

## Supporting Information

### **Nd Doping Bismuth Ferrite to Tune Electromagnetic Properties and Increase Microwave Absorption by Magnetic-dielectric Synergy**

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#### **Experimental section**

##### **Materials**

Sol-gel method was employed to prepare BFO and BNFO nanoparticles.  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ ,  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  and  $\text{Nd}(\text{NO}_3)_3 \cdot 6\text{H}_2\text{O}$  as raw materials in stoichiometric proportions were dissolved in 2-methoxyethanol. Citric acid in 1:1 molar ratio with respect to the metal nitrates was added to the solution, followed by polyethylene glycol as a dispersant. The mixture was stirred at 50 °C for 30 min to obtain the sol, and then it was dried at 80 °C to form the gel. The dried gel was calcined at 300 °C for 0.5 h. The calcined powders were sintered at 500 °C for 2 h.

##### **Characterizations and measurements**

The structure of the samples was collected using a X'Pert PRO system (Cu- $K_\alpha$ ). Rietveld refinement of the XRD patterns was performed by using the FullProf Program. The morphologies were measured using S-4800 SEM system. The microstructure were determined by a JEM-2100 TEM system. Raman spectra were obtained using a HR800 Raman spectrometer. XPS spectra were measured on a PHI Quantera system. The DC conductivity measurements were carried out using an Keithley 2401A-6517B multi-meter. Magnetic properties were measured by a Lakeshore 7407

vibrating sample magnetometer. The complex permittivity and permeability was measured on an Anritsu 37269D vector network analyzer by the waveguide method in X-band.

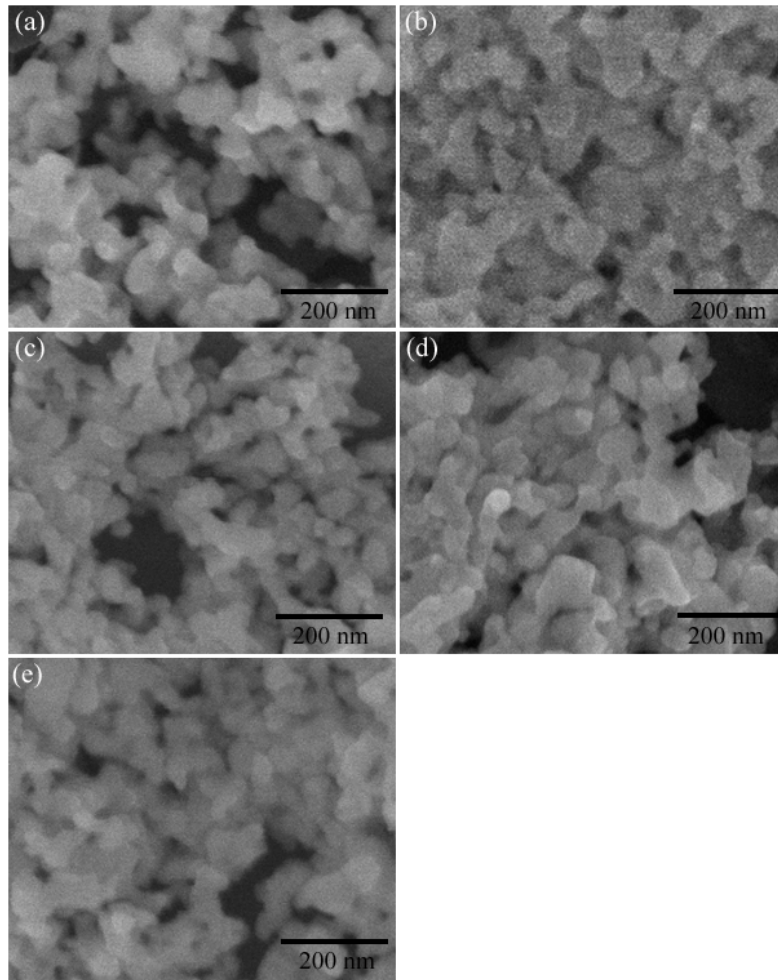
### First-principles calculations

The calculations on density of states were performed using the CASTEP program code based on the first-principles plane-wave pseudo-potential method. The generalized gradient approximation (GGA) was adopted along with the exchange-correlation function realized by Perdow-Burke-Emzerhof (PBE). The plane wave cutoff energy of 500 eV and  $2 \times 2 \times 2$   $K$ -point Monkhorst-Pack grid were applied to guarantee a well-converged structure under study. A  $2a \times 2b \times c$  supercell was adopted for all the calculations.

