

## Prominent electrochemical performance of $\text{Li}_3\text{VO}_4/\text{C-Ni}$ anode via hierarchically porous architecture design

Jicheng Zhang,<sup>a</sup> Shibing Ni,<sup>\*a,b</sup> Tao Kang,<sup>a</sup> Jun Tang,<sup>a</sup> Xuelin Yang<sup>\*a,b</sup>, Lulu Zhang<sup>a,b</sup>

<sup>a</sup>College of Materials and Chemical Engineering, Three Gorges University, 8 Daxue Road, Yichang, Hubei 443002, China.

<sup>b</sup>Hubei Provincial Collaborative Innovation Center for New Energy Microgrid, China Three Gorges University.

\*E-mail: shibingni07@126.com.

### Experimental section

#### Sample preparation

The chemicals were analytical grade and purchased from Shanghai Chemical Reagents. Ni foam (100 PPI pore size,  $380 \text{ g m}^{-2}$  surface density, 1.5 mm thick) was purchased from Changsha Lyrun New Material corporation. In a typical procedure, 1 mmol  $\text{V}_2\text{O}_5$ , 3 mmol  $\text{Li}_2\text{CO}_3$  and 5 mmol hexamethylenetetramine were added in 30 ml distilled water. After stirring for 20 minutes, the homogeneous yellowy suspension was transferred into a 50 ml teflonlined autoclave, distilled water was subsequently added to 80% of its capacity. The autoclave was at last sealed and placed in an oven, heated at  $120 \text{ }^\circ\text{C}$  for 24 h. The final transparent solution was transferred in a culture dish, 0.05 g citric acid was added in the solution. After stirring for 2h, Ni discs were immersed into the solution for 1h. After that, the Ni discs were dried and annealed in  $\text{N}_2$  atmosphere at  $500 \text{ }^\circ\text{C}$  for 5h. The weight of active  $\text{Li}_3\text{VO}_4/\text{C}$  on Ni disc can be estimated according to the weight difference between the Ni disc and the obtained  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  ( $m_{\text{Li}_3\text{VO}_4/\text{C}} = m_{\text{Li}_3\text{VO}_4/\text{C-Ni}} - m_{\text{Ni}}$ ).

## **Structure and morphology characterization**

The structure and morphology of the resulting products were characterized by X-Ray powder diffraction (Rigaku Ultima IV Cu K $\alpha$  radiation  $\lambda=1.5406$  Å), field-emission scanning electron microscopy (FE-SEM JSM 7500F, JEOL), and transmission electron microscopy (TEM, FEI, Tecnai G2 F30) equipped with selected area electron diffraction (SAED).

## **Electrochemical characterization**

For preparing anode, the as-prepared Li<sub>3</sub>VO<sub>4</sub>/C-Ni foam discs (diameter of 14 mm) were dried at 120 °C for 24 h in vacuum oven. 2025 coin-type half cells of Li/1 M LiPF<sub>6</sub> in ethylene carbonate, dimethyl carbonate and diethyl carbonate (EC/DMC/DEC, 1:1:1 v/v/v)/Li<sub>3</sub>VO<sub>4</sub>/C-Ni were assembled in an argon-filled dry box (MIKROUNA, Super 1220/750, H<sub>2</sub>O<1.0 ppm, O<sub>2</sub><1.0 ppm). A Celgard 2400 microporous polypropylene was used as the separator membrane. For the fabrication of LiFePO<sub>4</sub> cathode, a mixture of commercial LiFePO<sub>4</sub>, acetylene black, and polyvinylidene fluoride (PVDF) dissolved in N-methylpyrrolidone (NMP) solution (0.02 g mL<sup>-1</sup>) with weight ratio of 7.5:1.5:1 were coated on aluminium foil and cut into disc electrodes with a diameter of 14 mm. 2025 coin-type full cells with the LiFePO<sub>4</sub> cathode and the Li<sub>3</sub>VO<sub>4</sub>/C-Ni anode were assembled using the same technology of half cells. Considering the initial irreversible lithium ions consumption in charging, a capacity ratio of 1.2:1 between cathode and anode in full cells was adopted. Before the matching, half cell of LiFePO<sub>4</sub> cathode and Li<sub>3</sub>VO<sub>4</sub>/C-Ni was resembled and measured, respectively. Galvanostatic charge/discharge test was

characterized on a multichannel battery test system (LAND CT2001A) in the voltage region 0.02~3 V for half cells and 0.8~3.5 V for full cells. The cyclic voltammetry measurement of the electrodes (vs. Li<sup>+</sup>/Li) was carried out on a CHI660C electrochemical workstation at a scan rate of 0.2 mV s<sup>-1</sup> between 0 and 3 V. Electrochemical impedance spectroscopy measurements were performed on CHI660C electrochemical workstation under open circuit conditions over a frequency range from 0.01 Hz to 100 kHz by applying an AC signal of 5 mV in amplitude throughout the tests. Equivalent circuit fitting of EIS data was carried out via ZsimpWin software.

