# **Supporting information**

# Exploration of VPO<sub>4</sub> as a new anode material for sodium-ion batteries

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#### **Experimental Section**

#### Synthesis of VPO<sub>4</sub>@C composite.

Stoichiometric ratio (1:1:0.2:0.5) of vanadium (III) acetylacetonate ( $C_{10}H_{14}O_5V$ ), ammonium phosphate (( $NH_4$ )<sub>2</sub>HPO<sub>4</sub>), glucose ( $C_6H_{12}O_6$ ) and oxalic acid ( $C_2H_2O_4$ ) were sequentially dissolved in deionized water under continuous stirring and heating. The  $C_{10}H_{14}O_5V$  was added as vanadium source and carbon source simultaneously. After forming a sol precursor, then the mixed solution was continuous stirred and heated at 80°C to achieve the gel, followed by drying in a vacuum oven at 100 °C overnight (freeze drying). Finally, the as-prepared powders were obtained by sintering at 300°C for 3 hours and 900°C for 1 hours in Ar/H<sub>2</sub> (95:5) atmosphere.

### Material characterizations

X-ray diffraction (XRD) was collected with Bruker Advance D8 diffractometer using Cu Kα source. Thermogravimetric analysis (TGA) was carried out by a Mettler Toledo TGA/DSC-1100. Raman test was conducted by a LabRAM Aramis spectrophotometer with a laser wavelength of 532 nm. X-ray photoelectron (XPS) was recorded on a LabRAM Analyzer with wavelength of 514 nm. Scanning electron microscopy (SEM) and Transmission electron microscopy (TEM) were performed on JEOL JSM-7500FA and JEOL-2011 at 200 kV equipped with energy-dispersive Xray spectroscopy (EDS), respectively.

The *in-situ* XRD measurement was recorded by home-design cell and as reported previously.<sup>S1,S2</sup> The cell was made of stainless steel and inset with an internal slot with 12-mm diameter, while the configuration was illustrated in Fig.S1. Specifically,

the beryllium foil served as transition window to allow X-ray passage, and the carbon paper acted as a current collector. The electrode was obtain by mixing the active material (VPO<sub>4</sub>/C) and polyvinylidene fluorid binder (PVDF) with ratio of 8:2, which were homogeneously dispersed by the N-methylpyrrolidinone (NMP) solvent and then cast into the carbon paper. Each XRD pattern was performed in the step incremental of 0.02° and scaned between  $2\theta=10^{\circ}-60^{\circ}$  at the rate of 0.08° s<sup>-1</sup>. There was 30 s for interval for each required pattern. The corresponding charge/discharge measurement was carried out at the rate of 100 mA g<sup>-1</sup>, ensuring that at least 40 scans can be recorded for a entire sodiation/desodiation cycle.

#### **Electrochemical measurements**

The as-prepared electrodes were prepared by mixing VPO<sub>4</sub> (70%), carbon black (20%), and PVDF (10%) in NMP solvent. The slurry was coated on copper foil, followed by drying at vacuum oven. The electrodes were punched into round disks with 1.0-1.2 mg cm<sup>-2</sup> loading of active material, and then assembled in the glovebox by sodium tablet, glass fiber (Whatman GF/D) and 1.0 M NaCF<sub>3</sub>SO<sub>3</sub> dissolved in diethylene glycol dimethylether (DEGDME) as reference/counter electrode, separator and electrolyte, respectively. The galvanostatic tests were conducted on a Land System (CT2001A) in the potential range of 0.01-3.0 V. Cyclic voltammogram (CV) at a scan rate of 0.1 mV s<sup>-1</sup> and electrochemical impedance spectroscopy (EIS) measurements over the frequency range from 100 kHz to 0.01 Hz were performed on a CHI660E electrochemical work-station and IM6 (Zahner) electrochemical station, respectively.



Fig. S1. The configuration of *in-situ* battery cell.



Fig. S2. Rietveld refinement XRD pattern of bare VPO<sub>4</sub> (A) and VPO<sub>4</sub>@C (B).

Table S1. Experimental lattice parameters calculated from the Rietveld refinement for bare  $VPO_4$  and  $VPO_4@C$  composites.