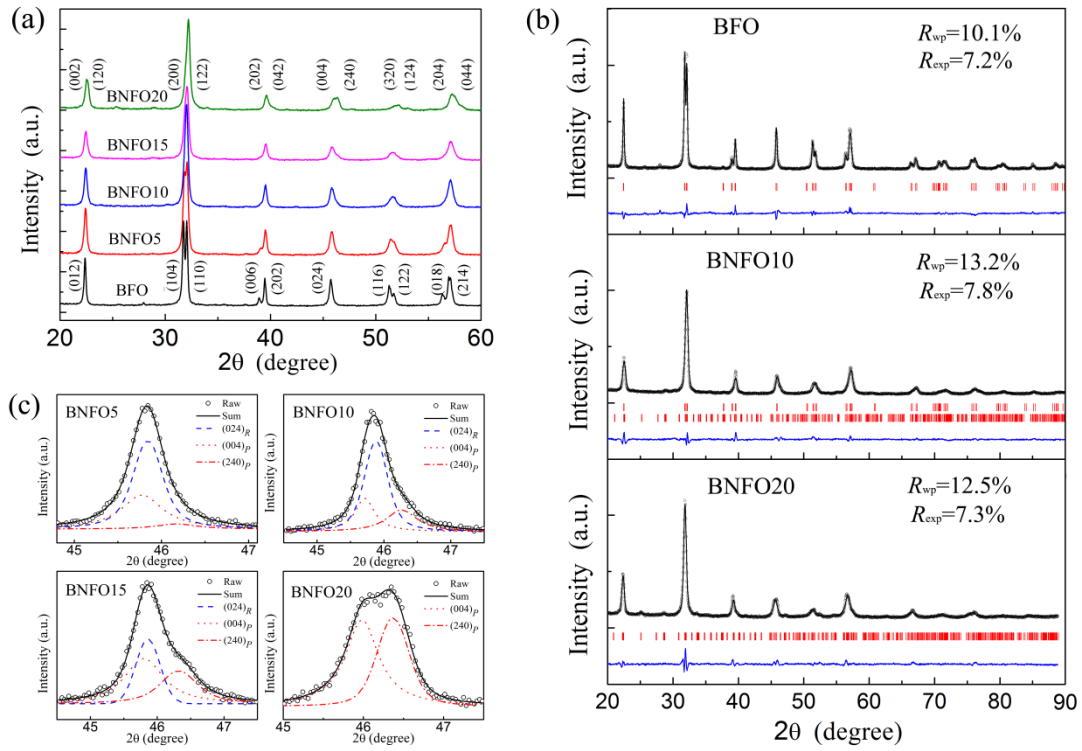
**Table S1.** Microwave absorption performance of representative bismuth ferrite materials

Samples	$f$ (GHz)	Min. $RL$ (dB)	$BW_{eff}$ (GHz)	$BW_p$ (%)	$d$ (mm)	Refs
BiFeO <sub>3</sub> nanoparticles	12.4–18	-26	4.8	85.7	3.5	23
BiFeO <sub>3</sub> ceramics	0–18	-17	0.8	4.4	5	24
BiFeO <sub>3</sub> ceramics	12.4–18	-23	3.5	62.5	1.2	25
Bi <sub>0.85</sub> Ho <sub>0.15</sub> FeO <sub>3</sub> /paraffin wax composite	2–18	-11	0.5	3.1	3	26
Bi <sub>0.8</sub> La <sub>0.2</sub> FeO <sub>3</sub> nanoparticles	8.2–12.4	-30	1	17.8	6	27
Bi <sub>0.8</sub> Nd <sub>0.2</sub> FeO <sub>3</sub> nanoparticles	8.2–12.4	-42	3	71.4	2.3	This work

