

Supporting information

Colloidal Heterostructured Quantum Dots Sensitized Carbon

Nanotubes-TiO₂ Hybrid Photoanode for High Efficiency Hydrogen

Generation

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1 **Experimental**

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3 **Synthesis of CdSe QDs and core/shell QDs**

4 CdSe QDs were synthesized using the hot-injection approach¹. Deposition of CdS layers
5 on CdSe QDs followed procedures described elsewhere². Typically, in a 50-mL round-
6 bottom flask, OLA (5 ml), ODE (5 mL) and CdSe QDs ($\sim 2 \times 10^{-7}$ mol in hexane) were
7 degassed at 110 °C for 30 min. The reaction flask was re-stored with N₂ and the
8 temperature was further raised to 240 °C with stirring. The Cd(OA)₂ dispersed in ODE (0.25
9 mL, 0.2 M) was added dropwise and the mixture was allowed to react for 2.5 h, followed
10 by dropwise addition of 0.2 M sulfur in ODE with the same volume. The shell was further
11 annealed for 60 min. All subsequent shells were annealed at 240 °C for ~ 10 min following
12 the injection of sulfur and ~ 2.5 h following dropwise addition of the Cd(OA)₂ in ODE.
13 Sulfur/Cd(OA)₂ addition volumes for shell addition cycles 1-6 were as follows: 0.25, 0.36,
14 0.49, 0.63, 0.8, and 0.98 mL, respectively.

15 Graded alloyed shells were synthesized by tailoring the molar ratio of S:Se during *in situ*
16 growth of each layer CdSe_xS_{1-x} (x = 0.9~0.1) over the CdSe core QDs. Subsequently another
17 two layers of CdS were coated on the alloyed shell. The reaction was cooled to room
18 temperature using ice water. Ethanol was added, and then the suspension was
19 centrifuged and the supernatant was removed. The QDs were then dispersed in toluene
20 for further characterization.

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22 **Anode preparation**

23 A thin and compact TiO₂ blocking layer was deposited on ultrasonically cleaned FTO glass
24 substrates by hydrolysis of 0.50 mM TiCl₄ solution at 70 °C for 30 min. It was then annealed
25 at 500 °C for 30 min under ambient atmosphere and left to cool down to room
26 temperature.

27 A good dispersion of MWCNTs (an average length of 10 μm) in ethanol was prepared by mixing
28 6 mg of MWCNTs in 15 mL of ethanol and sonicated for 3 hours. TiO₂-MWCNTs hybrid pastes
29 with different concentration of MWCNTs were prepared by mixing the precise amount of
30 ethanolic suspension of MWCNTs into a known weight of TiO₂ paste composed of small (20 nm
31 in diameter) and large (up to 450 nm in diameter) size anatase particles (18NR-AO).
32 Subsequently, the above prepared TiO₂-MWCNTs hybrid pastes were deposited on top of the
33 compact TiO₂ layer by tape casting. A drying process was followed for 15 min at ambient
34 conditions and then placed on a hot plate for 6 min at 120 °C. A second layer was then deposited
35 on the top, following the same procedure³. All the photoanodes were then annealed at 500 °C
36 for 30 min under ambient conditions. For a systematic comparison, bare TiO₂ photoanodes were

1 also prepared under the same conditions. The thickness of all photoanodes was measured using
2 a profilometer.

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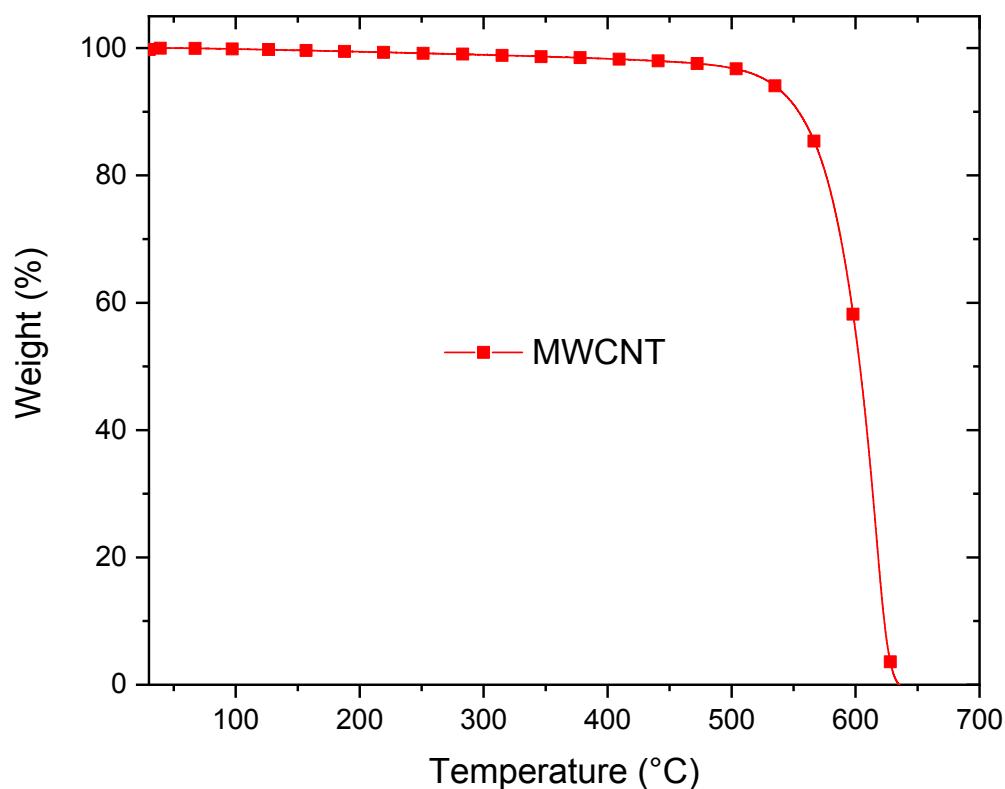
4 **ZrO₂ film preparation**

