# **Supporting Information**

Trapping [PMo<sub>12</sub>O<sub>40</sub>]<sup>3-</sup> clusters into pre-synthesized ZIF-67 toward Mo<sub>x</sub>Co<sub>x</sub>C particles Confined in Uniform Carbon Polyhedron for Efficient Overall Water Splitting

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## 1. Experimental Section

### The synthesis procedure of other control samples

The experimental parameters, including the solvothermal time, the amount of PMo<sub>12</sub> and the kind of Mo source, were tuned to regulate the micro-structure of final materials. The solids obtained after solvothermal reaction at 100°C for 2 h and 10h were denoted as PMo/ZIF-67-2 and PMo/ZIF-67-10. The PMo/ZIF-67-6 (0.066), PMo/ZIF-67-6 (0.165) and PMo/ZIF-67-6 (0.660) were also prepared under the same condition with PMo/ZIF-67-6, excepting the use of 0.066, 0.165 and 0.660 mmol of PMo12 instead of 0.330 mmol of PMo<sub>12</sub>. The Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6 was obtained by using Na<sub>2</sub>MoO4 as Mo source instead of PMo<sub>12</sub>. Unless otherwise specified, the PMo/ZIF-67-X represents the samples which was prepared by solvothermal treatment of ZIF-67 in the presence of 0.330 mmol PMo<sub>12</sub>.

# Synthesis of S-PMo/ZIF-67

S-PMo/ZIF-67 was synthesized through the simultaneous addition of POMs, 2-MIM and  $Co^{2+}$  salts in solution (a route similar to that used for the synthesis of POMOF). After reaction at room temperature for 6h, the solid was separated by filtration, and was washed with methanol for three times. And then the solid was dried at 60 °C in a vacuum oven for 12 h. (S in S-PMo/ZIF-67 represented "simultaneous" )

#### Synthesis of S-PMo/ZIF-67-6N

The procedure was similar to the synthesis of PMo/ZIF-67-6-6N excepting the replacement of PMo/ZIF-67-6 with S-PMo/ZIF-67.

### 2. The content calculated based on ICP

(1) The ICP test indicated the amount of Mo in PMo/ZIF-67-6 is about 21.5%. So, the amount of  $PMo_{12}$  in PMo/ZIF-67-6 can be calculated as follows (MW is Molecular weight):

 $PMo_{12}\% = Mo\% \times MW \text{ of } PMo_{12}/12 \text{ MW of } Mo = 21.5\% \times 1825.25/1151.52=34.1\%$ 

(2) We have analyzed Co content from Co NPs in PMo/ZIF-67-67-6N based on ICP analysis. For this test, the PMo/ZIF-67-67-6N was firstly treated by inaqua regia to solve Co and Mo components. The solution was set the certain volume with deionized water, and the total concentration of Co  $(Co_{(T)})$  and Mo  $(Mo_{(T)})$  ions in solution were analyzed by ICP analyser. The amount of Co ions from Co<sub>6</sub>Mo<sub>6</sub>C (Co in Co<sub>6</sub>Mo<sub>6</sub>C) is calculated based on the followed formula:

Content (Co) in  $Co_6Mo_6C = Mo_{(T)} \times 58.93/95.94 = 0.614Mo_{(T)}$ 

The content of Co from Co NPs can be calculated by deducting Co in  $Co_6Mo_6C$  from  $Co_{(T)}$ . The Co mass from Co NPs is about 26% based above analyses.