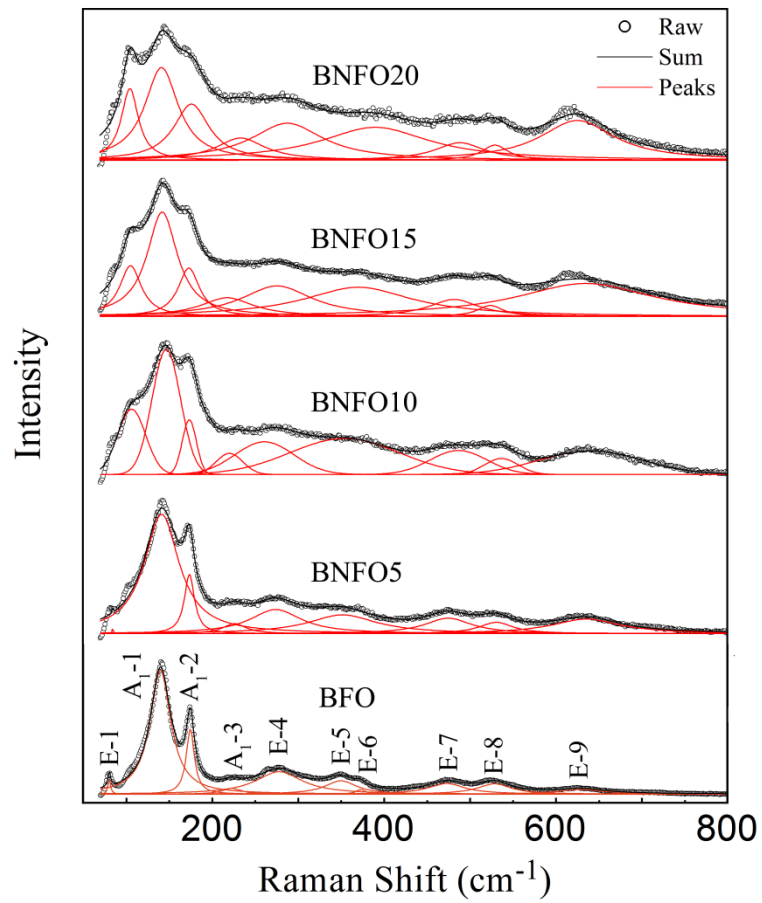
$f$ , frequency range; Min. $RL$ , minimum reflection loss;  $BW_{eff}$ , effective bandwidth ( $RL \leq -10$  dB);  $BW_p$ , the proportion of effective bandwidth in the frequency range investigated;  $d$ , thickness.



