

Supplementary materials

S1 - SAXS analysis

The Porod invariant is defined in the Equation S1.

$$Q = \int_0^{\infty} q^2 I(q) dq = 2\pi^2 V_0 (\Delta\rho^2) \quad \text{Equation S1}$$

Where, $(\Delta\rho^2) = (\rho_1 - \rho_2)^2 \phi_1 \phi_2$ is the contrast in binary system⁵³. The ρ_1 and ρ_2 correspond to the density of both phase 1 and 2, respectively, while ϕ_1 and ϕ_2 represent the volume ratio of each phase. In other words, the beam scattering happened at the interface of the different phases. In a typical ex-situ experiment, the vacancy of porous material was filled with air⁶¹. So the ρ_2 is the density of air while ϕ_2 is the porosity of material.

The contrast $(\Delta\rho^2)$ was therefore modified in order to satisfy ternary phase system (Equation S2).

$$(\Delta\rho^2) = (\rho_1 - \rho_2)^2 \phi_1 \phi_2 + (\rho_1 - \rho_3)^2 \phi_1 \phi_3 + (\rho_3 - \rho_2)^2 \phi_3 \phi_2 \quad \text{Equation S2}$$

Where the ρ_1 and ϕ_1 represent the density and volume ratio of pristine brass phase respectively; while the ρ_2 and ϕ_2 represent those of the vacancy phase which is filled by solution and the ρ_3 and ϕ_3 correspond to the ligament phase. A number of assumptions were introduced to simplify the data analysis.

The Kratky plot was calculated from Equation S3 and S4.

$$Q = \int_0^{0.2} q^2 I(q) dq \quad \text{Equation S3}$$

Then,

$$Q(t) = \int_0^{0.2} q^2 I(q,t) dq \quad \text{Equation S4}$$

In Equation S5, t corresponds to the duration. At $t = 0$, then $I(q,t) = I_0(q)$, which yields the background invariant Q_0 . The porous layer invariant $Q_p(t)$ is therefore calculated by the difference between $Q(t)$ and Q_0 .

$$Q_p(t) = Q(T) - Q_0 = \int_0^{0.2} q^2 [I(q,t) - I_0(q)] dq \quad \text{Equation S5}$$

Since only the porous part will scatter in the given q range observed the difference value between $I(q,t) - I_0(q)$ is theoretically null, except in the peak area. In other words, only the area under the knee is relevant and may be integrated, to further simplify the system, from 0 to $0.2 \text{ \AA}^{-1}$ (Equation S6).

$$Q_p(t) = \int_{\text{Peak area}} q^2 [I(q,t) - I_0(q)] dq \quad \text{Equation S6}$$

The Q invariant was determined from the Kratky plot data. Specifically, $I''(q,t)$ was defined as, $I''(q,t) = (I'(q,t) + I'b(t)) * I_0(q)$. Since the value interval of original $I_0(q)$ is ranging from $0.003 \text{ \AA}^{-1}$ to $0.2 \text{ \AA}^{-1}$, the $I_0'(q)$ was obtained by extrapolating $I_0(q)$ with the Boltzmann equation, which yielded Equation S7.

$$Q_p(t) = \int_{\text{Peak start}}^{\text{Peak end}} q^2 [I''(q,t) - I_0'(q)] dq \quad \text{Equation S7}$$

The $Q_p(t)$ is regarded as the scattering energy on the de-alloyed part of the sample if assuming the background is constant (assumption still valid in this case, largely within experimental error). Therefore, the scattering volume of the de-alloyed part was obtained by inputting the $Q_p(t)$ into Equation 4.1. Regarding the dense part of the sample, or the non-dealloyed part, which is still part of the background signal, the contrast $(\Delta\rho)^2$ can be simplified into a simple binary system, which is $(\Delta\rho^2) = (\rho_1 - \rho_2)^2 \phi_1 \phi_2$.

