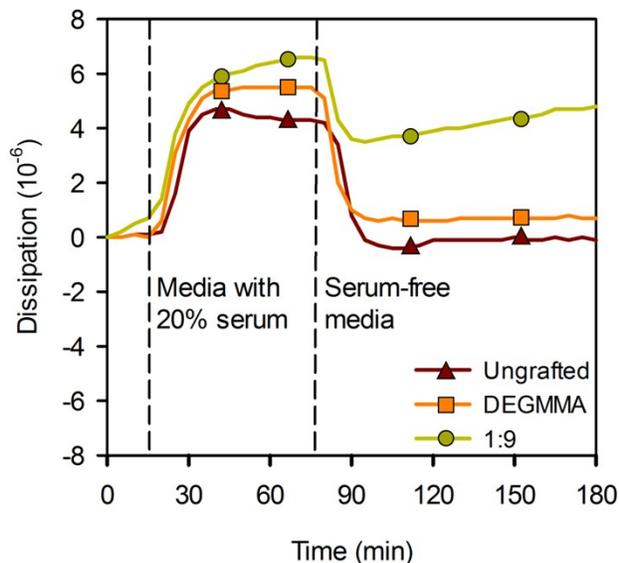


Figure S14. Dissipation response of ungrafted PBrEDOT, PBrEDOT-g-P(DEGMMA) and PBrEDOT-g-P(PEGMMA-co-DEGMMA, 1:9) corresponding to the frequency response shown in Figure 8, recorded in in serum-free media and 20% serum. All three samples showed an increase in dissipation on addition of serum, due to the bulk shift caused by the solution.

1. Baba, A.; Lübber, J.; Tamada, K.; Knoll, W., Optical Properties of Ultrathin Poly(3,4-ethylenedioxythiophene) Films at Several Doping Levels Studied by In Situ Electrochemical Surface Plasmon Resonance Spectroscopy. *Langmuir* **2003**, *19* (21), 9058-9064.
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3. Kim, C.-H.; Pyun, S.-I.; Kim, J.-H., An investigation of the capacitance dispersion on the fractal carbon electrode with edge and basal orientations. *Electrochim. Acta* **2003**, *48* (23), 3455-3463.
4. Chen, Y.; Hong, T.; Gopal, M.; Jepson, W. P., EIS studies of a corrosion inhibitor behavior under multiphase flow conditions. *Corros. Sci.* **2000**, *42* (6), 979-990.