5 ZrO₂ films were prepared by using a commercial ZrO₂ nanopowder (Aldrich, particle size
6 < 100 nm). A single layer of ZrO₂ film was deposited on FTO glass by tape casting, then
7 annealed in air at 500 °C for 30 min and cooled down to room temperature. We studied
8 the electron transfer rate by using transient fluorescence spectroscopy on QDs deposited
9 into TiO₂ or ZrO₂ mesoporous films. The ZrO₂/QDs film serves as a benchmark sample, in
10 which the energy levels do not favour electron/hole transfer. The hole transfer rate in this
11 experiment was monitored by immersing the ZrO₂/QDs film into the Na₂S/Na₂SO₃ solution
12 (pH ~13) as a hole scavenger.

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14 **Results and Discussion**

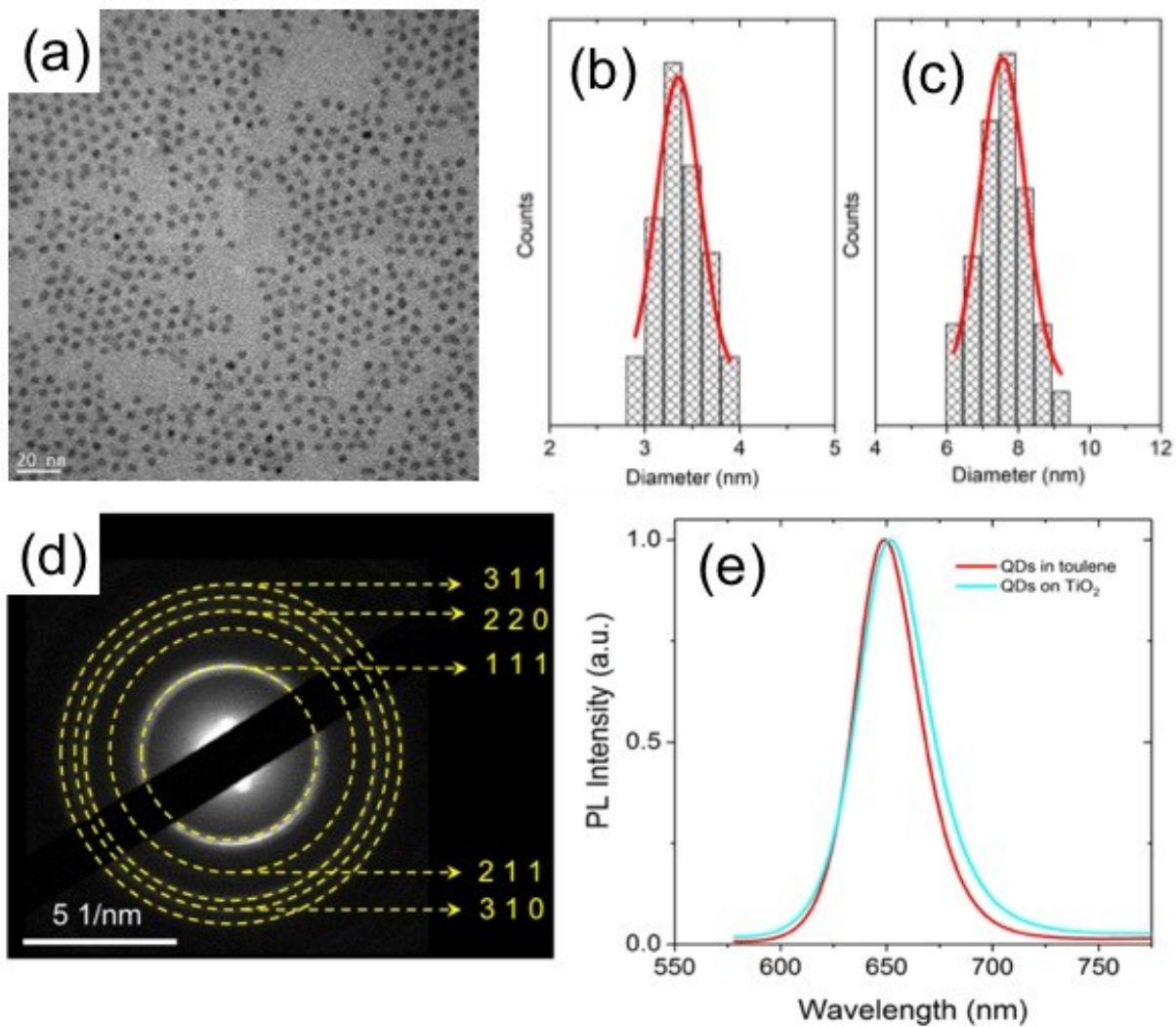
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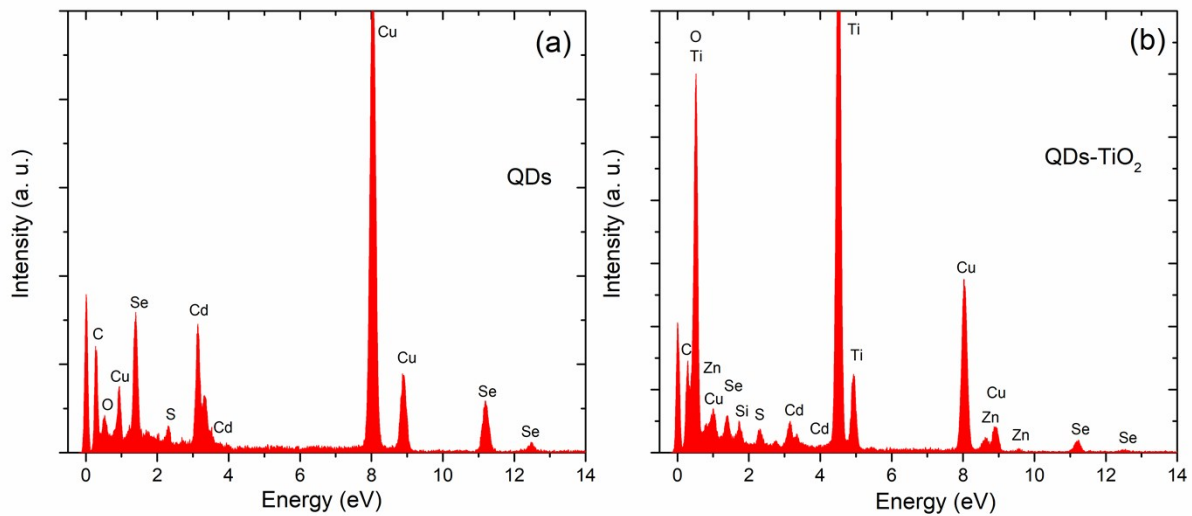
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Figure 1S. Thermogravimetric analysis of MWCNTs powder.



