

# **New monolayer ternary In-containing sesquichalcogenides BiInSe<sub>3</sub>, SbInSe<sub>3</sub>, BiInTe<sub>3</sub>, and SbInTe<sub>3</sub> with high stability and extraordinary piezoelectric properties**

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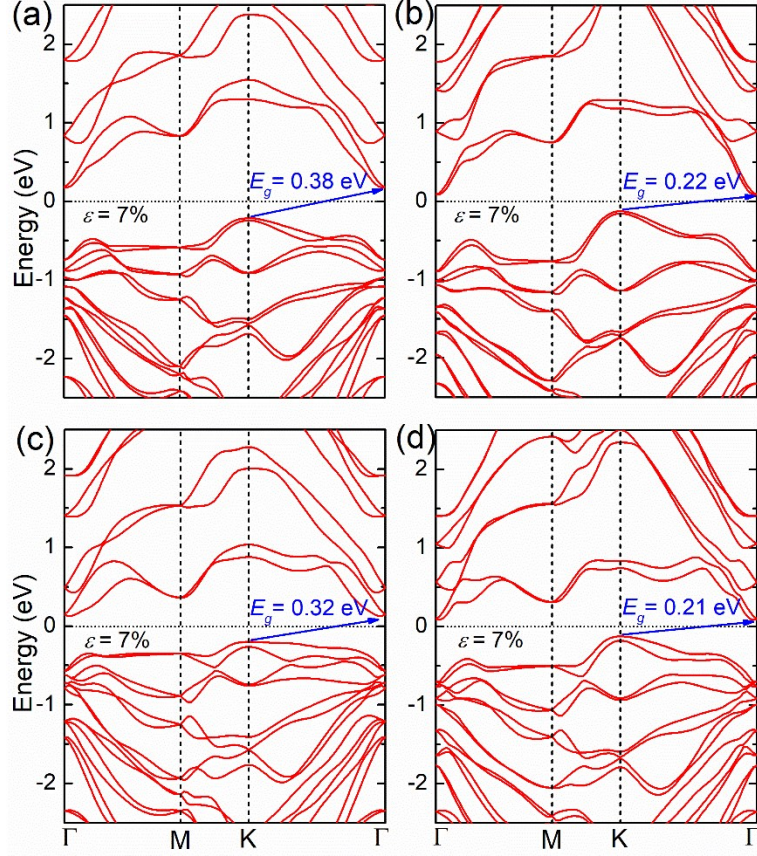
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## Computational details for piezoelectric coefficients using VASP code.

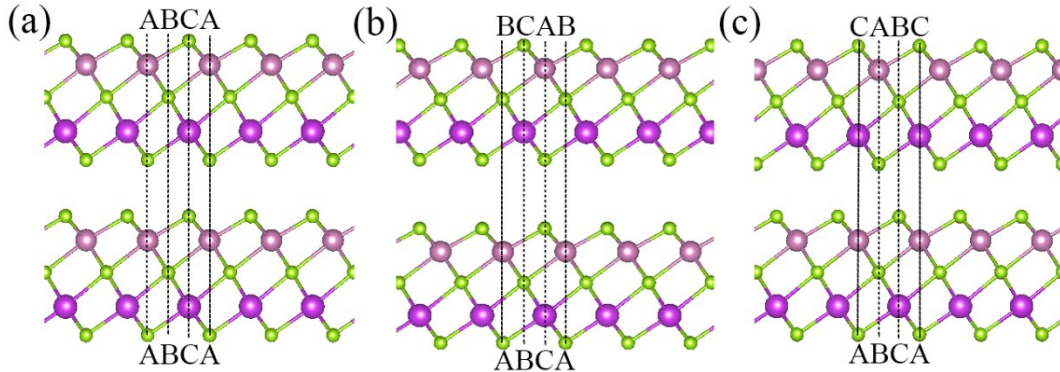
According to Eqn (1) and (2) in the main text, we obtain the piezoelectric coefficients  $e_{ijk}$  and  $d_{ijk}$  from the polarization tensor  $P_i$ , strain tensor  $\varepsilon_{jk}$ , and stress tensor  $\sigma_{jk}$ . The indices  $i=1, 2$ , and  $3$  for polarization tensor  $P_i$  indicate the polarization components along  $x$ -,  $y$ -, and  $z$ -directions, respectively. In Voigt notation, for  $\varepsilon_{jk}$  and  $\sigma_{jk}$  tensors, the indices  $j$  and  $k$  can also be labeled as  $1 = xx$ ,  $2 = yy$ ,  $3 = zz$ ,  $4 = yz$ ,  $5 = zx$ , and  $6 = xy$ . However, in 2D materials, we only consider the in-plane strains and stresses, such as those with  $1 (xx)$ ,  $2 (yy)$ , and  $6 (xy)$  components. Therefore, the second-rank  $\varepsilon_{jk}$  tensors of quasi-2D systems under study can be indicated as only three forms, i.e.,  $\varepsilon_{11}$ ,  $\varepsilon_{22}$ , and  $\varepsilon_{12}$ , where  $1$  or  $2$  corresponds to  $x$ - or  $y$ -direction as defined in Figure 1 in the main text and the relationship of  $\varepsilon_{12} = \varepsilon_{21}$  could exist. Therefore, the 2D elastic and piezoelectric tensors become simplified forms of  $C_{ij}$  and  $e_{ij}$ . In the DFT calculations performed using VASP code, we renormalize the 2D coefficients  $C_{ij}^{2D}$  and  $e_{ij}^{2D}$  by multiplying the lattice parameter along the  $z$ -direction because of the existence of vacuum space in the direction that is perpendicular to the 2D layer.

