

# **Simple Structured Polyetheramines, Jeffamines, as Efficient Cathode Interfacial Layers for Organic Photovoltaics Providing Power Conversion Efficiencies up to 9.1%**

Bing-Huang Jiang, Ya-Juan Peng, Chih-Ping Chen\*

Department of Materials Engineering, Ming Chi University of Technology, 84 Gunjuan Road, Taishan, New Taipei City, 24301, Taiwan

## **Supporting Information**

### *Device fabrication and characterization*

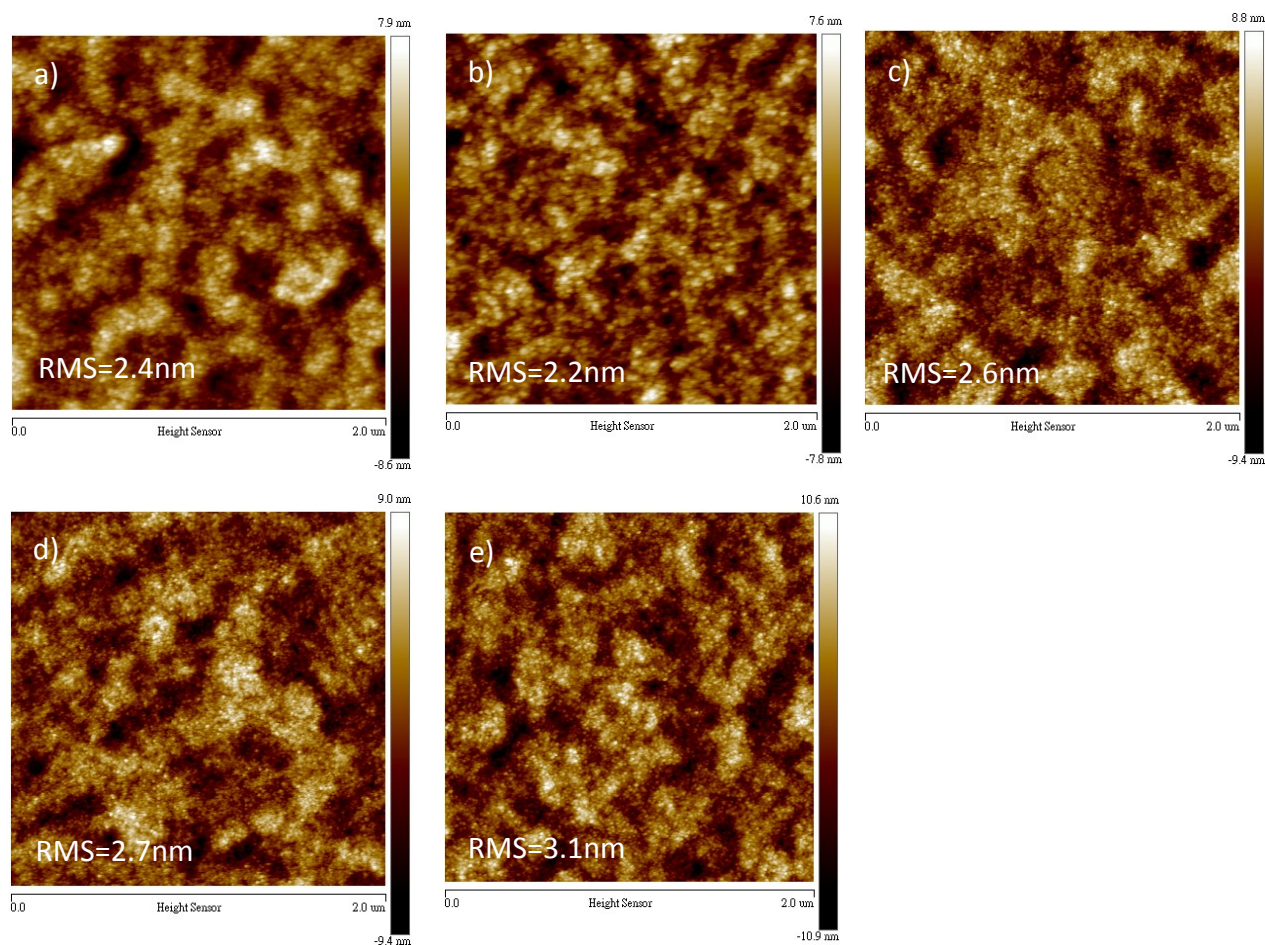
ITO glass substrates [Sanyo, Japan ( $8 \Omega/\text{square}$ )] were patterned lithographically, washed (detergent), treated ultrasonically [using acetone, water, and isopropyl alcohol (IPA)], dried ( $140 \text{ }^\circ\text{C}$ , 10 min), and further cleaned with  $\text{O}_2$  plasma for 5 min. The zinc oxide (ZnO) layer (50 nm) was prepared from a 0.5 M zinc acetate precursor solution in 2-methoxyethanol (2-ME). The ZnO was annealed at  $160 \text{ }^\circ\text{C}$  for 30min in air prior to use. Solutions of Jeffamine materials were prepared by dissolving each polymer in 2-ME to achieve a desired concentration. These solutions were deposited on top of the ZnO layers through spin-coating (5000 rpm) in air and then the samples were dried ( $100 \text{ }^\circ\text{C}$ , 10 min) in a glove box. Active layer solution of PTB7 and  $\text{PC}_{71}\text{BM}$  (1:1.5) (10mg/ml for PTB7) was stirred in chlorobenzene (CB) (with 3vol% of 1,8-diodooctane (DIO)) overnight, filtered through a polytetrafluoroethylene (PTFE; 0.2-

