

Supporting Information

Ca²⁺ and Ga³⁺ doped LaMnO₃ perovskite as highly efficient and stable catalyst for two-step thermochemical water splitting

Lulu Wang,^a Mohammad Al-Mamun,^a Yu Lin Zhong,^a Lixue Jiang,^a Porun Liu,^a Yun Wang,^a

Hua Gui Yang^b and Huijun Zhao^{*ac}

^aCentre for Clean Environment and Energy, Griffith University, Gold Coast Campus,

QLD 4222, Australia

^bKey Laboratory for Ultrafine Materials of Ministry of Education, School of Materials

Science and Engineering, East China University of Science and Technology, 130 Meilong

Road, Shanghai 200237 China

^cCentre for Environmental and Energy Nanomaterials, Institute of Solid State Physics,

Chinese Academy of Sciences, Hefei 230031, P. R. China

E-mail: h.zhao@griffith.edu.au; Fax: +61 7 5552 8067; Tel: +61 7 5552 8261

I Experimental Section

Perovskites synthesis

All the perovskite oxides were synthesised by a modified Pechini method.¹ In brief, stoichiometric amounts of metal nitrate precursors were dissolved in deionized water and heated at 70 °C with continuous stirring. Citric acid was then added to the solution with a molar ratio of 1.5:1 with respect to the metal cations. After 10 min, ethylene glycol (ethylene glycol: citric acid molar ratio = 1:1) was added to the mixture and heated at 90 °C until a brownish gel was obtained. The resulting gel was transferred to an electric oven and kept at 120 °C for 12 h to obtain a resin. The resin was then collected and ground with a mortar and pestle. The final perovskite oxide powder was prepared by calcining the resin at 1300 °C for 6 h to remove any remaining organics or nitrates.

Characterisations

The X-ray diffraction (XRD) patterns of the samples were collected with a Bruker D8 X-ray diffractometer operated at 40 kV with the current of 25 mA using a Cu K α radiation. The measurements of 2 θ symmetrical scans were carried out in the range of 20–80 ° with a step size of 0.02 ° and a scanning rate of 2 °/min, respectively. Phase identification was conducted through comparisons with the standard Inorganic Crystal Structure Database (ICSD). Scanning electron microscopy (SEM, JSM- JSM-7100), energy dispersive X-ray (EDX, JSM-7100) mapping and transmission electron microscope (TEM, Tecnai F20) techniques were also employed to characterise the surface morphology and elemental distribution within the samples.

Thermochemical H₂O splitting test

The two-step thermochemical H₂O splitting activity was investigated using a laboratory-scale fix-bed reactor, as shown in Fig. S4. To facilitate effective testing, the freshly-prepared perovskite samples (~ 0.5 g) were transferred into a solid porous monolith structure by mixing with isopropanol then fired at 1300 °C for 6 h in air.² The prepared redox sample was then packed at the centre of an alumina tubular reactor coupled within a high temperature programmable electric furnace (MTI, KSL-1800X-S60). Ar (ultra-high purity ~ 99.99%) was used as a carrier gas and the flow rate was controlled by a mass flow controller (Alicat Scientific). The thermal reduction step was carried out by heating the perovskite sample to 1300 °C and holding for 1 h under an Ar flow rate of 200 sccm (the oxygen partial pressure was ~20 ppm). The temperature was then allowed to decrease to 900 °C for H₂O splitting. The

temperature increase and decrease rates were fixed at 10 °C/min. The water was introduced into the chamber by a water pump (Easypump, BT100N), then vaporised and mixed with Ar gas (the volume ratio of vapour in the mixture gas was ~ 40%). The whole H₂O splitting process was maintained for 1 h to complete the re-oxidation process. During the thermal reduction and H₂O splitting processes, the O₂ and H₂ produced were detected by mass spectrometer (MS, Omni Star GSD 320). For quantitative measurement of evolved gas, the ion current signal of the MS result was calibrated by standard O₂ and H₂ gases.

The re-oxidation yield (α) of thermochemical H₂O splitting test was calculated as follow,

$$\alpha = (n_{H_2} / 2n_{O_2}) \times 100\%$$

where n_{O_2} and n_{H_2} represent the total amount (mole weight) of O₂ and H₂ evolved during two-step thermochemical H₂O splitting process.

