

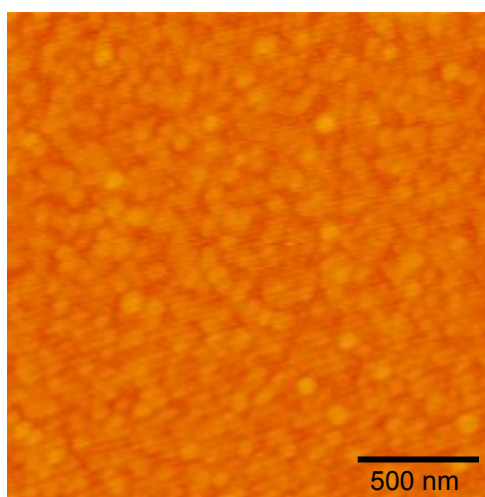
**Supplementary Information:**

Graphene/nitrogen-functionalized graphene quantum dots hybrid  
broadband photodetectors with buffer layer of boron nitride  
nanosheets

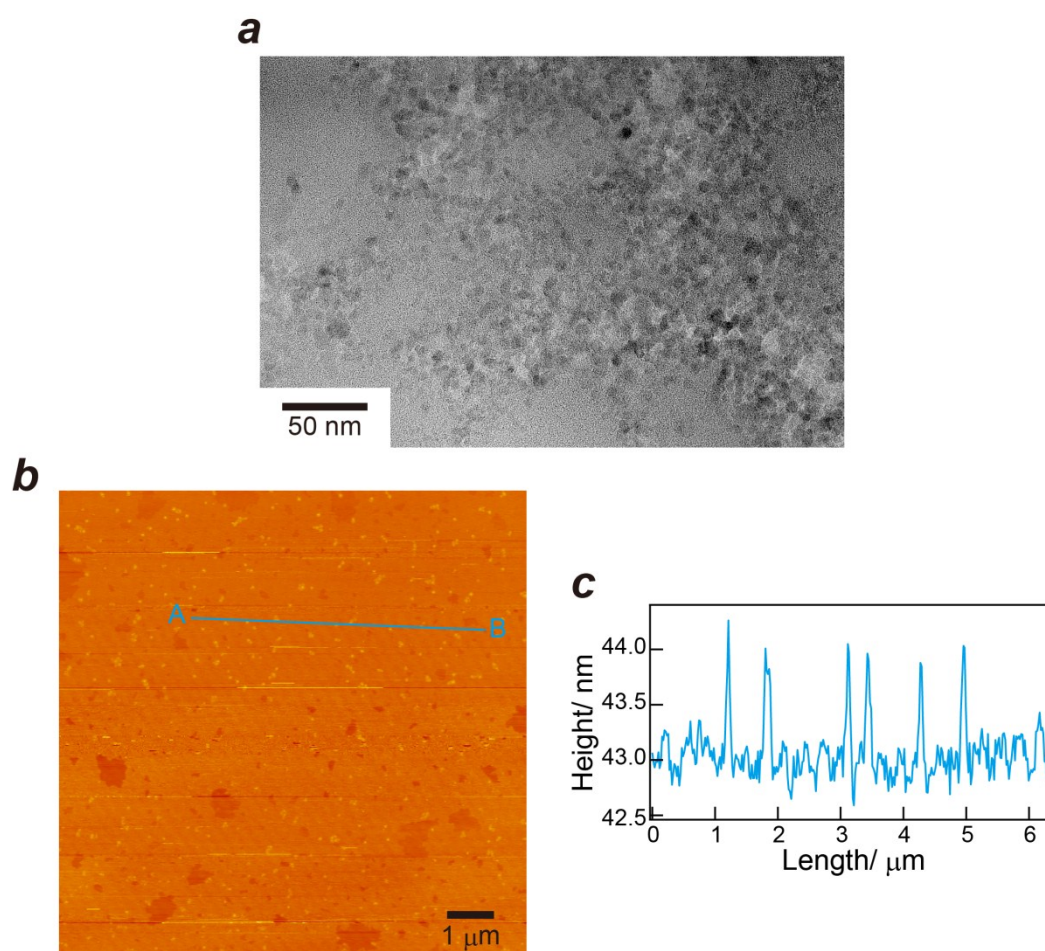
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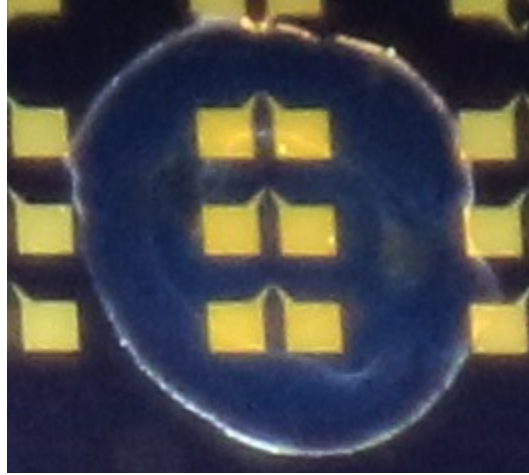
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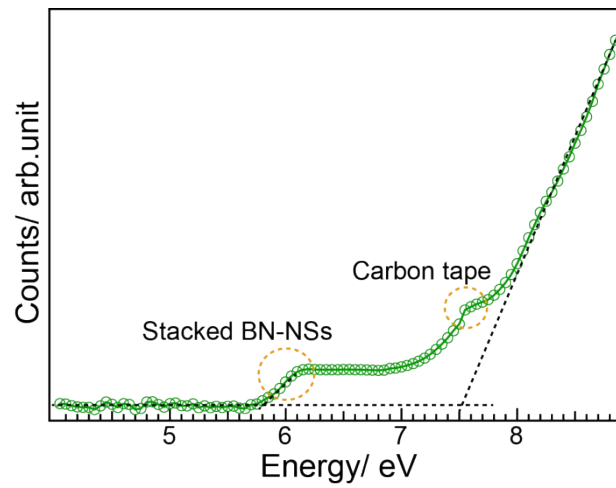
**Fig. S1** AFM image of BN-NSs layer coated on graphene.



