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## Supporting Information

 $Cs^+$  incorporation into  $CH_3NH_3PbI_3$  perovskite: Broad miscibility gap and stability enhancement

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Figure E1: An example of a solar cell library (13 x 13 cells) with a gradient perovskite layer after dipping into a CsI in IPA solution.

Table E1: Fittings of the (	(110)	) reflection of	during	in-situ	cation	exchange.
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Time	$2\Theta [^{\circ}]$
0min	14.13
$3 \min$	14.16
8min	14.17
$18 \min$	14.17
$28 \min$	14.18
$38 \mathrm{min}$	no peak



Figure E2: A summary is given of the photovoltaic parameters over dipping time for two different batches, namely open-circuit voltage  $V_{\rm OC}$ , short-circuit current  $I_{\rm SC}$ , fill factor FF and power conversion efficiency PCE, respectively. (a+c) shows the full parameter distribution with each cell indicated as a dot, while (b+d) shows the distribution of those parameters for the native sample in form of a histogram. The qualitative trend is the same in that the  $V_{\rm OC}$  increases upon Cs-incorporation and  $I_{\rm SC}$  (and therefore PCE continuously drop for increasing dipping times. The blank sample shown in (d) shows a broader distribution of most PV parameters, but the qualitative response of the parameters for the Cs<sup>+</sup> dipping remains the same.



Figure E3: Synthesis and analysis of Cs+ incorporated MAPbI3 solar cells. A gradual dipping conversion was done on 7–7 cm2 substrates. The optical, electronic and structural properties of the material were tested in a high-throughput fashion. In-situ measurements were taken to clarify the conversion mechanism and show the improved stability of CsxMA1-xPbI3. Analysis of the thin-films and solar cells gave information about the material and its photovoltaic performance.



Figure E4: full XRD spectra of the in-situ cation exchange shows the complete conversion within the 38min of the experiment. The inset shows zoomed-in diffractograms at different stages of the conversion process.

Table E2: Peak position of (110) reflection found in diffractions done for different  $Cs_xMA_{1-x}PbI_3$  compositions made with a one-step deposition. Peak position obtained using fittings of the (110) reflection peak to a calculated pseudoVogit intensity profile done in CrystalMaker. The full diffractograms can be found in fig. 2 in the main document.

Composition	$2\Theta [^{\circ}]$
x=0	14.17
x=5	14.19
x=10	14.21
x = 15	14.23
x=20	14.20
x=25	14.20
x=30	14.21
x=40	14.19
x = 50	14.23
x = 75	14.34
x=100	-



Figure E5: (a) UV-Vis spectra of different conversion stages of the native thinfilms (native MAPbI<sub>3</sub> and different conversion times in CsI solution). (b) The subtracted spectra before baseline removal. The baseline corrected spectra are shown in fig. 5 in the main document.



Figure E6: SEM images taken at different stages of the peeling process on  $MAPbI_3$  ting(before, after one peeling and ten peelings). Results show the incremental removal of the capping layer.

Table E3: Structural parameters of the plain perovskites MAPbI<sub>3</sub> and CsPbI<sub>3</sub> as well as the mixed cation Cs<sub>x</sub>MA<sub>1-x</sub>PbI<sub>3</sub> in their cubic phase. Namely the cation radii  $r_{cation}$  [1] and unit cell volume  $V_{UC,cubic}$  [2] were taken from literature. The values of the (110) reflection in XRD (2 $\Theta$ ) and corresponding d-spacing as measured in this study.

Perovskite cation		MA	$\rm Cs_xMA_{1\text{-}x}$	$\mathbf{Cs}$	Rel. diff.
Literature	r <sub>cation</sub> [1]	2.17  Å		1.81 Å	-16.6~%
	$V_{UC,cubic}$ [2]	$251.6 \text{ Å}^3$		$248.8 \text{ Å}^3$	-1.12 %
Measured	$2 \Theta (110)$	14.14 °	14.18°		0.30~%
	d-spacing	6.26 Å	6.24 Å		-0.30~%



Figure E7: UV-Vis absorption spectra of thin-films (native MAPbI<sub>3</sub> above and  $Cs_xMA1 - xPbI_3$  below) dipped into a solution of IPA with different concentrations of water. The absorption in the blue regime dropped significantly as the thin-films decomposed into PbI<sub>2</sub>, alongside an increase of the red regime, caused by scattering. A drop of the signal at 450 nm of 80 % was defined as the time of degradation.  $Cs_xMA1 - xPbI_3$  in solutions of water/IPA < 0.02 were stable over the full duration of the measurement (10h) and remained without visible color change for more than a week.

## Error analysis

The propagation of error calculations were calculated individually for the three different approaches that were taken in this study (Tauc plot, EDAX and XRD).

