## Facile preparation of high-quality perovskites for efficient solar cells

## via a fast conversion of wet PbI<sub>2</sub> precursor films

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## **Supplementary Figures**



**Fig. S1.** Cross-sectional SEM images of the annealed MAPbI<sub>3</sub> films on mesoporous  $TiO_2$  substrate prepared from their respective PbI<sub>2</sub> precursor films. The scale bars are 500 nm. The thickness of a MAPbI<sub>3</sub> capping layer on the top of the mp-TiO<sub>2</sub> prepared from c-PbI<sub>2</sub>, p-PbI<sub>2</sub>, and w-PbI<sub>2</sub> precursor film were determined to be 200, 197 and 185 nm, respectively. The increased thickness of capping layer of c-PbI<sub>2</sub>-based perovskite film can be attributed to volume expansion occurred in the two-step sequential deposition process based on well-crystallized and compact PbI<sub>2</sub>, which is because of the growth of perovskite crystals with the intercalation of organic iodides into PbI<sub>2</sub> skeleton.<sup>1, 2</sup>



**Fig. S2.** Grain size distribution of the SEM images to the left. The perovskite films were prepared from c-Pbl<sub>2</sub> (a1), w-Pbl<sub>2</sub> (b1) and p-Pbl<sub>2</sub> precursor films (c1), respectively. The scale bars in the SEM images are 200 nm. The average grain size of perovskites prepared with c-Pbl<sub>2</sub>, w-Pbl<sub>2</sub> and w-Pbl<sub>2</sub> precursor films are determined to be 127.1±64.7, 172.7±74.7 and 148.6±71.7 nm, respectively.



Fig. S3. Top-view SEM images of the as-prepared annealing-free perovskite films prepared from their respective  $PbI_2$  precursor films. The scale bars are 1 $\mu$ m.



**Fig. S4.** AFM topographical 3D images (2  $\mu$ m ×2  $\mu$ m) of our perovskite films fabricated from their respective PbI<sub>2</sub> precursor films. The perovskite film prepared from annealing-free, wet PbI<sub>2</sub> (w-PbI<sub>2</sub>) is uniform and ultra-smooth surface (RMS is 19.5 nm) due to the direct substitution of DMF molecules by MAI molecules in wet PbI<sub>2</sub> precursor film.



**Fig. S5.** Box charts of photovoltaic parameters for the perovskite solar cells fabricated with different MAPbI<sub>3</sub> films prepared from the various  $PbI_2$  precursor films. The data were obtained from 20 individual cells.



**Fig. S6.** (a) The J–V curves of a best MAPbI<sub>3</sub> solar cell prepared from w-PbI<sub>2</sub> precursor film with the highest efficiency of 15.49% ( $J_{sc}$ =20.74 mA/cm<sup>2</sup>,  $V_{oc}$ =1.004 V, FF=0.744) under simulated 1 sun illumination. (b) Stability of power conversion efficiency as a function of time for the same cell measured at 0.84 V under simulated 1 sun illumination. (c) The external quantum efficiency (EQE) spectrum. The Jsc value is consistent with the EQE spectrum. It is worth to mention that there are reports showing that the integrated  $J_{sc}$  from EQE doesn't always match well with Jsc extracted from J-V measurement, which is presumably related to the mismatch (approximately 10%) between the EQE spectra and the photon flux AM 1.5 G spectrum,<sup>3</sup> and the difference in device characteristics (e.g., stability during measurement outside the glovebox) for solar cells.<sup>2, 4</sup>

PbI <sub>2</sub> precursor films	$PbI_2$ (001) peak	MAPbI <sub>3</sub> (110) peak	
		Annealing-free	Annealed
c-PbI <sub>2</sub>	0.414	0.525	0.369
w-PbI <sub>2</sub>	0.472	0.470	0.270
p-PbI <sub>2</sub>	0.382	0.495	0.362

**Table S1.** The full width at half maximum (FWHM) of MAPbI<sub>3</sub> (110) diffraction peak and PbI<sub>2</sub> (001) diffraction peak extracted from the XRD patterns (Fig. 3 in manuscript).

It is well known that the smaller FWHM of diffraction peak generally means the higher crystallinity. For w-PbI<sub>2</sub> film, the largest FWHM (0.472) of PbI<sub>2</sub> (001)diffraction peak indicates the lowest crystallization, which is attributed to that the presence of high boiling DMF solvent in the as-prepared wet PbI<sub>2</sub> film retards the growth of PbI<sub>2</sub> crystals. After annealing at 70 °C for 30 min, the crystallinity of PbI<sub>2</sub> film increases, as the decreased FWHM (0.414) of c-PbI<sub>2</sub> film, which is due to the evaporation of solvent DMF molecules. The lowest FWHM (0.382) of PbI<sub>2</sub> (001) diffraction peak presents the highest crystallization of PbI<sub>2</sub> crystals in the p-PbI<sub>2</sub> film, which can be ascribed to an effective extraction of residual DMF by an anti-solvent of IPA molecules. For annealing-free MAPbI<sub>3</sub> films, tens of seconds at room temperature is effective and sufficient to transform PbI<sub>2</sub> into well-crystallized MAPbI<sub>3</sub> films with the lowest FWHM (0.382) of MAPbI<sub>3</sub> (110) diffraction peak, which can be attributed to the retarded crystallization of w-PbI<sub>2</sub> and a direct molecular exchange of DMF molecules intercalated in PbI<sub>2</sub> with MAI. Upon annealing, the FWHM of MAPbI<sub>3</sub> (110) diffraction peak for all samples significantly decreases, which indicates the increased crystallinity of annealed perovskites.

## References

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