

Supporting Information

High-Performance Lead-free Piezoelectrics with Local Structural Heterogeneity

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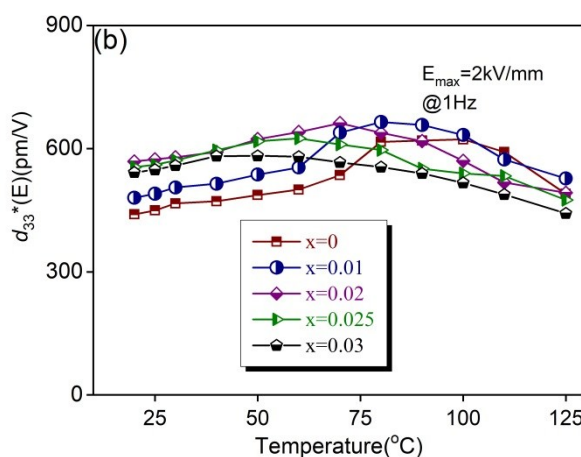
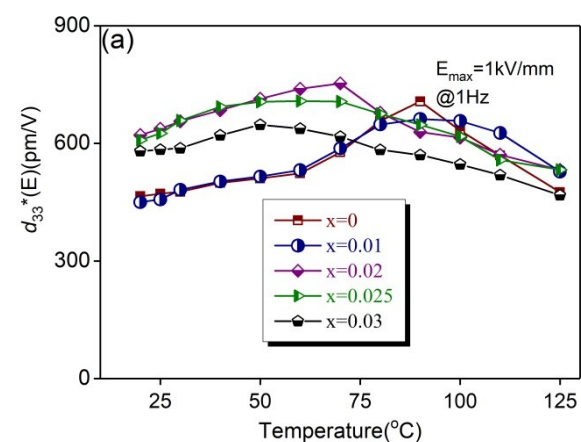


Figure S1. The temperature stability of the d_{33}^* calculated by S_{\max}/E_{\max} , $E_{\max}=1\text{kV/mm}$ (a) and $E_{\max}=2\text{kV/mm}$ (b) for the $\text{KNNS}_x\text{-5BZ-2BNH-1Mn}$ samples.

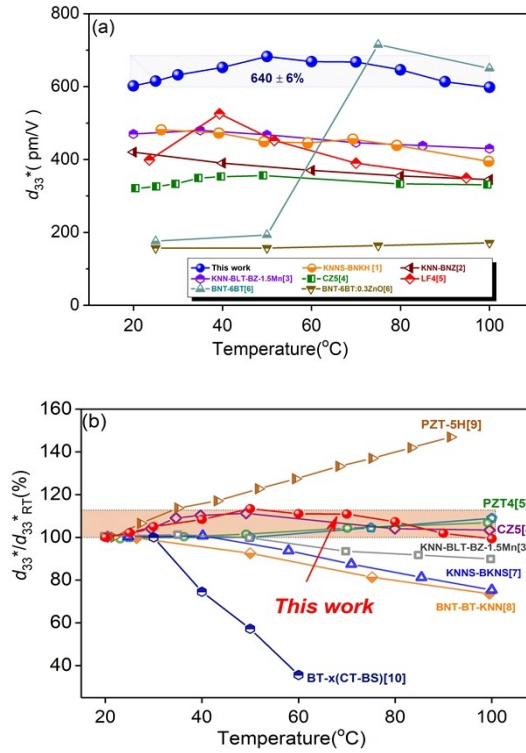


Figure S2. (a). Comparison of piezoelectric strain coefficient d_{33}^* from room temperature to 100 $^{\circ}\text{C}$ under certain electric fields of several representative systems. [1-6] (b). Comparison of the temperature dependences of large signal d_{33}^* for various ceramics as normalized to its room temperature value $d_{33}^*_{\text{RT}}$. [3-5, 7-10]