Entry	Samples	Solvothermal (trapping) Time (h)	Calcination temperature (°C)	Calcination Time (h)	n (Mo source) (mmol)
1	ZIF-67-6N	0	600	4	0 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
2	ZIF-67-7N	0	700	4	0 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
3	PMo/ZIF-67-6-5N	6	500	4	0.330 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
4	PMo/ZIF-67-6-6N	6	600	4	0.330 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
5	PMo/ZIF-67-6-7N	6	700	4	0.330 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
6	PMo/ZIF-67-6-8N	6	800	4	0.330 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
7	PMo/ZIF-67-2-6N	2	600	4	0.330 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
8	PMo/ZIF-67-10-6N	10	600	4	0.330 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
9	PMo/ZIF-67-2-7N	2	700	4	0.330 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
10	PMo/ZIF-67-10-7N	10	700	4	0.330 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
11	PMo/ZIF-67-6-6N (0.066)	6	600	4	0.066 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
12	PMo/ZIF-67-6-6N (0.165)	6	600	4	0.165 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
13	PMo/ZIF-67-6-6N (0.660)	6	600	4	0.660 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
14	PMo/ZIF-67-6-7N (0.066)	6	700	4	0.066 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
15	PMo/ZIF-67-6-7N (0.165)	6	700	4	0.165 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
16	PMo/ZIF-67-6-7N (0.660)	6	700	4	0.660 (H <sub>3</sub> PMo <sub>12</sub> O <sub>40</sub> )
17	Na2MoO4/ZIF-67-6 -6N	6	600	4	3.960 (Na <sub>2</sub> MoO <sub>4</sub> ·2H <sub>2</sub> O)
18	Na <sub>2</sub> MoO <sub>4</sub> /ZIF-67-6 -7N	6	700	4	3.960 (Na <sub>2</sub> MoO <sub>4</sub> ·2H <sub>2</sub> O)

**Table S1.** The samples and their corresponding synthesis parameters.

\* In all syntheses, the amount of ZIF-67 is 0.452 mmol (0.1g). The total volume of methanol solvent is 60 mL.



Figure S1. SEM image of S-PMo/ZIF-67.

Figure S1 shows SEM image of S-PMo/ZIF-67 from the simultaneous addition of POMs, 2-MIM and  $Co^{2+}$  salts in solution (a route similar to POMOF). It can be obviously seen the presence of un-uniform sphere, which is obviously different from the regular polyhedron morphology of PMo/ZIF-67 from "trapping" route. The results indicate the less controllability of the "direct" route on the structure due to difficult in coordinating the reaction rate both of  $Co^{2+}$  and PMo<sub>12</sub> with 2-MIM.



Figure S2. (a) UV-vis spectra of Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O and (b) 2-MIM in methanol with

differrent concentration. (c) UV-vis spectra of the supernatant separated from (A) hot and (B) cooling dispersion from treating methanol solution of ZIF-67 at 100 °C for 6h in water bath condition. (d) UV-vis spectra of PMo<sub>12</sub> in 60mL methanol and the supernatant separated from dispersion of PMo/ZIF-67-2, PMo/ZIF-67-6 and PMo/ZIF-67-10.

The ZIF-67 have large cage with size of 1.16 nm. The size of PMo<sub>12</sub> is about 1 nm. So, in principle, the cage of ZIF-67 can accommodate PMo<sub>12</sub> with the fashion of one PMo<sub>12</sub> in one cage. However, the window aperture of ZIF-67 is too small (0.34 nm) to allow the access of  $PMo_{12}$  into pre-synthesized ZIF-67. Opening the window will provide the large avenue to allow the assess of PMo<sub>12</sub> into the cage. It is known that there is the presence of the precipitation-dissolution equilibrium in the dispersion of the solid. For MOF solid, the process should also exist, and is called as disassembly-reassembly equilibrium. The disassembly process will result in the presence of dissociative Co<sup>2+</sup> and 2-MIM in solution. Based on UV-vis test, we have found that the methanol solution of Co(NO<sub>3</sub>)<sub>2</sub> and 2-MIM have shown obvious adsorption peaks in the range of 200-250 nm (Figure S2a, b). The adsorption intensity increases with the increase of concentration of both Co<sup>2+</sup> and 2-MIM, although it have not linear relation between the concentration and adsorption. So, we have studied the degree of disassembly based on UV-vis methods. To this, the dispersion liquid of ZIF-67 (0.452 mmol) in methanol (60 mL) was heated in water bath (100°C) for 6h. After heating, the dispersion liquid of ZIF-67 in methanol was treated by two different ways.