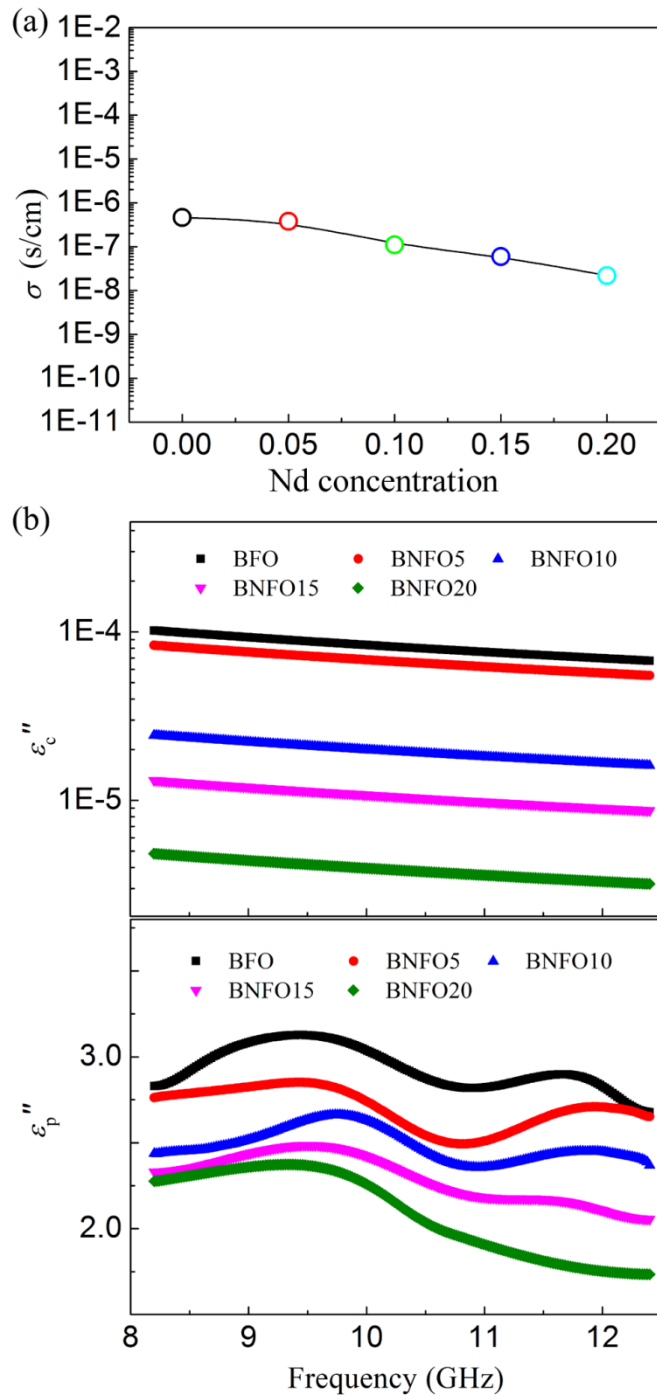
**Fig. S1 SEM images of (a) BFO, (b) BNFO5, (c) BNFO10, (d) BNFO15 and (e) BNFO20 powders.**