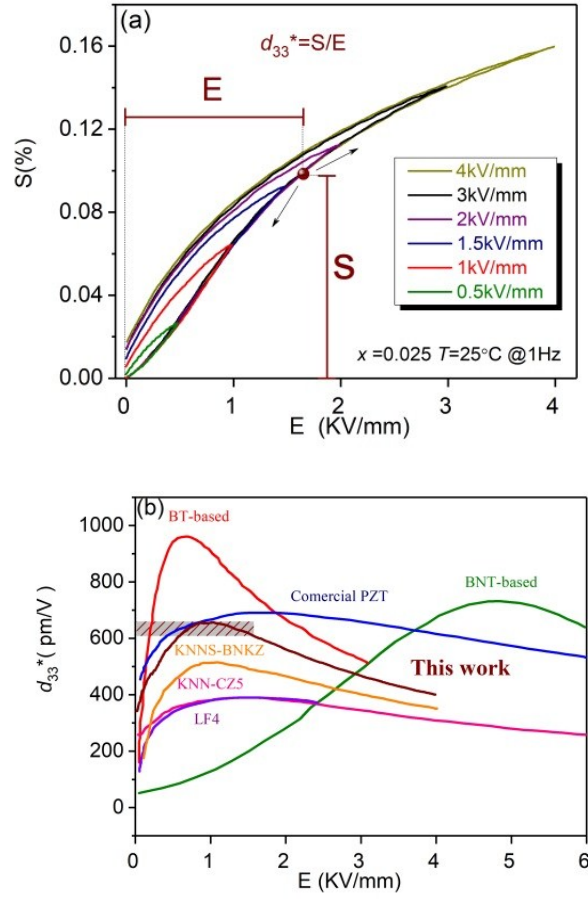


Figure S3. (a) Unipolar electric-field-induced strain curves for the $x=0.025$ sample under different electric fields. (b) The converse d_{33}^* at room temperature as a function of field for this work and other representative lead-free piezoelectric as well as commercial PZT (PIC 151). [4, 5, 7, 11-16]

To give further insights into the relationship between d_{33}^* and electric field, the converse d_{33}^* as a function of electric field was calculated by S/E according to the S-E curve measured under a triangular-shaped base waveform with $E_{\text{max}}=4\text{kV/mm}$ at room temperature for this work as shown in **Figure S3(a)**, and the relationship between d_{33}^* and electric field for other representative lead-free piezoelectrics as well as commercial PZT (PIC 151) were obtained from literature as shown in **Figure S3(b)**. [4,

5, 7, 11-16] A large d_{33}^* above 600pm/V could be obtained at the low electric field ($1 \text{ kV/mm} \leq E \leq 1.5 \text{ kV/mm}$) in this work, which is in analogy with PZT, while a larger driving electric field is expected in BNT-based ceramics. [17] BT-based ceramics show a higher level d_{33}^* but the application temperature range could be limited by their low T_c . Considering the high level of the d_{33}^* and the reliable feature in analogy with PZT, this material exhibits the promising potential in the actuator applications.

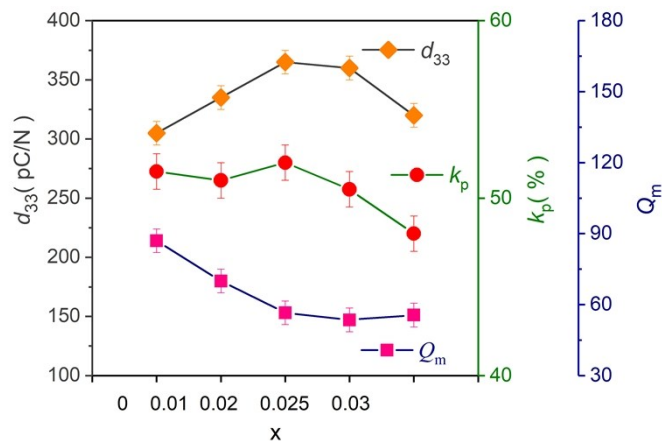


Figure S4. Planar electro-mechanical coupling factor k_p , piezoelectric voltage coefficient d_{33} and the mechanical quality factor Q_m of of poled KNNS_x-5BZ-2BNH ceramics.