In the first way, the dispersion liquid was cooled to room temperature, and then centrifuged to separate solid and solution. Figure S2c shows UV-vis spectra of supernatant. We can see the obvious absorbance at ~220nm. The absorbance should be from Co ions and 2-MIM disassembly from ZIF-67. The total concentration is about 0.5 mmol L<sup>-1</sup> by comparing with UV-vis spectra of Co<sup>2+</sup> (Figure S2a) and 2-MIM (Figure S2b) solution. Assuming all Co<sup>2+</sup> and 2-MIM are disassemblied from 0.452 mmol of ZIF-67 (0.1g) in 60mL methanol, the concentration of Co<sup>2+</sup> and

2-MIM is about 7.53 mmol L<sup>-1</sup> and 15.06 mmol L<sup>-1</sup>. UV-vis spectra test indicates that a part of Co<sup>2+</sup> and 2-MIM disassembled from ZIF-67. The removal of a number of linkers or metal nodes can not cause collapse of the framework structure due to the high degree of connectivity of MOF<sup>[1]</sup>. The loss of ligand or metal ions can result in the formation of large avenue in MOF (UIO-66), facilitating the developement of catalytic activity<sup>[2]</sup>, adsorbance of water molecular<sup>[3]</sup> and accommodation of guest molecular<sup>[4]</sup>. In our work, the disassembly process can also make the formation of some defect, thus opening the window of ZIF-67 and allowing the access of PMo<sub>12</sub> into ZIF-67 cage.

In the second way, the solid was separated fast and directly from hot reaction solution after reaction. The supernatant have shown stronger absorbance at  $\sim 220$ nm than that separated from cooling dispersion, implying the high concentration of Co<sup>2+</sup> and 2-MIM in hot solution (Figure S2c). The trapping process was performed at higher temperature (100°C under solvothermal condition), which can result in the release of more Co<sup>2+</sup> and 2-MIM from ZIF-67 (more defect). The test implies the reassembly of a part of Co<sup>2+</sup> and 2-MIM on ZIF-67 under cooling process, which facilitate the encapsulation of guest molecular.

Based on above analysis, the trapping of  $PMo_{12}$  into ZIF-67 can be proposed as follows: there are presence of dynamic diassembly-reassembly process after dispersion of ZIF-67 into methanol under heating. Some Co<sup>2+</sup> and 2-MIM can be released from ZIF-67 to solution. The process can make the "opening" of the window of ZIF-67, thus creating the large avenue. The cage can accommodate  $PMo_{12}$  after opening the window due to the suitable size (1.16nm vs 1.0 nm). Also, the residual 2-MIM in cage can coordinate with  $PMo_{12}$ , making the stabilization of  $PMo_{12}$  in cage. The reassembly of Co<sup>2+</sup> and 2-MIM on ZIF-67 (especially under cooling process) and the coordination between  $PMo_{12}$  with residual 2-MIM in cage can make the stabilization and encapsulation of  $PMo_{12}$  in cage.

Figure S2d shows the UV-vis spectra of PMo<sub>12</sub> in 60mL methanol and the supernatant separated from dispersion of PMo/ZIF-67-2, PMo/ZIF-67-6 and

PMo/ZIF-67-10. We can see that peaks of PMo<sub>12</sub> in solution gradually decrease with prolonging the solvothermal time, implying its trapping in ZIF-67. The intensive adsorption peaks in Figure S2d indicated that more  $Co^{2+}$  and 2-MIM were deassembled from ZIF-67 under solvothermal condition at 100°C, being favorable for more effective trapping of PMo<sub>12</sub> into ZIF-67.