Tab. S1 The summarization of discharge capacity versus cycle number of various  $\text{Li}_3\text{VO}_4$  electrodes. (the references are in accordance with those in the manuscript).

<b>material</b>	<b>current (mA/g)</b> <b>1C<math>\approx</math>394 mA/g</b>	<b>reversible capacity</b> <b>mAh g<sup>-1</sup>/cycle no.</b>	<b>ref.</b>
$\text{Li}_3\text{VO}_4$	20 mA/g	283/25	1
$\text{Li}_3\text{VO}_4/\text{graphene}$	20 mA/g	378/50	2
$\text{Li}_3\text{VO}_4/\text{graphite}$	156 mA/g	468/100	3
$\text{Li}_3\text{VO}_4/\text{CNTs}$	2000 mA/g	250/2000	4
$\text{Li}_3\text{VO}_4/\text{C}$	0.8 C	363/40	5
$\text{Li}_3\text{VO}_4/\text{graphene}$	5 C	163/5000	6
$\text{Li}_3\text{VO}_4/\text{C}$	1 C	394/100	7
$\text{Li}_3\text{VO}_4/\text{graphene}$	0.2 C	453/200	8
$\text{Li}_3\text{VO}_4/\text{C}$	20 mA/g	245/50	9
$\text{Li}_3\text{VO}_4/\text{C}$	0.1 C	401/50	10
$\text{Li}_3\text{VO}_4/\text{C}@EG$	2000 mA/g	205/2000	11
	100 mA/g	405/200	
$\text{HP-Li}_3\text{VO}_4/\text{C}$	200 mA/g	381/300	12
	4000 mA/g	275/500	
$\text{Li}_3\text{VO}_4/3\text{DGNs}$	2000 mA/g	259/2500	19
$\text{Li}_3\text{VO}_4$	0.25 C	396/100	21
$\text{Li}_3\text{VO}_{4-\delta}$	200 mA/g	286/200	23
$\text{Li}_3\text{VO}_4$	0.5 C	250/100	24
$\text{Li}_3\text{VO}_4$	1 C	190/100	25
$\text{Li}_3\text{VO}_4$	2000 mA/g	280/500	26
$\text{Li}_3\text{VO}_4$	0.1 C	311/50	27
$\text{Li}_3\text{VO}_4/\text{C-Ni}$	10 C	325/ 2000	this work

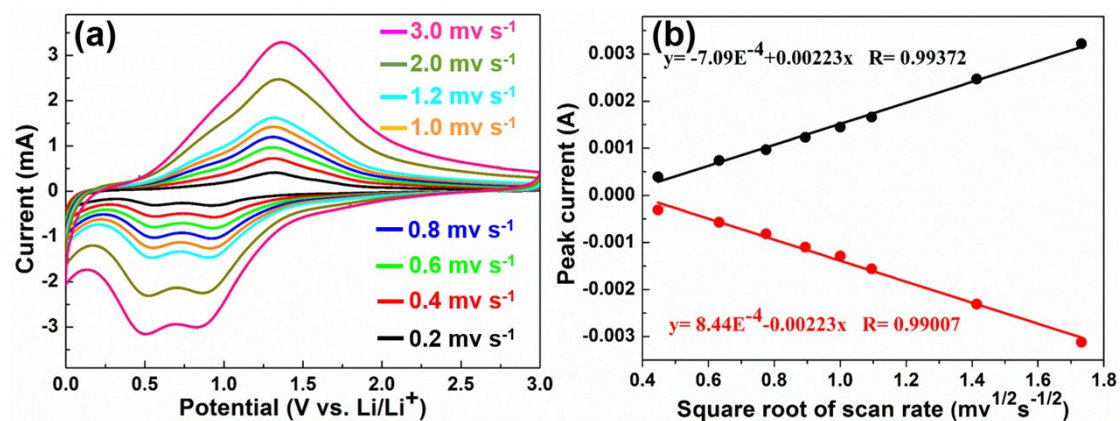


Fig. S1 (a) CV curves of the  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  electrode at different scan rates between 0 and 3 V. (b) Dependence of peak current on the square root of the scan rate for the  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  electrode.