The modern theory of polarization based on the Berry's phase approximation, which has been proved to be the most effective method in evaluating the piezoelectricity of 2D materials, are applied to calculate relaxed-ion piezoelectric coefficients,  $e_{ij}$ , including both the ionic and electronic contributions. By definition, we perform the polarization vector calculations by using the **LCALCPOL** tag in VASP code. Once we obtain the polarization changes under a very small strain range, a linear fitting method is used to calculate the piezoelectric coefficients  $e_{11}$  and  $e_{31}$ , which reflect the polarization changes  $\Delta P_1$  along the  $x$ -direction and  $\Delta P_3$  along the  $z$ -direction under the in-plane uniaxial strains  $\varepsilon_{11}$  along the  $x$ -direction, respectively. Because of the noncentrosymmetric  $C_{3v}$  point group symmetry of these four monolayer ternary compounds, BiInSe<sub>3</sub>, SbInSe<sub>3</sub>, BiInTe<sub>3</sub>, and SbInTe<sub>3</sub>, only the piezoelectric  $e_{11}$  and  $e_{31}$  coefficients are non-zero and independent. As described in Eqn (3) and (4) in the main text, the relationships of  $e_{12} =$

$$-e_{11}, e_{32} = e_{31}, \text{ and } e_{26} = -\frac{e_{11}}{2} \text{ could exist.}$$



**Fig. S1** PBE+SOC band structures of (a) BiInSe<sub>3</sub>, (b) SbInSe<sub>3</sub>, (c) BiInTe<sub>3</sub>, and (d) SbInTe<sub>3</sub> ternary monolayers under an in-plane 7% tensile strain. The strain is defined as  $\varepsilon = (a - a_0)/a_0$ , where  $a_0$  is the lattice constant in plane and  $a$  is that under external strain. The band gaps are highlighted in blue. The Fermi level is at 0 eV.



**Fig. S2** Three types of binary stackings: (a) AA, (b) AB, and (c) AC. The atomic layers in one unit cell are illustrated as A, B, and C.

**Table S1** Calculated averaged relaxed-ion piezoelectric coefficients  $e_{11}/N$  and  $e_{31}/N$  of bilayer  $\text{BiInTe}_3$  with three different stacking models.  $N$  is the layer number. The coefficient is in unit of

$10^{-10} \text{ C/m}$ .

	stacking models	$N$	$e_{11}/N$	$e_{31}/N$
monolayer	A	1	22.83	1.55
bilayer	AA	2	22.42	1.44
	AB	2	22.94	1.53
	AC	2	22.75	1.57