$\mu\text{m}$ ) filter, and then spin-coated (1000 rpm, 30 s) onto the ZnO layer (with or without IFLs). All devices were completed by depositing layers of MoO<sub>3</sub> (5 nm) and Ag (100 nm) at pressures below 10<sup>-6</sup> torr. Each device had an active area of 10 mm<sup>2</sup>. Cell performances were measured in a glove box. The current density–voltage ( $J$ – $V$ ) curves of the devices were determined using a computer-controlled Keithley 2400 source measurement unit and a Newport solar simulator (Oriel® Sol2A Class ABA) with AM 1.5 G illumination (1000 W m<sup>-2</sup>). A standard Si secondary reference cell cover with a KG-5 filter was used to calibrate the illumination intensity. IPCEs were measured using an Enlitech QE-R spectral response measurement system to calibrate the current densities of the devices. The morphologies of the blend films were analyzed through AFM, using a Bruker Dimension Edge microscope operated in dynamic force mode at ambient temperature. We fabricated these SCLC electron-only devices using the optimal blend film conditions. When we applied a sufficient voltage to these devices, the transport of electrons through the BHJ layer was limited by the accumulated space-charge. The equation of SCLC is described by

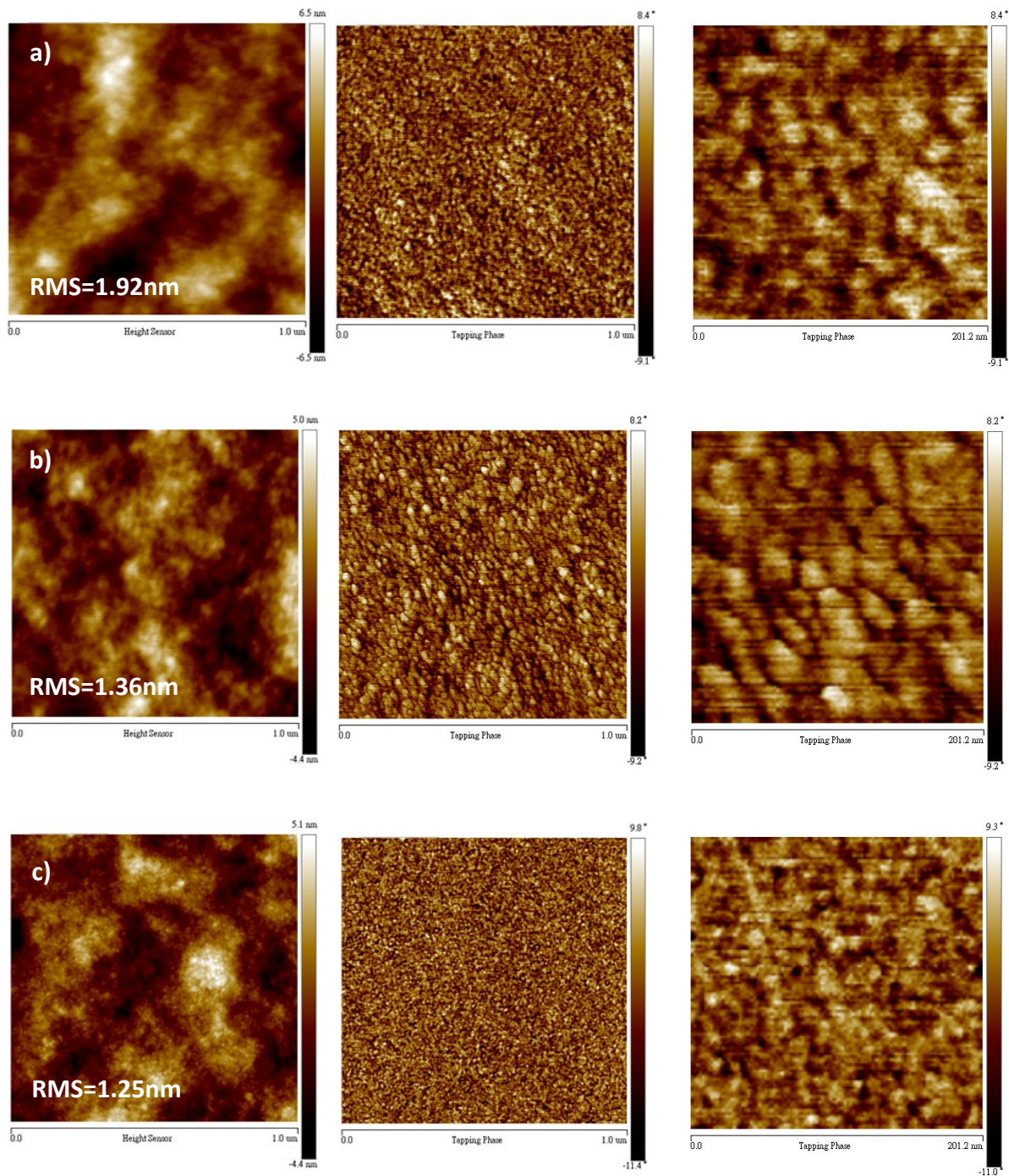
$$J = \frac{9}{8} \varepsilon_r \varepsilon_0 \mu_e \frac{V^2}{L^3}$$

where  $\varepsilon_r$  is the dielectric constant of PCBM (3.9),  $\varepsilon_0$  is the permittivity of free space ( $8.85 \times 10^{-12}$  F m<sup>-1</sup>),  $\mu_e$  is the electron mobility,  $V$  is the device's applied voltage, and  $L$  is the active layer thickness (120 nm). The work function of the ZnOs surfaces were

measured by an ultraviolet photoelectron spectrometer (UPS) using a ULVAC-PHI PHI 5000 Versaprobe II system with a windowless He discharge light source that provided He(I) emission at 21.2 eV. The take-off angle of the photoelectron was set at 90° and a pass energy of 0.58 eV was used. The samples were biased at -5 V dc to drive low-energy secondary electrons into the detector.



**Figure S1.** Tapping-mode AFM images of (a) a ZnO film and (b–e) ZnO films modified with (b) M2005, (c) M2070, (d) D2000, and (e) ED2003.



**Figure S2.** AFM topographic ( $1 \mu\text{m} \times 1 \mu\text{m}$ ) and phase ( $1 \mu\text{m} \times 1 \mu\text{m}$ ) and enlarge phase images ( $200 \text{ nm} \times 200 \text{ nm}$ ) images of a) M2005, b) M2070, and c) ED2003.