CV curves of the  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  electrode at scan rate from 0.2 to 3.0  $\text{mV s}^{-1}$  are shown in Fig. S1(a). As seen, the reduction peaks shift to low potential regions with increasing scan rate, whereas the oxidation peak shifts to a high potential region, suggesting higher polarization under a high scan rate. Fig. S1(b) shows the relationship between peak current and the square root of the scan rate obtained from the experimental data in Fig. S1(a). As found, both the anodic and cathodic peak currents show a linear dependence on the square root of scan rate from 0.2 to 3  $\text{mV s}^{-1}$ , suggesting a Li-ion diffusion controlled mechanism in the discharge and charge process [1]. High symmetry of the slope of the fitted lines indicates a highly reversible insertion and extraction of lithium ions in cycling.

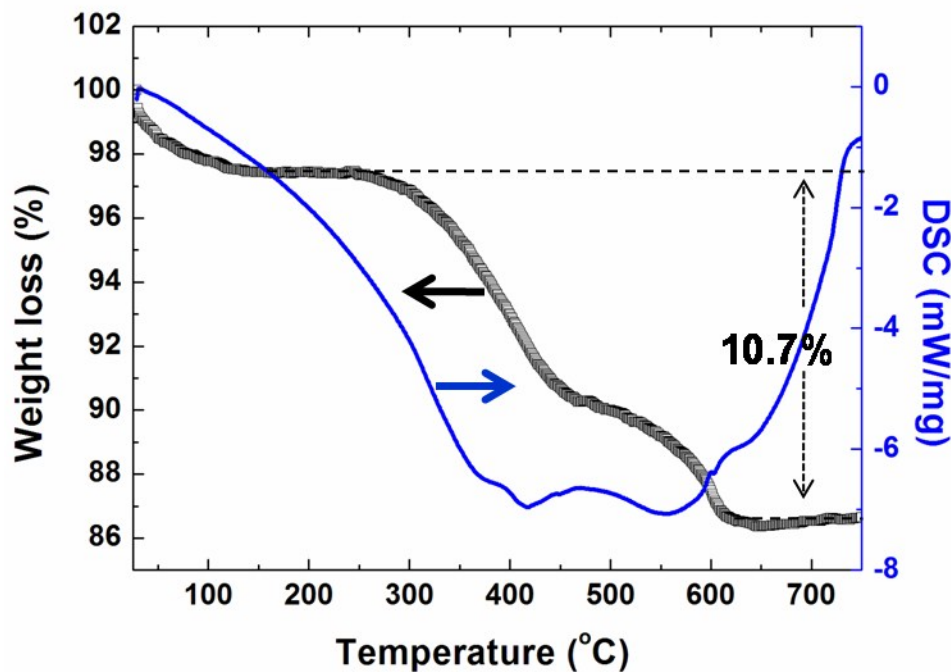


Fig. S2 TG and DSC curves of  $\text{Li}_3\text{VO}_4/\text{C}$  obtained under the same condition of  $\text{Li}_3\text{VO}_4/\text{C}$ -Ni electrode.

Fig. S2 shows the TG and DSC curves of the  $\text{Li}_3\text{VO}_4/\text{C}$  obtained under the same condition of  $\text{Li}_3\text{VO}_4/\text{C}$ -Ni electrode. The weight loss below 150 °C corresponds to the elimination of absorbed water, and the weight loss in temperature region 250~700 °C corresponds to the escape of C in  $\text{Li}_3\text{VO}_4/\text{C}$ . According to the TG results, the weight of C in  $\text{Li}_3\text{VO}_4$  is about 10.7%. Considering the low specific capacity of amorphous C of about 300 mAh  $\text{g}^{-1}$ , the final specific capacity ( $\sim 550$  mAh  $\text{g}^{-1}$ ) of the  $\text{Li}_3\text{VO}_4/\text{C}$  is reasonable within the acceptable measuring error.

The morphology of the  $\text{Li}_3\text{VO}_4/\text{C}$ -Ni electrode after 2000 cycles is shown in Fig. S3. As seen, hierarchically porous characteristics of the electrode are kept, though the integral morphology of the electrode differs little from that before cycling. Such morphology variation originates from the reduction of particle-size of  $\text{Li}_3\text{VO}_4/\text{C}$

owing to electrochemical activation and electrochemical reconstruction, which is similar to that in previous study [2].

