

## Supporting Information

### **Contact electrification Induced Interfacial Reactions and the Electrochemical Nanoimprint Lithography Directly in n-Type Gallium Arsenate Wafer**

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## 1 Experimental section

### *Chemicals and materials*

All chemicals used in the experiments ( $\text{KMnO}_4$ ,  $\text{H}_2\text{SO}_4$ ) are analytical grade or better and provided by Sinopharm Co., China. Silicon doped (n-type) GaAs (100) wafers with doping level between  $(0.8\text{--}2.3) \times 10^{18} \text{ cm}^{-3}$  were purchased from China Crystal Technologies Co., China. Before experiments, n-GaAs was rinsed with acetone and deionized water. All aqueous solutions were prepared with deionized water ( $18.2 \text{ M}\Omega\cdot\text{cm}$ , Milli-Q, Millipore Co.).

### *Instruments and procedures*

To investigate the change of interfacial potentials during the isolated and contact status in ECNL processes, a Pt mold electrode (area:  $1 \text{ cm}^2$ ) shown in Figure 3a was controlled to bring into contact with the n-GaAs fixed at the bottom of the electrolytic cell by a micromanipulation system (SECM workstation, CHI 920c, CHI Co. USA). The interfacial potential of both the mold electrode and GaAs in the working solution containing  $0.040 \text{ mol/L KMnO}_4$ ,  $1.84 \text{ M H}_2\text{SO}_4$  at constant temperature ( $35\text{--}37^\circ\text{C}$ ) was monitored by the potentiostatic of CHI 920c and another potentiostat (CHI 760d), respectively. The LSV of  $\text{KMnO}_4$  at Pt electrode was studied by using a Pt disk electrode (diameter:  $2 \text{ mm}$ ) inserted in the same solution. Similarly, The LSV of n-GaAs in  $\text{KMnO}_4$  solution was studied by using an n-GaAs electrode (area:  $0.6 \text{ cm}^2$ ) inserted in the same solution. The Tafel experiments in the same solution were performed by the use of  $2\text{-mm-diameter Pt}$  disk electrode and  $0.6\text{-cm}^2\text{-area GaAs}$  electrode. The LSV and Tafel experiments were performed with an electrochemical workshop (CHI 920c, CHI Co. USA).

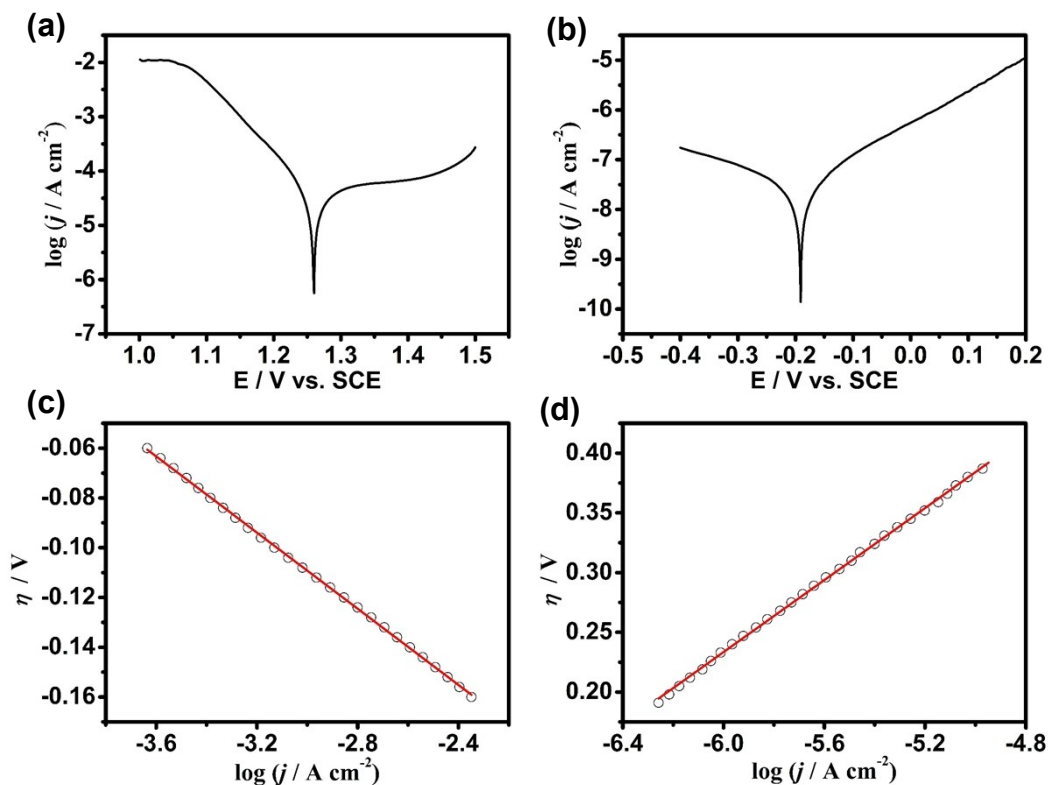
All the PMMA mold used in the experiments were prepared by nanoimprint lithography, which were then coated with titanium (thickness:  $10 \text{ nm}$ ) and platinum (thickness:  $100 \text{ nm}$ ) by magnetron sputtering method. The contact status is very

