
* Electronic supplementary information

Universal description of heating-induced reshaping preference in core-shell bimetallic nanoparticles

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Supplementary Methods

1. Theoretical model

Based on the effective Ising Hamiltonian model, the definition of Δh_i^{eff} , Δh_i^{size} , and V_{ij} before introducing T can be written as

$$\Delta h_i^{eff} = \chi \Delta \gamma(\alpha) = \chi [V_i^B(\alpha) - r_i^A(\alpha)] \quad (\text{S1})$$

$$\Delta h_i^{size} = \Delta E_r = z_i^B \lambda \exp[-r^B / \nu] - z_i^A \lambda \exp[-r^A / \nu] \quad (\text{S2})$$

$$V_{ij} = \frac{1}{2} [E_{b(ij)}^{AA} + E_{b(ij)}^{BB} - 2E_{b(ij)}^{AB}] \quad (\text{S3})$$

$$E_{b(ij)} = E_c + E_{pro} + E_{pol} \quad (\text{S4})$$

where $\gamma(\alpha)$ is the surface energy at different shape factors α . χ is the second moment approximation coefficients.¹ Z_i is the coordination number of an atom at site i . r is the atom radius. λ and ν are the constants determined by fitting to the bulk modulus and atomic volume.² E_c and E_{pol} are the cohesive energy and polarisation energies from literature.³ E_{pro} is the promotion energy required to take the atom from its ground-state configuration to the magnetic ground-state configuration of the prepared atom, which can be obtained from experimental data⁴.

2. The TB-SMA Potential and Monte Carlo (MC) simulation

Monte Carlo (MC) simulation is firstly used to obtain the equilibrium structures of bimetallic nanoparticles as the initial configurations in the following MD simulation. The interaction between metal atoms was modeled semi-empirically based on the well-established TB-SMA potential⁵. It is noted that TB-SMA potential has been successfully used to model the thermodynamic properties of bimetallic nanoparticles effectively.⁶⁻⁷ In the Gupta potential, the metal-metal (M-M) interaction energy, E_{M-M} , is given by

$$E_{M-M} = \sum_i E_R^i + E_B^i \quad (\text{S5})$$

where E_R^i and E_B^i are the Born-Mayer ion-ion repulsion and band interactions, respectively. These two terms for an atom i can be represented based on

$$E_R^i = \sum_{j \neq i} A e^{-p(r_{ij}/r_0 - 1)} \quad (\text{S6})$$

$$E_B^i = - \left\{ \sum_j \xi^2 e^{-2q(r_{ij}/r_0 - 1)} \right\} \quad (\text{S7})$$

where r_{ij} is the distance between atoms i and j in the cluster, r_0 is the nearest-neighbor distance in the pure metals, and N is the number of the metal atoms. The parameters A , r_0 , ξ , p , and q for the Ag-Cu, Au-Pd, Ag-Au, and Pd-Cu nanoparticles used in this study are taken from the literature^{5, 8} as listed in Table S1, which were employed in previous studies with satisfactory results⁹⁻¹⁰ and proved by experiment elsewhere¹¹⁻¹². In MC simulations, the integration time step was set to 1 fs and 4 ns are adopted for the total simulation length to achieve the most stable structure. The former 2 ns are performed to make the system equilibrium and the next 2 ns were used for the statistical averaging. When this method is used to study the bimetallic nanoparticles, two trials are introduced in the sampling scheme: (1) random displacement of each atom from its original position and (2) random interchange of two atoms with different species in the cluster.

2. Molecular dynamics simulations

We identify the melting of bimetallic nanoparticles from the caloric curve of the bimetallic nanoparticles and the corresponding heat capacity C_v per atom. According to the position of the maximum peak value in the heat capacity C_v and the step jump

in the caloric curve, and then the melting temperature can be predicted. The heat capacity Cv per atom is defined by

$$Cv = \frac{\langle\langle E^2 \rangle\rangle - \langle E \rangle^2}{nk_B T^2} \quad (\text{S8})$$

where E is the potential energy, k_B is the Boltzmann constant, n is the total number of the atoms in the cluster and T is the temperature. The transition probability of atoms from state i to state j

$$p_{ij} = \frac{p_i}{p_j} = \exp\left[-(E_i - E_j)/k_B T\right] \quad (\text{S9})$$

Where E_i and E_j are the energy for an atom of type A at site i and an atom of type B at site j. In the BOMD simulation, the Perdew–Burke–Ernzerhof (PBE) exchange–correlation functional are used in this work as implemented in the CP2K package¹³. The spatial dimension of the simulation super cell is $55 \times 55 \times 55 \text{ \AA}$ with a vacuum distance 18 \AA , which is large enough to neglect interaction among adjacent periodic images of the nanoparticles. All the nanoparticles is fully relaxed before the BOMD simulation. An energy cutoff of 300 Ry is set for the plane wave basis set and VDW long range correction is added. The time step is 1 fs and the total simulation time for each system ranges from 20 ps, assuring that the system can reach the global optimal.