**Tauc plot error:** The shift of the bandgap was determined via Tauc plot. Therefore, we fitted a first order polynomial to the linear region of the Tauc plot of the  $Cs_xMA_{1-x}PbI_3$  after cation conversion (Tauc 1 with bandgap  $T_1$ ) and for the blank MAPbI<sub>3</sub> (Tauc 2 with  $T_2$ ). The linear fit parameters for Tauc 1 plot curve with standard deviation are  $a_1 = 0.1775 \pm 0.0009 \cdot 10^{-23}$  and  $c_1 = -0.2851 \pm 0.0014 \cdot 10^{-23}$ . The intersection with the x axis and corresponding error function are:

$$T_1 = \frac{c_1}{a_1} \qquad \Rightarrow \Delta T_1 = \pm \sqrt{\left(\Delta c_1 \frac{1}{a_1}\right)^2 + \left(\Delta a_1 \frac{c_1}{a_1^2}\right)^2}$$
$$\Rightarrow T_1 = 1.618 \pm 0.012 \text{ eV}$$

For the blank MAPbI<sub>3</sub> sample we performed and analogous calculation with the fit parameters of the linear region of Tauc 2 of  $a_2 = 0.1842 \pm 0.00015 \cdot 10^{-23}$  and  $c_2 = -0.2946 \pm 0.0002 \cdot 10^{-23}$ . The intersection with the x axis and corresponding error function are:

$$T_2 = \frac{c_2}{a_2} \qquad \Rightarrow \Delta T_2 = \pm \sqrt{\left(\Delta c_2 \frac{1}{a_2}\right)^2 + \left(\Delta a_2 \frac{c_2}{a_2^2}\right)^2}$$
$$\Rightarrow T_2 = 1.598 \pm 0.0018 \text{ eV}$$

The value for the bandgap of pure  $\text{CsPbI}_3$  is taken from literature[2]  $T_3 = 1.73 \text{ eV}$ . Combining these values we can calculate the composition from our UV-Vis measurements and the corresponding error function:

$$x_{Tauc} = \frac{T_2 - T_1}{T_3 - T_1} \qquad \Rightarrow \Delta x_{Tauc} = \pm \sqrt{\left[\Delta T_2 \frac{T_1 - T_2}{(T_3 - T_2)^2}\right]^2 + \left(\Delta T_1 \frac{1}{T_3 - T_2}\right)^2} \\ \Rightarrow \Delta x_{Tauc} = \pm 0.081$$

**EDAX error:** The constant regime (*step I*) of the absorbance at 450 nm is evaluated by forming the mean average  $c_1 = 2.850 \pm 0.047$ , while the slope that occurs during the *step II* conversion is fitted with a first order polynomial  $(y = a \cdot x + b)$  with  $a_2 = 0.407 \pm 0.021$  and  $b_2 = -1.47 \pm 0.18$ . The slope of the EDAX Cs signal was also fitted with a first order polynomial with  $a_3 = -0.0466 \pm 0.0018$  and  $b_3 = 0.602 \pm 0.016$ . The formula to determine composition  $x_{EDAX}$  derives from simple geometric consideration to:

$$x_{EDAX} = a_3 \frac{c_1 - b_2}{a_2} + b_3$$
$$\Rightarrow \Delta x_{EDAX} =$$

$$\pm \sqrt{\left[\Delta a_3 \frac{c_1 - b_2}{a_2}\right]^2 + \left[\Delta c_1 \frac{a_3}{a_2}\right]^2 + \left[\Delta b_2 \frac{a_3}{a_2}\right]^2 + \left[\Delta a_2 \frac{a_3}{a_2^2}(c_1 - b_2)\right]^2 + \Delta b_3^2}$$
$$\Rightarrow \Delta x_{EDAX} = \pm 0.042$$

**XRD error:** The miscibility gap estimation from XRD was done by measureing the (110) reflection peak of different stoichiometric  $Cs_xMA_{1-x}PbI_3$  thinfilms. The shifting reflection peaks were fitted with a first order polynomial with a slope  $a = 0.4012 \pm 0.048$ . This slope was compared to the obtained shift during the cation conversion, with (110) reflection before conversion of  $\Theta_1 = 14.137 \pm 0.0037$ ° and after 28 min conversion  $\Theta_2 = 14.184 \pm 0.0098$ °. The function to calculate the composition and the according error function are:

$$x_{XRD} = \frac{\Theta_2 - \Theta_1}{a}$$
$$\Rightarrow \Delta x_{XRD} = \pm \sqrt{\left[\Delta a \frac{\Theta_2 - \Theta_2}{a^2}\right]^2 + \left(\Delta \Theta_1 \frac{1}{a}\right)^2 + \left(\Delta \Theta_2 \frac{1}{a}\right)^2}$$
$$\Rightarrow \Delta x_{XRD} = \pm 0.030$$

## References

- G. Kieslich, S. Sun, and T. Cheetham. An Extended Tolerance Factor Approach for Organic-Inorganic Perovskites. *Chem. Sci.*, 6:3430–3433, 2015.
- [2] C. C. Stoumpos, C. D. Malliakas, and M. G. Kanatzidis. Semiconducting tin and lead iodide perovskites with organic cations: phase transitions, high mobilities, and near-infrared photoluminescent properties. *Inorganic chemistry*, 52(15):9019–9038, aug 2013.