important for the electrochemical nanoimprint experiments. If the Pt mold electrode doesn't contact well with the GaAs substrate, it is difficult to realize the contact electrification. The MacEtch process will never occur. Our experimental condition was optimized with the contact pressure at 0.5 atm. The alignment is also crucial for the nanoimprint experiments. The key issue is to make the pressure distribution homogeneous on the GaAs substrate. For this, a gas-pressure-driven system (Obducat Eitre-6 Nanoimprint lithography system, Obducat Technologies AB, Sweden) was employed to perform the electrochemical nanoimprint experiments. The self-adapting alignment function will let the Pt mold electrode and the GaAs substrate align symmetrically with each other even if they are a little bit deviated from the horizon level. Actually, the workbench is always aligned to the horizon level before experiments. To perform the ECNL, a special electrolytic cell is adopted to hold the workpiece and the working electrolyte. A force sensor was used to monitor the contact force between mold electrode and workpiece. During the ECNL process, the contact pressure was kept constant at 0.5 atm. When the ECNL process was finished, the workpiece was rinsed with deionized water and dried under nitrogen for the further characterization.

### ***Characterization***

Confocal laser microscope (VK-X200, KEYENCE Co.), scanning electron microscope (Hitachi S-4800, Hitachi High-Technologies Co.), atomic force microscopy (Nanoscope III, Digital Instrument Co.) were employed to characterize the morphology of the fabricated micro- and nano-structures in the semiconductors. X-ray photoelectron spectroscopy (XPS) was carried out on an Omicron Sphera II hemispherical electron energy analyzer (Monochromatic Al K $\alpha$  with 1486.6 eV operating at 15 kV and 300 W). The base pressure of the systems was  $5.0 \times 10^{-9}$  mbar.

**2 Tafel curves of n-GaAs and Pt electrode in the working solution** (Figures are the same as in the text, more information is given here.)



**Figure S1.** (a-b) the Tafel curves of n-GaAs electrode and 2-mm-diameter Pt electrode in the working solution containing 0.040 mol/L  $\text{KMnO}_4$  and 1.84 mol/L  $\text{H}_2\text{SO}_4$  at 35~37 °C. (c-d) the semi-logarithmic relationship between the potential and the current density.

Tafel experiments were performed to figure out the kinetic rate of  $\text{MnO}_4^-$  reduction at the Pt/electrolyte interface. From the semi-logarithmic relationship between the overpotential and the current density, exchange current density ( $j_0$ ) and the product of electron transfer coefficient and electron transfer number ( $\alpha n$ ) can be obtained by Tafel equation:<sup>1</sup>

$$\eta = -\frac{2.303RT}{\alpha nF} \lg j + \frac{2.303RT}{\alpha nF} \lg j_0$$

For  $\text{KMnO}_4$  reduction on Pt electrode,  $j_0 = 3.74 \times 10^{-5} \text{ A/cm}^2$ .

