## General solution-processed formation of porous transition-metal oxides on exfoliated molybdenum disulfides for highperformance asymmetric supercapacitors<sup>†</sup>

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## Electronic Supplementary Information

## Figure caption:

**Fig. S1.** (a) Digital images showing the as-obtained (left) and diluted (right) suspension of  $f-MoS_2$  in NMP. (b) Digital images showing the as-obtained (left) and diluted (right) suspension of  $f-MoS_2$  in DI water upon the solvent-exchange method.

Fig. S2. The direct sonication of  $b-MoS_2$  in water cannot produce exfoliated  $f-MoS_2$  suspension.

**Fig. S3.** (a) Nitrogen sorption isotherms and (b) pore size distributions for  $Ni(OH)_2$  and NiO. (c) Nitrogen sorption isotherms and (d) pore size distributions for  $MoS_2$ -NiO,  $MoS_2$ -Ni(OH)<sub>2</sub> and f-MoS<sub>2</sub>.

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**Fig. S20.** Comparison of galvanostatic charge/discharge curves of  $MoS_2$ -NiO, NiO and f-MoS<sub>2</sub> at a current density of 0.5 A g<sup>-1</sup>.

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**Fig. S24.** CV curves of (a)  $f-MoS_2$  and (b)  $Co_3O_4$  at different scan rates, and the galvanostatic discharge curves of (c)  $f-MoS_2$ , (d)  $Co_3O_4$  and (e)  $MoS_2-Co_3O_4$  at different current densities.

**Fig. S25.** CV curves of (a) f-MoS<sub>2</sub>, (b)  $Fe_2O_3$  and (c)  $MoS_2$ - $Fe_2O_3$  at different scan rates. Galvanostatic charge/discharge curves of (d) f-MoS<sub>2</sub>, (e)  $Fe_2O_3$  and (f)  $MoS_2$ - $Fe_2O_3$  at different current densities.

**Fig. S26.** (a) Comparison of CV curves of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub> and f-MoS<sub>2</sub> at a scan rate of 5 mV s<sup>-1</sup>. (b) Specific capacitances of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub> and f-MoS<sub>2</sub> at various current densities. (c) Cycling performance of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub> and Fe<sub>2</sub>O<sub>3</sub> at a current density of 2 A g<sup>-1</sup>.

Supplementary information (ESI) for Journal of Materials Chemistry This journal is <sup>6</sup> The Royal Society of Chemistry 2010 **Table S1.** The BET surface areas and pore size distributions of Ni(OH)<sub>2</sub> and MoS<sub>2</sub>-Ni(OH)<sub>2</sub>.

Samples	$S_{BET}$ $[m^2 g^{-1}]$	Total pore volume [mL g <sup>-1</sup> ]	Micropore volume [mL g <sup>-1</sup> ]	Mesopore volume [mL g <sup>-1</sup> ]
Ni(OH) <sub>2</sub>	39.9	0.10	~ 0	0.10
MoS <sub>2</sub> -Ni(OH) <sub>2</sub>	45.6	0.14	~ 0	0.14

**Table S2.** Comparison of electrochemical capacitive performances of MoS<sub>2</sub>-NiO hybrids with NiO-based hybrids in other literatures.

Electrode materials	Specific capacitance		Ref.
	(F g <sup>-1</sup> )		
	1 A g <sup>-1</sup>	20 A g <sup>-1</sup>	
NiO@PPy	595	401.2	[S1]
NiO nanotube arrays	523.4	230.1	[S2]
MoS <sub>2</sub> @Ni(OH) <sub>2</sub>	657	242.2	[S3]
MoS <sub>2</sub> @PANI	864.5	263.2	[S4]
NiO coated graphene/PANI	1260.2	521	[S5]
CNT arrays@NiO	1120.2	385.2	[S6]
CNT@NiO	1098.4	516.8	[S7]
MoS <sub>2</sub> -NiO-2	1080.6	668.4	This work



Fig. S1. (a) Digital images showing the as-obtained (left) and diluted (right) suspension of  $f-MoS_2$  in NMP. All the two samples are left standing for 2 weeks before taking the photograph. (b) Digital images showing the as-obtained (left) and diluted (right) suspension of  $f-MoS_2$  in DI water upon the solvent-exchange method, which are left standing for 24 h before taking the photograph.



