Electronic Supplementary Information

Highly efficient electrocatalytic oxidation of urea on Mnincorporated Ni(OH)₂/carbon fiber cloth for energy-saved rechargeable Zn-air battery

Xian Zhang,^{ab} Guoqiang Liu,^{ab} Cuijiao Zhao,^{ab} Guozhong Wang,^a Yunxia Zhang,^a

Haimin Zhang*a and Huijun Zhaoac

^a Key Laboratory of Materials Physics, Centre for Environmental and Energy Nanomaterials, Anhui Key Laboratory of Nanomaterials and Nanotechnology, CAS Center for Excellence in Nanoscience, Institute of Solid State Physics, Chinese Academy of Sciences, Hefei 230031, China. E-mail: zhanghm@issp.ac.cn; Fax: +86 (0)551 65591434; Tel: +86 (0)551 65591973

^b University of Science and Technology of China, Hefei 230026, China.

^c Centre for Clean Environment and Energy, Griffith University, Gold Coast Campus, QLD 4222, Australia

Experimental section

Fabrication of Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC, Mn-hydroxide/CFC

electrocatalysts. All of the chemical regents were analytical grade (AR) and were used without further purification. First, CFC (carbon fiber cloth, 1.0 cm×2.0 cm) was immersed in concentrated nitric acid (65 wt.%) for 48 h to adequately remove the impurity and enhance surface functional groups. Then, the CFC was rinsed with ethanol and water adequately. The Ni²⁺, Mn²⁺ adsorbed CFC was obtained by immersing the pre-treated CFC in a solution containing Ni²⁺ and Mn²⁺ (NiCl₂:MnCl₂) with different molar ratios (1:0 or 2:1 or 0:1) for overnight at room temperature. The total amount of Ni²⁺ and Mn²⁺ in aqueous solution was 6.0 mM and the molar ratio of Ni²⁺ to Mn²⁺ was varied systematically. Subsequently, the as-obtained proportional Mn²⁺, Ni²⁺ adsorbed CFC was immersed in a 1.0 M KOH solution to convert Mn²⁺, Ni²⁺ adsorbed CFC into Ni-Mn hydroxide/CFC. The obtained electrocatalysts in the molar ratios 1:0, 2:1 and 0:1 were denoted as Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC and Mn-hydroxide/CFC respectively.

Similarly, other molar ratios (Ni²⁺/Mn²⁺=1:5, 1:2, 1:1, 5:1) Ni-Mn hydroxide/CFC were prepared. The obtained electrocatalysts in the molar ratios 1:5, 1:2, 1:1 and 5:1 were denoted as Ni-Mn hydroxide/CFC-1:5, Ni-Mn hydroxide/CFC-1:2, Ni-Mn hydroxide/CFC-1:1 and Ni-Mn hydroxide/CFC-5:1, respectively.

Fabrication of powder formed Ni-hydroxide, Mn-Ni-hydroxide and Mn-hydroxide. The powder formed Ni-hydroxide, Mn-Ni-hydroxide and Mn-hydroxide samples were prepared by one-step solution reaction route. Firstly, the proportional Ni²⁺, Mn²⁺ (NiCl₂:MnCl₂) solution was added into 1.0 M KOH solution under stirring for 30 min to obtain power hydroxides. The total amount of Ni²⁺ and Mn²⁺ in aqueous solution was 6.0 mM and the molar ratios of Ni²⁺ and Mn²⁺ were 1:0, 2:1 and 0:1 respectively. After collection by filtration, washing with deionized water and freeze drying for 24 h, the powder formed power hydroxides were obtained and denoted as Ni-hydroxide, Mn-Ni-hydroxide and Mn-hydroxide.

Characterization. The crystalline structures of samples were identified by X-ray diffraction analysis (XRD, Philips X'pert PRO) using Nifiltered monochromatic CuKa radiation ($\lambda K \alpha 1 = 1.5418$ Å) at 40 kV and 40 mA. The morphology and structure of samples were characterized by field emission scanning electron microscopy (FESEM, Quanta 200FEG) and transmission electron microscopy (TEM, JEOL 2010) with an energy dispersive X-ray spectrometer (EDS Oxford, Link ISIS).