For the anodic dissolution of n-GaAs at the n-GaAs/electrolyte interface, the anodic current density are given in a semi-logarithmic plot versus overpotential, which should abide by:<sup>2</sup>

$$j = j_0 \left[ \exp\left(\frac{nF\eta}{RT}\right) - 1 \right]$$

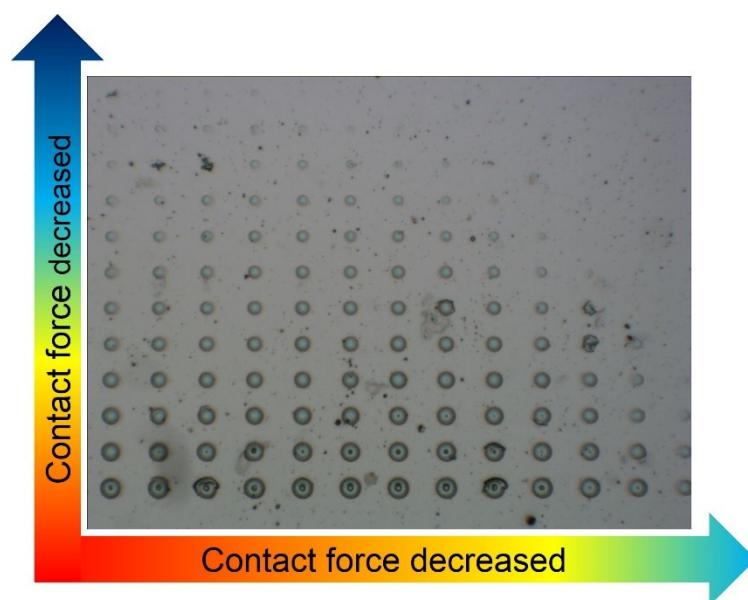
For anodic polarization ( $\exp\left(\frac{nF\eta}{RT}\right) \gg 1$ ), the equation is simplified to:

$$\eta = \frac{2.303RT}{nF} \lg j - \frac{2.303RT}{nF} \lg j_0$$

For n-GaAs anodic dissolution,  $j_0 = 2.82 \times 10^{-8} \text{ A/cm}^2$ . Because the electron transfer for semiconductor can occur only at the conductive band minimum or valence band maximum, the exchange current density of semiconductor electrode is much less than that of the metallic electrode.<sup>2</sup> Since the exchange current density of  $\text{MnO}_4^-$  reduction at the Pt/electrolyte interface is much higher than that of anodic dissolution of n-GaAs at the n-GaAs/electrolyte interface, the rate determining step should be the anodic dissolution of GaAs.

### 3 Optimizing the contact pressure of ECNL

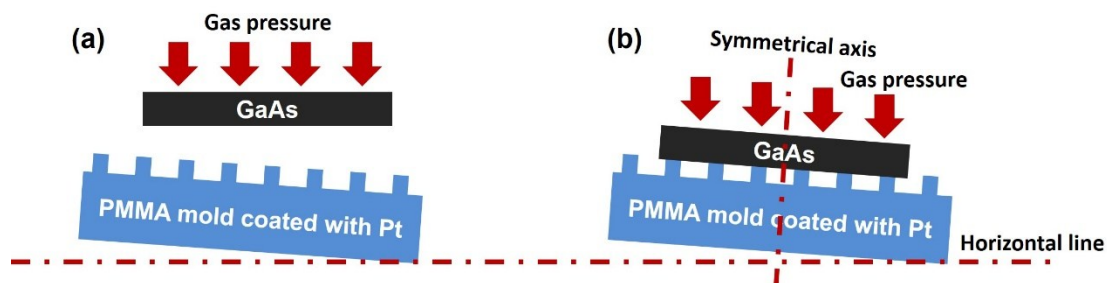
The alignment is also crucial for the nanoimprint experiments. The key issue is to make the pressure distribution homogeneous on the GaAs substrate. For the traditional MatEtch, the metal film is patterned directly on the surface of semiconductors, which ensures a good contact between the metal film and semiconductor. For the situation of ECNL herein, the contact between the mold and semiconductor is realized by the external pressure. A non-uniform contact will lead to a non-uniform etching rate, which results in non-uniform micro-nanostructures fabricated (Figure S2).



**Figure S2.** the non-uniform concave hemispheres fabricated under a with bad alignment. The contact pressure is 0.5 atm.