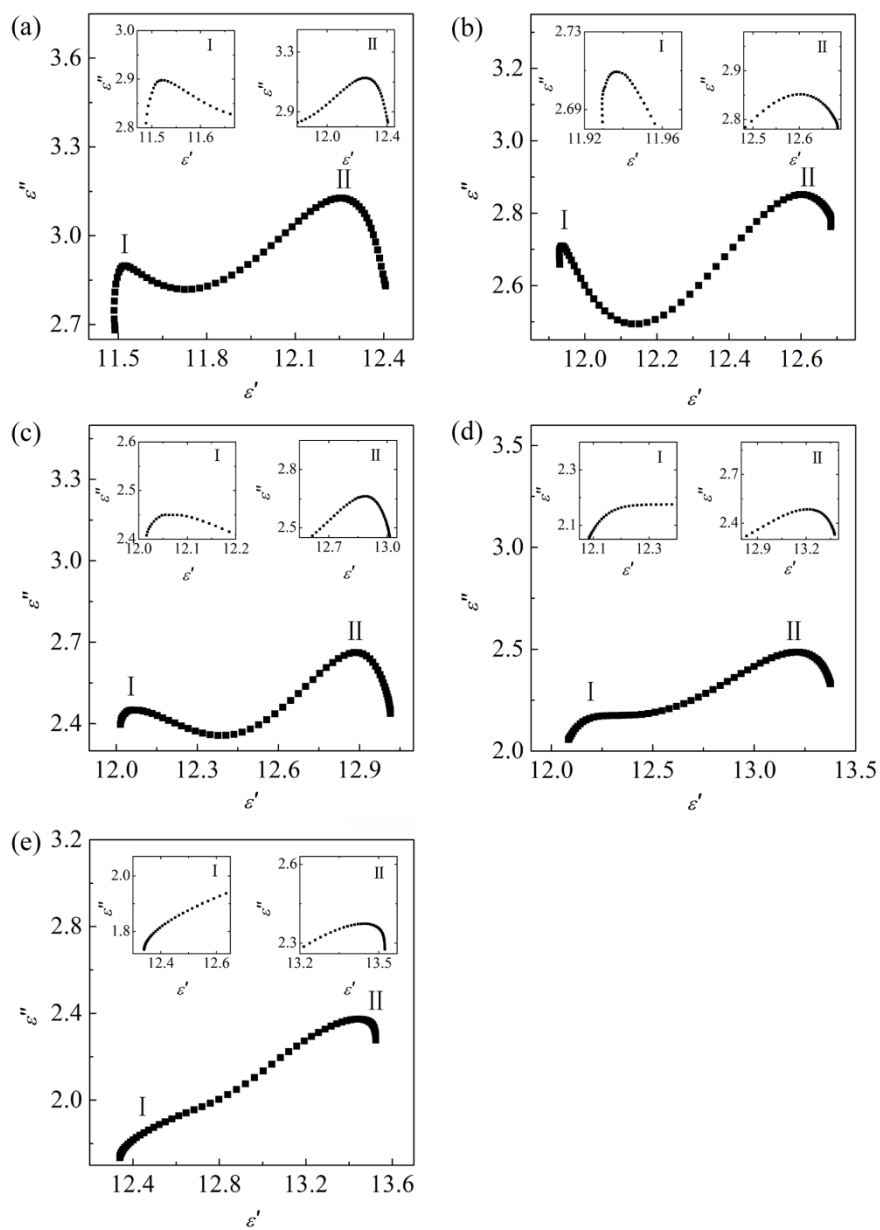
**Fig. S2 (a)** The XRD patterns of BFO and BNFO. **(b) Rietveld refinement of BFO, BNFO10 and BNFO20:** experimental (circle), calculated (black line). **(c) The separated diffraction peaks in the range  $2\theta = 45-47^\circ$ .** The XRD patterns and Rietveld refinement of BFO and BNFO demonstrate that BFO is single phase perovskite with the  $R3c$  space group, while BNFO20 possesses the  $\text{PbZrO}_3$ -like structure as confirmed using the  $Pbam$  space group. In the intermediate compositions, the  $R3c$  phase and the  $Pbam$  phase coexist. The separated diffraction peaks in the range  $2\theta = 45-47^\circ$  show that the intensity of  $(0\ 0\ 4)_P$  and  $(2\ 4\ 0)_P$  of the  $Pbam$  phase increases while the intensity of  $(0\ 2\ 4)_R$  of the  $R3c$  phase decreases with the increase of Nd concentration, indicating the proportion of the  $Pbam$  phase increases gradually.



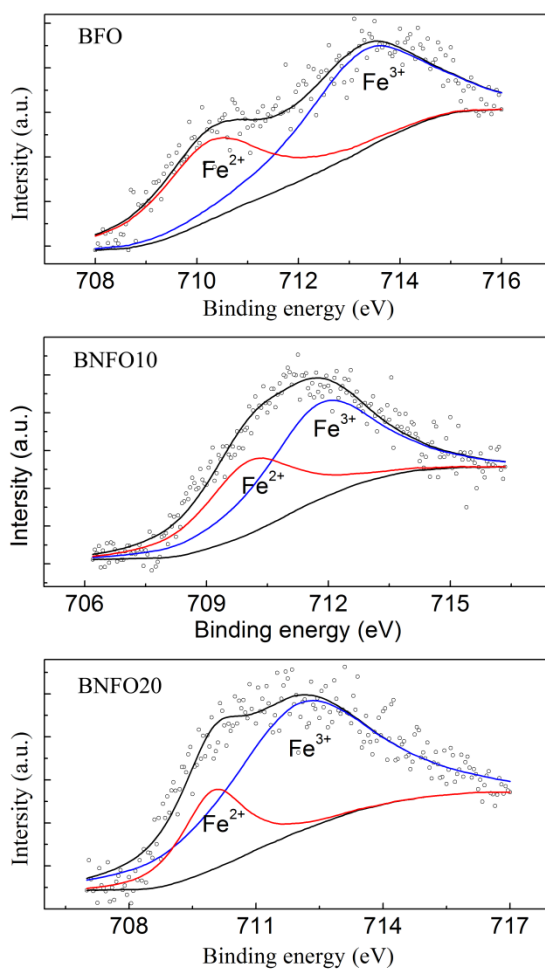
**Fig. S3 Raman spectra of BFO and BNFO at room temperature.** The  $E-1$ ,  $A_{1-1}$ ,  $A_{1-2}$  and  $A_{1-3}$  modes dominated by the Bi–O covalent bonds weaken with the increase of Nd concentration, indicating weakening of the hybridization of Bi  $6s$  and O  $2p$  orbitals.