Fig. S2. The direct sonication of  $b-MoS_2$  in water cannot produce exfoliated  $f-MoS_2$  suspension.



**Fig. S3.** (a) Nitrogen sorption isotherms and (b) pore size distributions for  $Ni(OH)_2$  and NiO. (c) Nitrogen sorption isotherms and (d) pore size distributions for  $MoS_2$ -NiO,  $MoS_2$ -Ni(OH)<sub>2</sub> and f-MoS<sub>2</sub>.



**Fig. S4.** (a) XRD patterns of b-MoS<sub>2</sub> and f-MoS<sub>2</sub>. (b) XRD patterns of MoS<sub>2</sub>-Ni(OH)<sub>2</sub>, Ni(OH)<sub>2</sub> and f-MoS<sub>2</sub>. For the intermediates of MoS<sub>2</sub>-Ni(OH)<sub>2</sub> hybrids, Ni(OH)<sub>2</sub> shows XRD patterns centered at  $2\theta = 18.5$ , 31.3, 38.8, 51.5, 58.3, 62.1, 69.7 and  $71.9^{\circ}$ , which can be ascribed to the (002), (100), (101), (102), (110), (111), (103) and (112) planes, respectively, validating the formation of  $\alpha$ -phase hexagonal structure of Ni(OH)<sub>2</sub> (JCPDS card No.14-0117).



**Fig. S5.** XPS spectra of (a) survey and (b) O 1s of the MoS<sub>2</sub>-NiO-2 sample. A binding energy peak at 529.0 eV corresponds to O atoms from NiO, while the peak located at 530.5 eV can be ascribed to the absorbed  $O_x^-$  ions (O<sup>-</sup> and  $O_2^-$  ions) in the oxygendeficient regions within the NiO.



Fig. S6. SEM image of b-MoS<sub>2</sub>.



Fig. S7. TEM images of f-MoS $_2$  at (a) low and (b) high magnifications, respectively.



Fig. S8. SEM images of (a, b) MoS<sub>2</sub>-NiO-1 and (c, d) MoS<sub>2</sub>-NiO-3 at low and high magnifications, respectively.



Fig. S9. SEM images of NiO at (a) low and (b) high magnifications, respectively.



**Fig. S10.** SEM image and the corresponding elemental mappings of MoS<sub>2</sub>-NiO-2 sample.



Fig. S11. TEM images of MoS<sub>2</sub>-Co<sub>3</sub>O<sub>4</sub> hybrid at different magnifications.



**Fig. S12.** SEM images of  $Co_3O_4$  at (a) low and (b) high magnifications, respectively, which show a nanowire morphology (50 ~ 70 nm in diameter; ~ 500 nm in length) with staggered distributions.



**Fig. S13.** (a) XRD patterns of MoS<sub>2</sub>-Co(OH)<sub>2</sub>, Co(OH)<sub>2</sub> and f-MoS<sub>2</sub>. (b) XRD patterns of MoS<sub>2</sub>-Co<sub>3</sub>O<sub>4</sub>, Co<sub>3</sub>O<sub>4</sub> and f-MoS<sub>2</sub>. The diffraction peaks of the as-obtained products with sharp and slender peaks are well indexed into brucite  $\beta$ -Co(OH)<sub>2</sub> (JCPDS card No. 45-0031) for the intermediates and cubic Co<sub>3</sub>O<sub>4</sub> (JCPDS card No. 42-1467) for the final products, which imply their good crystallinity within the hybrids.



Fig. S14. TEM image of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub> at low magnification.



Fig. S15. SEM images of  $Fe_2O_3$  at (a) low and (b) high magnifications, respectively. Neat  $Fe_2O_3$  shows a nanoparticle morphology with a uniform size.