Samples	a (Å)	b (Å)	c (Å)	$V(Å^3)$	$\mathbf{R}_{\mathrm{wp}}$	R <sub>p</sub>
VPO <sub>4</sub>	5.2212093	7.7634015	6.2760865	254.39705	25.67	20.13
VPO <sub>4</sub> @C	5.2284616	7.7828670	6.2657795	254.96974	20.32	16.59

Table S2. Structural parameters calculated from the Rietveld refinement for bare

 $VPO_4$ 

Atom	site	Х	У	Z	Occ.
V1	4a	0.00000	0.00000	0.00000	1
P1	4c	0.00000	0.35055	0.25000	1
01	8g	0.24720	0.47002	0.25000	1
02	8f	0.00000	0.24518	0.03811	1

Table S3. Structural parameters calculated from the Rietveld refinement for the

VPO<sub>4</sub>@C

Atom	site	Х	у	Z	Occ.
V1	4a	0.00000	0.00000	0.00000	1
P1	4c	0.00000	0.35497	0.25000	1
01	8g	0.24826	0.46801	0.25000	1
02	8f	0.00000	0.24148	0.04237	1



**Fig. S3.** TGA measurement of bare VPO<sub>4</sub>, and VPO<sub>4</sub>@C composites in the temperature range of 30-700 °C in the flowing of air atmosphere (A), XRD patterns for final product of VPO<sub>4</sub> sintered at 700 °C under air atmosphere (B).

TGA test is operated in air flow to calculate the carbon content of VPO<sub>4</sub>@C (Fig. S3A). The apparent increasing mass of bare VPO<sub>4</sub> is corresponded to the oxidation of VPO<sub>4</sub> to VOPO<sub>4</sub> (Fig. S3B). In contrast, the rapid mass loss starting from 400°C for VPO<sub>4</sub>@C sample is related to the removal of carbonous materials.



Fig. S4. Raman spectrum of bare VPO<sub>4</sub> and VPO<sub>4</sub>@C composites.



Fig. S5. XPS full pattern (A) and core level of V 2p (B) for VPO<sub>4</sub>@C composites.



Fig. S6. XPS core level of V 2p (A) and P 2p (B) for bare VPO<sub>4</sub>.

It is clearly observed that the binding energy of P 2p spectrum for bare VPO<sub>4</sub> is 133.9 eV, which is higher than that of VPO<sub>4</sub>@C composite. The difference of P 2p spectrum between VPO<sub>4</sub> and VPO<sub>4</sub>@C is ascribed to that the P 2p spectrum of bare VPO<sub>4</sub> can only be fitted into one peak, which is assigned to the P-O bond. The result is consistent with other literatures.<sup>S3,S4</sup>



Fig. S7.  $N_2$  adsorption–desorption isotherms and (inset) the pore size distributions of bare VPO<sub>4</sub>, and VPO<sub>4</sub>@C composite.

According to the Brunauer-Emmett-Teller (BET) measurements, the corresponding specific surface area of the VPO<sub>4</sub>@C is 13.51 m<sup>2</sup> g<sup>-1</sup>, which is larger than that of VPO<sub>4</sub> ( $6.22 \text{ m}^2 \text{ g}^{-1}$ ). For VPO<sub>4</sub>@C composite, the larger surface area with porous structure is beneficial to Na<sup>+</sup> transportation and electrolyte diffusion.



Fig. S8. SEM (A, B), TEM and (C) HRTEM (D) images of VPO<sub>4</sub>.



**Fig. S9.** The CV plots of  $VPO_4$  in the first 4 cycles between 0.01 V and 3.0 V at a scanning rate of 0.1 mV s<sup>-1</sup>.



Fig. S10. The charge/discharge curves of  $VPO_4$  in the first 4 cycles between 0.01 V and 3.0 V at rate of 50 mA g<sup>-1</sup>.