In the Porod's law, the intensity $I(q)$ is proportional to the scattering vector q in high q , while the relative surface area S is the coefficient (Equation S8).

$$I(q) \sim S q^{-4} \quad \text{Equation S8}$$

The interval of validity of the Porod law should in theory be smaller than $0.1 \text{ \AA}^{-1}$ since the scattering features in this range would be attributed to interfaces rather than atoms or bonds. Here, the Porod's law was used to reveal the surface area of the ligaments. Therefore, the scattering vector interval of interest was taken for the Porod law outside the range of the main knee area, and close to $0.1 \text{ \AA}^{-1}$.

The Guinier analysis allows for the direct estimation of two SAXS invariants, the radius-of-gyration, R_g , and the extrapolated intensity at zero scattering angles, $I(0)$ (Equation S9).

$$\ln I(q) = \ln I(0) - \frac{R_g^2}{3} \cdot q^2 \quad \text{Equation S9}$$

Where the R_g in Radius of Gyration, the q is scattering vector and I is the scattering intensity which is the function of q . The $q \cdot R_g$ should typically be smaller than 1.3 according to the

most common definition of R_g ^{52, 62}. Since $I(0)$ is constant, the can be simplified and converted into Equation S10.

$$\ln I(q) \sim -\frac{R_g^2}{3} \cdot q^2 \quad \text{Equation S10}$$

The probed volume V_0 of the sample corresponds to the beam size multiplied by the thickness of the sample. So, the scattering volume $V_s = V_0 \cdot \varphi_1(t)$. Therefore, as shown in Equation S11:

$$SQ \sim V_s(t) \varphi_2(t) \lim_{q \rightarrow \infty} (q^4 I(q)) \quad \text{Equation S11}$$

Where,

$$\lim_{q \rightarrow \infty} (q^4 I(q)) \rightarrow \text{constant}$$

Then,

$$AQ \sim V_s(t) [1 - \varphi_1(t)] \quad \text{Equation S12}$$

The plot $A \times Q$ against the process duration, is provided in Figure 6.

During the Au-Ag in-situ dealloying tests, the solid phase ratio increased with process progressing but the increase rate was negatively correlated to the solution temperature. The dissolution rate of atoms is proportional to the solution temperature.

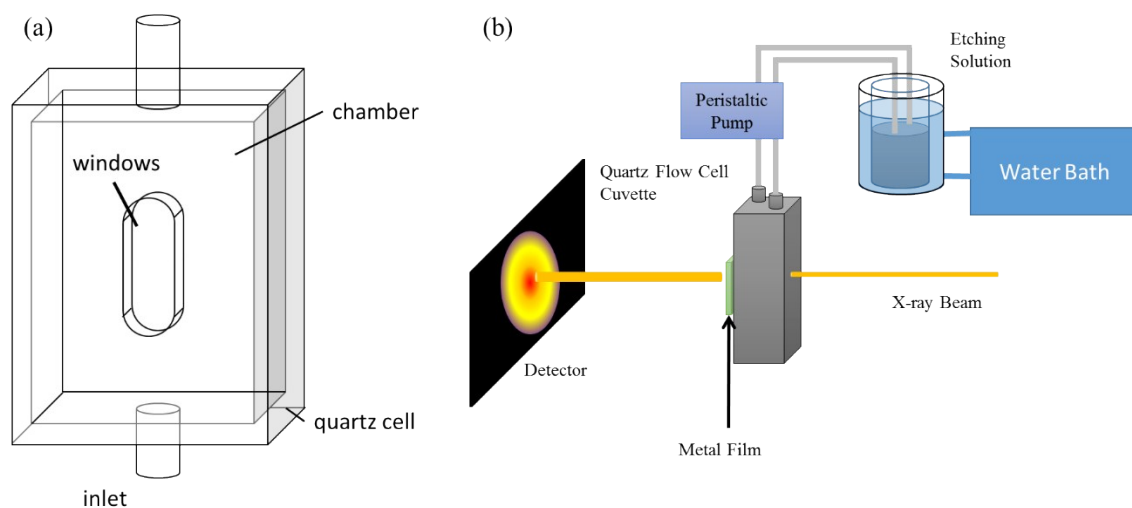


Figure S1. (a) Schematic of the flow cell and (b) schematic of the in-situ chemical DA experiment on the SAXS beam line