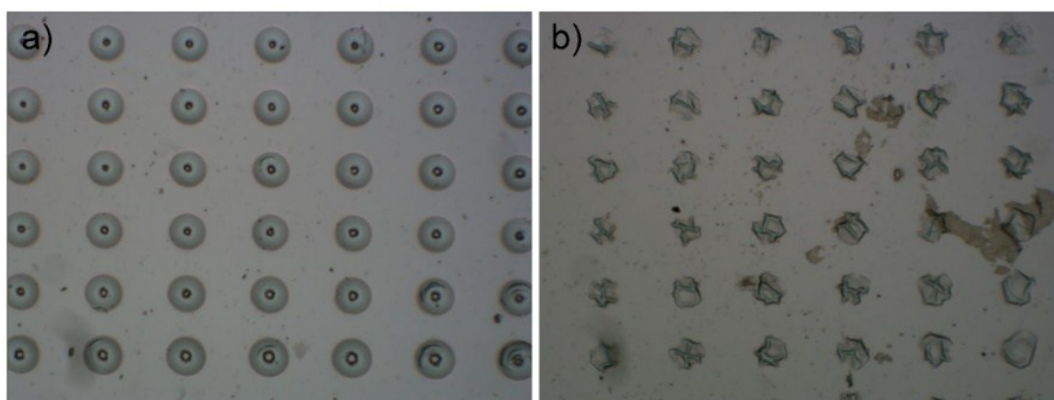
To ensure a uniform contact between mold electrode and GaAs, a gas pressure driven contact method in the nanoimprint equipment (Eitre-6, Obducat Technologies AB, Sweden) was used in the ECNL process. Before the ECNL, the mold was fixed at the bottom of the electrolytic cell, and was leveled by the bubble level. The leveling method by bubble level has some error (Figure S3). But this error has no effect on the uniformity of the contact performed by the gas pressure subsequently. The self-

adapting leveling between mold and substrate was achieved by the gas pressure driven attitude adjustment of GaAs workpiece (Figure S3). Therefore, the mold was well aligned to GaAs to ensure a uniform contact.



**Figure S3.** the schematic of self-adapting leveling between mold and substrate.

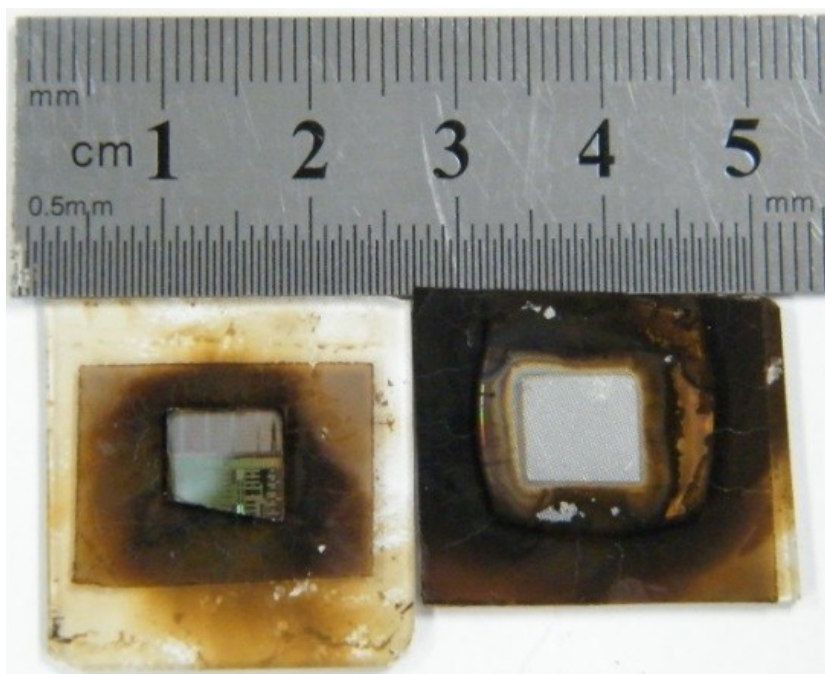
Usually, large contact pressure ( $> 1$  atm) will lead to the deformation and wrinkles on the imprint mold, which results in the failure of microfabrications (Figure S4). Different contact pressure are tried and the optimized contact pressure is 0.5 atm.