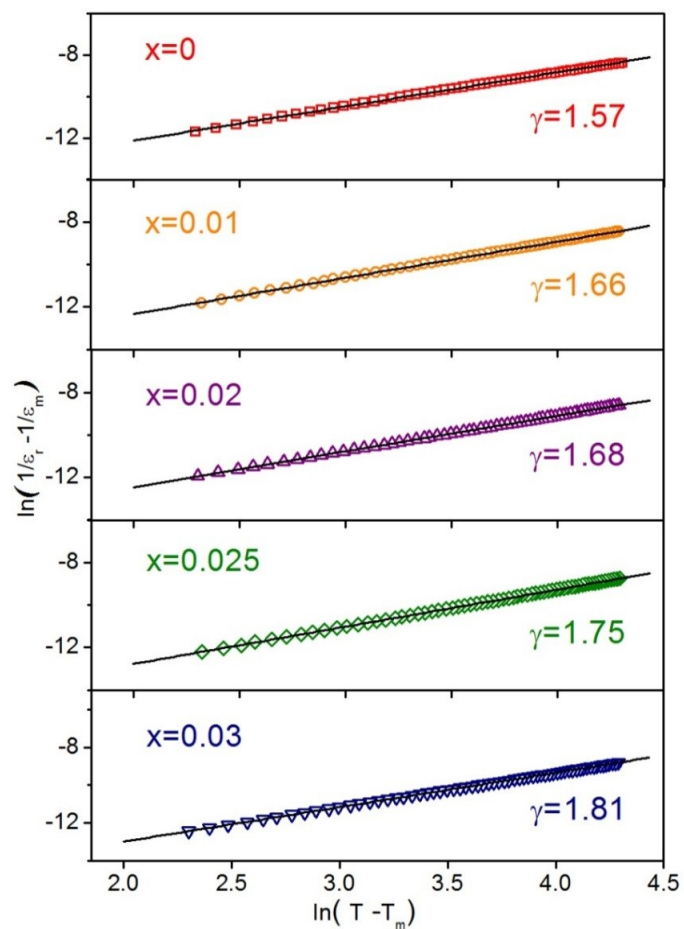


Figure S5. $\ln(1/\epsilon_r - 1/\epsilon_m)$ vs. $\ln(T - T_m)$ figures of the KNNS_x-BBNZH-1Mn samples for determining the calculated degree of diffuseness γ .

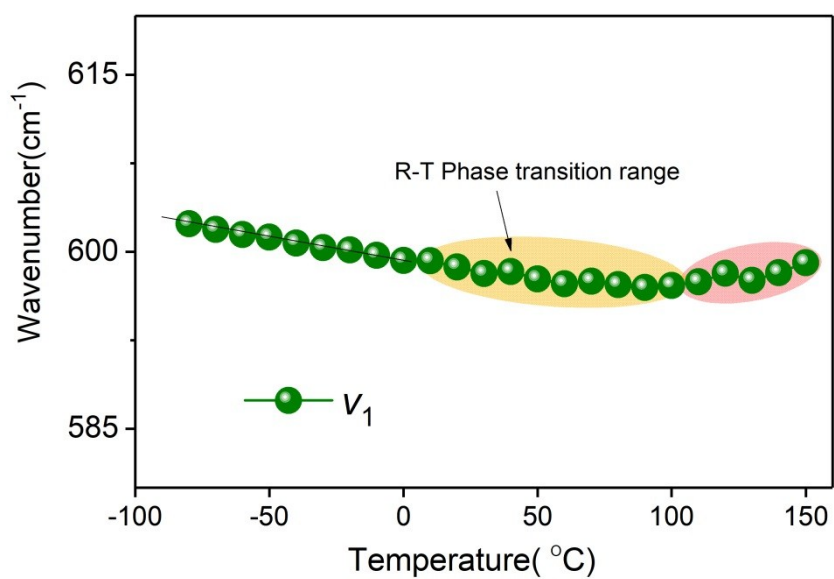


Figure S6. The temperature dependence of the ν_1 peak positions.