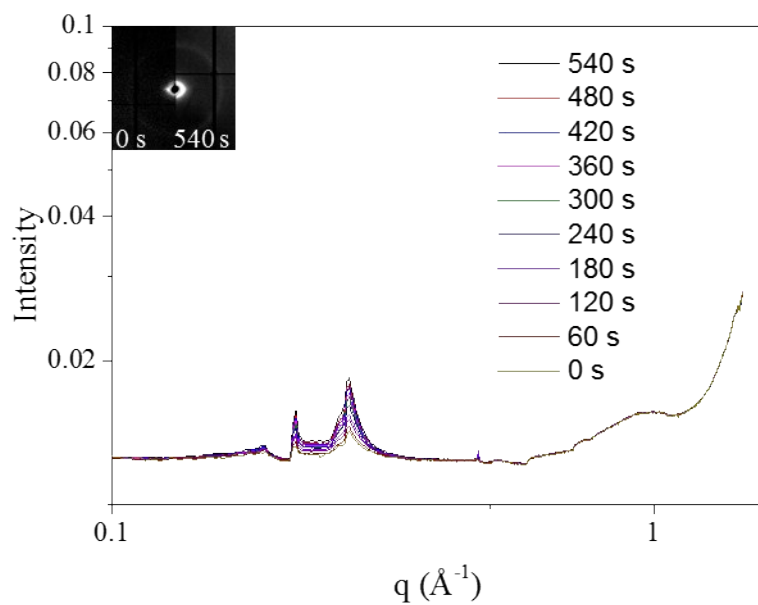


Figure S2. The SAXS patterns of in-situ dealloying test on Au-Ag alloy with 0.6 m camera lens. De-alloyed with 10% HNO_3 at 20 $^{\circ}\text{C}$