**Figure S4.** The optical images of concave hemispheres fabricated in n-GaAs with humps and wrinkles fabricated under a contact pressure of 1 atm (a) and 2 atm (b).

#### 4 Optimizing the working temperature of ECNL

It should be noted that the decomposition rate of  $\text{KMnO}_4$  is affected by the acidity and temperature of the working electrolyte. High concentration and high temperature should be avoided. When the temperature is higher than  $40\text{ }^\circ\text{C}$ , oxygen bubbles and  $\text{MnO}_2$  precipitant were observed obviously in the solution containing  $100\text{ mM}$   $\text{KMnO}_4$ . To ensure the fabrication efficiency and quality, the working temperature should be kept no higher than  $37\text{ }^\circ\text{C}$  in the experiments.

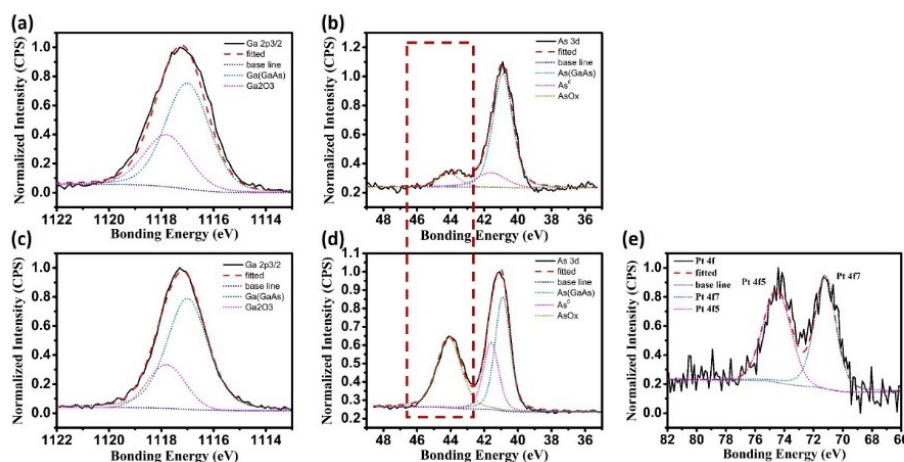


**Figure S5.** The precipitates of  $\text{MnO}_2$  observed on the surface of mold after 40-min etching in the presence of  $100\text{ mM}$   $\text{KMnO}_4$ . The temperature was kept at  $40\text{ }^\circ\text{C}$ .



## 5 XPS analysis of the n-GaAs before and after ECNL

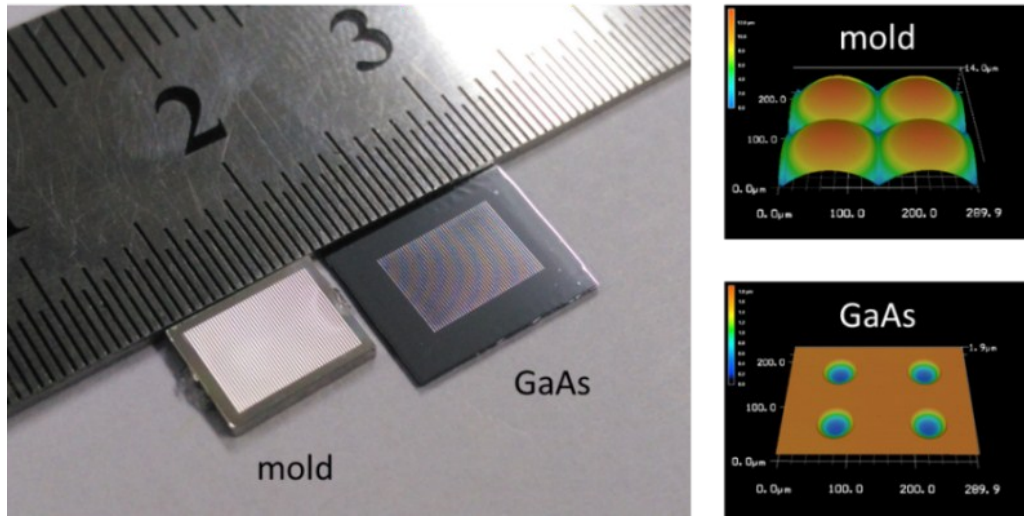
n-GaAs can be oxidized in the air to form a little oxides on surface.<sup>3</sup> The bonding energy at 41.6 eV could be due to the elemental arsenic ( $\text{As}^0$ ). While the bonding energy at 44.1 eV could be due to a lower arsenic oxide oxidation state (arsenic non-stoichiometric oxide,  $\text{AsO}_x$ ).<sup>4-7</sup> After ECNL experiments, the elemental arsenic and lower arsenic oxide oxidation state increased dramatically (Figure S6d). The lower arsenic oxide oxidation state was always generated at the earlier oxidation of pure arsenic.<sup>4, 7</sup> The oxides  $\text{As}_2\text{O}_3$  and  $\text{As}_2\text{O}_5$  with bonding energy at 44.8 eV and 46.2 eV was not found on the n-GaAs surface. Generally, arsenic oxides are more insoluble than gallium oxides.<sup>8, 9</sup> Therefore, the arsenic oxides were mainly formed on the surface.