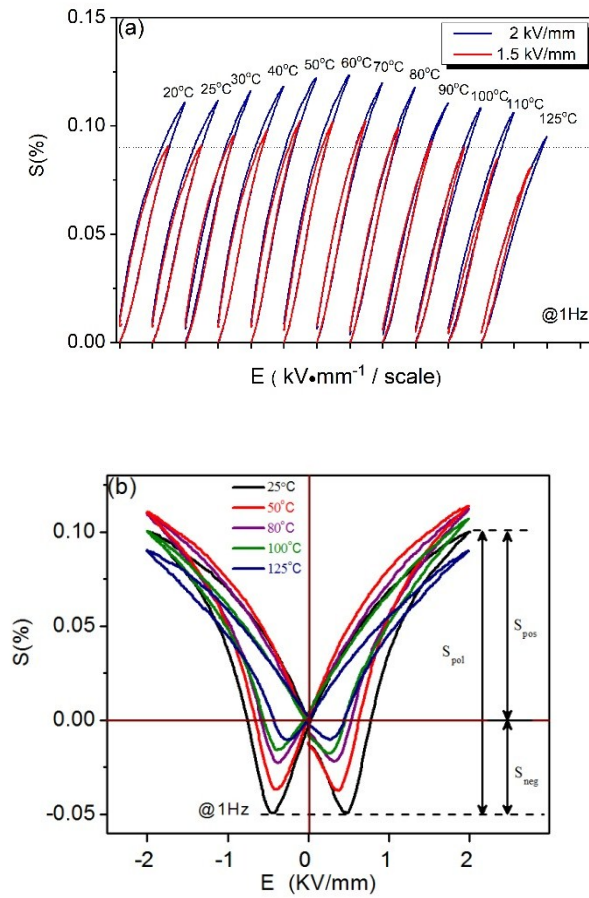


Figure S7. (a) Unipolar electric-field-induced strains under different electric fields ($E_{\text{max}}=1.5\text{kV/mm}$ and 2kV/mm) in the temperature range $20\text{-}125^\circ\text{C}$ for the $x=0.025$ sample. (b) Bi-polar electric-field-induced strain curves of the $x=0.025$ sample under a triangular-shaped base waveform with a maximum electric field 2kV/mm .

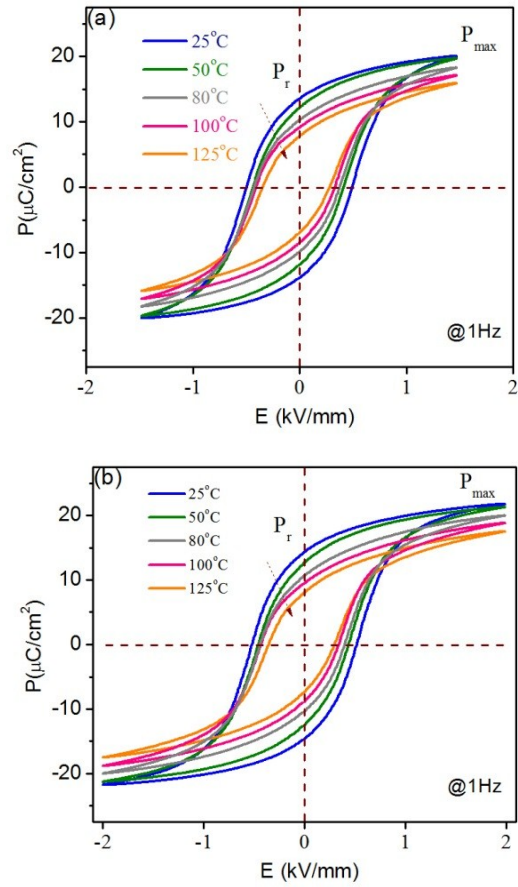


Figure S8. P - E loops for $x=0.025$ sample under electric fields with $E_{\text{max}}=1.5\text{ kV}/\text{mm}$

(a) and $E_{\text{max}}=2\text{ kV}/\text{mm}$.