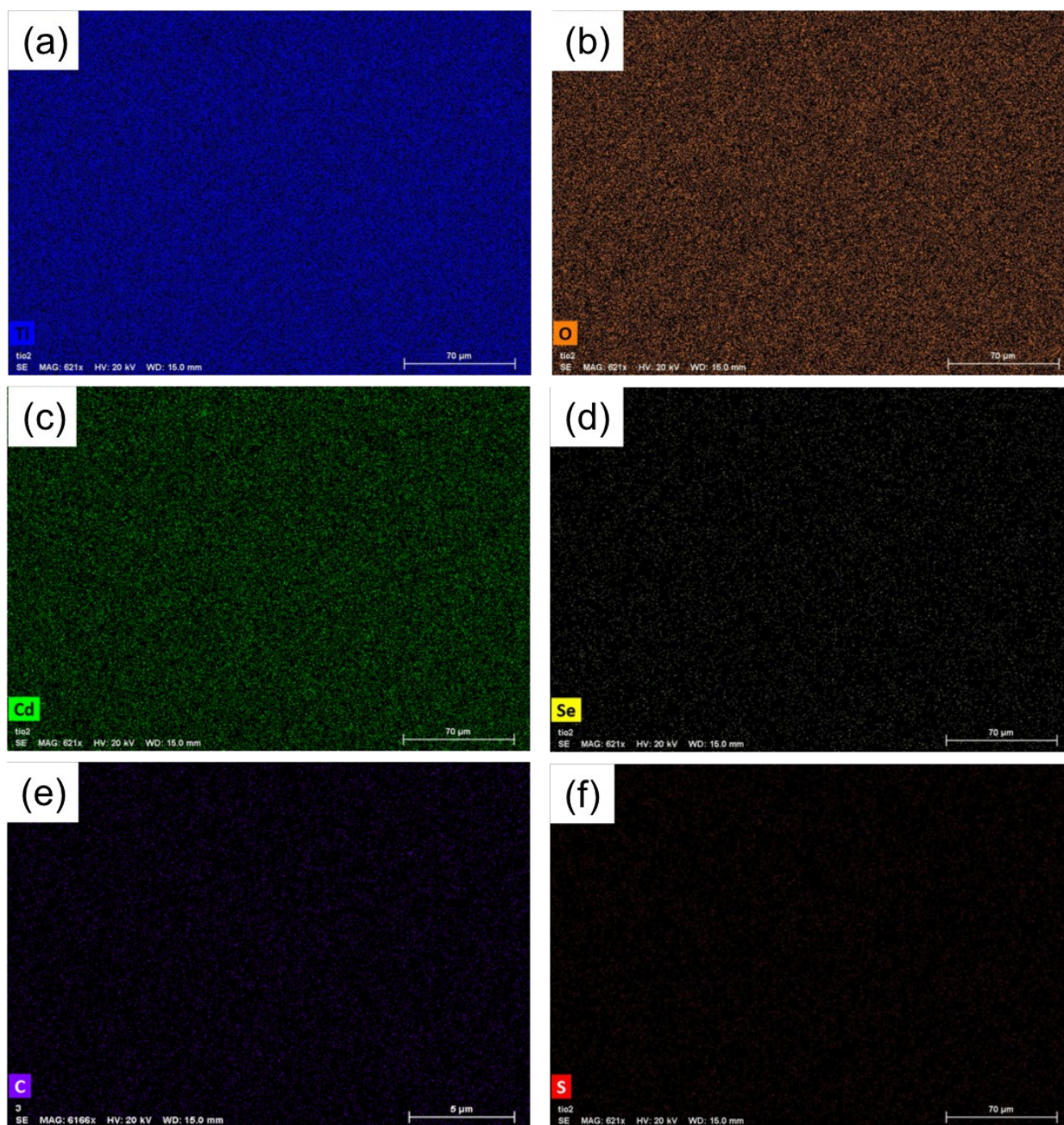
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Figure 2S. Structural properties of colloidal QDs: (a) TEM image of as-synthesized CdSe core QDs before the growth of alloyed shell. Size distribution (solid lines are Gaussian fit of the experimental data): (b) CdSe core QDs; (c) gradient alloyed heterostructured CdSe/(CdSe_xS_{1-x})₅/(CdS)₂QDs; (d) Selected area electron diffraction pattern (SEAD) of gradient alloyed heterostructured CdSe/(CdSe_xS_{1-x})₅/(CdS)₂QDs. (e) PL spectra of as-synthesized gradient alloyed heterostructured CdSe/(CdSe_xS_{1-x})₅/(CdS)₂QDs in toluene and deposited on TiO₂ mesoporous film.



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 2 **Figure 3S.** EDS spectra of: (a) colloidal heterostructured QDs; (b) TiO₂/QDs/ZnS/SiO₂ mesoporous
 3 film.