**Fig. S4 (a) The DC conductivity versus Nd concentration at room temperature. (b) The conduction part ( $\epsilon_c''$ ) and polarization part ( $\epsilon_p''$ ) of imaginary permittivity for BFO and BNFO versus frequency. The  $\epsilon_c''$  is obtained based on Debye equation, i.e.  $\epsilon_c'' = \sigma/(\epsilon_0\omega)$ , where  $\epsilon_0$  is vacuum permittivity,  $\omega$  is angular frequency and  $\sigma$  is conductivity of medium. Subtracting the  $\epsilon_c''$  from the imaginary permittivity, the  $\epsilon_p''$  is gained.**

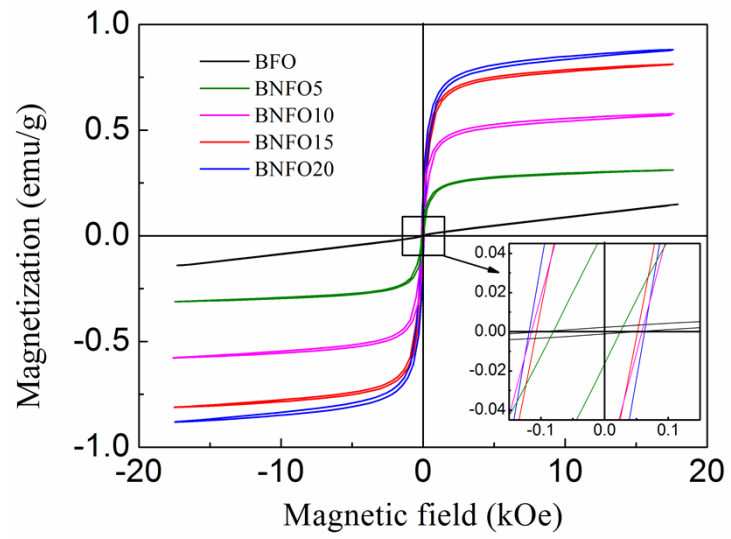


**Fig. S5 The Cole-Cole plots of (a) BFO, (b) BNFO5, (c) BNFO10, (d) BNFO15 and (e) BNFO20.**



**Fig. S6** High-resolution XPS of Fe  $2p_{3/2}$  core levels for BFO, BNFO10 and BNFO20. Circle, black line, blue and red lines represent experimental data, fitting data, the peak-fitting simulations of  $\text{Fe}^{3+}$  and  $\text{Fe}^{2+}$ , respectively. The appearance of  $\text{Fe}^{2+}$  reveals oxygen vacancies exist in BFO and BNFO, because  $\text{Fe}^{2+}$  and oxygen vacancies appear simultaneously for charge compensation in BFO materials.