**Fig. S2** Characterization of diaminonaphthalene-functionalized graphene quantum dots (DAN-GQDs): (a) TEM image of DAN-GQDs. (b) AFM image of aggregated DAN-GQDs obtained by spin-casting a suspension of the DAN-GQDs in DMF onto mica. (c) Height profile along the line A–B shown in (b).



**Fig. S3** Optical image of DAN-GQDs/BN-NSs@GFET hybrid photodetector.



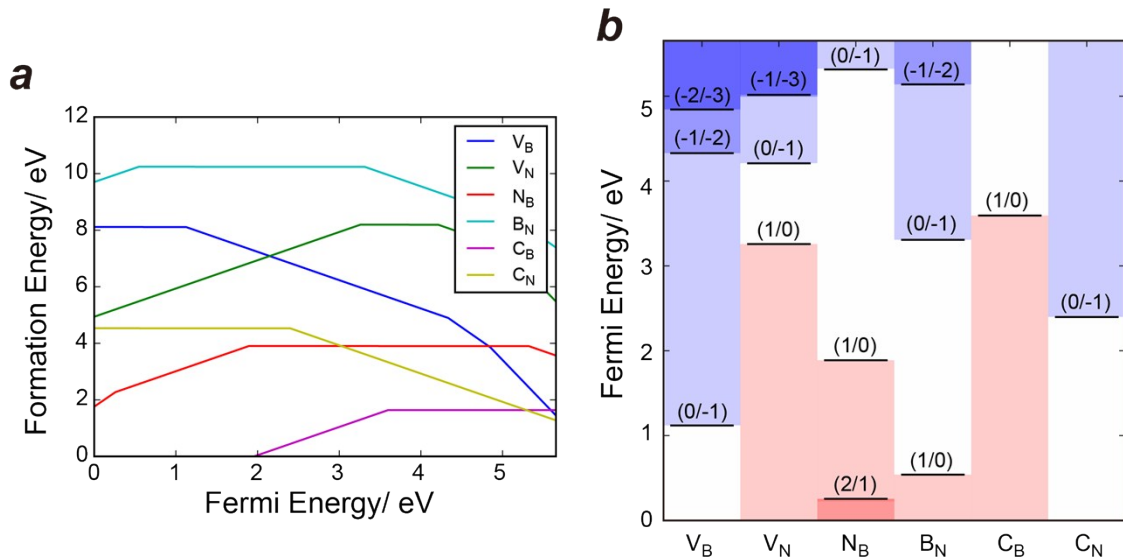
**Fig. S4** Photoelectron yield spectra measured for dried BN-NS powder fixed on carbon tape.