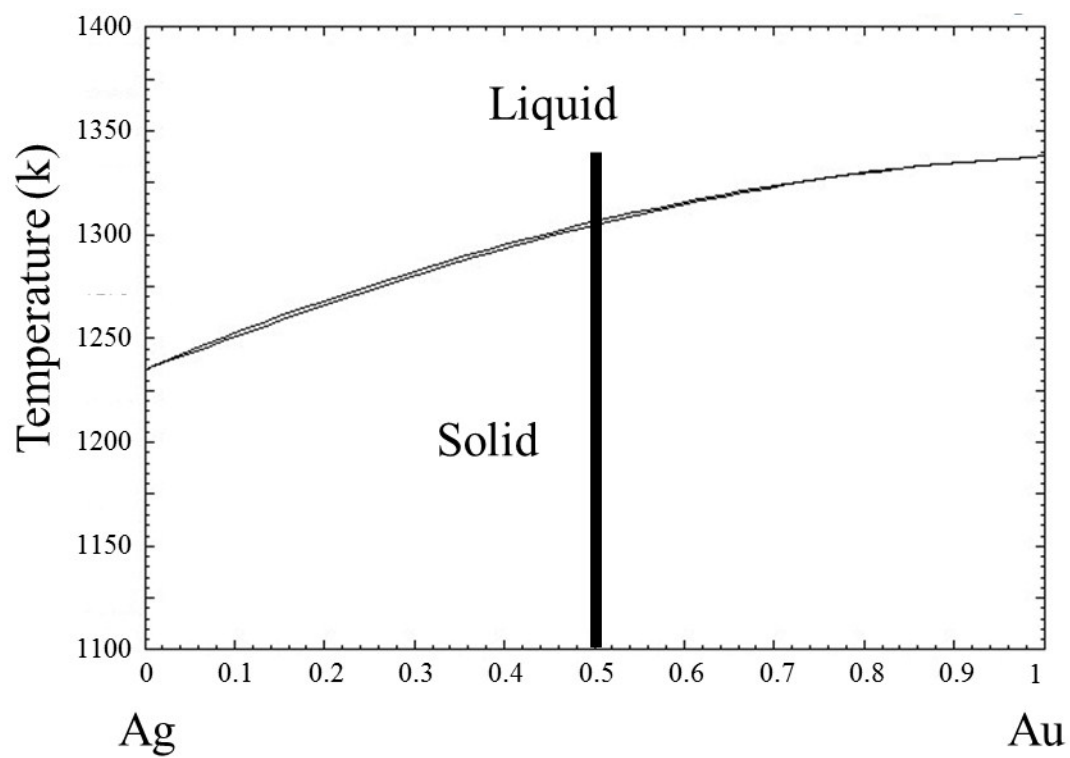


Figure S3. Phase diagram of Au-Ag (re-phase from ⁵⁶). Solid line: AuAg50

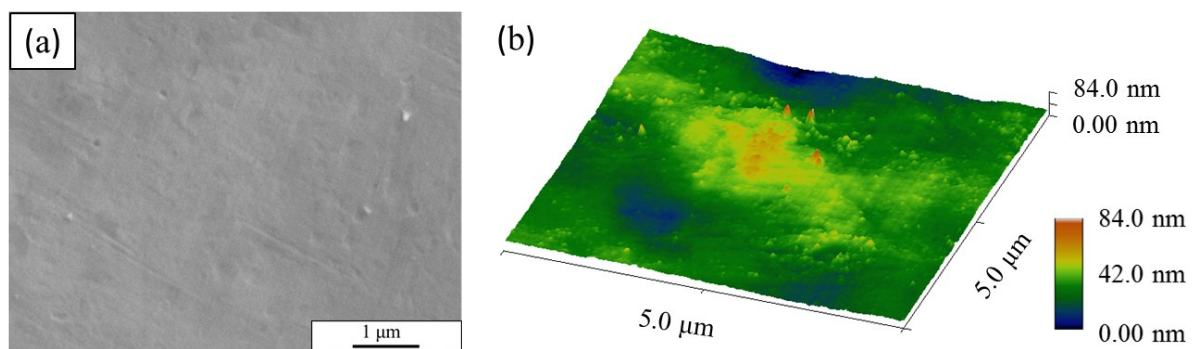


Figure S4. Surface morphology of pristine 120 nm AuAg50 ultra-thin foil. (a) SE IMAGES image; (b) AFM mapping

