# Supporting information: Highly efficient and stable MoS<sub>2</sub> FETs with reversible *n*-doping using a dehydrated poly(vinyl-alcohol) coating

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### **Electrical characterization of PVA**

To ensure that the PVA coating makes no substantial contribution to the source drain current of the PVA-doped  $MoS_2$  FET, we measured the current characteristics of a PVA film in a similar transistor configuration. The transfer characteristics of this PVA device are shown in Figure S1. The total current, I<sub>D</sub>, which is composed of the current flowing through PVA and the back gate leakage current, was obtained from a device with L<sub>CH</sub> 2.5 µm and W<sub>CH</sub> of 28.1 µm with 47 nm of PVA bridging the source and drain contacts. The current was orders of magnitude smaller than the corresponding value of the PVA-doped MoS<sub>2</sub> FET (|I<sub>D</sub>/W|<500 fA), as expected for an insulating material.<sup>S1</sup>



Figure S1: Plot of  $I_D$  against  $V_{GS}$  for a FET containing 1% PVA as a channel.

#### FTIR spectroscopic and XRD characterization of PVA films

Fourier transform infrared (FTIR) spectroscopy was used to extract information about the composition of the deposited PVA films. First a thick film (~100  $\mu$ m) deposited by drop casting on a 55-nm-thick SiO<sub>2</sub> film on top of a silicon wafer was assessed for reference. The FTIR spectrum of this PVA film (Figure S4) clearly contained absorption peaks from O-H, C-H and C=O chemical groups.<sup>52</sup>



Figure S2: FTIR spectrum of a thick (~100 μm) PVA film.

Next, attenuated total reflectance FTIR (ATR-FTIR) spectroscopy was used to evaluate a 20-nm-thick PVA film spin coated on a 55-nm-thick  $SiO_2/Si$  substrate under the same conditions that were used in the experiments in the main paper. This ATR-FTIR spectrum is presented in Figure S5.



Figure S3: ATR-FTIR spectrum of a 50-nm-thick PVA film.

The same chemical groups can be identified in the ATR-FTIR spectrum of the 20-nm-thick film as in the FTIR spectrum of the thicker film. This is further proof of the formation of a thin PVA film by the described process. However, the peaks at lower wavenumbers were poorly resolved because of the small thickness of the material being sampled.

Next, grazing incidence X-ray diffraction (GI-XRD) patterns of thick PVA films (~100  $\mu$ m) were collected to obtain information about the crystallinity of the material. Three different PVA films were assessed: A) as-deposited, B) as-deposited + annealed for 20 min at 90 °C in N<sub>2</sub> atmosphere, and C) as-deposited + annealed for 20 min at 90 °C (N<sub>2</sub>) + aged for 1 day at room temperature (Figure S4).



Figure S4: GI-XRD patterns of a thick (~100 μm) PVA film under different post-treatment conditions. The figures are shifted for better visualization.

The peak at around  $2\theta = 20^{\circ}$  corresponds to the 101 plane of semi-crystalline PVA.<sup>S3</sup> Thus, the height of this peak can be used as an indication of the degree of crystallization of the PVA film.<sup>S4</sup> The height of this peak did not change drastically when the sample was annealed at 90 °C for 20 min. However, when the annealed sample was aged for 1 day, the relative height of the peak increased almost 51%, indicating that the crystallization of the PVA film increased. This observation supports the argument made in the main text that the device performance is enhanced by the crystallization of PVA.

## Influence of surface passivation on doping of MoS<sub>2</sub>

It has been shown that surface passivation of thin-film MoS2 can partially alter its electrical performance <sup>S5</sup>. Thus, to ensure that the charge modulation observed in our PVA-treated MoS2 FETs is mainly originated from doping effect rather than simply from surface passivation, two tests were performed:

First, as described in the methodology section, all the devices are measured in N2 atmosphere. This minimizes the time that the device is exposed to air. Thus, quenching of the intrinsic n-doping of the  $MoS_2$  film is not likely. Moreover, to demonstrate that the ambient air does not affect the device substantially, it was measured in vacuum condition (10<sup>-7</sup> mbar) after degassing for 2 days. As shown in figure S5-a, the threshold voltage (V<sub>TH</sub>) shows a negligible variation when switching from N2 to vacuum condition, while after coating with PVA and measuring the device in N2 we observe a large negative shift in V<sub>TH</sub>, which indicates n-doping.

Next, to ensure that the doping is not originating from a passivation effect of PVA but rather from its doping effect, we applied poly-trifluoroethylene (TrFE), a highly non-reactive thermoplastic fluoropolymer, on our  $MoS_2$  devices and compare them with PVA doped ones. As shown in figure S5-b TrFE passivation has minimum impact on the device  $V_{TH}$  (and thus its doping level) while the PVA doped device shows a strong  $V_{TH}$  shift. We believe these results rule out substantial contribution arising from surface passivation.



*Figure S5:* Id-Vg of a MoS2 FET measured in N2 and vacuum conditions before and after PVA doping. (b) PVA vs TrFE coated MoS2 FET

#### **References:**

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