Supplementary Tables

Table S1: Surface energy, surface energy temperature coefficient, bond energy, atom size, excess surface enthalpy, and thermal expansion coefficient for the different metal elements.¹⁴⁻¹⁷

Categor y	mental	$\gamma(J/m^2)$	$-d\gamma/dT$	$H_s(J/m^2)$	$\alpha(K^{-1} \times 10^{-6})$	$E_b(eV)$	$r_0(\text{\AA})$
3-d serials	Fe	2.188	0.23	2.175	12.3	7.27	1.16
	Co	2.339	0.25	2.251	13.36	6.02	1.16
	Ni	2.538	0.28	2.415	13.3	4.92	1.15
	Cu	1.830	0.21	1.512	16.5	3.83	1.17

	Zn	1.197	0.18	1.200	25	1.89	1.25
4-d serials	Ru	3.075	0.34	3.079	9.1	8.21	1.25
	Rh	2.600	0.22	2.525	8.40	6.77	1.25
	Pd	1.690	0.17	1.662	11.2	5.19	1.28
	Ag	1.179	0.17	1.088	19.2	3.15	1.34
	Cd	0.819	0.13	0.803	29.8	1.27	1.41
5-d serials	Os	3.713	0.38	3.904	6.1	10.41	1.26
	Ir	3.028	0.31	3.009	6.8	8.17	1.27
	Pt	2.601	0.26	2.385	9	6.28	1.30
	Au	1.542	0.18	1.340	14.16	4.2	1.34

Table S2: The alpha-dependence of the surface energies for metals ($\alpha=1.00$, **1.12, and **1.49** corresponding to the sphere, icosahedron, and star-like shape, respectively.)**

mental	α	$\gamma(J/m^2)$	mental	α	$\gamma(J/m^2)$	mental	α	$\gamma(J/m^2)$
Fe	1.00	2.228	Ru	1.00	3.829	Os	1.00	4.655
	1.12	2.443		1.12	4.320		1.12	5.151
	1.49	2.617		1.49	4.765		1.49	5.807
Co	1.00	3.031	Rh	1.00	2.524	Ir	1.00	2.796
	1.12	3.070		1.12	2.749		1.12	2.844
	1.49	3.095		1.49	2.906		1.49	2.908
Ni	1.00	2.023	Pd	1.00	1.829	Pt	1.00	2.185
	1.12	2.406		1.12	2.131		1.12	2.569
	1.49	2.462		1.49	2.357		1.49	2.892
Cu	1.00	1.893	Ag	1.00	1.169	Au	1.00	1.338
	1.12	2.077		1.12	1.212		1.12	1.571
	1.49	2.173		1.49	1.243		1.49	1.703
Zn	1.00	1.098	Cd	1.00	0.694			
	1.12	1.135		1.12	0.746			
	1.49	1.269		1.49	0.793			

Table S3: The bond energy (eV) of atomic species A and B (A and B represent the type of atoms in bimetallic system).

	Fe	Co	Ni	Cu	Zn	Ru	Rh	Pd	Ag	Cd	Os	Ir	Pt	Au
Fe	--	6.29	5.88	5.79	5.15	7.10	6.82	6.73	5.70	5.68	8.55	7.81	7.63	6.86
Co	--	--	5.24	5.15	4.52	6.46	6.18	6.09	5.07	5.04	7.91	7.18	7.00	6.23
Ni	--	--	--	4.74	4.10	6.05	5.77	5.68	4.65	4.63	7.50	6.76	6.58	5.81
Cu	--	--	--	--	4.02	5.96	5.68	5.59	4.56	4.54	7.41	6.67	6.50	5.73
Zn	--	--	--	--	--	5.32	5.05	4.96	3.93	3.91	6.78	6.04	5.86	5.09
Ru	--	--	--	--	--	--	6.99	6.90	5.87	5.85	8.72	7.98	7.80	7.03
Rh	--	--	--	--	--	--	--	6.62	5.59	5.57	8.44	7.70	7.53	6.76
Pd	--	--	--	--	--	--	--	--	5.50	5.48	8.35	7.61	7.43	6.67
Ag	--	--	--	--	--	--	--	--	--	4.45	7.32	6.59	6.41	5.64
Cd	--	--	--	--	--	--	--	--	--	--	5.88	5.51	5.42	5.04
Os	--	--	--	--	--	--	--	--	--	--	--	9.44	9.26	8.49
Ir	--	--	--	--	--	--	--	--	--	--	--	--	8.52	7.75
Pt	--	--	--	--	--	--	--	--	--	--	--	--	--	7.57
Au	--	--	--	--	--	--	--	--	--	--	--	--	--	--

Table S4: Parameters of the Gupta potential for the Ag-Ag, Cu-Cu, Au-Au, Pd-Pd, Ag-Cu, Au-Pd, Ag-Au, and Pd-Cu interactions.

	A (eV)	ξ (eV)	p	q	r_0 (Å)
Ag-Ag	0.1043	1.1940	10.79	3.190	2.890
Cu-Cu	0.0855	1.224	10.960	2.278	2.556
Au-Au	0.2061	1.790	10.229	4.036	2.884
Pd-Pd	0.1746	1.718	10.867	3.742	2.749
Ag-Cu	0.0944	1.2089	10.8746	2.734	2.723
Au-Pd	0.1896	1.7536	10.5431	3.889	2.816
Ag-Au	0.1490	1.4874	10.494	3.607	2.887
Pd-Cu	0.1221	1.4501	10.9135	3.01	2.6525

Table S5: The calculated Δh_i^{eff} , Δh_i^{size} and V_{ij} (in eV/atom) of Ag-Au bimetallic nanoparticles with the size (3 nm) and composition (A3B1) at different temperatures (400K, 600K, 800K).