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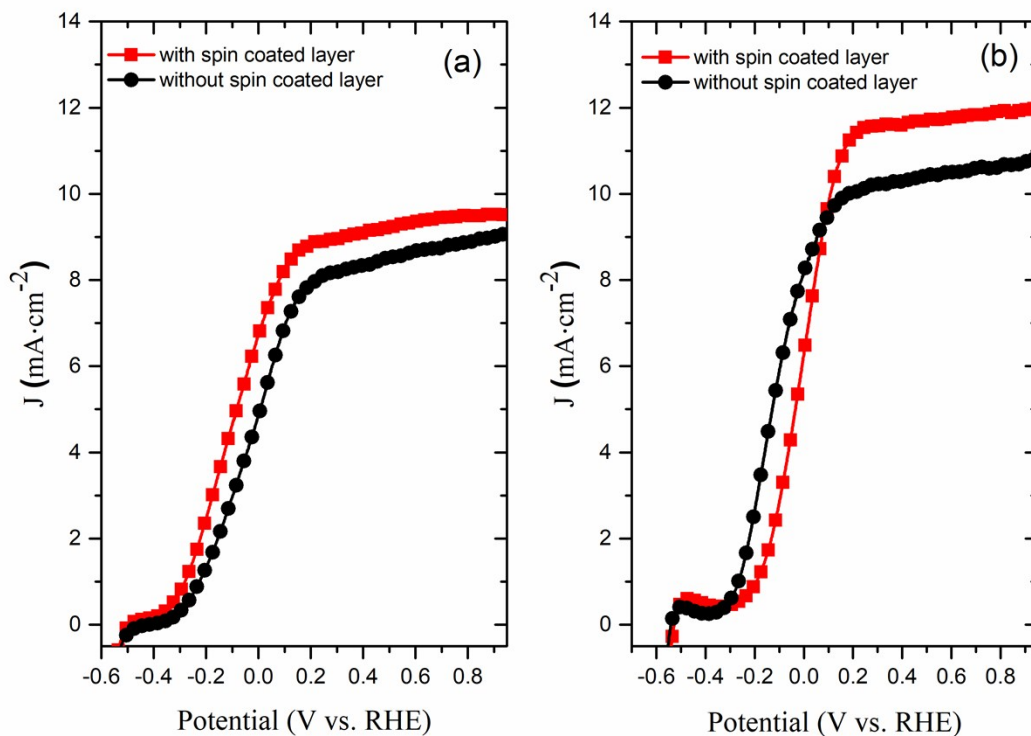
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3 **Figure 4S.** Elemental mapping by EDS spectroscopy of TiO₂/QDs: (a) Ti; (b) O; (c) Cd; (d) Se; (e) C
4 (f) and (f) S.

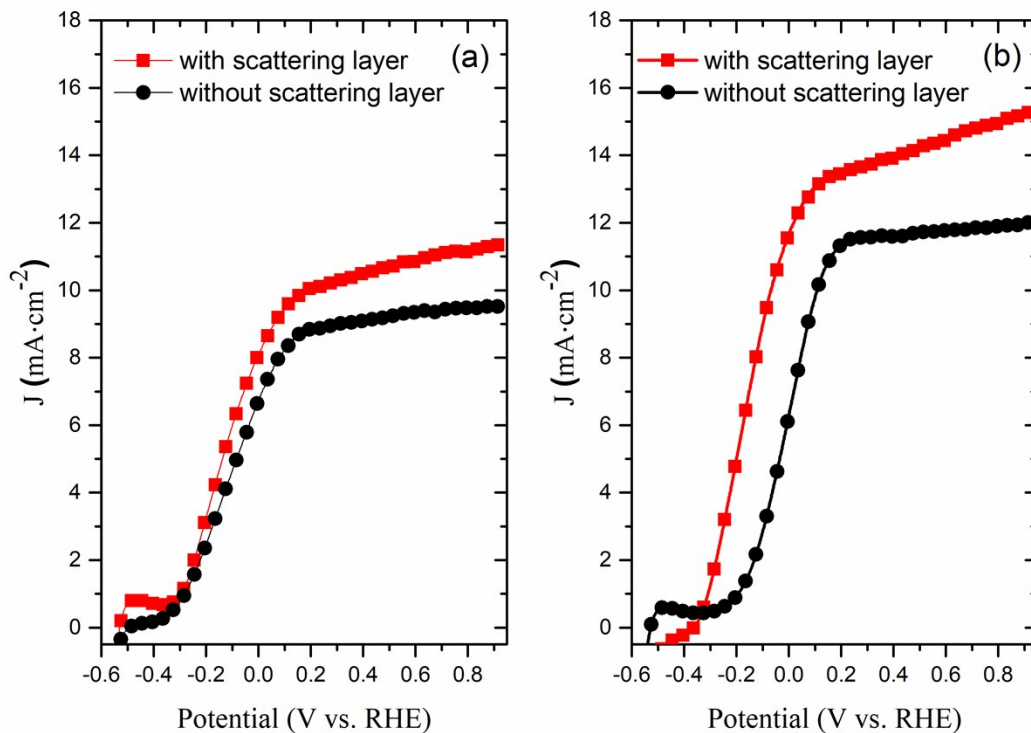
5 **Table 1 S.** The wt % of the different elements of colloidal QDs sensitized TiO₂ coated photoanode,
6 according to EDS measurements.

