

### Supporting Information

## **Influence of the Substrate on the Bulk Properties of Hybrid Lead Halide Perovskite Films**

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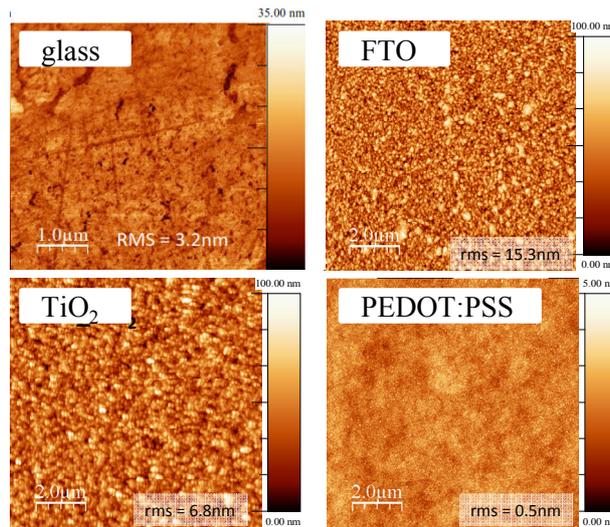
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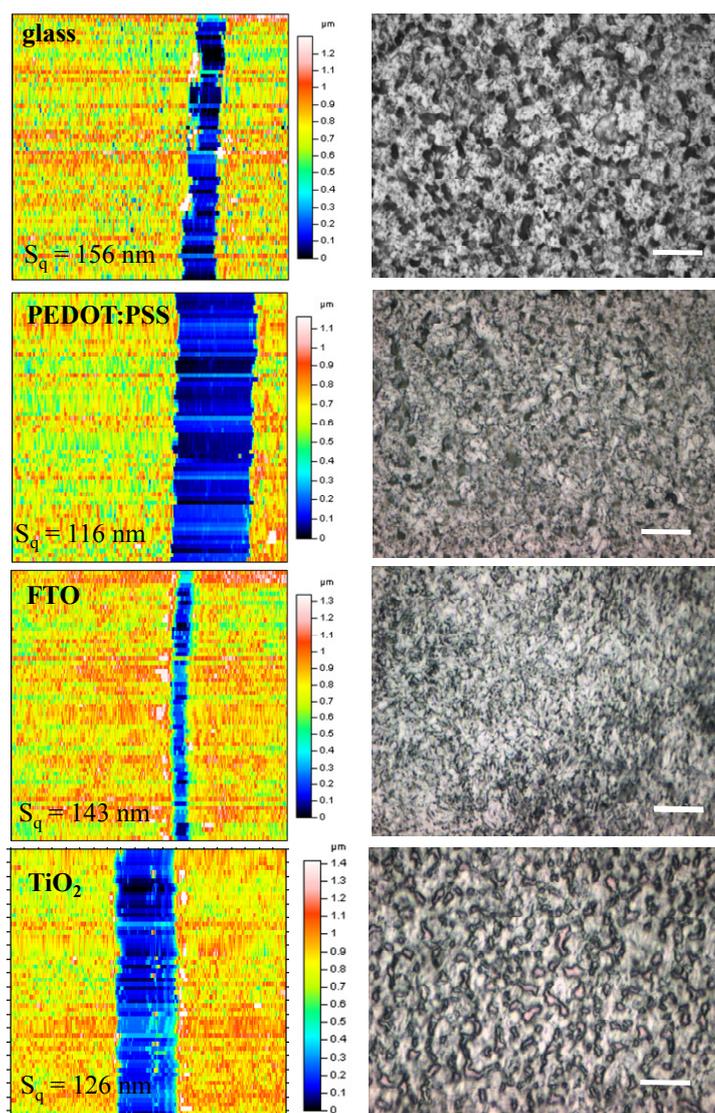
### Profilometry

Atomic force microscopy AFM images of the used substrates are in **Figure S1**. Measurements were performed using a commercial equipment and software from Nanotec<sup>1</sup> under ambient conditions. Images have been acquired in tapping mode using commercial silicon tips from Nanosensors (PPPQNCHR, 20–30 nm tip diameter). The measured roughness of the substrates are: 3.2, 15.3, 0.8 and 6.8 nm for bare glass, FTO, PEDOT:PSS and TiO<sub>2</sub> respectively.