Temperature	Δh_i^{eff}	Δh_i^{size}	V_{ij}
400	0.446	0.0292	-0.0315
600	0.613	0.0317	-0.0331
800	0.958	0.0363	-0.0359

Table S6: The comparison between the prediction results in this work and experiment results from literature.

Prediction in this work				
System	Chemical configuration	Heating temperature	Reshaping preference	Shape properties
Ag-Au	Core-shell	700K	Yes	star shape (core-shell)
Ag-Cu	Core-shell	750K	Yes	star shape (core-shell)
Au-Pd	Core-shell	870K	Yes	star shape (core-shell)
Au-Pt	Core-shell	910K	Yes	star shape (core-shell)
Pd-Cu	Core-shell	950K	No	Sphere (alloy)
Au-Cu	Core-shell	830K	No	Sphere (alloy)

Experiment results				
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System	Chemical configuration	Heating method and temperature	Reshaping preference	Ref
Ag-Au	Core-shell	resistive heating (45°C)	Yes (star shape)	¹⁸
Ag-Cu	Core-shell	In situ STEM heating experiments (150°C)	Yes (star shape)	¹⁹
Au-Pd	Core-shell	calcination in static air (400°C)	Yes (star shape)	²⁰
Au-Pt	Core-shell	hot solution pipette (160°C)	Yes (star shape)	²¹
Pd-Cu	Core-shell	Eurotherm programmable temperature controller (280°C)	No (Sphere-like alloy)	²²
Au-Cu	Core-shell	temperature-dependent X-ray absorption fine (280°C)	No (Sphere-like alloy)	²³

Supplementary Figures

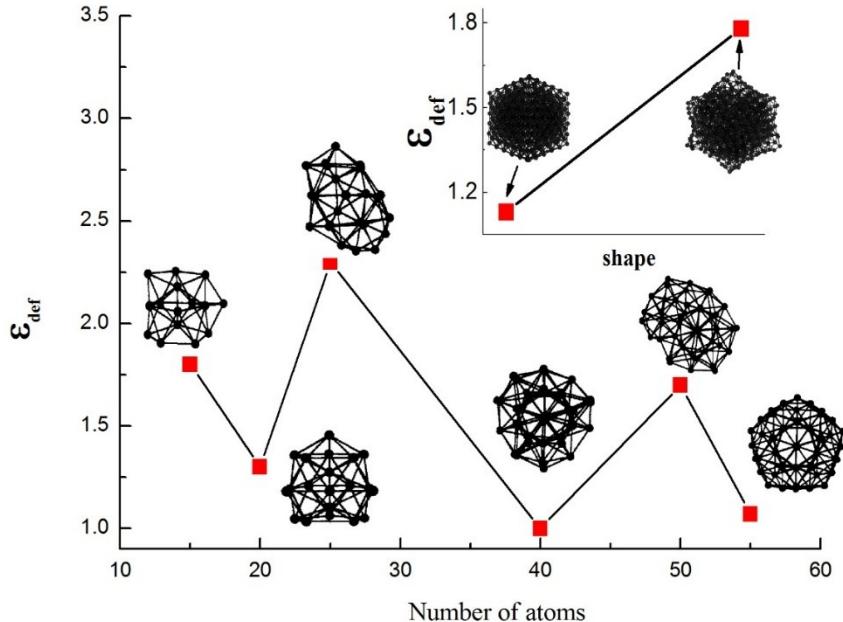


Fig. S1 The deformation parameter ε_{def} and responding snapshots of Ag clusters at different sizes.