X-ray photoelectron spectroscopy (XPS) analysis was performed on an ESCALAB 250 X-ray photoelectron spectrometer (Thermo, America) equipped with Al K $\alpha_{1,2}$ monochromatized radiation at 1486.6 eV X-ray source. Fourier transform infrared (FT-IR) spectroscopy of the sample was performed on a Perkin-Elmer TGA7 infrared spectrometer to identify the functional groups of the sample. Quantitative determination of the metal ions (Ni and Mn) content was performed by inductively coupled plasma (ICP, Thermo, America).

Electrochemical measurements. Electrochemical measurements were performed on an electrochemical workstation (CHI 760D, CH Instruments, Inc., Shanghai, China) coupled with a PINE rotating disk electrode (RDE) system (Pine Instruments Co. Ltd. USA). A standard three-electrode electrochemical cell equipped with gas flow system was employed during measurements. The potentials *vs*. Ag/AgCl were converted to the reversible hydrogen electrode (RHE) using the following relationship.

$$E_{\rm RHE} = E_{\rm Ag/AgCl} + 0.059 \rm pH + E^{\circ}_{\rm Ag/AgCl}$$

where E_{RHE} is the converted potential *vs*. RHE, $E^{\circ}_{\text{Ag/AgCl}} = 0.199 \text{ V}$ at 25°C, and $E_{\text{Ag/AgCl}}$ is the experimentally measured potential against Ag/AgCl reference.

The UOR and OER polarization curves were measured with a scan rate of 5.0 mV s^{-1} at room temperature in 1.0 M KOH with or without 0.5 M urea, respectively. All potentials were without *iR*-compensated. And the presented current density was normalized to the geometric surface area. All the polarization curves were the steady-state ones after several cycles. The electrochemical impedance spectroscopy (EIS)

measurements were performed by applying an AC voltage with 10 mV amplitude in a frequency range from 100,000 to 1 Hz and recorded at 1.45 V vs. RHE electrode.

ORR measurements: The Mn-Ni(OH)₂/CFC electrode was first grinded into powder form. 4.0 mg Mn-Ni(OH)₂/CFC powder catalyst was dispersed ultrasonically in 1000 μ L of solvent mixture of Nafion (5%), absolute ethanol, deionized water (V: V: V=1: 1: 8), to obtain a well-dispersed catalyst ink. Subsequently, 20 μ L of catalyst ink was cast onto the glassy carbon (GC) electrode surface and dried at room temperature. RDE experiments were carried out in an O₂-saturated 0.1 M KOH solution at room temperature, a sweep rate of 10 mV s⁻¹ under different rotating rates (400, 625, 900, 1225 and 1600 rpm). The electron transfer number (*n*) *per* oxygen molecule at different potentials during the oxygen reduction reaction (ORR) was calculated based on the Koutecky-Levich (K-L) equation:

$$J^{-1} = J_{L}^{-1} + J_{k}^{-1} = J_{K}^{-1} + (B\omega^{1/2})^{-1}$$
(1)

$$B = 0.2nF(D_0)^{2/3}v^{-1/6}C_0$$
⁽²⁾

$$J_k = nFkC_0 \tag{3}$$

where *J* is the measured current density, $J_{\rm L}$ and $J_{\rm K}$ are the diffusion- and kinetic-limiting current densities, ω is the rotation speed, *n* represents the number of electrons transferred *per* oxygen molecule, *F* is the Faraday constant (96485 C mol⁻¹), C_0 is the O₂ concentration in the electrolyte (1.26×10⁻⁶ mol cm⁻³), D_0 is the diffusion coefficient of O₂ in the electrolyte (1.9×10⁻⁵ cm² s⁻¹), and *v* is the kinetic viscosity (0.01 cm² s⁻¹).

Zinc-air batteries measurement: The measurements of Zn-air batteries were performed on home-built electrochemical cells. All data were collected from the as-fabricated cell with a Zahner-Zennium electrochemical workstation at room temperature. Briefly, Zn foil was used as anode and Mn-Ni(OH)₂/CFC catalysts loaded on the gas diffusion layer was used as the air cathode. The electrolyte was 1.0 M KOH+0.2 M Zn(Ac)₂ with or without 0.5 M urea. The Mn-Ni(OH)₂ mass loading on CFC was about 0.18 mg cm⁻².