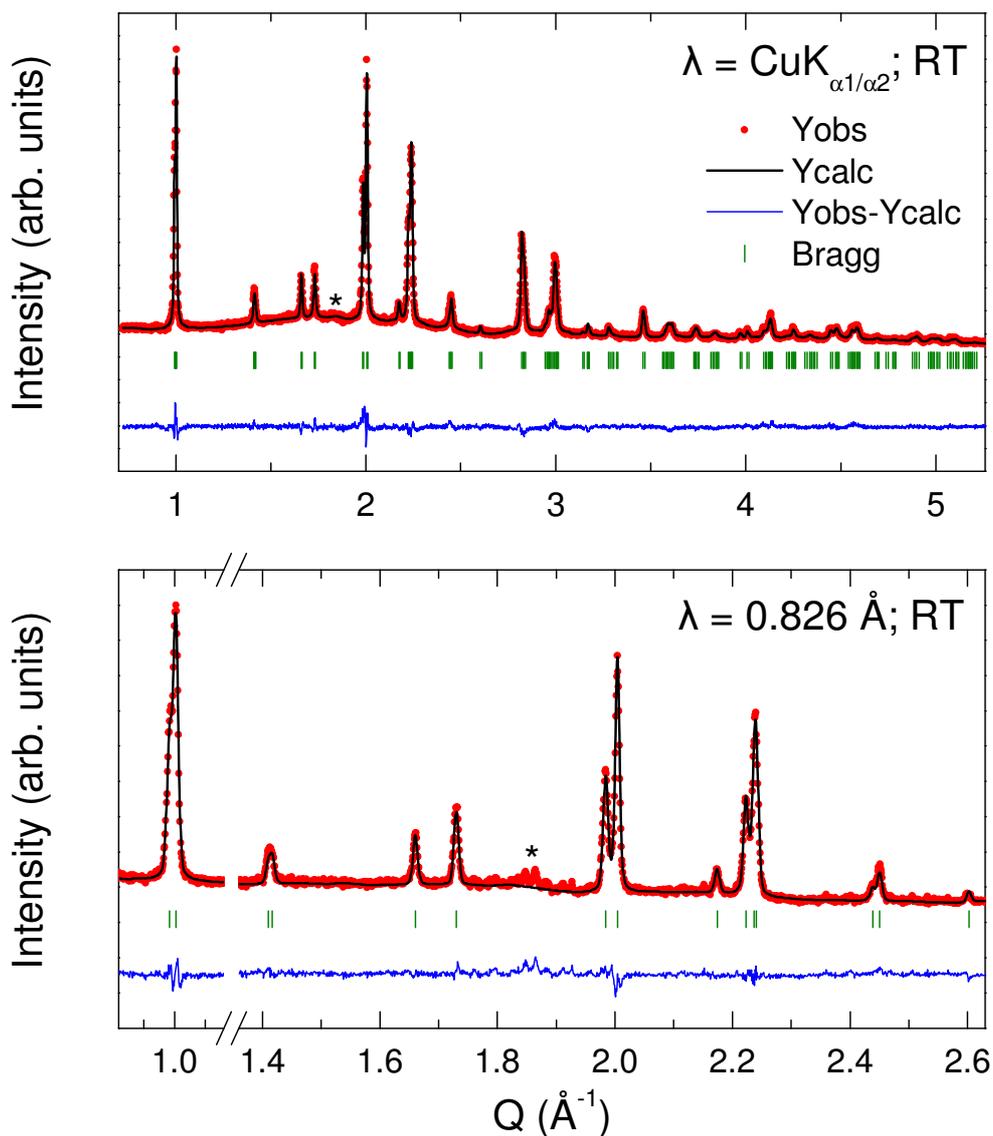
**Figure S1.** AFM profilometry showing the root mean square (rms) roughness of the four substrates on which the  $\text{MAPbCl}_x\text{I}_{3-x}$  are deposited: (a) glass, (b) FTO, (c)  $\text{TiO}_2$  and (d) PEDOT:PSS.