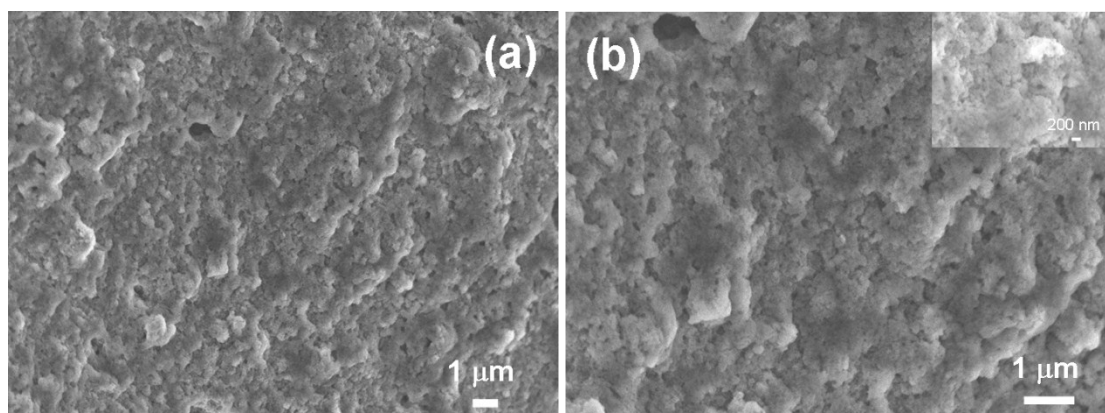


Fig. S3 SEM images of the  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  electrode after 2000 cycles with low (a) and high (b) magnification. The inset of (b) is a magnified SEM image.

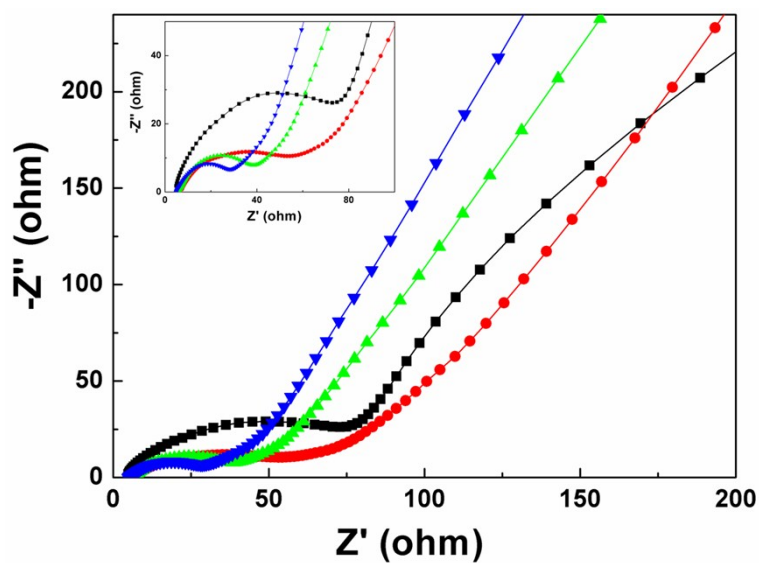


Fig. S4 AC impedance spectra of the  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  electrode under different state. The inset shows the magnified part of the spectra.

Fig. S4 shows the electrochemical impedance spectra of the  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  under different state. The intercept in high-frequency can be attributed to the contact resistance caused by SEI film and/or electrolyte, the medium-frequency semicircle is due to the charge-transfer impedance on electrode/electrolyte interface, and the

inclined line in low-frequency corresponds to the Li-ion diffusion process within electrodes [3]. The Nyquist plots were fitted via an R(C(RW)) equivalent circuit, and the contact and charge-transfer resistance of the electrode ( $R_e$  and  $R_{ct}$ ) were obtained from the fitting results (Tab. S2). As seen,  $R_e$  increases obviously after 2 cycles and then decreases along with the increasing of cycle number, which may be relevant to the gradual formation of stable SEI in the first few cycles. In addition,  $R_{ct}$  for the  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  decreases obviously along with the increasing of cycle number, suggest reduced charge transfer resistance. Such phenomena can ascribe to the originally high electronic conductivity based on the hierarchically porous architecture design and the unique electrochemical reconstruction in cycling that leads to enhanced contact between the integrate  $\text{Li}_3\text{VO}_4/\text{C}$  and the electric Ni foam substrate.

Tab. S2 Electrode kinetic parameters obtained from equivalent circuit fitting of Nyquist plots for  $\text{Li}_3\text{VO}_4/\text{C-Ni}$  electrode at different state.

$\text{Li}_3\text{VO}_4/\text{C-Ni}$	$R_e$ ( $\Omega$ )	$R_{ct}$ ( $\Omega$ )
fresh electrode	5.78	42.55
after 2 cycles	11.92	37.10
after 5 cycles	7.43	21.73
after 10 cycles	6.02	14.84

## References

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