II Supporting Figures

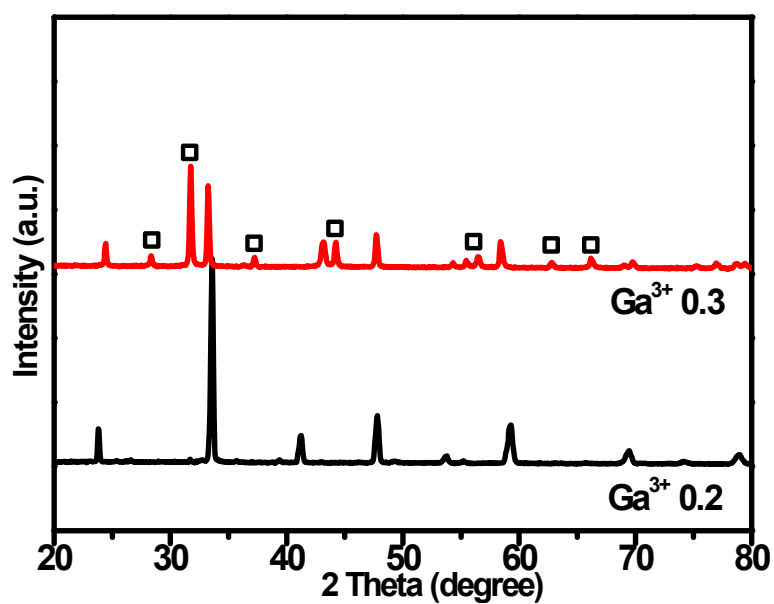


Fig. S1 XRD patterns of LaMnO₃ based perovskites with the Ga³⁺ doping content of 0.2 (black) and red (0.3) on the B site. The square represents the impurity of Ga₂O₃ in the fabricated perovskite sample.

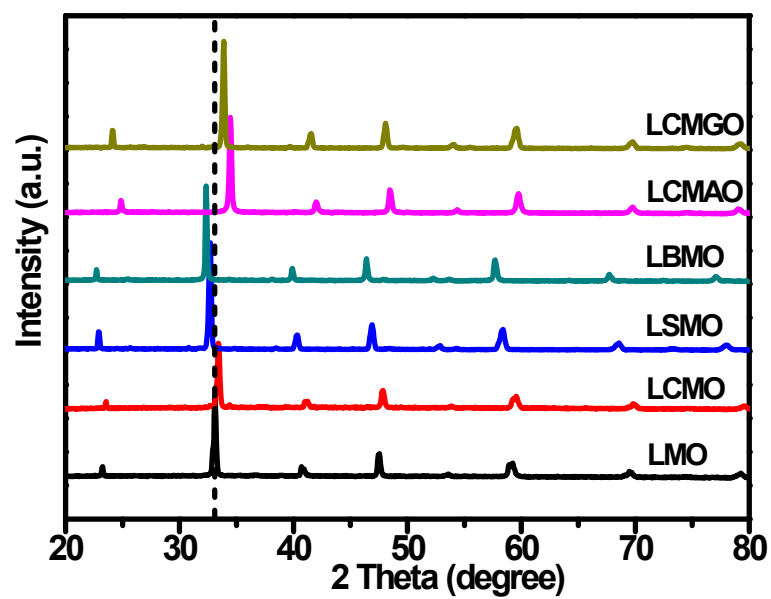


Fig. S2 XRD patterns of parent LaMnO_3 and A site (Ca^{2+} , Sr^{2+} , Ba^{2+}) and B site (Al^{3+} and Ga^{3+}) doped LaMnO_3 perovskites.