The films thickness of the  $\text{MAPbI}_{3-x}\text{Cl}_x$  films has been evaluated by contact profilometry (Alpha step 200 profilometer, Tenkor Instrument), obtaining around 650, 550, 650 and 750 nm for the thin film growth on glass, glass/FTO/PEDOT:PSS, glass/FTO and glass/FTO/ $\text{TiO}_2$  respectively. Profilometry and topography analysis on 500x500 microns area size on each sample and Apex 3D software pack gives us a Root Mean Square Roughness ( $S_q$ ) of 156, 116, 143 and 126 nm (**Figure S2**, left panel). The different morphologies observed by optical microscopy of  $\text{MAPbI}_{3-x}\text{Cl}_x$  thin films on the four substrates are shown in Figure S2, right panel.



**Figure S2.** Profilometry and topography analysis (Left panel) on  $500 \times 500 \mu\text{m}^2$  area size on each sample showing height profile and Root Mean Square Roughness ( $S_q$ ) for the  $\text{MAPbI}_{3-x}\text{Cl}_x$  thin films on (a) glass, (b) glass/FTO/PEDOT:PSS and (c) glass/FTO and (d) glass/FTO/ $\text{TiO}_2$ . The images show the groove (blue) made to measure the height step. Right panel: Optical images ( $73 \times 55 \mu\text{m}$ ) of  $\text{CH}_3\text{NH}_3\text{PbI}_{3-x}\text{Cl}_x$  thin films growth on (a) glass, (b) glass/FTO/PEDOT:PSS and (c) glass/FTO and (d) glass/FTO/ $\text{TiO}_2$ . The scale bar is equivalent to  $10 \mu\text{m}$ .

## Crystal structure



**Figure S3.** Observed (circles), calculated (solid line) and difference (lower solid line) XRPD ( $\lambda = \text{Cu K}\alpha_{1/\alpha_2}$ ) and SXRPD ( $\lambda = 0.826 \text{ \AA}$ ) profiles of MAPbI<sub>3</sub> powder at RT. Tick marks indicate the position of allowed reflections of MAPbI<sub>3</sub> in *I4cm* (upper). The asterisk indicates the presence of a minor amount of PbI<sub>2</sub>.