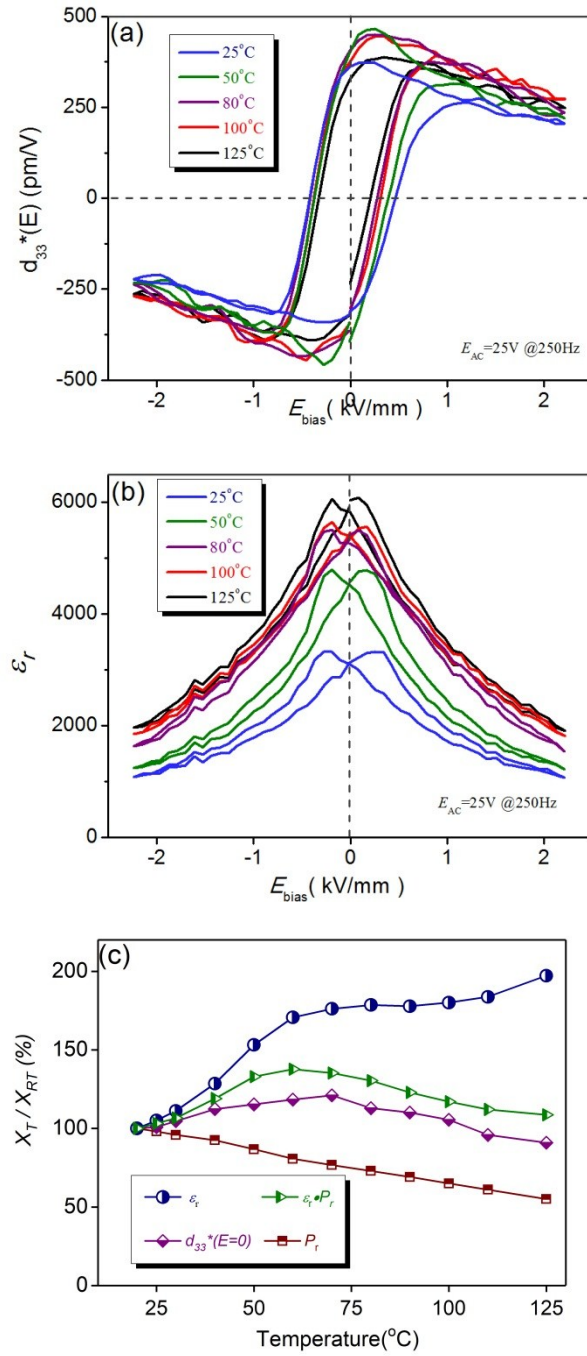


Figure S9. (a) $d_{33}^*(E)$ hysteresis loops at different temperatures for the $x=0.025$ sample. (b) $\epsilon_r - E_{bias}$ curves at different temperatures. (c) Summary of the temperature dependence of the relative dielectric coefficient ϵ_r , remanent polarization P_r , $\epsilon_r \cdot P_r$ and the small signal $d_{33}^*(E=0)$.

The small signal d_{33} exhibits a strong relationship with dielectric and ferroelectric properties as written in the equation: [1, 18-21]

$$d_{33} \sim \alpha \epsilon_r \cdot P_r.$$

The temperature dependence of small signal d_{33} is almost in tune with the variation of $\epsilon_r \cdot P_r$. It is considered that the good thermal stability of small signal d_{33} should be attributed to the balance between reduced P_r and enhanced dielectric property due to the phase transition.

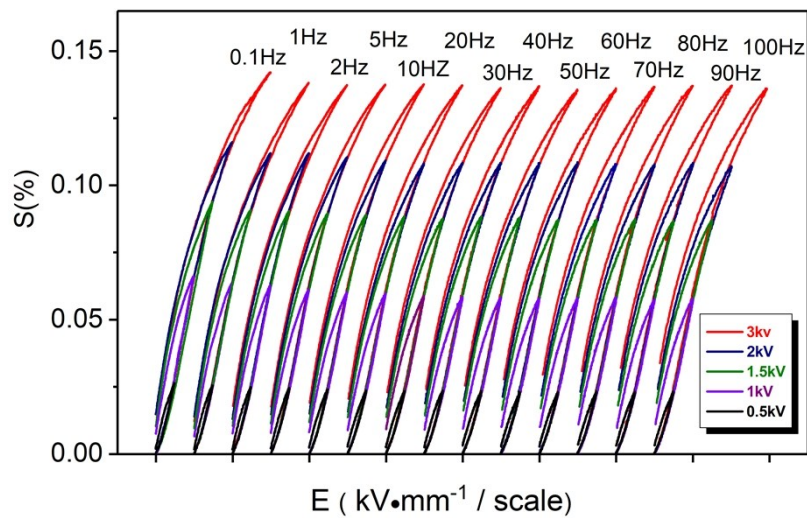


Figure S10. The S-E curves of the $x=0.025$ sample measured at different frequencies and electric fields.