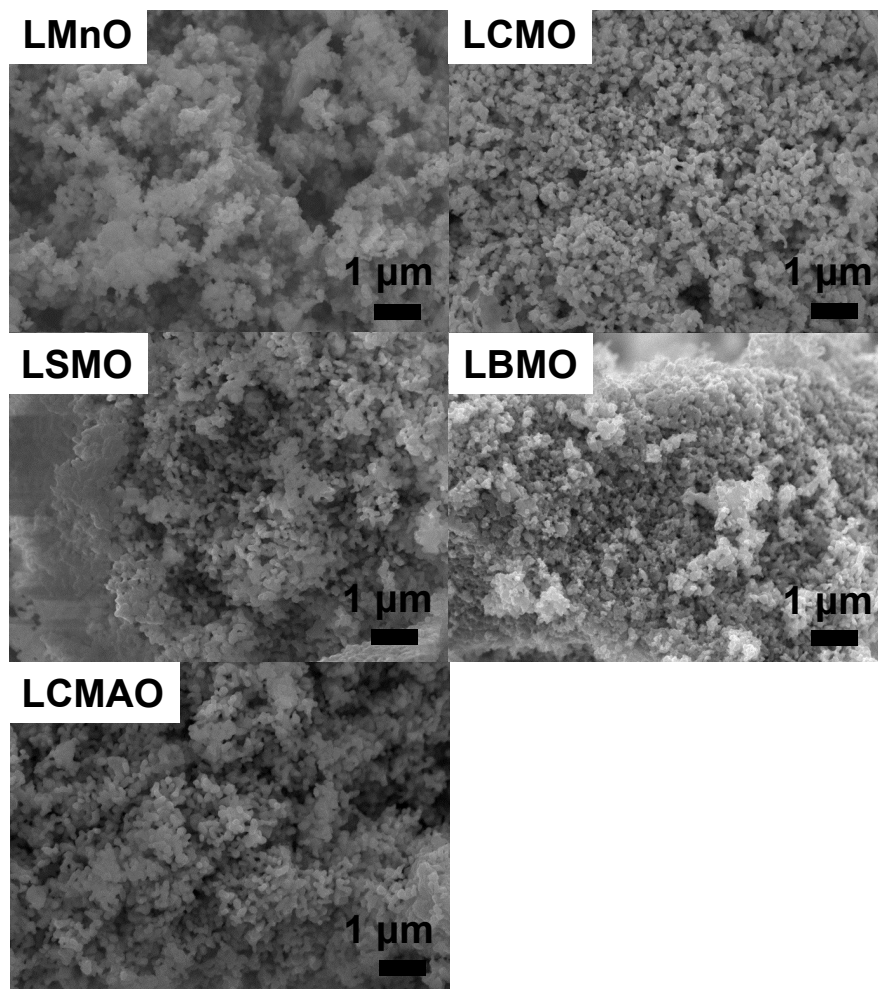


Fig. S3 SEM images of of parent LaMnO_3 and A site (Ca^{2+} , Sr^{2+} , Ba^{2+}) and B site (Al^{3+} and Ga^{3+}) doped LaMnO_3 perovskites..

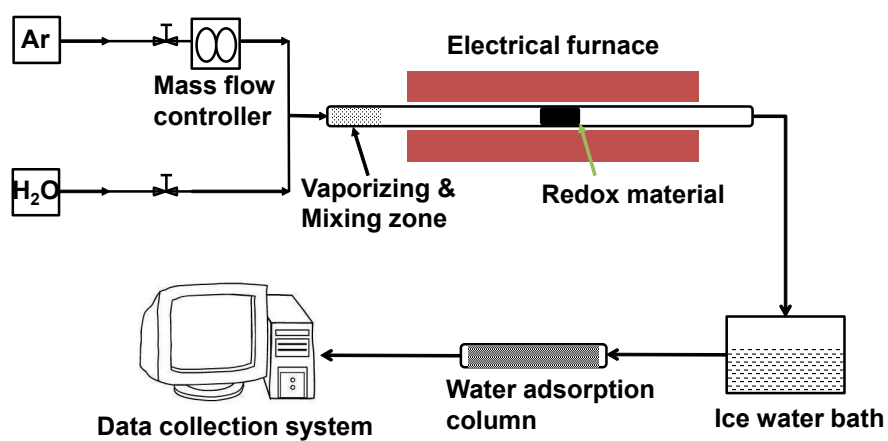


Fig. S4 Two-step thermochemical H₂O splitting set-up for H₂ production.

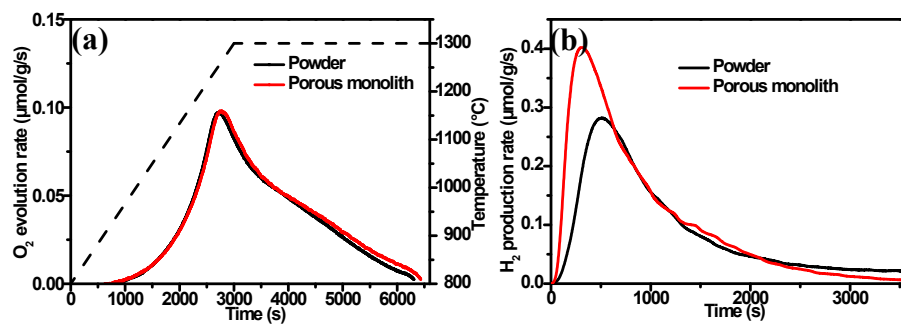


Fig. S5 A comparison of O_2 evolution and H_2 production between perovskite powder and porous monolith during two-step thermochemical H_2O splitting process.