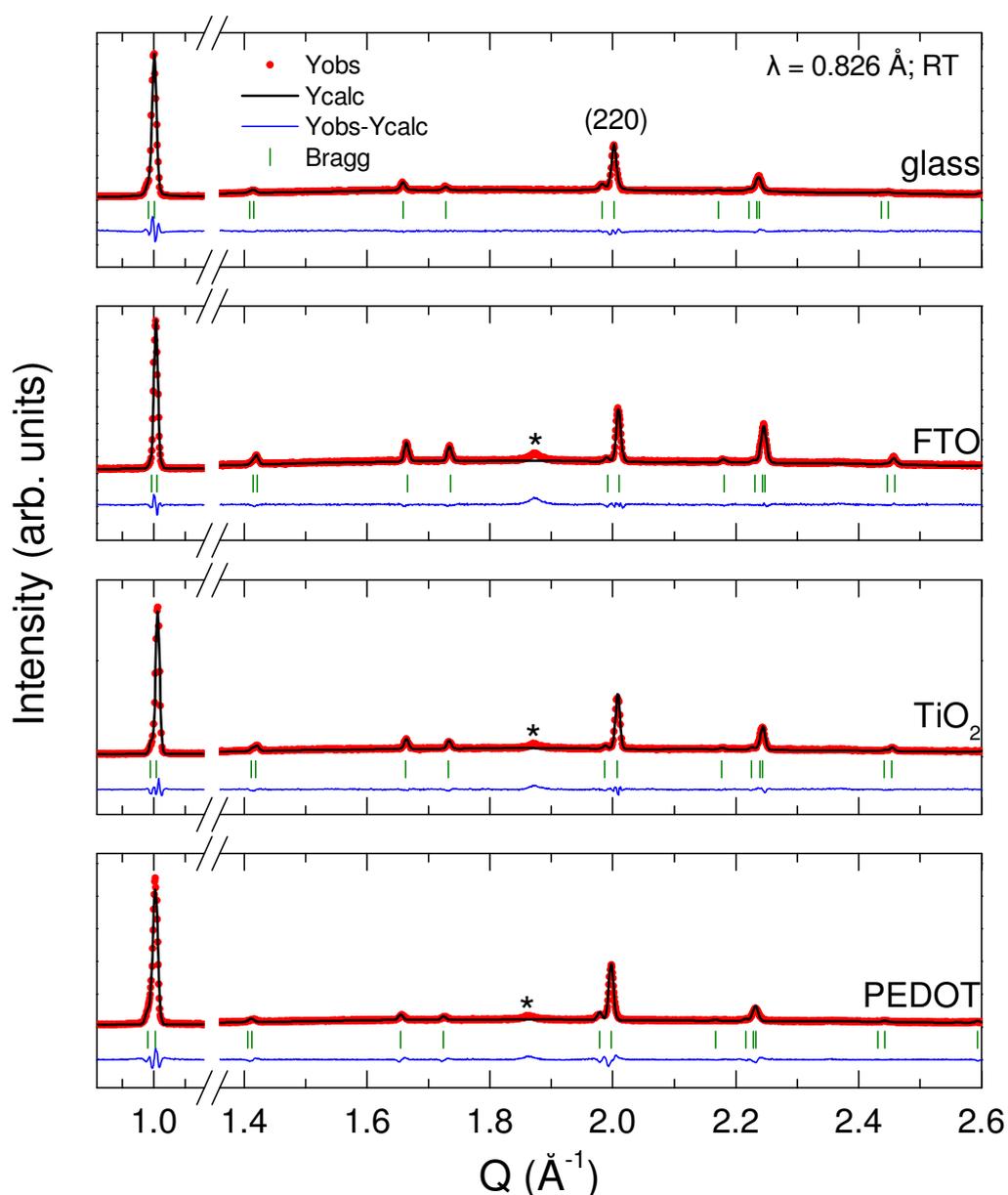
In order to refine the crystal structure of the MAPbI<sub>3</sub> powder, the non-centrosymmetric *I4cm* space group was chosen. These refinements were based on the polar tetragonal structural model previously reported by Stoumpos *et al.* for MAPbI<sub>3</sub> single-crystal, where the C–N bonds of the methylammonium cations are fixed parallel to the *c*-axis.<sup>2</sup> The used experimental data were collected on a laboratory diffractometer (D8 Focus,

Bruker; Cu  $K_{\alpha 1/\alpha 2}$ ), and on a six-circle diffractometer (15 KeV; 0.826 Å) at the BM25B (SpLine) beamline at the European Synchrotron facility (ESRF) at RT. The Rietveld plots and the crystallographic data are given in **Figure S3** and **Table S1**, respectively. The C and N atomic positions of the methylammonium cations were not refined taking into account the quality of the collected data.

