

Supporting information

Tailoring the structure and thermoelectric properties of BaTiO₃ via Eu²⁺ substitution

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Polar modes comprise all the symmetry-breaking atomic displacements causing ferroelectric transitions, normally cation off-centerings with respect to their coordination cage, which relate to the centre (Γ point) of the Brillouin zone; to non-polar modes, or zone-boundary modes (R point), belong, for example, the rotational modes that cause the antiferrodistortive transition in both EuTiO₃ and SrTiO₃. The Γ_4^- transformation, which is responsible for the ferroelectric transition in BaTiO₃¹, can lead to the space groups *P4mm*, *Amm2*, or *R3m* depending on the direction of the order parameter, i.e. the off-centering of Ti or Ba/Eu. This means that an electric dipole could form, respectively, in the directions $\langle 100 \rangle$, $\langle 110 \rangle$, or $\langle 111 \rangle$ with respect to the parent cubic structure.

A pure BaTiO₃ sample has been prepared and the transport properties have been evaluated under the same conditions as the substituted ones. Regarding the very large electrical resistivity of BaTiO₃ at room temperature, it was not possible to measure the electrical transport properties below 473 K with our ZEM measurement system. The electrical conductivity of BaTiO₃ was around 16 S/m at 1123 K, which is extremely lower than that of our Eu²⁺ substituted samples. Compared with Eu²⁺ substituted samples, BaTiO₃ sample possesses a much higher Seebeck coefficient in the entire investigated temperature range. The carrier concentration of BaTiO₃ was estimated to be $8.5 \times 10^{18} \text{ cm}^{-3}$ at 1123 K according to Heikes formula, and the calculated carrier mobility was $0.13 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, which is consistent with reference data ^{2,3}. As shown in Figure 1(a), the thermal conductivity of BaTiO₃ sample

possess a similar trend as $\text{Ba}_{1-x}\text{Eu}_x\text{TiO}_3$ ($x \leq 0.3$). The transition around 390 K is due to both a phase transition of tetragonal to cubic and a Curie transition⁴. The lattice thermal conductivity of BaTiO_3 (Figure 2(b)) also follows the trend which we expected.

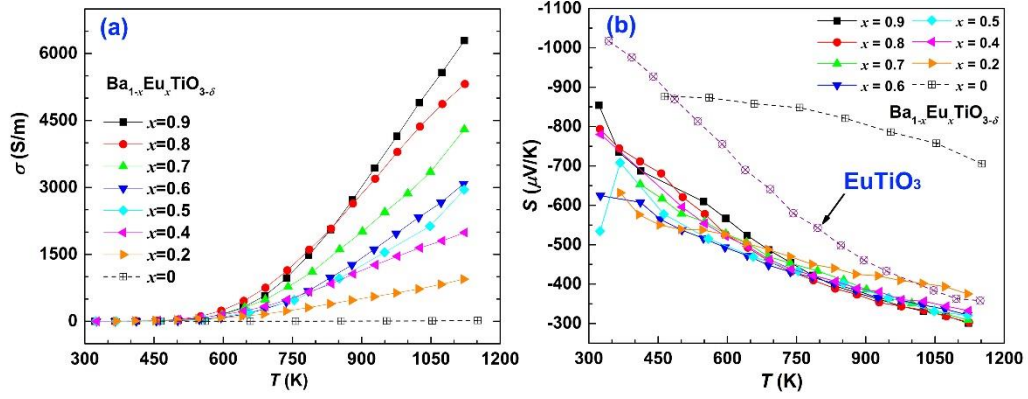


Fig.S1. Temperature dependence of the electrical conductivity (a) and Seebeck coefficient (b) of $\text{Ba}_{1-x}\text{Eu}_x\text{TiO}_{3-\delta}$ samples

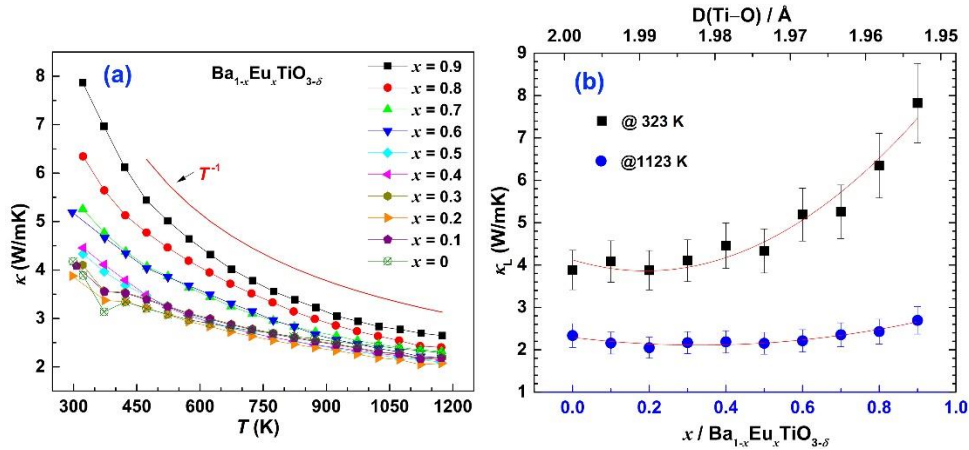


Fig.S2. Temperature dependence of the thermal conductivity κ (a) and the lattice thermal conductivity κ_L (b) of $\text{Ba}_{1-x}\text{Eu}_x\text{TiO}_{3-\delta}$ as a function of the Eu^{2+} content x and Ti–O distance at 323 K and 1123 K.

References

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