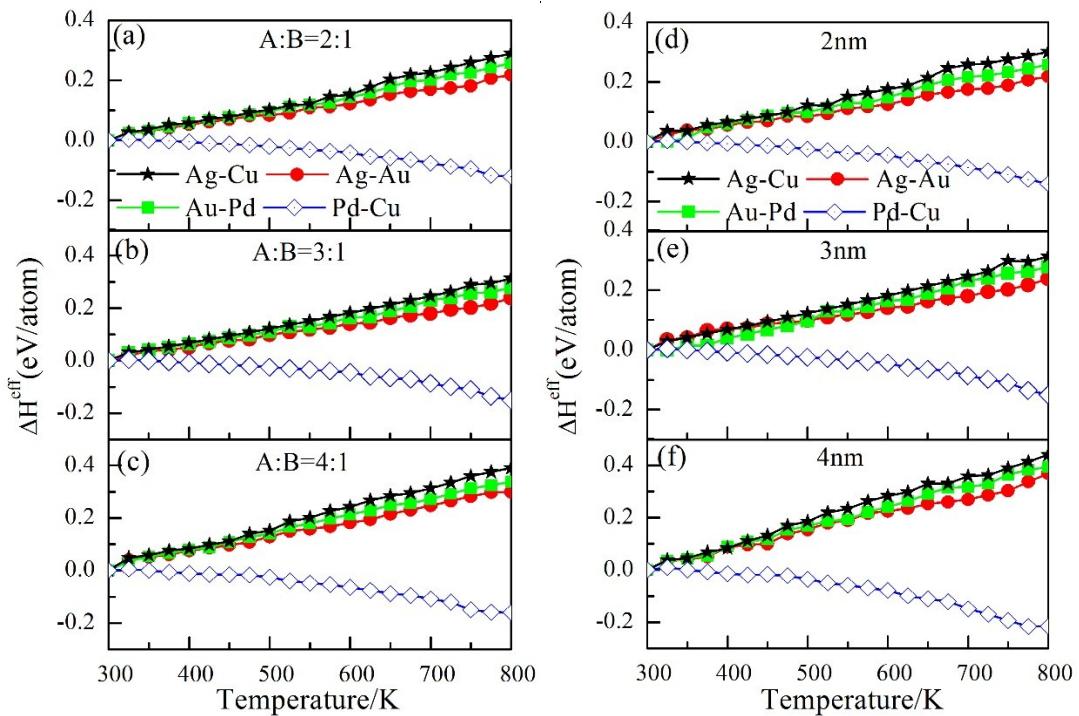


Fig. S2 The temperature dependent of ΔH^{eff} for Ag-Cu, Au-Pd, Ag-Au, and Pd-Cu bimetallic nanoparticles with the different compositions of A_2B_1 (a), A_3B_1 (b), and

A_4B_1 (c) (A represent Ag, Au, Ag, Pd, and B represent Cu, Pd, Au, Cu, respectively) and sizes of 2 nm (d), 3nm (e), and 4 nm (f) (fixed compositions: A_3B_1).

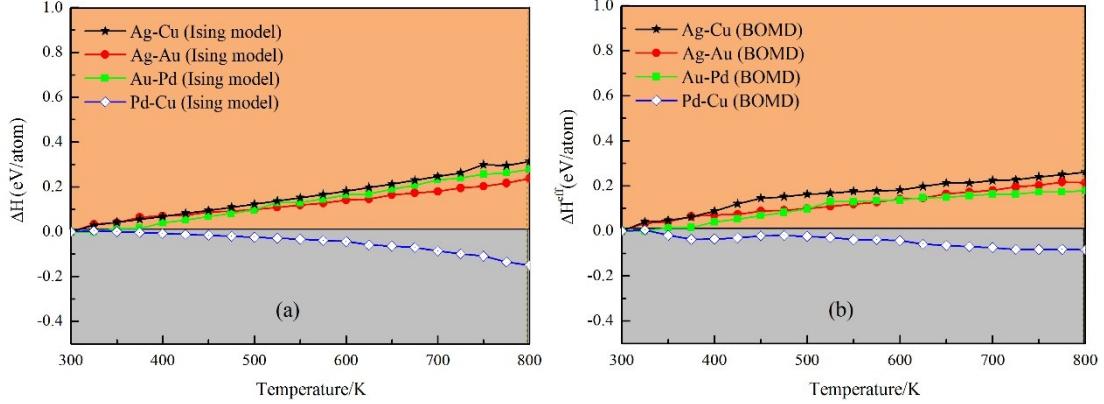


Fig. S3. The temperature dependent of ΔH for $Ag_{414}Cu_{147}$, $Au_{414}Pd_{147}$, $Ag_{414}Au_{147}$, and $Pd_{414}Cu_{147}$ core-shell bimetallic nanoparticles (3nm) with the different method including Ising method and BOMD.

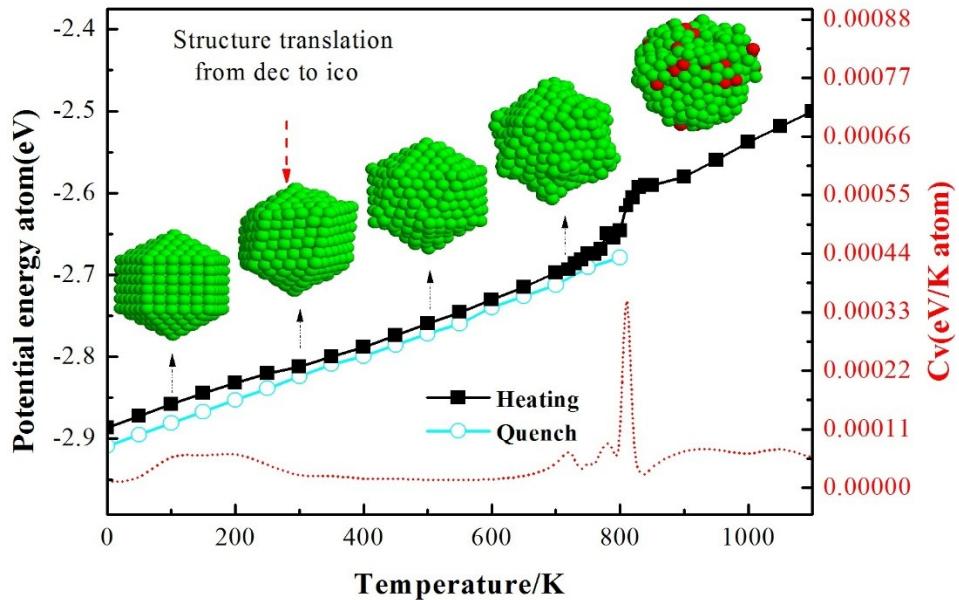


Fig. S4 The temperature dependent of caloric curves, the heat capacities Cv per atom, and corresponding snapshots (green: silver, red: copper) of decahedral (starting structure) $Ag_{414}Cu_{147}$ nanoparticles (3nm).