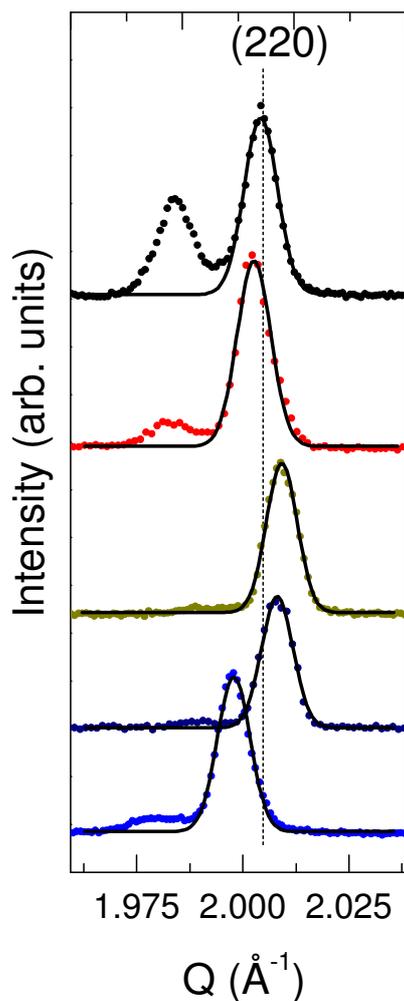
CH <sub>3</sub> NH <sub>3</sub> PbI <sub>3</sub>		<i>I4cm</i>
<i>a</i> (Å)		8.8695(1)
<i>c</i> (Å)		12.6624(2)
V (Å <sup>3</sup> )		996.13(3)
Pb (4 <i>a</i> )	U <sub>iso</sub> (Å <sup>2</sup> )	0.0386(9)
I_1 (4 <i>a</i> )	<i>z/c</i>	0.2524(2)
	U <sub>iso</sub> (Å <sup>2</sup> )	0.099(3)
I_2 (8 <i>c</i> )	<i>x/a</i>	0.2162(3)
	<i>y/b</i>	0.7162(3)
	<i>z/c</i>	0.01249(9)
	U <sub>iso</sub> (Å <sup>2</sup> )	0.0857(16)
C (4 <i>b</i> )	U <sub>iso</sub> (Å <sup>2</sup> )	0.112(14)
N (4 <i>b</i> )	U <sub>iso</sub> (Å <sup>2</sup> )	0.112
R <sub>Bragg</sub> (%)		8.74
R <sub>wp</sub> (%)		5.63

**Table S1.** Refined lattice parameters, atomic positions and atomic displacement parameters of CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub> in tetragonal *I4cm* (#108) at RT. Agreement factors (R<sub>Bragg</sub> and R<sub>wp</sub>) are also given. A minor amount (<4 wt.%) of PbI<sub>2</sub> impurity was found in these samples.

## Determination of lattice parameters and preferred orientation degree for the films



**Figure S4.** *Le Bail* profile fits (solid line) of SXRPD ( $\lambda = 0.826 \text{ \AA}$ ) data (circles) of MAPb<sub>3</sub> films on glass, glass/FTO, glass/FTO/TiO<sub>2</sub> and glass/FTO/PEDOT:PSS at RT. Tick marks indicate the position of allowed reflections in  $I4cm$ . The asterisk indicates the (110) reflection of FTO.



**Figure S5.** SXRPD ( $\lambda = 0.826 \text{ \AA}$ ) data of  $\text{MAPbI}_3$  on glass (red), glass/FTO (olive), glass/FTO/ $\text{TiO}_2$  (dark blue), glass/FTO/PEDOT:PSS (blue) and  $\text{MAPbI}_3$  powder (black) at RT. Solid black lines indicate the (220) reflection Voigt fit.

The ratio between the integrated intensity of the (220) reflection (Figure S4 and S5) and the total integrated intensity between  $14.5^\circ$  and  $19.5^\circ$  in two-theta (Figure 2 and S4) has been used as figure of merit of the 110 preferred orientation degree using the **equation S1**,