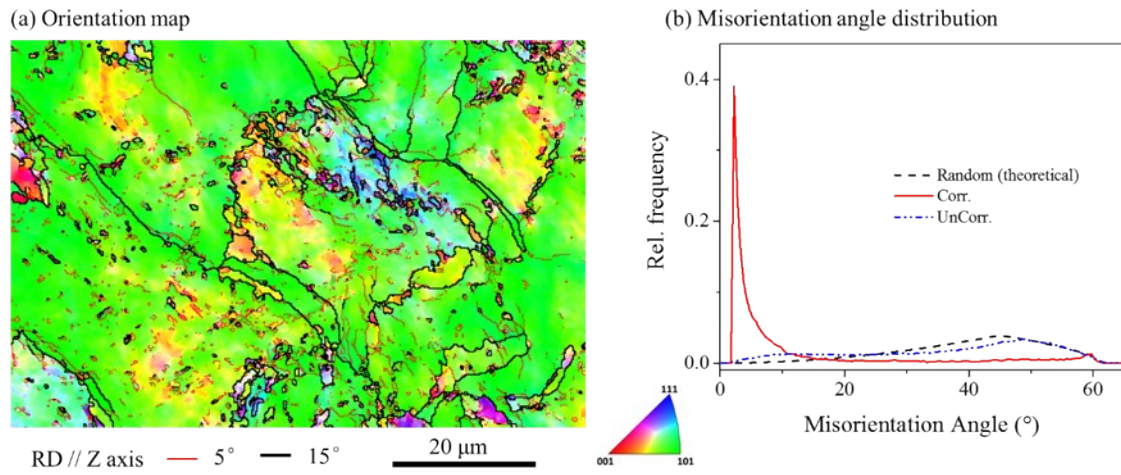
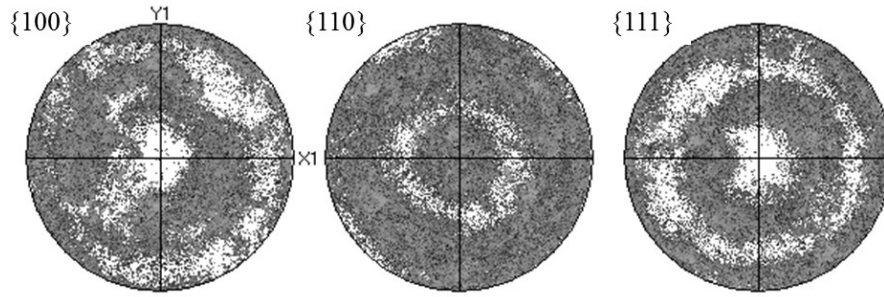
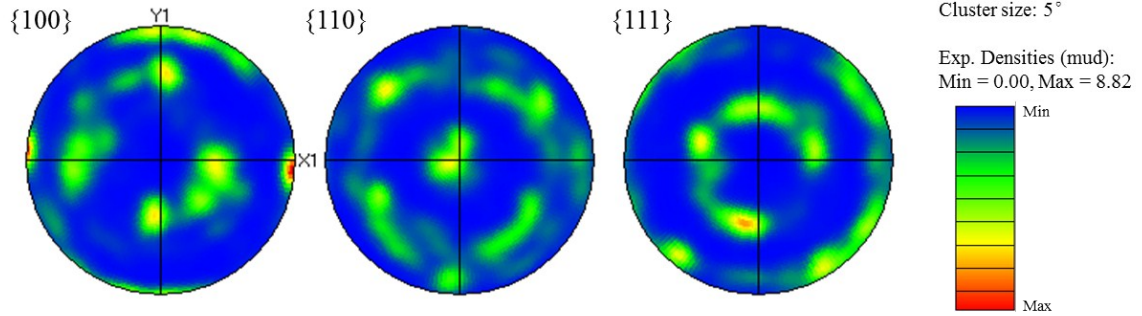


Figure S5. (a) EBSD orientation map of pristine 120 nm AuAg50 ultra-thin foil; (b) Misorientation angle distribution