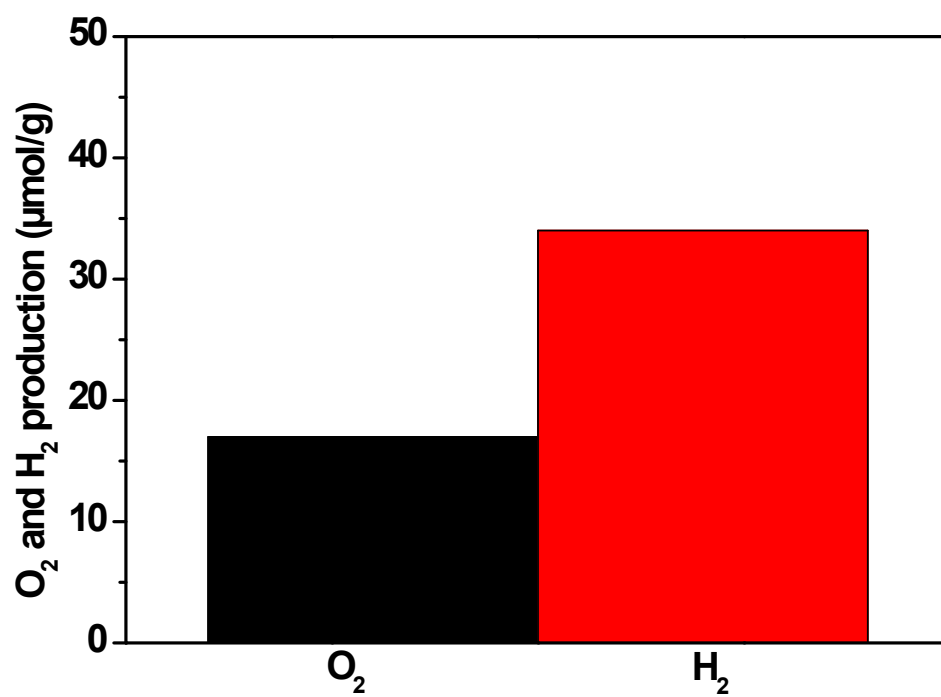


Fig.S6 O₂ and H₂ production results of CeO₂ from thermochemical H₂O splitting operated between 1300 and 900 °C.

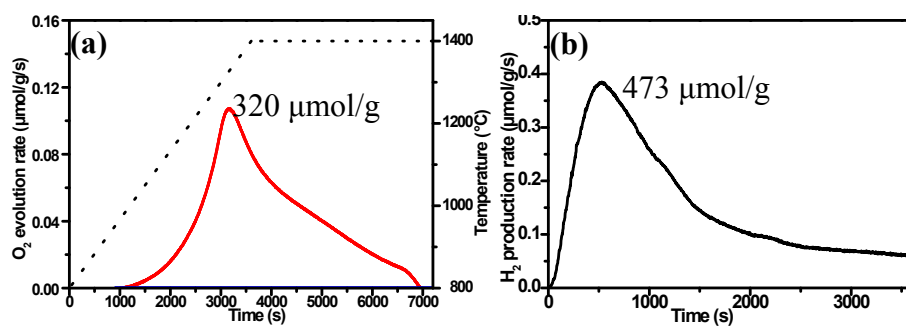


Fig. S7 (a) O_2 evolution and (b) H_2 production curves of $\text{La}_{0.6}\text{Ca}_{0.4}\text{Mn}_{0.8}\text{Ga}_{0.2}\text{O}_3$ perovskite oxide with the two-step thermochemical H_2O splitting carried out between 1400 and 900 $^\circ\text{C}$.