**Figure S6.** The X-ray photoelectron spectra of n-GaAs before (a-b) and after (c-e) the ECNL experiments. (a) and (c) XPS spectra of Ga 2p<sub>3/2</sub>. (b) and (d) XPS spectra of As 3d. (e) XPS spectra of Pt 4f.

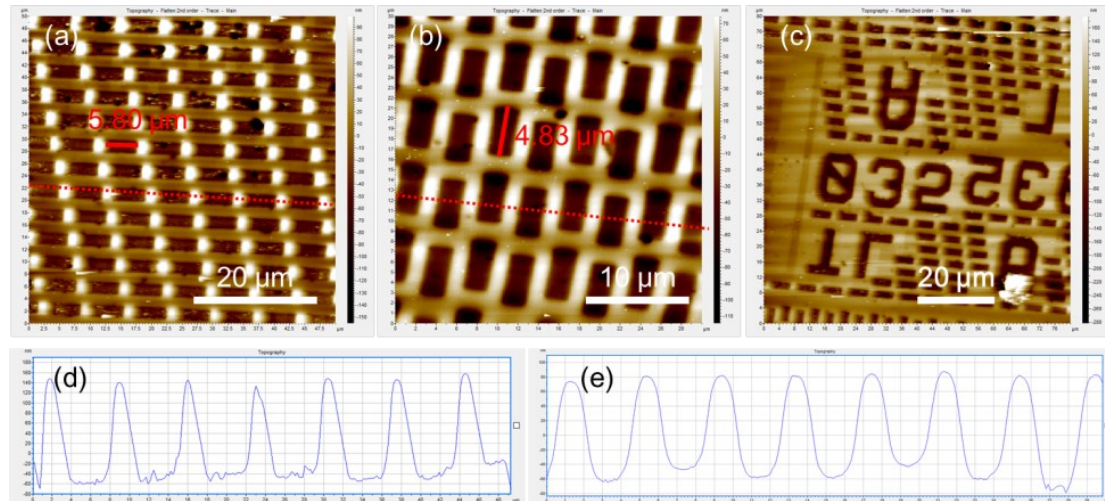
For traditional MatEtch, the patterned metal on semiconductor surface will sink into the structures and should be removed by suitable etchant for the further device applications.<sup>10</sup> It has been well known that Pt is a better catalyst than Au and Ag in MatEtch. However, Pt is too stable to be removed than Au and Ag.<sup>10, 11</sup> That is why Pt is not so popular as Au. Nevertheless, in the ECNL processes, from the experimental results of the X-ray photoelectron spectroscopy (XPS) as shown in Figure S6e, little Pt is residual on the n-GaAs wafer. The average atomic concentration of platinum is almost 0.29 %. This is a distinct advantage of ECNL.

## 6 Large area duplication by ECNL



**Figure S7.** The left shows the large-area ( $\sim\text{cm}^2$ ) mold with convex microlens array and the fabricated concave microlens array on n-GaAs substrate. The rights show the confocal laser microscopic images of the convex microlens on the mold and the concave microlens fabricated on the GaAs substrate.

## 7 More examples of ECNL on n-GaAs



**Figure S8.** (a-c) Different microstructures fabricated by ECNL on n-GaAs, (d) and (e) the profiles corresponding to the red lines in (a) and (b).

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