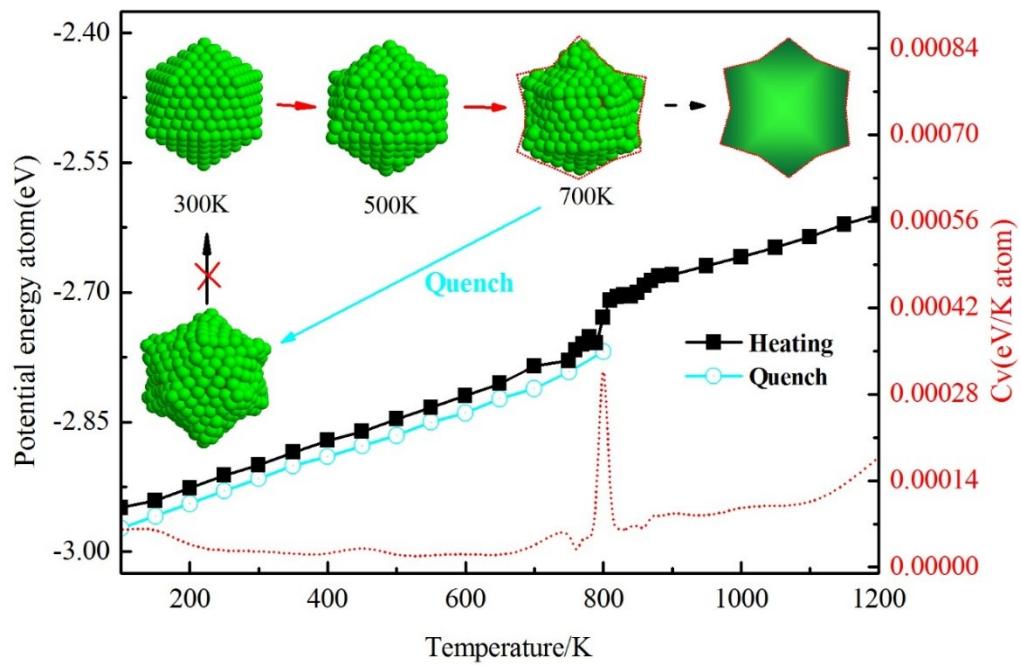


Fig. S5 The caloric curves, the heat capacities C_V per atom, and corresponding snapshots (green: silver, red: copper) of core-shell $\text{Ag}_{414}\text{Cu}_{147}$ nanoparticles (3nm) at different temperature.

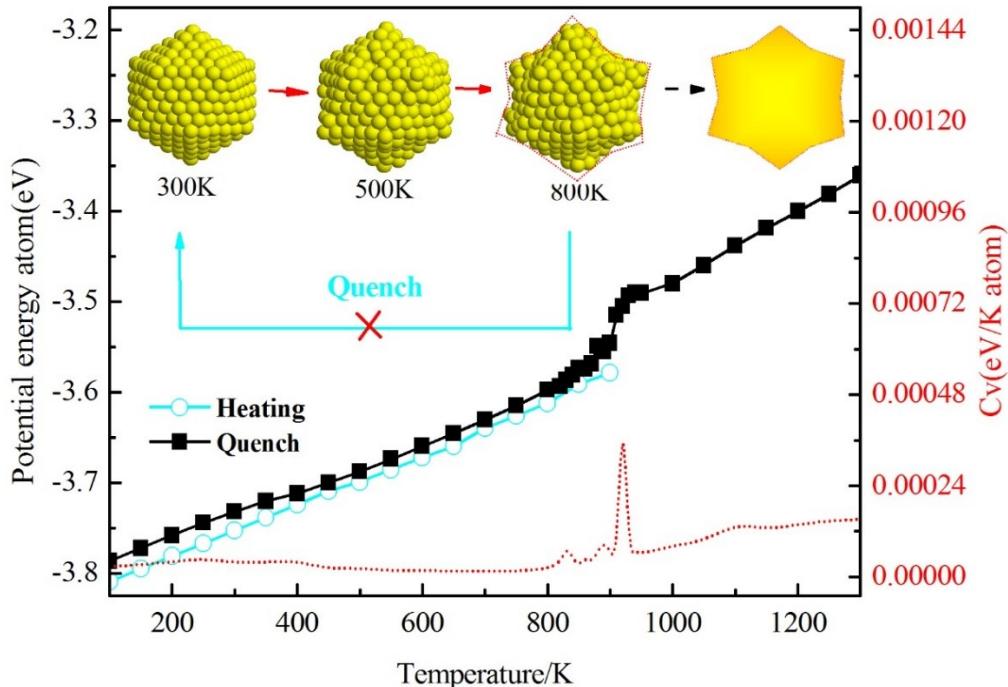


Fig. S6 The caloric curves, the heat capacities C_V per atom, and corresponding snapshots (yellow: gold, red: palladium) of core-shell $\text{Au}_{414}\text{Pd}_{147}$ nanoparticles (3nm) at different temperature.