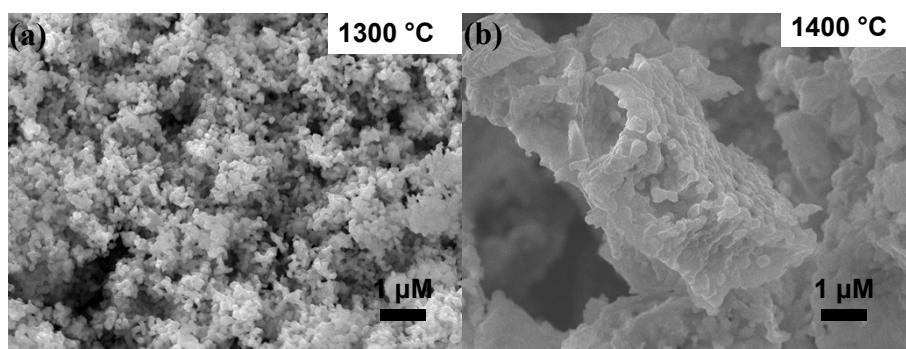


Fig. S8 SEM images of $\text{La}_{0.6}\text{Ca}_{0.4}\text{Mn}_{0.8}\text{Ga}_{0.2}\text{O}_3$ perovskite oxide after thermally reduced at (a) 1300 °C and (b) 1400 °C.

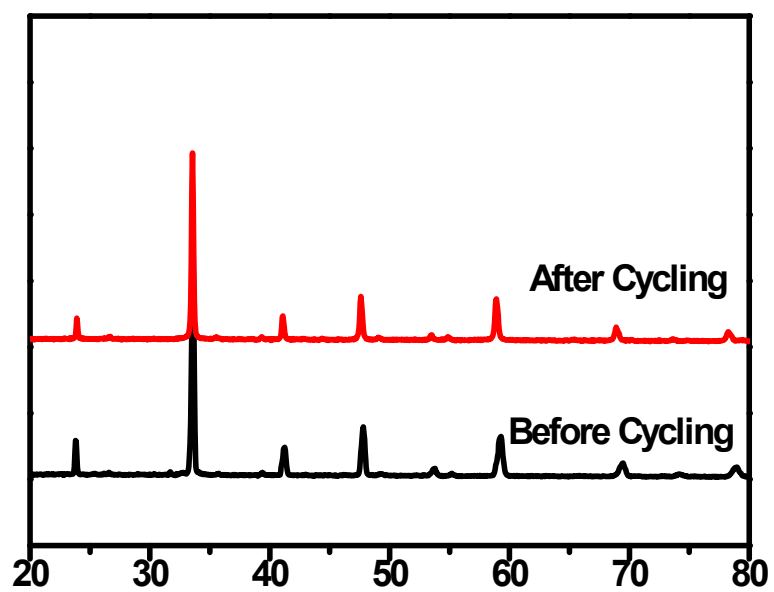


Fig. S9 XRD patterns of LCMGO before (black) and after (red) the thermochemical cycling measurements

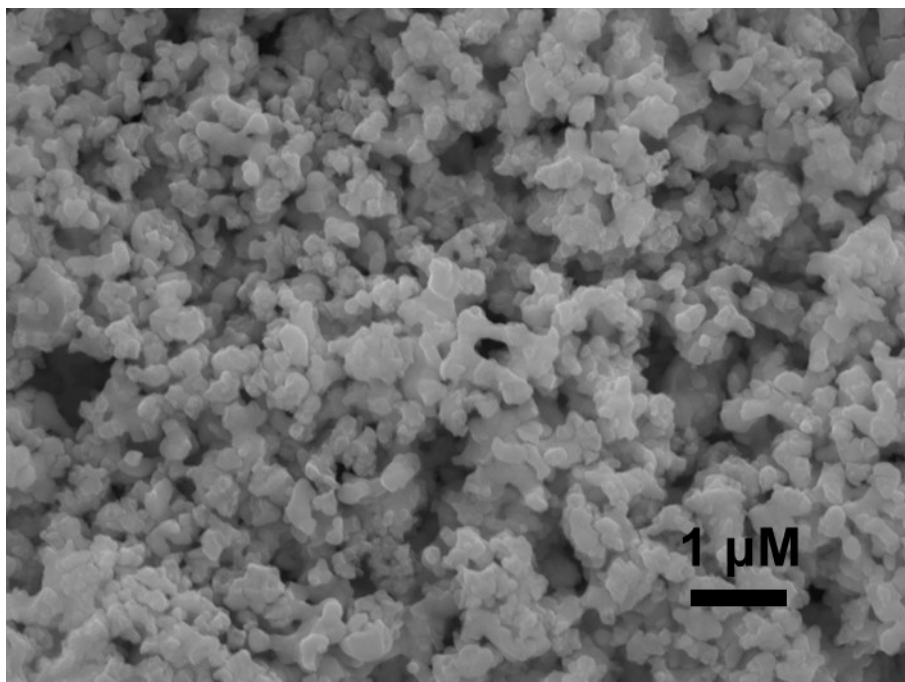


Fig. S10 SEM image of LCMGO after the two-step thermochemical H₂O splitting cycling test operated between 1300 and 900 °C.