Elements	Ti	O	Cd	Se	S	C
Wt. %	28.93	65.19	0.39	0.11	0.09	5.29

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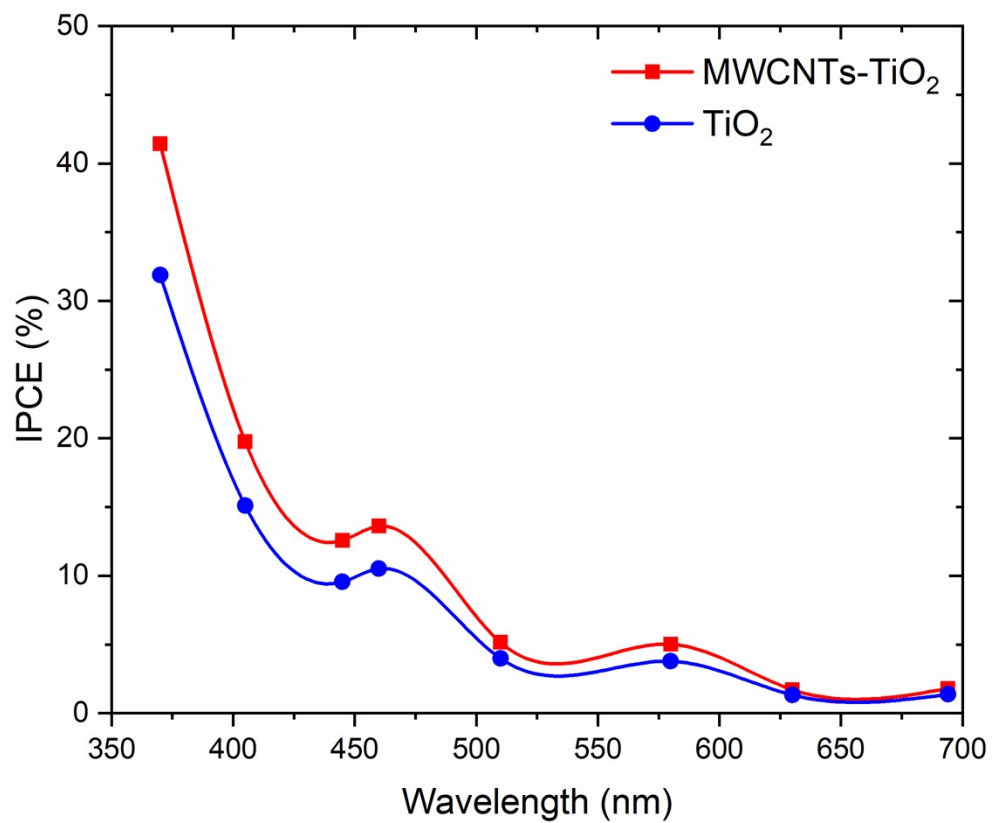


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5 **Figure 5S.** Comparison of photocurrent density versus bias potential (versus RHE) of PEC systems
6 based on colloidal heterostructured QDs sensitized photoanodes with (red square) and without
7 (black circle) spin coated layer of 20 nm nanoparticles under one sun continuous illumination
8 (AM 1.5 G, 100 mW·cm⁻²): (a) pristine TiO₂/QDs; (b) TiO₂/QDs-MWCNTs hybrid photoanodes.
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Figure 6S. Comparison of photocurrent density versus bias potential (versus RHE) of PEC systems based on colloidal heterostructured QDs sensitized photoanodes with (red square) and without (black circle) scattering layer of 150-400 nm nanoparticles under one sun continuous illumination (AM 1.5 G, 100 mW·cm⁻²): (a) pristine TiO₂/QDs; (b) TiO₂/QDs-MWCNTs hybrid photoanodes.



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2 **Figure 7S.** Incident photon-to-current efficiency (IPCE) spectra of PEC devices based on T/Q-
 3 MWCNTs (0.015 wt.%) hybrid (red line) and T/Q bare (blue line) photoanodes measured at 1.0 V
 4 vs RHE under one sun illumination (AM 1.5 G, 100 mW.cm⁻²).