**Fig. S7 Hysteresis loops of BFO and BNFO at room temperature.**

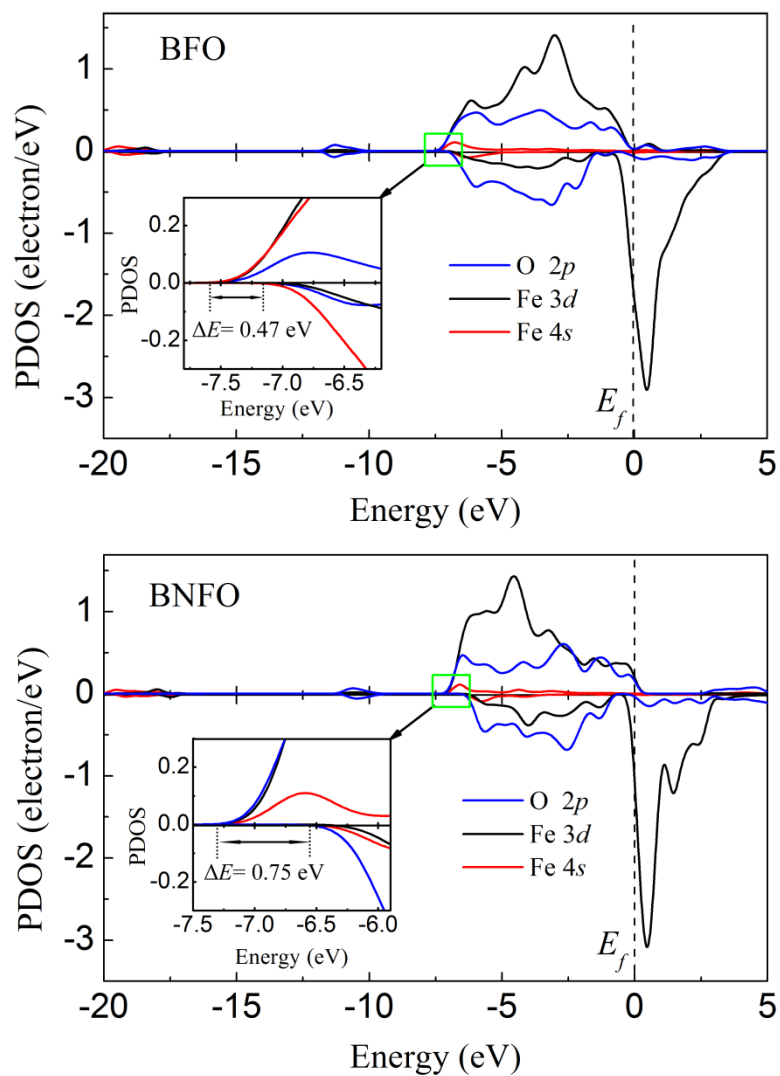


Fig. S8 The densities of states for Fe 3d, 4s and O 2p electrons of BFO and BNFO.