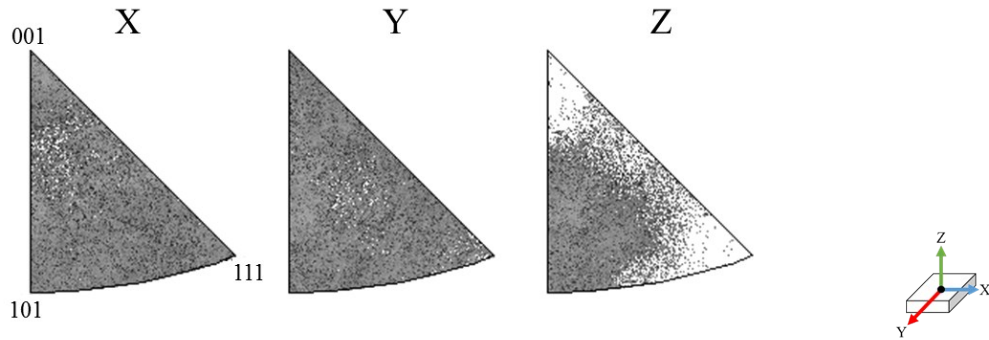
(a) Pole figure



(b) Pole figure – contoured



(c) Inverse pole figure



(d) Inverse pole figure –contoured

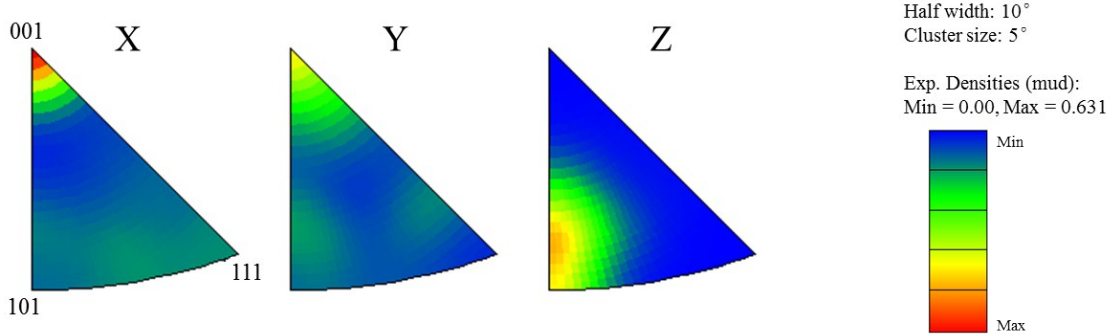
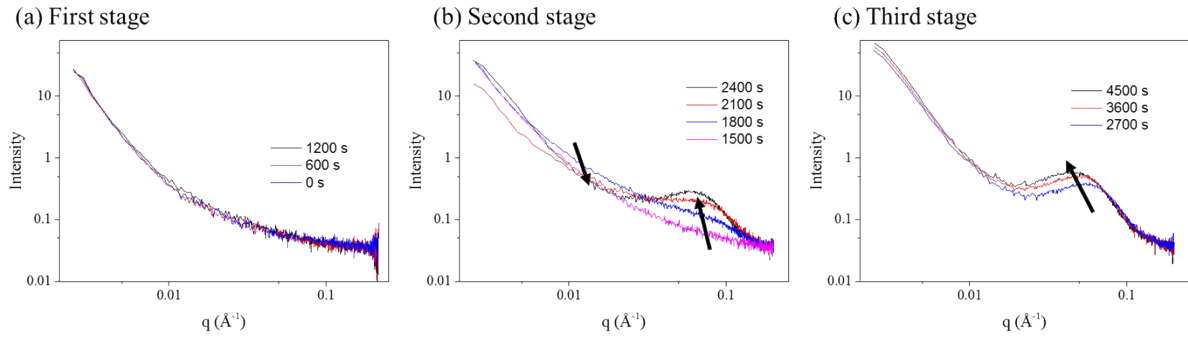


Figure S6. Pole figures and Inverse pole figure of pristine 120 nm AuAg50 ultra-thin foil for fibre texture analysis. (a) PF; (b) contoured PF; (c) IPF; (d) contoured IPF



Temperature	Start	End
5 °C	1500 s	2400 s
20 °C	450 s	900 s
40 °C	30 s	240 s
60 °C	0 s	180 s

Figure S7. Example of three stages of in-situ test shown in SAXS pattern. The sample SAXS pattern is the 5 °C de-alloyed AuAg50 leaf, de-alloyed with 10% HNO₃. The arrow displayed on (b) and (c) shown to the trend of intensity change with process progressing. The table shows the boundaries of transformation period for each sample.