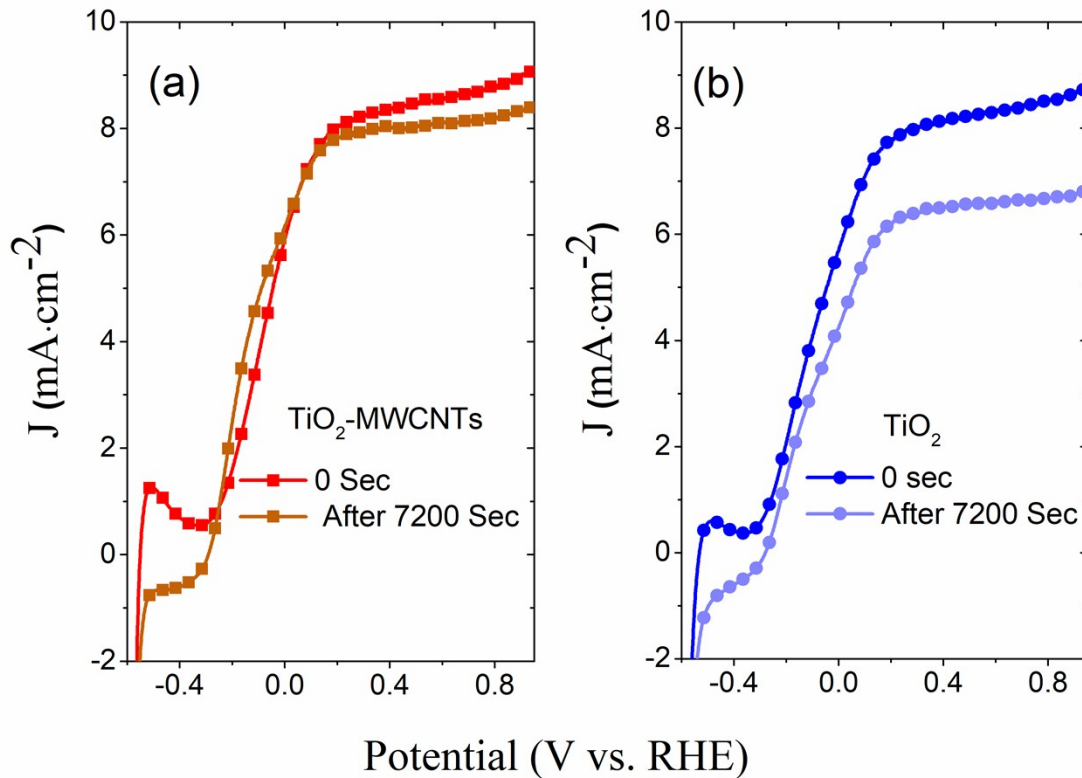
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2 **Figure 8S.** Comparison of the photocurrent density versus bias potential (versus RHE) of PEC
 3 systems under one sun illumination (AM 1.5 G, 100 mW·cm⁻²) based on: (a) TiO₂/QDs-MWCNTs
 4 hybrid fresh (red square) and after 7200 s (dark red square); (b) TiO₂/QDs fresh (blue circle) and
 5 after 7200 s (light blue circle).

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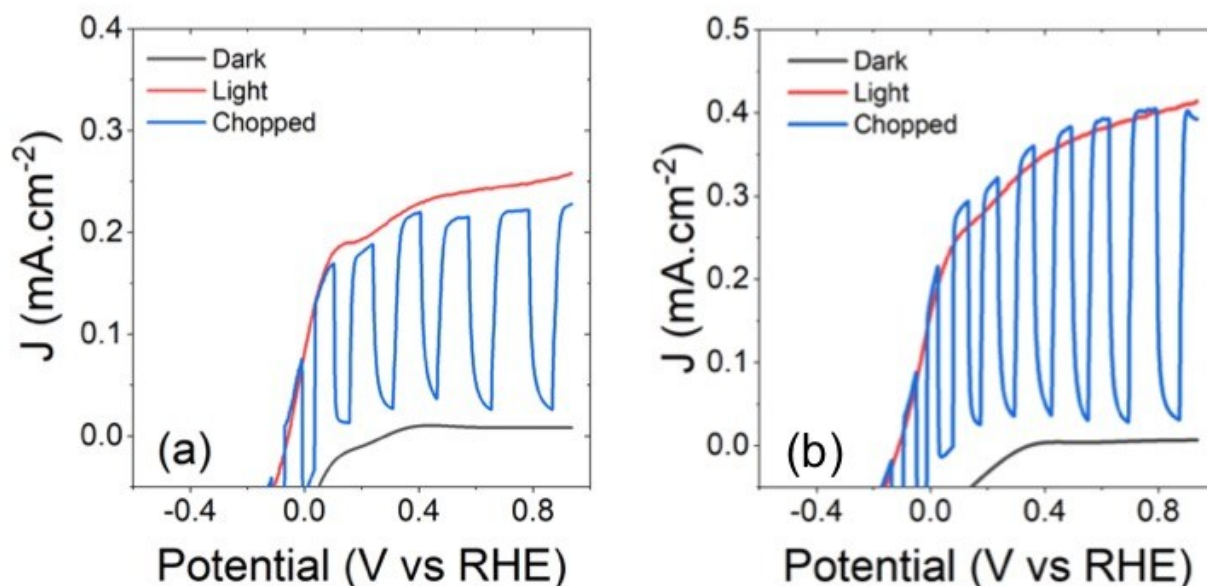
7 **Table S3.** Comparison of photocurrent density of PEC devices based on colloidal core/thick-shell
 8 QDs sensitized wide band gap semiconductors of this work and reported in the literature.