**Fig. S16.** XRD patterns of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub> and f-MoS<sub>2</sub> before (a) and after (b) annealing. XRD patterns of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub> hybrids before and after annealing are both indexed well with that of monoclinic Fe<sub>2</sub>O<sub>3</sub> (JCPDS no. 33-0664). This might be due to the instability of the Fe(OH)<sub>3</sub>, which will be directly converted into Fe<sub>2</sub>O<sub>3</sub> during the reaction process, but the post-annealing process makes the Fe<sub>2</sub>O<sub>3</sub> in the hybrids with higher crystallinity.



Fig. S17. Comparison of CV curves of graphite paper current collector and  $MoS_2$ -NiO-2 at a scan rate of 50 mV s<sup>-1</sup>.



**Fig. S18.** (a) CV curves of NiO at various scan rates, and (b) plot of the current densities of the cathodic peak for the  $MoS_2$ -NiO-2 and NiO electrodes as functions of the square roots of the scan rates.



Fig. S19. CV curves of (a) MoS<sub>2</sub>-NiO-2, (b) f-MoS<sub>2</sub>, and (c) NiO electrodes in electrochemical double-layer region at scan rates of 5, 10, 20, 50, 75 and 100 mV s<sup>-1</sup>.
(d) The ratio of the discharge current at 0.05 V with various scan rates of MoS<sub>2</sub>-NiO-2, f-MoS<sub>2</sub>, and NiO electrodes.



**Fig. S20.** Comparison of galvanostatic charge/discharge curves of MoS<sub>2</sub>-NiO, NiO and f-MoS<sub>2</sub> at a current density of 0.5 A g<sup>-1</sup>. For the NiO and MoS<sub>2</sub>-NiO electrodes, two variation regions are observed in the charge/discharge curves, in which a linear variation of potential/time dependence (below ~ 0.37 V) indicates an entire double layer capacitive behavior from the charge separation at the electrode/electrolyte interface. By contrast, a potential plateau of potential vs. time (0.25 ~ 0.35 V) indicates a typical pseudocapacitive behavior, which is caused by a redox reaction at the electrode/electrolyte interface. Meanwhile, the ideal straight line for the f-MoS<sub>2</sub> electrode demonstrates a typical double layer capacitive behavior.



**Fig. S21.** Comparison of galvanostatic charge/discharge curves of  $MoS_2$ -NiO-2 at various current densities. The potential plateau can be observed in all charge/discharge curves which are in good accordance with the data from the CV measurements. The specific capacitances are calculated by the following equation:

$$C = \frac{It}{mV}$$

where I is the charge/discharge current, t is the discharge time, V is the voltage range, and m is the mass of the active materials in the electrode.



Fig. S22. Nyquist plots of MoS<sub>2</sub>-NiO, NiO and f-MoS<sub>2</sub> electrodes in the frequency ranging from 1 MHz to 0.01 Hz (inset is the enlargement of the high-frequency region). No semicircles can be observed in high-frequency regions for all the electrodes, indicating extremely low charge-transfer resistances (R<sub>et</sub>) of the Faradaic process for all electrodes. The intercept of the curve at Z'-axis at the high frequency demonstrates the resistance of the electrochemical system (R<sub>s</sub>, which includes the intrinsic resistance of the substrate, the ionic resistance of electrolyte, and the contact resistance between current collector and active material). The MoS<sub>2</sub>-NiO-2 electrode displays the lowest R<sub>s</sub> (0.38  $\Omega$ ) among all the electrodes (0.42, 0.55, 0.65 and 1.6  $\Omega$ for MoS<sub>2</sub>-NiO-1, MoS<sub>2</sub>-NiO-3, f-MoS<sub>2</sub> and NiO, respectively), suggesting that the MoS<sub>2</sub>-NiO-2 electrode owns a lower internal resistance. MoS<sub>2</sub>-NiO-2 electrode shows a more ideal linear in the low-frequency region, which indicates a more efficient electrolyte and proton diffusion. The above results distinctly demonstrate that the rational construction of heterostructured MoS<sub>2</sub>-NiO electrode shows the favorable charge-transfer kinetics, the full utilization of the electroactive materials, and the expedite penetration of the electrolyte within the electrode, which adequately explain the increase of the electrochemical performances of the as-synthesized hybrids as described above.