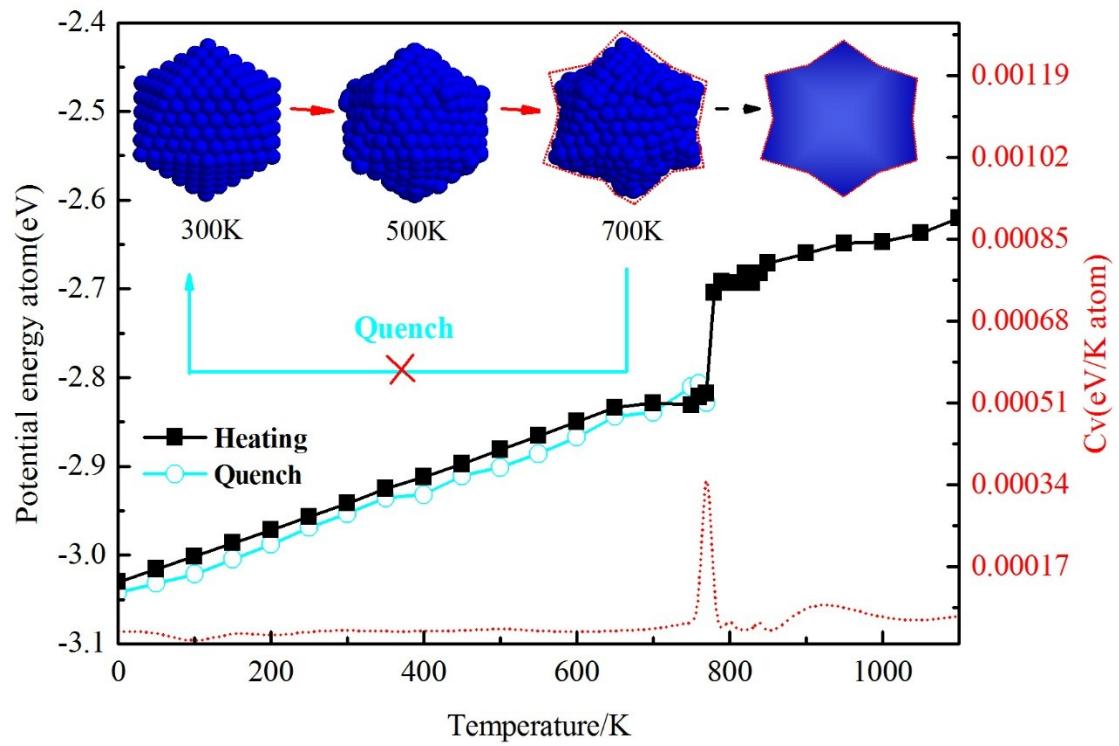


Fig. S7 The caloric curves, the heat capacities C_v per atom, and corresponding snapshots (blue: silver, red: gold) of core-shell Ag₄₁₄Au₁₄₇ nanoparticles (3nm) at different temperature.