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Photoanode structure	QDs shape	Saturated current density (mA·cm ⁻²)	Ref.
CdSe/(CdSe _x S _{1-x})/(CdS) QDs-TiO ₂ -MWCNTs	Spherical	15.9	This work
CdSe/CdSe _x S _{1-x} /CdS QDs-TiO ₂	Pyramidal	12.0	4
CdSe/Pb _x Cd _{1-x} S/CdS QDs- TiO ₂	Spherical	10.2	5
CdSe/CdS QDs-TiO ₂	Spherical	10.0	6

CuInSe/CuInS QDs-TiO ₂	Spherical	3.1	7
CISES/CdSeS/CdS QDs-TiO ₂	Pyramidal	5.5	8
PbS/CdS QDs-TiO ₂	Spherical	5.3	9

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3 **Figure 9S.** Photocurrent density-potential curves of PEC devices under dark, continuous and
 4 chopped illumination (AM 1.5 G, 100 mW.cm⁻²): (a) TiO₂ anode; (b) TiO₂-MWCNTs hybrid with
 5 0.015 wt% concentration of MWCNTs.

6 Literature comparison of the performance of PEC devices based on TiO₂ with TiO₂-MWCNTs.

7 To compare the performance of PEC devices based on TiO₂ for H₂ evolution reported in the
 8 literature with the PEC devices based on bare TiO₂ and hybrid TiO₂-MWCNTs mesoporous film
 9 prepared in this work, a series of PEC devices were fabricated by using bare TiO₂ and hybrid
 10 mesoporous film without colloidal heterostructured QDs sensitization. A systematic comparison
 11 of photocurrent density of PEC devices based on bare TiO₂ and hybrid TiO₂-MWCNTs mesoporous
 12 film (reported in this work) and PEC device based on TiO₂ (reported in the literature), calculated
 13 from current density-voltage curves under one sun illumination irradiation (AM 1.5 G, 100 mW
 14 cm⁻²) is reported in Table S2. The photocurrent density of PEC device based on hybrid TiO₂-
 15 MWCNTs (0.015 wt.%) is 0.44 mA.cm⁻² at 1.0 V vs RHE, which is ~70% higher than the PEC device
 16 based on bare TiO₂ mesoporous photoanode (see Figure 9 S). Similarly, Morais et al. reported
 17 photocurrent density of 0.11 mA.cm⁻² at 1.23 V vs RHE under one sun illumination irradiation
 18 (AM 1.5 G, 100 mW cm⁻²), which is further boosted to 0.20 mA.cm⁻² at 1.23 V by introducing

1 reduced graphene oxide (0.1 wt%) in TiO₂¹⁰. Furthermore, due to better electron transport
 2 properties of nanowires than nanoparticles, Hwang et al.¹¹ fabricated PEC devices by using rutile
 3 TiO₂ nanowires and obtained the photocurrent density of 0.73 mA.cm⁻² at 1.5 V vs RHE. This
 4 photocurrent density is significantly enhanced to 1.1 mA.cm⁻² at 1.5 V vs RHE with atomic layer
 5 deposition (ALD) of epitaxial rutile TiO₂ shell. Also, Kang et al.¹² reported the photocurrent density
 6 of 0.19 mA.cm⁻² at 1.23 V vs RHE for PEC device based on TiO₂ nanotube arrays and further
 7 boosted to 0.73 mA.cm⁻² at 1.23 V vs RHE by reducing the TiO₂ nanotube arrays with NaBH₄.

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10 **Table S4.** Comparison of performance of PEC devices based on TiO₂-MWCNTs in this work and
 11 PEC based on TiO₂ reported in the literature.

Photoanode structure	Electrolyte	J_{sc} (mA.cm ⁻²) @ V vs RHE	Ref.
TiO ₂	Aqueous 0.25 M Na ₂ S and 0.35 M Na ₂ SO ₃ (pH~13)	0.26 mA.cm ⁻² at 1.0 V	This work
TiO ₂ -MWCNTs (0.015 wt%)		0.44 mA.cm ⁻² at 1.0 V	
pristine TiO ₂	Aqueous H ₂ SO ₄ (0.5 mol L ⁻¹)	0.11 mA.cm ⁻² at 1.23 V	10
Rutile TiO ₂ nanowires	Aqueous 1 M NaOH	0.73 mA.cm ⁻² at 1.5 V	11
TiO ₂ nanotube arrays	Aqueous 1 M NaOH (pH=13.9)	0.19 mA.cm ⁻² at 1.23 V	12

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