**Fig. S11.** Performance comparison of VPO<sub>4</sub>@C with other recently reported anodes for SIBs

	Rate capability	Cycling stability	Reference	
Sample	Capacity/current	Capacity/current/cycles	s	
	$(mAh g^{-1}/mA g^{-1})$	$(mAh g^{-1}/mA g^{-1}/n)$		
VPO <sub>4</sub> @C	204.8/2000	245.3/1000/200	This work	
С	100/2000	160/100/100	S5	
TiO <sub>2</sub>	82.7/2000	160/20/50	<b>S</b> 6	
$V_2O_5$	140/1280	177/40/100	<b>S</b> 7	
NaV <sub>3</sub> (PO <sub>4</sub> ) <sub>3</sub>	107/2200	126/220/100	<b>S</b> 8	
$Na_3V_2(PO_4)_3$	103/117	136/12/50	S9	
NaTi <sub>2</sub> (PO <sub>4</sub> ) <sub>3</sub>	85/2660	77/1330/1000	S10	
Na <sub>2</sub> Ti <sub>3</sub> O <sub>7</sub>	71/885	125/35.4/50	S11	
NaAlTi <sub>3</sub> O <sub>3</sub>	65/250	62/25/100	S12	
Na <sub>2.65</sub> Ti <sub>3.35</sub> Fe <sub>0.65</sub> O <sub>9</sub>	74.2/100	110/40/100	S13	
$Na_{2/3}Ni_{1/6}Mg_{1/6}Ti_{2/3}O_2$	41/96	80.4/9.6/100	S14	

**Table S4.** Comparison of electrochmical performance for  $VPO_4@C$  with other materials reported elsewhere as the anode materials for SIBs.



**Fig. S12.** The recorded impedance spectra of VPO<sub>4</sub> (A) and VPO<sub>4</sub>@C (B) before and after various cycles, linear fitting to Z' versus  $\omega^{-1/2}$  plots in the low-frequency range (C), equivalent circuit used for fitting the experimental EIS data (D).

Samples	Cycle number, n	R <sub>s</sub> , Ω	$R_{ct}, \Omega$	$\sigma_{w}, \Omega \ s^{-1}$	$D_{Na}$ , cm <sup>2</sup> s <sup>-1</sup>
VPO <sub>4</sub>	Pristine	13.01	4.26	235.5	8.19×10 <sup>-15</sup>
	10	14.83	19.15		
	20	14.98	64.52		
	50	14.68	93.25		
VPO4@C	Pristine	8.61	2.58	138.9	2.35×10 <sup>-14</sup>
	10	8.97	4.15		
	20	8.84	5.46		
	50	9.32	14.17		

**Table S5.** Result of electrochemical impedance and Warburg coefficient in FigureS13.



**Fig. S13.** Selected 2 $\theta$  regions plot of *in-situ* XRD results of VPO<sub>4</sub>/C electrode against the voltage profile during the initial cycle.

For *in-situ* XRD measurement, the cell was galvanostatic charged/discharged at a current rate of 100 mA g<sup>-1</sup>, while each XRD patterns were collected at different desodiation/sodiation state in the first cycle and stacked together sequently. Meanwhile, the coresponding contour plot of *in-situ* patterns is displayed in Fig. 4A. The phase compositions are color-coded to have a better distinction for desodiation/sodiation process. The red color means low intensity, and the blue color means high intensity, which are shown in the right side of Fig. 4A. It is better to observe the phase transformation by combination of charge/discharge curves and XRD pattern at various desodiation/sodiation states.

The intermediate phase is a new material with peaks located at 20 of 24° and  $30.5^{\circ}$ , which are neither Na<sub>3</sub>PO<sub>4</sub> nor VPO<sub>4</sub>, probably ascribed to the combination of Na<sup>+</sup> and VPO<sub>4</sub> and formation of Na<sub>x</sub>V(PO<sub>4</sub>)<sub>y</sub>. The Na<sub>x</sub>V(PO<sub>4</sub>)<sub>y</sub> could be indexed into the Na<sub>3</sub>V<sub>3</sub>(PO<sub>4</sub>)<sub>3</sub> or Na<sub>4</sub>V<sub>2</sub>(PO<sub>4</sub>)<sub>3</sub>, which can be found in the other polyanion anode materals of sodiated NaV<sub>3</sub>(PO<sub>4</sub>)<sub>3</sub> <sup>S15,S16</sup> and Na<sub>3</sub>V<sub>2</sub>(PO<sub>4</sub>)<sub>3</sub>, <sup>S17,S18</sup> respectively. However, the intermediate Na<sub>x</sub>V(PO<sub>4</sub>)<sub>y</sub> is metastable and hardly to determine the accurate

structure, and this phenomenon usually occurs in the halfway of sodiation/desodiation process.<sup>S15-S18</sup> Under the limit of experiment condition, other characterizations should be carried out to verify the intermediate phase.

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