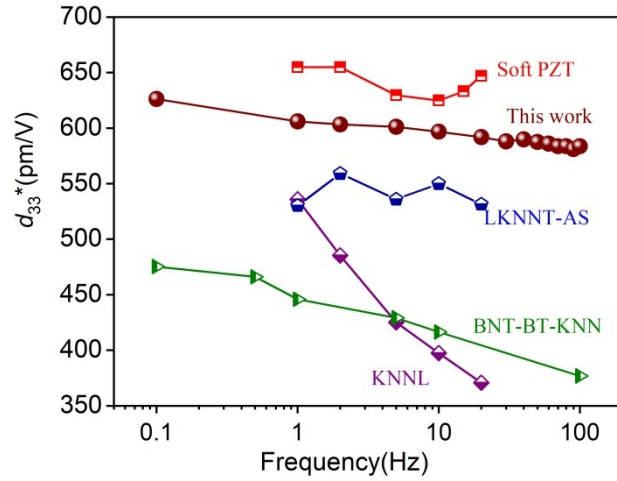


Figure S11. Frequency dependence d_{33}^* of the optimal composition in this work compared with that of KNN-based ceramics, typical BNT-based ceramics and soft PZT as reported in References. [22-24]

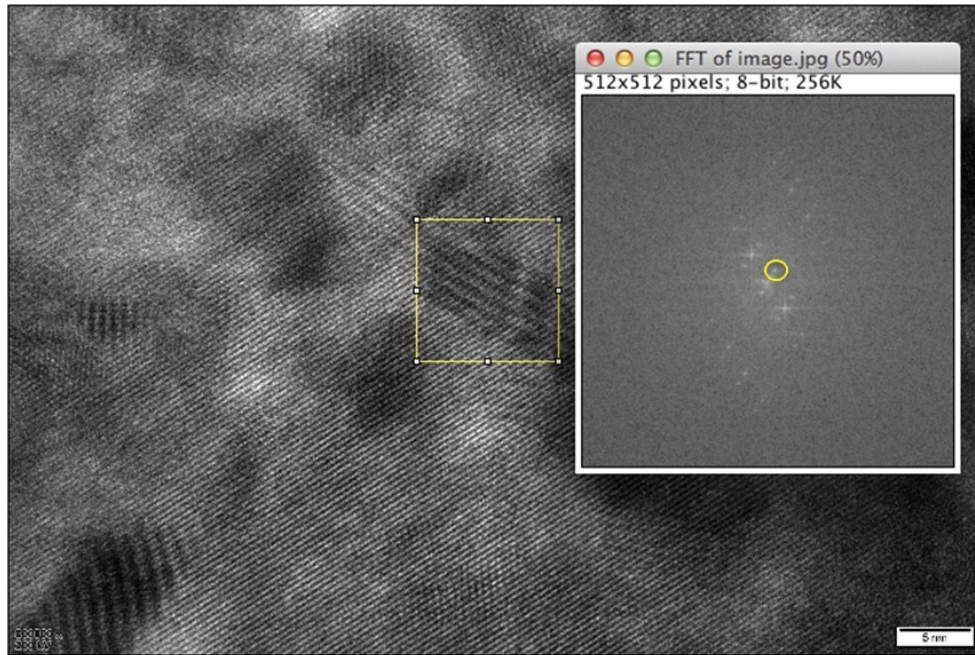


Figure S12. Fast Fourier transform (FFT) of a selected nanosized pattern showing moiré fringes. The diffraction spot indicated by the yellow circle is corresponding to the periodicity of the fringes inside the pattern.

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