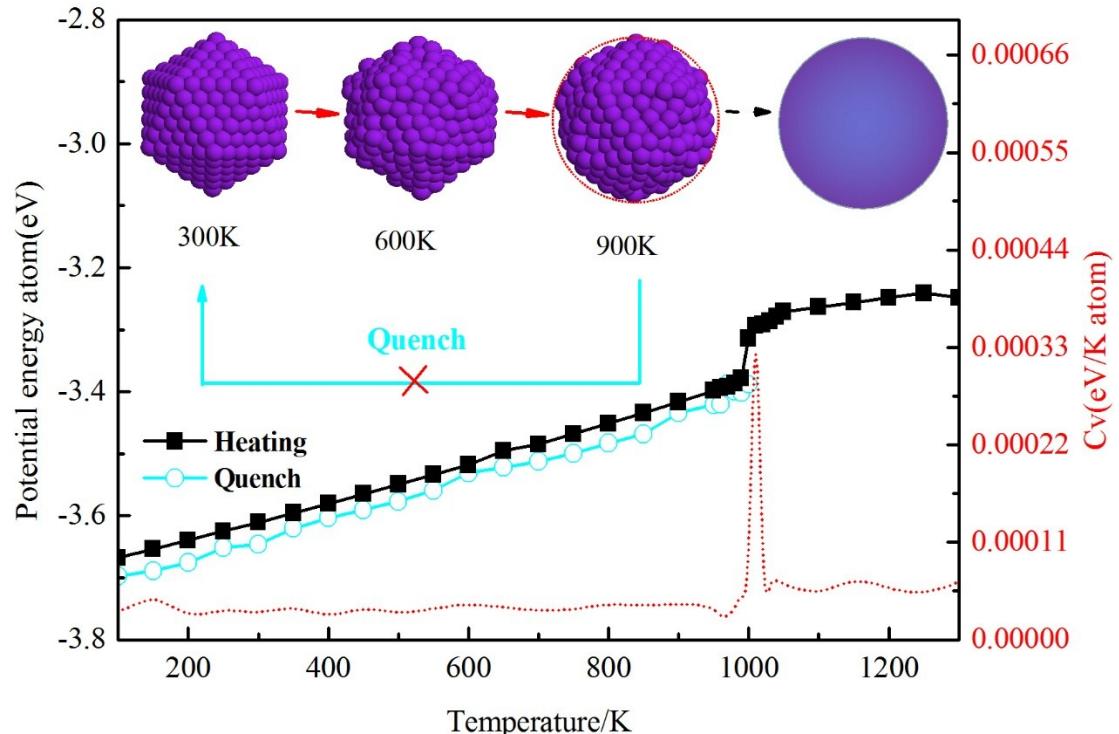


Fig. S8 The caloric curves, the heat capacities C_v per atom, and corresponding snapshots (purple: palladium, red: copper) of core-shell Pd₄₁₄Cu₁₄₇ nanoparticles (3nm) at different temperature.

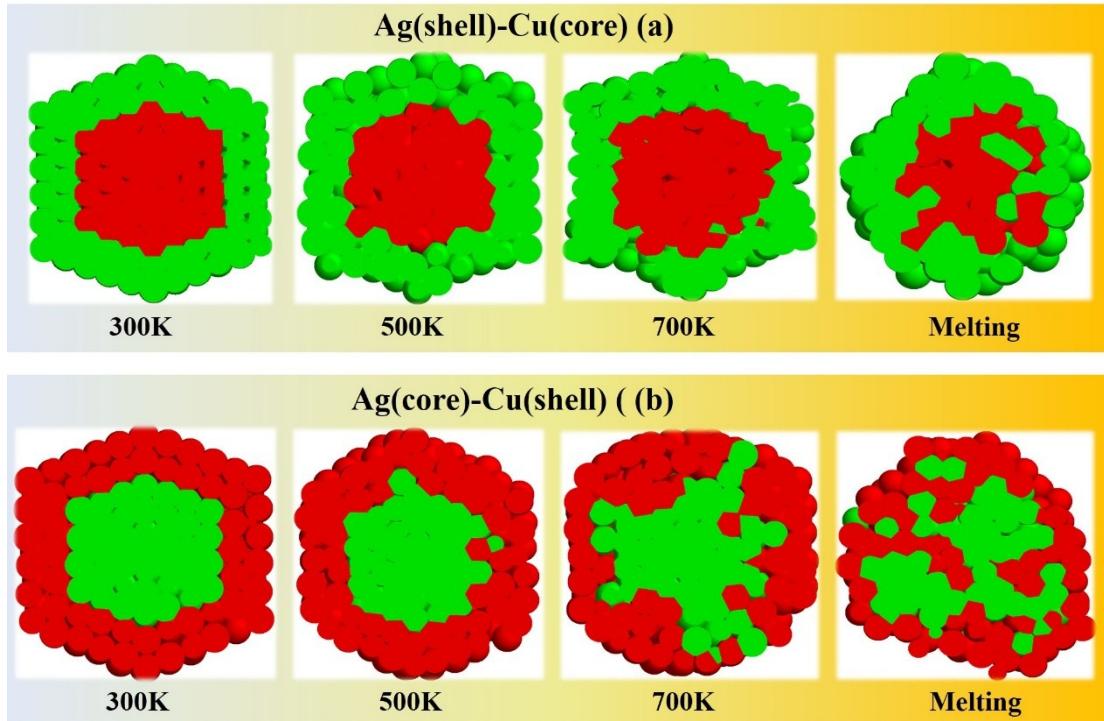


Fig. S9 The comparison of the structure evolution of $\text{Ag}_{\text{shell}}\text{Cu}_{\text{core}}$ and $\text{Ag}_{\text{core}}\text{Cu}_{\text{shell}}$ bimetallic nanoparticles in different temperatures (green: Ag atoms; red: Cu atoms).

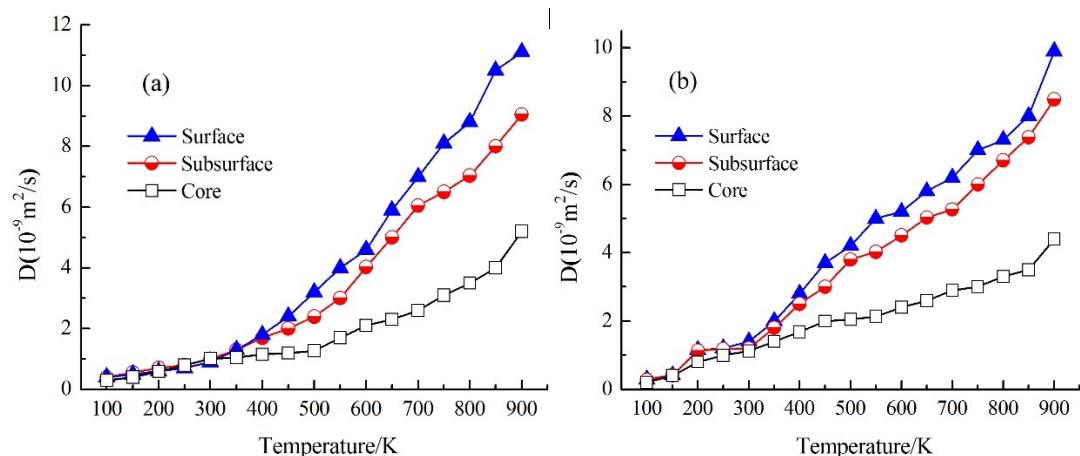


Fig. S10 The self-diffusivity coefficient, D of the (a) Au-Pd and Ag-Au bimetallic nanoparticles varying with different temperature.

