## Supporting Information

## Tuning of intrinsic antiferromagnetic to ferromagnetic ordering in microporous $\alpha$ -MnO<sub>2</sub> by inducing tensile strain

G.Kruthika<sup>a,b</sup>, J.Karthikeyan<sup>a,b</sup>, and P.Murugan<sup>a,b\*</sup>

<sup>a</sup>Functional Materials Division, CSIR Central Electrochemical Research Institute (CECRI), Karaikudi–630 003, Tamil Nadu, India

<sup>b</sup>Academy of Scientific & Innovative Research, CECRI, India

\*E-mail: murugan@cecri.res.in.



Fig. S1: Polyhedral model of optimized structure of  $\alpha$ -MnO<sub>2</sub> 0.375H<sub>2</sub>O compound. It is observed that the tunnel accommodated with two water molecules distorts the crystal structure. Thus, it gives lesser stability as compared to  $\alpha$ -MnO<sub>2</sub> 0.25H<sub>2</sub>O compound.

Table. S1. Number of sites obeying GKA rule (two Mn atoms that are connected through  $O(sp^2)$  atoms) and triangular ( $\Delta$ ) rule (in the double chain) in different types of magnetic configurations of  $\alpha$ -MnO<sub>2</sub>.0.25 H<sub>2</sub>O.

Туре	AFM-C	AFM-C2	AFM-A2	FM	FIM
$\Delta$ rule	All 4 sites are not obeyed	All 4 sites are obeyed	All 4 sites are not obeyed	All 4 sites are obeyed	All 4 sites are obeyed
GKA rule	All 4 sites are obeyed	All 4 sites are obeyed	All 4 sites are not obeyed	All 4 sites are not obeyed	Only 2 sites are obeyed



Fig. S2 (a). Exchange energies (J1, J2) are shown with respect to triaxial strain. (b) Optimized structure distorted at 11% tensile strained  $\alpha$ -MnO<sub>2</sub>.0.25H<sub>2</sub>O. It is observed that at 11% tensile strain, the linkage between double chains is broken.



Fig. S3. Exchange energy (J2) for unhydrated  $\alpha$ -MnO<sub>2</sub> compound. It shows that the FM ordering is induced by an application of 3% tensile strain.



Fig. S4: Variation of (a) Mn-Mn distance and (b) Mn-O-Mn angle with respect to the triaxial strain introduced  $\alpha$ -MnO<sub>2</sub>.0.25H<sub>2</sub>O compound. All the values are given for AFM-C2 configuration.

Table. S2. Comparison of various parameters of unstrained and 4% tensile strained  $\alpha$ -MnO<sub>2</sub>. 0.25H<sub>2</sub>O compound, calculated using different exchange correlation functionals (PW91 and PBE).

		Unstrained	4% tensile strain
Moment on	PW91	3.2	3.4
Mn atoms	PBE	3.1	3.3
$(\mu_{\rm B})$			
Exchange	PW91	-16	10
Energy			
$(\Delta = E_{afmc2-fm})$	PBE	-9	19
(meV/f.u.)			



Fig. S5: The distortion in geometry of  $MnO_6$  octahedra are shown for various triaxial strained  $\alpha$ -MnO<sub>2</sub> compounds. Mn-O bond distances (in Å) are given in the diagram.



Fig. S6: Charge transfer between sorbent and  $\alpha$ -MnO<sub>2</sub> framework ( $\Delta \rho = \rho_{\alpha-MnO2,0.25X} - \rho_{\alpha-MnO2} - \rho_X$ ) is shown for  $\alpha$ -MnO<sub>2</sub> 0.25*X* compound, where *X* = Xe (a) and NH<sub>3</sub> (b). It demonstrates that the small charge transfer is occurred between the tunnel species and lattice oxygen. Isosurface for respective compounds are plotted for values of 3×10<sup>-3</sup> and 4×10<sup>-3</sup> e/Å<sup>3</sup> respectively. Yellow (blue) isosurface represents charge accumulation (depletion) region.

Table. S3. Lattice parameters, relative energy of  $\alpha$ -MnO<sub>2</sub> compound having various tunnel species with respect to different (AFM-C2, FM) magnetic configurations and local magnetic moment (absolute average value) of Mn, O(sp<sup>3</sup>), O(sp<sup>2</sup>) atoms in their stable ground state magnetic configurations.

Tunnel species	Lattice parameters (Å)			Magnetic configurations (meV/f.u.)		Local magnetic moment $(\mu_B)$		
	a	b	с	AFM-C2	FM	Mn	O(sp <sup>3</sup> )	O(sp <sup>2</sup> )
Ar	9.83	9.83	2.90	0	12	3.2	0.2	0.0
Kr	9.86	9.87	2.92	0	8	3.3	0.2	0.0
Xe	9.99	9.99	2.96	28	0	3.4	0.2	0.1
NH <sub>3</sub>	10.04	9.73	2.89	20	0	3.4	0.2	0.1
CH <sub>4</sub>	9.69	9.89	2.90	0	13	3.2	0.2	0.0
H <sub>2</sub> S	9.92	9.98	2.79	0	83	3.0	0.1	0.0
Li <sub>2</sub> O	9.80	10.14	2.89	2	0	3.5	0.2	0.1