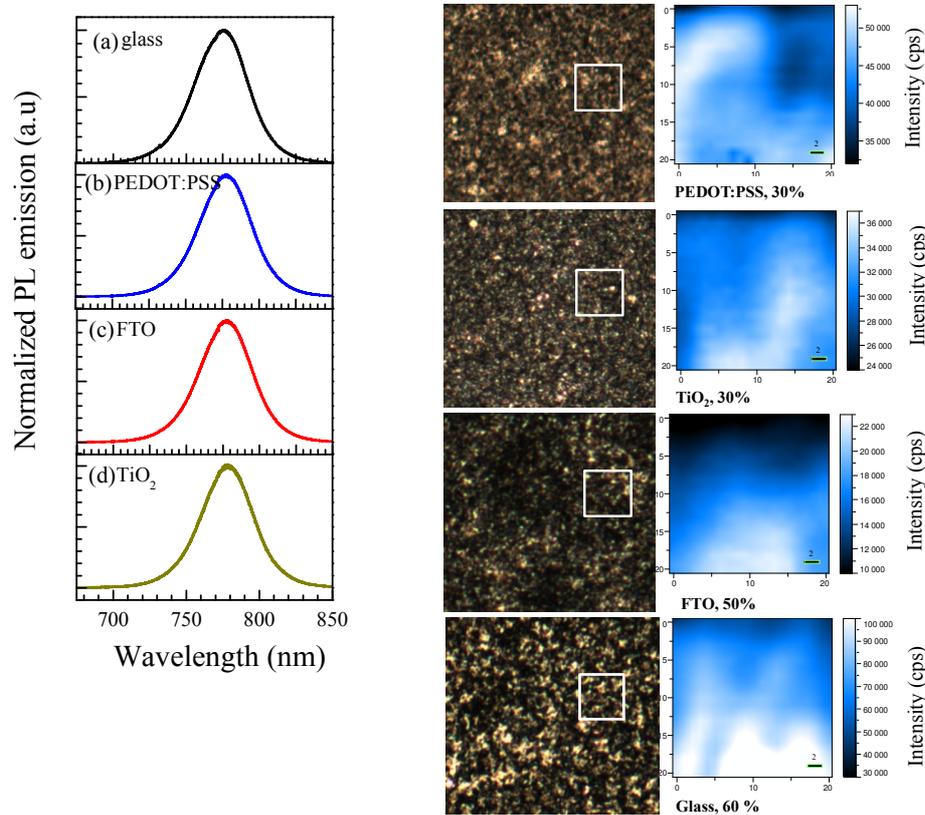
$$\left[ 1 - \frac{(A_{220}/A_{14.5^\circ-19.5^\circ})_{\text{powder}}}{(A_{220}/A_{14.5^\circ-19.5^\circ})_x} \right] \times 100$$

where  $A_{220}$  is the (220) integrated intensity,  $A_{14.5^\circ-19.5^\circ}$  represents the total integrated intensity between  $14.5^\circ$  and  $19.5^\circ$  in two-theta for  $\text{MAPbI}_3$  powder and  $\text{MAPbI}_3$  on  $x$  substrate (glass, glass/FTO, glass/FTO/ $\text{TiO}_2$ , and glass/FTO/PEDOT:PSS).

Note:  $(hkl)$  indicates a lattice plane,  $[uvw]$  a lattice direction and  $\langle uvw \rangle$  indicates a family of  $[uvw]$  directions, in that way the family of  $[110]$  directions is designated as  $\langle 110 \rangle$ .

### Photoluminescence

Prototypical steady PL emission of MAPbI<sub>3</sub>:Cl thin films on the different substrates exciting at 488 nm laser wavelength with very low power (6  $\mu$ W, 12 W/cm<sup>2</sup>) are shown in left panel of Figure S5. Taking into account reported absorption coefficient<sup>3</sup> we can assume the same sample volume evaluated for each thin film as the absorption depth for 488 nm wavelength was calculated to be around 85 nm. The band maximum arises at 775.5, 778, 775 and 778 nm for glass, PEDOT:PSS, FTO and TiO<sub>2</sub> respectively.



**Figure S6.** (Left panel) PL emission of the MAPbI<sub>3</sub>:Cl thin films deposited on glass (a), FTO/pedot:pss (b), FTO (c) and FTO/TiO<sub>2</sub> (d) compact layer. Right panel: variation of the PL intensity in a 20x20  $\mu$ m area (the white square in the corresponding microphotograph) size, showing % variation between the intensity maximum and minimum.