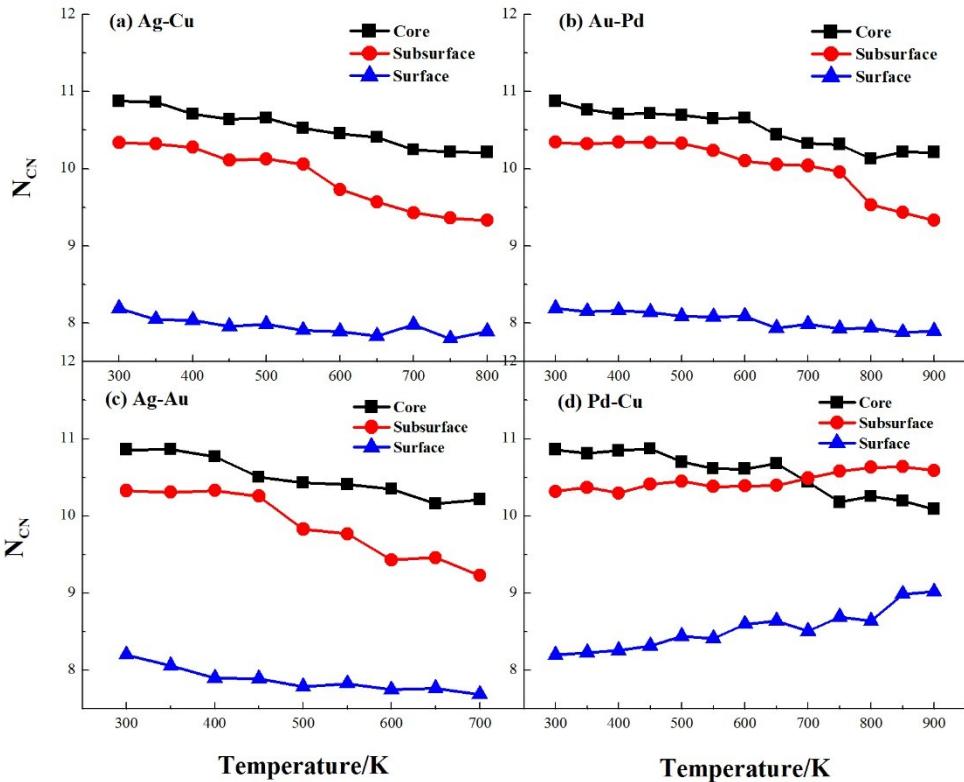


Fig. S11 The comparison of average coordination number (ACN) of surface, subsurface, and core atoms for $\text{Ag}_{414}\text{Cu}_{147}$ (a), $\text{Au}_{414}\text{Pd}_{147}$ (b), $\text{Ag}_{414}\text{Au}_{147}$ (c), and $\text{Pd}_{414}\text{Cu}_{147}$ (d) bimetallic nanoparticles, respectively.

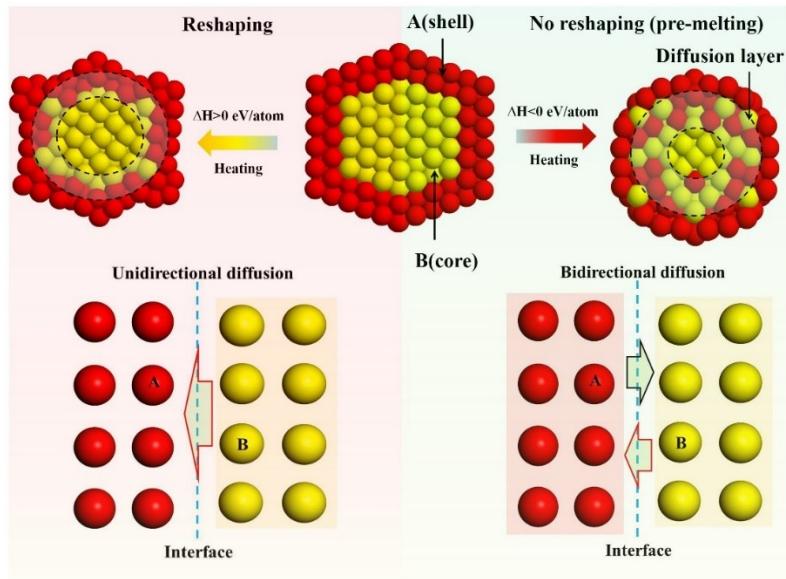


Fig. S12 Schematic of the diffusion process for bimetallic nanoparticles. Upon heating, the unidirectional diffusion (left) appears between A (shell) and B (core) and causes the reshaping of bimetallic nanoparticles. In contrast, the bidirectional diffusion (right) appear in bimetallic nanoparticles and cause the formation of spherical nanoparticles without reshaping.

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