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Templated direct growth of ultra-thin double-walled carbon nanotubes: supplementary information^{\dagger}

Lei Shi,^{*a} Jinquan Wei,^b Kazuhiro Yanagi,^c Takeshi Saito,^d Kecheng Cao,^e Ute Kaiser,^e Paola Ayala,^a and Thomas Pichler ^a

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^a University of Vienna, Faculty of Physics, 1090 Wien, Austria. E-mails: lei.shi@univie.ac.at

^b Tsinghua University, State Key Lab of New Ceramics and Fine Processing, School of Materials Science and Engineering, Beijing 100084, P. R. China

 $^{^{\}rm c}$ Tokyo Metropolitan University, Department of Physics, 1-1 Minami-Osawa, Hachiouji, Tokyo 192-0397, Japan

^d National Institute of Advanced Industrial Science and Technology (AIST), Nanomaterials Research Institute, Tsukuba 305-8565, Japan

^e Ulm University, Central Facility for Electron Microscopy, Electron Microscopy Group of Materials Science, Ulm 89081, Germany

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Fig. S1 RBM region of Raman spectra of pristine eDIPS-1.7 nm SWCNTs and annealed sample at 1500 °C excited by a 568 nm laser.



Fig. S2 RBM region of Raman spectra of annealed (a) and pristine (b) eDIPS-1.7 nm SWCNTs excited by multi-frequency Raman spectroscopy.



Fig. S3 Raman spectra of annealed samples at different temperatures excited by a 568 nm laser. (a) RBM, (b) D and G-bands, (c) 2D-band.



Fig. S4 RBM region of Raman spectra of annealed (a) and pristine (b) eDIPS-1.3 nm SWCNTs excited by multi-frequency Raman spectroscopy.



Fig. S5 Raman spectra of the pristine HiPco SWCNTs and annealed sample at 1500 °C excited by 568 and 633 nm lasers. (a) RBM, (b) D and G-bands, (c) 2D-band.



Fig. S6 TEM image ofeDIPS-1.3 SWCNTs annealed at 1500 °C. The sample has high purity. Only very few catalyst particles could be observed.



Fig. S7 Bundles of double-walled carbon nanotubes. The arrow in (b) indicates the transformation position between the double-walled carbon nanotube and triple-walled carbon nanotube with an addition inner tube.



Fig. S8 HRTEM images of several isolated double-walled carbon nanotubes.



Fig. S9 Optical absorption spectra indicate the high-purity of separated metallic and semiconducting SWCNTs.



Fig. S10 RBM region of Raman spectra of annealed semiconducting arc-discharge SWCNTs at 1500 °C excited by multi-frequency Raman spectroscopy.



Fig. S11 RBM region of Raman spectra of annealed metallic arc-discharge SWCNTs at 1500 °C excited by multi-frequency Raman spectroscopy.