## Ab initio calculations of defect and impurity states for BN

Calculations of the total energy were conducted using density functional theory (DFT) with plane waves as a basis set and the projector-augmented wave scheme as implemented in the VASP code. The Heyd-Scuseria-Ernzerhof (HSE) hybrid functional was used to accurately calculate the bandgap and the defect formation energies, while the generalized gradient approximation (GGA) usually underestimates the bandgaps of semiconductors. The calculated bandgap of 5.65 eV for h-BN determined using the HSE functional is larger than that with the PBE functional by 1.0 eV, which is consistent with the previous calculation.<sup>1</sup>

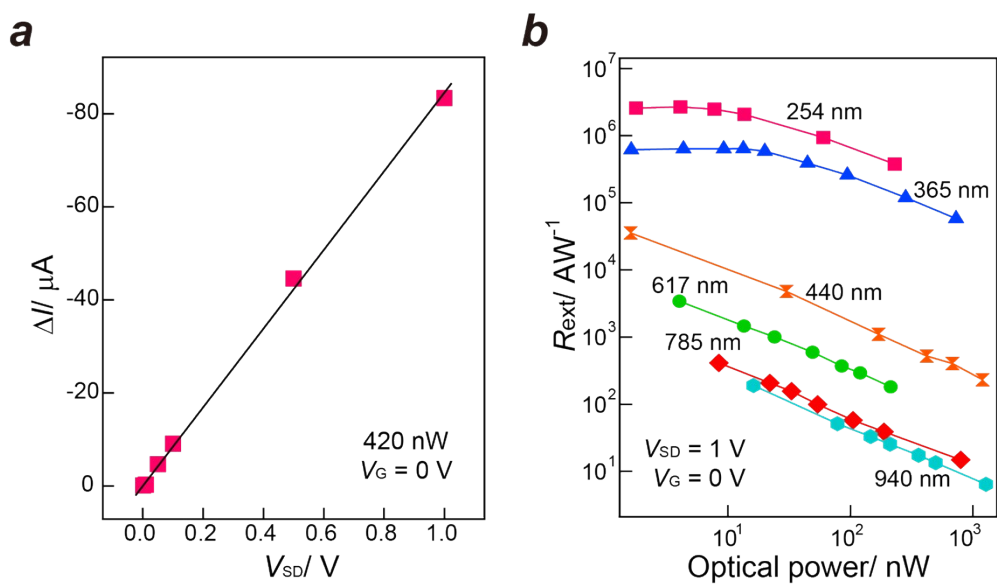
Defect formation energies were calculated using the  $7 \times 7 \times 1$  supercell with 98 atoms separated by a vacuum layer larger than 15 Å to eliminate the interaction between the periodic images. The HSE functional has a large computational cost; therefore, the  $\Gamma$  point was used for Brillouin-zone sampling in the supercell calculations. The errors of k-point sampling were estimated using the PBE functional to be 0.01 and 0.38 eV for neutral and +1 charged carbon impurities at B sites ( $C_B$ ), respectively.

The defect formation energies  $\Delta E^{D,q}(E_F)$ , in equilibrium with a  $N_2$  gas phase (N-rich condition) are shown in Fig. S5a. The slopes and kinks of  $\Delta E^{D,q}(E_F)$  correspond to the defect charge states and the charge transition levels  $\alpha(q/q')$ , respectively, as summarized in Fig. S5b. The present results are in agreement with the previous calculations for vacancies and carbon substitution.<sup>S1</sup> The results indicate  $C_B$  and nitrogen vacancies ( $V_N$ ) induce the deep levels observed in the CL measurements. The charge transition levels, which are located in the experimentally observed CL spectrum between 3.0 and 4.2 eV are  $\alpha(+1/0)$  of  $C_B$  at 3.6 eV,  $\alpha(+1/0)$  at 3.3 eV, and  $\alpha(0/-1)$  at 4.2 eV for N vacancy.  $\alpha(-2/-3)$  of  $V_B$  at 4.3 eV can be identified in the spectrum because of the larger error for the formation energies of largely charged states.