Fig. S1 FT-IR spectrum of CFC after HNO3 treatment and Mn-Ni(OH)2/CFC



Fig. S2 XRD patterns of powder formed Ni-hydroxide, Mn-Ni-hydroxide and Mn-hydroxide powers.

As shown in Fig. S2, pure Ni-hydroxide sample shows the diffraction peaks at 18.9° , 33.2° , 38.6° , 59.2° and $62,7^{\circ}$, corresponding to the (001), (100), (101), (110) and (111) crystalline planes of hexagonal Ni(OH)₂ (PDF no 00-001-1047), and pure Mn-hydroxide can be attributed to the $(Mn_{4.2x}Mn_x)Mn_8O_{16-x}(OH)_x$ phase (PDF no 00-012-0252), demonstrated that the Mn-Hydroxide/CFC was Mn oxyhydroxide obtained by partially dehydrated by Mn hydroxide.



Fig. S3 SEM images of Ni(OH)₂/CFC (a), (b); Mn-Ni(OH)₂/CFC (c), (d); Mnhydroxide/CFC (e), (f).

Fig. S3 shows the surface SEM images of the as-prepared Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC and Mn-hydroxide/CFC samples. As shown, Ni(OH)₂/CFC shows a small nanoparticle structure (Fig. S3a and b). Also, the Mn-Ni(OH)₂/CFC surface presents a nanoparticle morphology adhered on the CFC surface (Fig. S3c and d);

however, Mn-hydroxide/CFC displays a characteristic nanosheet structure (Fig. S3e and f).

In our study, ICP measurements were performed to obtain the component content information of samples. The ICP results indicate that the molar ratio of Ni (0.13 mg cm⁻¹) to Mn (0.057 mg cm⁻¹) in Mn-Ni(OH)₂/CFC is about 2.1:1, confirming the dominance of Ni component in sample, consistent with the results of XRD and electron microscopy characterizations.



Fig. S4 Transmission electron microscopy-energy dispersive spectroscopy (TEM-EDS) elemental mapping images of Mn-Ni(OH)₂/CFC.



Fig. S5 (a) UOR polarization curves with and without *iR* compensation of CFC, $Ni(OH)_2/CFC$, $Mn-Ni(OH)_2/CFC$ and Mn-hydroxide/CFC in 1.0 M KOH electrolyte containing 0.5 M urea. (b) UOR polarization curves without *iR* compensation of different molar ratio of Ni^{2+}/Mn^{2+} prepared samples in 1.0 M KOH electrolyte containing 0.5 M urea.



Fig. S6 OER polarization curves without *iR* compensation without *iR* compensation of different molar ratio of Ni²⁺/Mn²⁺ prepared samples in 1.0 M KOH electrolyte.



Fig. S7. Cyclic voltammetry curves of (a) Ni(OH)₂/CFC, (b) Ni-Mn hydroxide/CFC-5:1, (c) Mn-Ni(OH)₂/CFC, (d) Ni-Mn hydroxide/CFC-1:1 (e) Ni-Mn hydroxide/CFC-1:2, (f) Ni-Mn hydroxide/CFC-1:5.

As we known, the surface concentration of redox active Ni centers can be extracted from the slope of the linear relationship between the peak current of the Ni^{III}/Ni^{III}

reduction wave and the scan rate:

$$slope = n^2 F^2 A \Gamma o / 4RT$$

where n = 1, F is Faraday constant (96485 C mol⁻¹), A is the surface area of the electrode (1.0 cm²), Γ o is surface concentration (mol cm⁻²), R is ideal gas constant (8.314 J mol⁻¹ K⁻¹), T is temperature (298 K). From the figure S8, it can be seen the Ni(OH)₂/CFC displays the largest slope value, and with the Mn content increasing, the slope value gradually decreased. Therefore it is reasonable to believe that Ni(OH)₂/CFC possesses more Ni-related catalytic active sites, thus obtaining higher OER performance in comparison with other ratio electrocatalysts.



Fig. S8 EIS of CFC, Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC and Mn-hydroxide/CFC measured in 1.0 M KOH and 0.5 M urea electrolyte applying an AC voltage with 10 mV amplitude in a frequency range from 100,000 to 1 Hz and recorded at 1.45 V *vs.* RHE.