### Raman spectra of the MAPbI<sub>3</sub>:Cl thin films

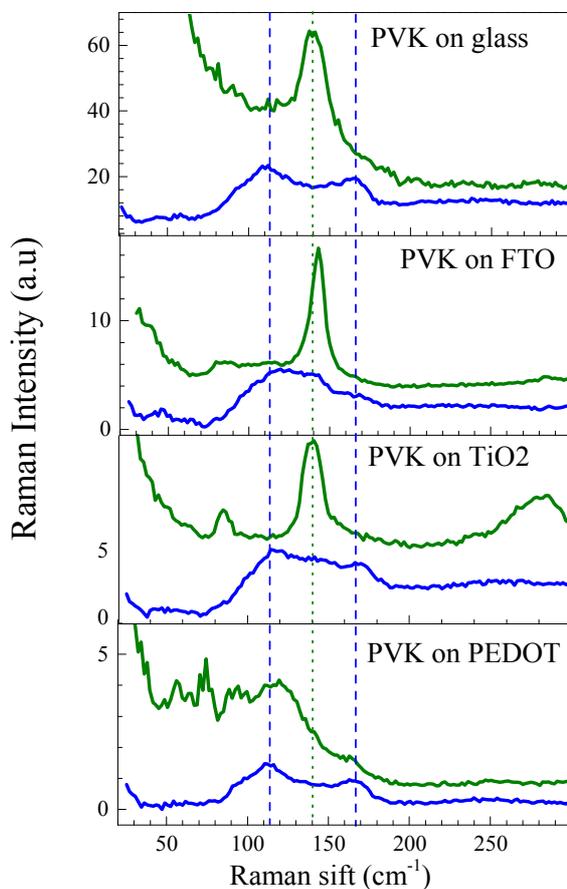
In **Table S2** are the measured low frequencies Raman modes (cm<sup>-1</sup>) at RT with 488 nm excitation of PbI<sub>2</sub>, PbO<sub>2</sub> and MAPbI<sub>3</sub>. Reported PbI<sub>2</sub> and MAPbI<sub>3</sub> modes are included for comparison. According to calculations, based on the harmonic approximation and neglecting spin-orbit coupling,<sup>4</sup> the band observed around 120-130 cm<sup>-1</sup> is due to MA normal modes coupled to some extent to the inorganic cage in the range 140-170 cm<sup>-1</sup> which are sensitive to the orientation of the MA<sup>+</sup> cation in tetragonal or orthorhombic structures. For the Raman mode at 237 cm<sup>-1</sup>, the most similar calculated frequencies are torsional modes of the organic cations predicted in the 200–370 cm<sup>-1</sup> range from more disordered or confined MA to more ordered structure (orthogonal).<sup>4,5,6</sup> It has been reported that the deformation of the MA molecule shifts the torsional mode toward higher frequencies and, on the contrary, the formation of specific hydrogen bonding interactions, typical of these compounds<sup>7,8</sup> shifts this mode toward lower frequencies.

PbI <sub>2</sub> this work	PbI <sub>2</sub> 9	PbO <sub>2</sub> This work	MAPbI <sub>3</sub> theory 4	MAPbI <sub>3</sub> This work (**)	MAPbI <sub>3</sub> 4	MAPbI <sub>3</sub> 10	MAPbI <sub>3</sub> 11	MAPbI <sub>3</sub> :Cl 6
	72	83	72 87		62 94	71 94	52	
115	95,113		141-150	127(85)	120	110	110	119
		138			154	140		150-160
210 (*)	214(*)	278	200-340	237(76)	270	270		250

**Table S2.** Measured low frequencies Raman modes (cm<sup>-1</sup>) at RT with 488 nm excitation of PbI<sub>2</sub>, PbO<sub>2</sub> and MAPbI<sub>3</sub>. PbI<sub>2</sub> and MAPbI<sub>3</sub> measured and calculated modes reported by different authors are included for comparison. (\*) only visible in resonant conditions. (\*\*) In parenthesis the FWHM of the fitted peaks in Figure 3d.

In **Figure S7** we can observe the Raman spectra for increasing incident power laser of MAPbI<sub>3</sub>:Cl films for the four different substrates in ambient conditions (a-d) at intermediate **stage** and final **stage 5**. We observe different evolution depending of the substrate. The intermediate stage (blue line) corresponds with the situation where PL emission is partially quenched. The bands around 110 cm<sup>-1</sup> and 165 cm<sup>-1</sup> are clearly detected and for higher laser power, in the final stage (green lines), the peak associated

to the formation of PbOx around 138-140  $\text{cm}^{-1}$  is dominating the spectra except for the case of PEDOT:PSS substrate.



**Figure S7.** (a-d) Micro Raman spectra for the MAPbI<sub>3</sub>:Cl thin films for the four substrates at intermediate (blue) and final (green) stages with increasing incident power.

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