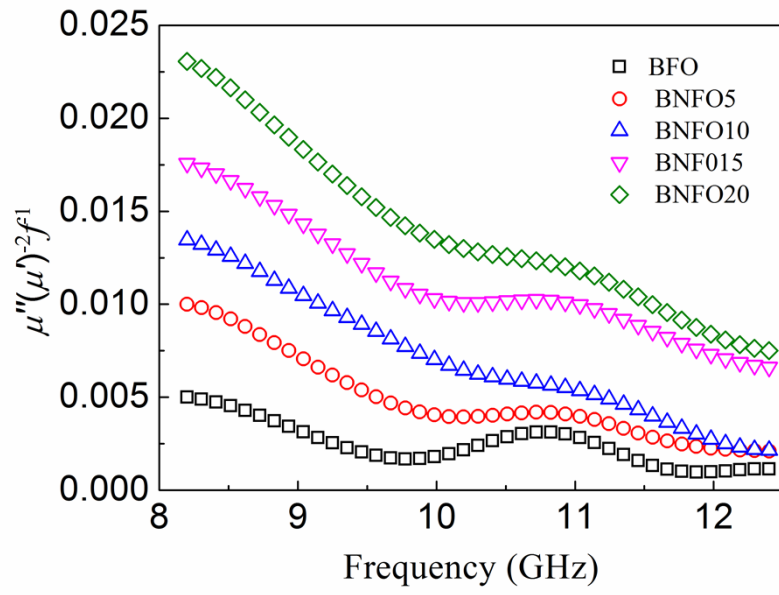
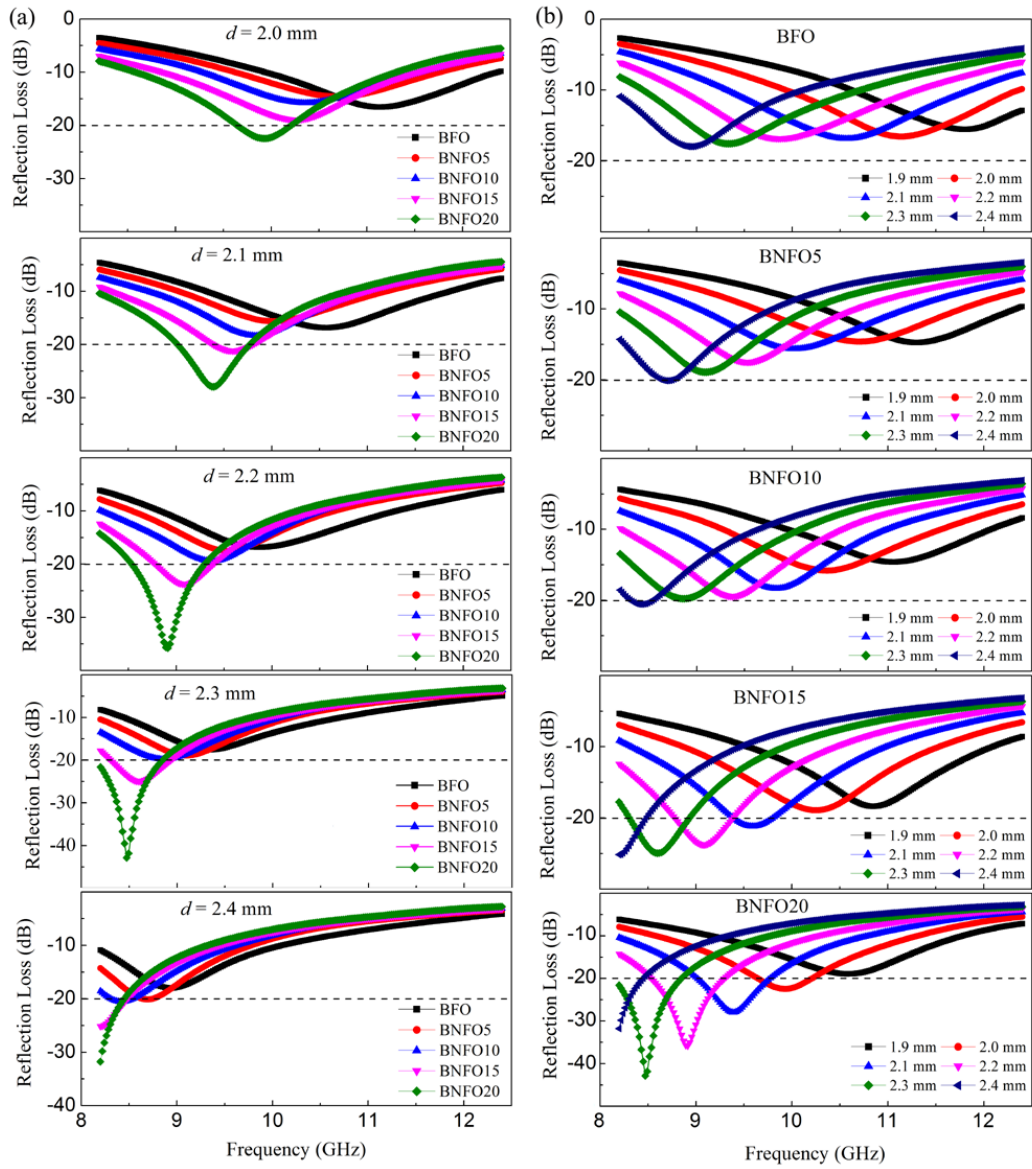


Fig. S9 The eddy current coefficients  $\mu''(\mu')^{-2}f^1$  of BFO and BNFO versus frequency.



**Fig. S10** The *RL* of BFO and BNFO at various (a) Nd concentrations and (b) thicknesses.