Table S1 BET surface area of LaMnO₃ based perovskites with various metal ions doped before and after two-step thermochemical H₂O splitting process.

Perovskite sample	BET surface area before thermochemical test (m ² /g)	BET surface area after thermochemical test (m ² /g)
La _{0.6} Ba _{0.4} MnO ₃	10.4	7.9
La _{0.6} Sr _{0.4} MnO ₃	11.5	9.4
La _{0.6} Ca _{0.4} MnO ₃	14.2	11.5
La _{0.6} Ca _{0.4} Mn _{0.8} Al _{0.2} O ₃	16.9	13.4
La _{0.6} Ca _{0.4} Mn _{0.8} Ga _{0.2} O ₃	19.6	15.2

Table S2 Comparison of thermochemical performances of the present study with those reported by other perovskite type catalyst and benchmark CeO₂ in the literature.

Redox Catalyst	Thermal reduction temperature (°C)	O ₂ evolution (μmol/g)	H ₂ production (μmol/g)	Re- oxidation yield (%)	Ref.
Ba _{0.5} Sr _{0.5} Co _{0.8} Fe _{0.2} O ₃	1000	600	83	7	3
La _{0.6} Sr _{0.4} Cr _{0.8} Co _{0.2} O ₃	1200	154	50	16	4
La _{0.5} Sr _{0.5} Co _{0.8} Fe _{0.2} O ₃	1200	503	90	9	3
La _{0.6} Ca _{0.4} Mn _{0.8} Ga _{0.2} O ₃	1300	212	401	95	This study
LaSrCoO ₄	1300	268	161	30	3
La _{0.6} Sr _{0.4} Mn _{0.4} Al _{0.6} O ₃	1350	120	220	92	5
La _{0.5} Sr _{0.5} Mn _{0.95} Sc _{0.05} O ₃	1400	390	250	32	6
La _{0.6} Sr _{0.4} MnO ₃	1400	219	397	91	7
La _{0.5} Sr _{0.5} MnO ₃	1400	298	195	33	3
La _{0.6} Ca _{0.4} MnO ₃	1400	272	407	75	8
Y _{0.5} Sr _{0.5} MnO ₃	1400	481	320	33	9
Y _{0.5} Ca _{0.5} MnO ₃	1400	593	310	26	9
CeO ₂	1400	66	144	109	10
CeO ₂	1640	111	178	80	11

Supporting References

1. Y.-S. Han and H.-G. Kim, *J. Power Sources*, 2000, **88**, 161.
2. Y. Hao, C. K. Yang and S. M. Haile, *Chem. Mater.*, 2014, **26**, 6073.
3. A. Demont, S. Abanades and E. Beche, *J. Phys. Chem. C*, 2014, **118**, 12682.
4. A. H. Bork, M. Kubicek, M. Struzik and J. L. M. Rupp, *J. Mater. Chem. A* 2015, **3**, 15546.
5. A. H. McDaniel, E. C. Miller, D. Arifin, A. Ambrosini, E. N. Coker, R. O'Hayre, W. C. Chueh and J. H. Tong, *Energy Environ. Sci.*, 2013, **6**, 2424.
6. S. Dey, B. S. Naidu and C. N. R. Rao, *Dalton Trans.*, 2016, **45**, 2430.
7. C. K. Yang, Y. Yamazaki, A. Aydin and S. M. Haile, *J. Mater. Chem. A*, 2014, **2**, 13612.
8. S. Dey, B. S. Naidu, A. Govindaraj and C. N. R. Rao, *Phys. Chem. Chem. Phys.*, 2015, **17**, 122.
9. S. Dey, B. S. Naidu and C. N. R. Rao, *Chem. Eur. J.*, 2015, **21**, 7077.
10. A. Le Gal, S. Abanades and G. Flamant, 2011, **25**.
11. W. C. Chueh, C. Falter, M. Abbott, D. Scipio, P. Furler, S. M. Haile and A. Steinfeld, *Science*, 2010, **330**, 1797.