**Fig. S5** (a) Calculated formation energies for various models with B or N defects, and B, N or C substitutional impurities:  $V_B$ , B defect;  $V_N$ , N defect;  $N_B$ , substitution of N at a B-site;  $B_N$ , substitution of B at an N-site;  $C_B$ , substitution of C at a B-site;  $C_N$ , substitution of C at an N-site. (b) Summarized charge transition levels.

[S1] B. Huang and H. Lee, *Phys. Rev. B*, 2012, **86**, 245406.



**Fig. S6** (a) Source-drain voltage dependency of the photocurrent for the DAN-GQDs/BN-NSs@GFET hybrid photodetector ( $V_G = 0 V$ , 440 nm, 420 nW). (b) Photoresponsivity as a function of the optical power for different wavelengths.

**Table S1.** Summary of the performances of our DAN-GQDs/BN-NSs/graphene hybrid phototransistor and photodetectors of various types, at different wavelengths.

Material	Responsivity (A/W) /Detectivity (Jones)		
	UV (<400)	Visible	Infrared (>700)
<b>This work</b>	<b><math>\sim 2 \times 10^6 / \sim 5 \times 10^{13}</math></b> <b>(254 nm)</b>	<b><math>\sim 2 \times 10^4 / \sim 8 \times 10^{11}</math></b> <b>(440 nm)</b>	<b><math>\sim 3 \times 10^2 / \sim 8 \times 10^9</math></b> <b>(940 nm)</b>
graphene/ZnO QDs <sup>S2</sup>	$\sim 2 \times 10^7 / \sim 5 \times 10^{13}$ (335 nm)	$\sim 1 \times 10^4 / \sim 1 \times 10^{11}$ (500 nm)	-
ZnO NWs <sup>S3</sup>	$\sim 1 \times 10^7 / \sim 8 \times 10^{17}$ (300 nm)	-	-
Organic-inorganic composites <sup>S4</sup>	$\sim 1 \times 10^3 / \sim 3 \times 10^{15}$ (360 nm)	-	-
Doped WSe <sub>2</sub> <sup>S5</sup>	-	$\sim 1 \times 10^4 / \sim 5 \times 10^{10}$ (520 nm)	-
Doped MoS <sub>2</sub> <sup>S5</sup>	-	$\sim 5 \times 10^3 / \sim 3 \times 10^8$ (520 nm)	-
MoS <sub>2</sub> NSs <sup>S6</sup>	-	$\sim 8 \times 10^2 / \sim 1 \times 10^7$ (561 nm)	-
Mo:ReSe <sub>2</sub> NSs <sup>S7</sup>	-	$\sim 5 \times 10^1 /$ (633 nm)	-
InSe NSs <sup>S8</sup>	-	$\sim 1 \times 10^1 / \sim 1 \times 10^{11}$ (450 nm)	$\sim 1 \times 10^0 /$ (785 nm)
graphene/PbS QDs <sup>S9</sup>	-	$\sim 5 \times 10^7 / \sim 7 \times 10^{13}$ (532 nm)	$\sim 5 \times 10^5 /$ (1,000 nm)
MoS <sub>2</sub> /BP <sup>S10</sup>	-	$\sim 2 \times 10^2 / \sim 3 \times 10^{11}$ (532 nm)	$\sim 0.1 / \sim 2 \times 10^9$ (1,550 nm)
Perovskite/Polymer <sup>S11</sup>	$\sim 0.01 / \sim 6 \times 10^9$ (365 nm)	$\sim 0.05 / \sim 1 \times 10^{10}$ (650 nm)	$\sim 0.02 / \sim 3 \times 10^9$ (835 nm)
Si photodiode	$\sim 0.15 / \sim 9 \times 10^{11}$ (300 nm)	$\sim 0.3 / \sim 1 \times 10^{12}$ (600 nm)	$\sim 0.3 / \sim 2 \times 10^{12}$ (900 nm)

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**Movie S1.** The operation of the infrared photoreflector.