**Fig. S23.** (a) Comparison of CV curves of  $MoS_2$ - $Co_3O_4$ ,  $Co_3O_4$  and f- $MoS_2$  at 5 mV s<sup>-1</sup>. (b) CV curves of  $MoS_2$ - $Co_3O_4$  at different scan rates. (c) Specific capacitances of  $MoS_2$ - $Co_3O_4$ ,  $Co_3O_4$  and f- $MoS_2$  at various current densities. (d) Cycling performance of  $MoS_2$ - $Co_3O_4$  and  $Co_3O_4$  at 2 A g<sup>-1</sup>.



Fig. S24. CV curves of (a)  $f-MoS_2$  and (b)  $Co_3O_4$  at different scan rates, and the galvanostatic discharge curves of (c)  $f-MoS_2$ , (d)  $Co_3O_4$  and (e)  $MoS_2-Co_3O_4$  at different current densities.



**Fig. S25.** CV curves of (a) f-MoS<sub>2</sub>, (b) Fe<sub>2</sub>O<sub>3</sub> and (c) MoS<sub>2</sub>-Fe<sub>2</sub>O<sub>3</sub> at different scan rates. Galvanostatic charge/discharge curves of (d) f-MoS<sub>2</sub>, (e) Fe<sub>2</sub>O<sub>3</sub> and (f) MoS<sub>2</sub>-Fe<sub>2</sub>O<sub>3</sub> at different current densities. The MoS<sub>2</sub>-Fe<sub>2</sub>O<sub>3</sub> hybrids show the largest loop area, indicating their superior electrochemical capacitive performance compared with neat f-MoS<sub>2</sub> and Fe<sub>2</sub>O<sub>3</sub>. The f-MoS<sub>2</sub> has a rectangular shape, while there is a pair of anodic and cathodic peaks at about -0.6 and -1.05 V, respectively, in the Fe<sub>2</sub>O<sub>3</sub> and MoS<sub>2</sub>-Fe<sub>2</sub>O<sub>3</sub> electrodes, which can be assigned to the redox process of reversible reaction between Fe<sup>2+</sup> and Fe<sup>3+</sup>, respectively. The MoS<sub>2</sub>-Fe<sub>2</sub>O<sub>3</sub> electrode presents a typical pseudocapacitive behavior with highly nonlinear charge/discharge curves.



**Fig. S26.** (a) Comparison of CV curves of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub> and f-MoS<sub>2</sub> at a scan rate of 5 mV s<sup>-1</sup>. (b) Specific capacitances of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub> and f-MoS<sub>2</sub> at various current densities. (c) Cycling performance of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub> and Fe<sub>2</sub>O<sub>3</sub> at a current density of 2 A g<sup>-1</sup>. The specific capacitances of  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub> electrodes are 647.8 and 362.5 F g<sup>-1</sup> at current densities of 1 and 20 A g<sup>-1</sup>, respectively, indicating an excellent rate performance exceeding neat Fe<sub>2</sub>O<sub>3</sub> and f-MoS<sub>2</sub>. The cycling stabilities of Fe<sub>2</sub>O<sub>3</sub> and  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub> electrodes are tested up to 6000 charge/discharge cycles. Neat Fe<sub>2</sub>O<sub>3</sub> as electrode materials can hardly achieve a satisfying cycling stability, but a rational combination of f-MoS<sub>2</sub> and Fe<sub>2</sub>O<sub>3</sub> into unique hybrids reaches a significantly enhanced cycling stability. For  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub> hybrids, the capacitance retention of the  $MoS_2$ -Fe<sub>2</sub>O<sub>3</sub> electrode decreases gradually to ~ 96% after 6000 cycles, while neat Fe<sub>2</sub>O<sub>3</sub> electrode displays a continuous decrease to ~ 54% after 6000 cycles.

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