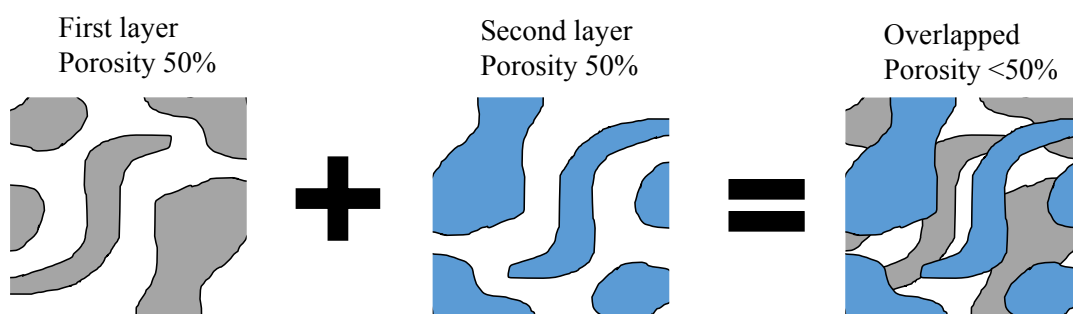


Figure S8. Schematic of the different overlapping dealloyed layers constituting the material. This aspect of the structure is rationalized to explain the difference in surface porosity calculated from the SE IMAGES analysis

Table S1. Analysed results of de-alloyed Au-Ag for both image analysis and Irena size distribution of pores

		ImageJ analysing		Irena modelling
	R_g (nm)	average radius (nm)	average diameter (nm)	mean diameter (nm)
5 °C	10.5	15.4	30.7	30.96
20 °C	15.8	17.1	34.2	34.56
40 °C	17.3	21.7	43.4	47.82
60 °C	24	27.3	54.5	54.90

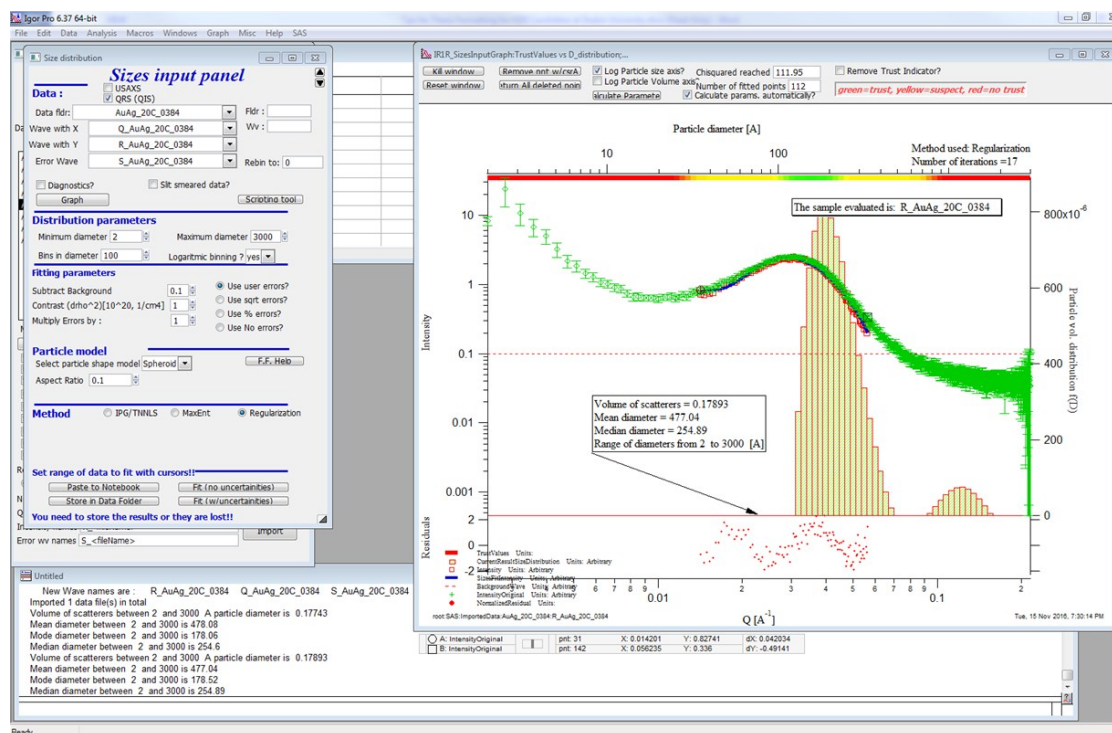


Figure S9. Screen-shot of modelling with Irena