As shown in Fig. S8, the smallest series resistance ($R_s=2.25 \Omega$) and charge transfer resistance ($R_{ct}=2.75 \Omega$) can be achieved for Mn-Ni(OH)₂/CFC, revealing its superior electrical conductivity and UOR activity by optimum Mn incorporation.



Fig. S9 (a) XPS survey spectra of fresh Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC after UOR testing and Mn-Ni(OH)₂/CFC after OER testing. (b) high resolution Mn 2p XPS spectrum of fresh Mn-Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC after UOR testing and Mn-Ni(OH)₂/CFC after OER testing. (c) high resolution Ni 2p XPS spectrum of fresh Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC, Mn-Ni(OH)₂/CFC after UOR testing and Mn-Ni(OH)₂/CFC after OER testing.



Fig. S10 XRD patterns of powder Mn-Ni(OH)₂ after OER/UOR testing.



Fig. S11 Durability test for the Mn-Ni(OH)₂/CFC and Pt/C samples in 1.0 M KOH solution with 0.5 M urea.



Fig. S12 (a) LSV curves of, pure CFC and Mn-Ni(OH)₂/CFC in O₂-saturated 0.1 M KOH solution at a scan rate of 10 mV s⁻¹ and a rotation speed of 1600 rpm, (b) LSV curves of Mn-Ni(OH)₂/CFC obtained at different rotating rates in O₂-saturated 0.1 M KOH solution at a scan rate of 10 mV s⁻¹, (d) Koutecky-Levich (K-L) plots of Mn-Ni(OH)₂/CFC derived from (b), (d) electron transfer number (n).

The ORR activity of Mn-Ni(OH)₂/CFC was therefore evaluated in 0.1 M KOH with a PINE rotating disk electrode (RDE) system. The LSV curves of Mn-Ni(OH)₂/CFC obtained from an O₂-saturated 0.1 M KOH solution at different rotation rates are shown in Fig. S12b. As shown, the investigated electrocatalyst exhibits the onset potential of ca. 0.83 V (vs. RHE), and the limiting current density increases with the rotation rate. Based on Fig. S12b, the derived Koutecky–Levich (K-L) plots are shown in Fig. S12c, exhibiting a good linear relationship. On the basis of the K–L plots, the average value of the number of transferred electrons (Fig. S12d) within 0.27-0.47 V (vs. RHE) was calculated to be 3.7 for Mn-Ni(OH)₂/CFC, indicating a near four electron ORR process.



Fig. S13 The charging energy saving efficiency curves of Zn-air battery with 0.5 M urea.



Fig. S14 (a) The Zn-air battery assembled by Mn-Ni(OH)₂/CFC as air cathode. (b) The urea-assisted charging Zn-air battery assembled by Mn-Ni(OH)₂/CFC as air cathode. (c) In charging process, the charging voltage of a rechargeable Zn-air battery is compensated by the potential difference between oxidation of urea and water.



Fig. S15 Recycling durability test of home-made rechargeable Zn-air battery.

References	Catalyst	Potential at 10	Mass	Electrolyte
		mA cm ⁻² (V <i>vs</i> .	loading	
		RHE)	$(mg cm^{-1})$	
Angew. Chem.,	S-MnO ₂	1.33	1.5	1 M KOH
2016, 128, 3868				+ 0.5 M
				Urea
Chem. Commun.,	Pt	1.50	2.5	5 M KOH
2009, 32, 4859				+ 0.33 M
				Urea
Angew. Chem.	M-Ni(OH) ₂	1.38	0.535	1 M KOH
Int. Ed., 2016, 55,				+ 0.33 M
12465				Urea
J. Power Sources,	NiO nanosheet	1.38	0.27	1 M KOH
2014, 272, 711	array			+ 0.33 M
				Urea
Electrochim.	NiMo sheet array	1.37	0.35	1 M KOH
Acta, 2015, 153,				+ 0.33 M
456				Urea
Sci. Rep., 2014,	NiCo alloy	1.53	10	1 M KOH
4, 5863				+ 0.33 M
				Urea
This work	Mn-Ni(OH) ₂ /CFC	1.30	0.18	1 M KOH
				+ 0.5 M
				Urea

 Table S1. Comparison of the UOR activities for recently reported highly active
 electrocatalysts.