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**Figure S3.** (a) XPS survey spectra and high-resolution spectra of (b) Mo 3d of PMo/ZIF-67-6.

The XPS survey spectrum shows the presence of Co, Mo, C, N, O, P in PMo/ZIF-67-6, implying a combination of  $PMo_{12}$  and ZIF-67. In the high-resolution spectrum of Mo 3d of PMo/ZIF-67-6, there are two peaks. The peak at 231.8 eV and 234.8eV are related to  $3d_{5/2}$  and  $3d_{3/2}$ , respectively. The peaks have shifted to lower energy region in comparison with  $PMo_{12}$ , which should be relative with the partial electronic interaction of  $PMo_{12}$  clusters with ZIF-67 host.



Figure S4. SEM image of ZIF-67.

Figure S4 shows SEM image of ZIF-67. It can be clearly observed that the formation of ZIF-67 with polyhedron morphology and uniform size.



Figure S5. XRD patterns of PMo/ZIF-67-6-6N.

**Figure S5** shows XRD patterns of PMo/ZIF-67-6-6N. The peaks located at 32.83, 35.88, 40.50, 53.08, 47.12, 70.67, 73.67, and 75.45 can be observed, which indexed to (400), (331), (422), (511), (440), (733), (822), and (555) planes of  $Mo_6Co_6C$ , respectively (JPCDS 65-8115). The peaks located at 44.2, 51.5 and 75.8 correspond to (111), (200) and (220) planes of Co (JPCDS 15-0806). The test indicates the formation of  $Mo_6Co_6C$  after calcinating PMo/ZIF-67-6 under N<sub>2</sub>.



Figure S6. XRD patterns of PMo/ZIF-67-6-6N taken on X-ray powder diffractometer with high power (6 KW).

To solidly prove the existence of  $Co_6Mo_6C$ , we have performed the XRD test on X-ray powder diffractometer with high power (6 KW). We can see the obvious diffraction peaks of  $Co_6Mo_6C$  in the pattern. The peaks located at  $2\theta = 43.1^{\circ}$ ,  $40.5^{\circ}$ ,  $35.8^{\circ}$ ,  $32.8^{\circ}$  and  $47.2^{\circ}$  and  $73.7^{\circ}$  can be indexed to (511), (422), (331), (400), (440) and (822) planes of  $Co_6Mo_6C$ . The peaks located at  $2\theta = 44.2^{\circ}$ ,  $51.5^{\circ}$ ,  $75.8^{\circ}$  should be indexed to (111), (200), (220) planes of metal Co.



Figure S7. SEM image of ZIF-67-6N.

Figure S7 shows SEM image of ZIF-67-6N. We can see the presence of polyhedron with uniform size. Being different from ZIF-67, the surface of polyhedron in ZIF-67-6N was very rough, implying the formation of nanoparticles.



Figure S8. EDS mapping of N and P for PMo/ZIF-67-6-6N. Scale bar is 500 nm.



Figure S9. TG curve of PMo<sub>12</sub>/ZIF-67-6-6N tested in air atmosphere.

Figure S9 shows TG curve of PMo<sub>12</sub>/ZIF-67-6-6N. A obvious loss weight between room temperature and 400°C should be ascribed to the decomposition of carbon. The increase of weight between 530 to 900°C should be due to the oxidation of Mo and Co species. The amount of carbon in PMo<sub>12</sub>/ZIF-67-6-6N is about 9.4% based on TG analysis.



Figure S10. High-resolution scans of N 1s and P 2p of PMo/ZIF-67-6-6N (M represented Mo or Co)

The high-resolution of N 1s XPS peaks can be divided into three peaks, the peaks at 397.8 eV, 399.3 eV and 399.2 eV denotes to the pyridinic-N, pyrrolic-N and graphitic-N respectively. Compared with the pyrrolic-N, Pyridinic-N and graphitic-N exhibit higher activity ,which is conducive to enhancing catalytic activity. After deconvolution of curve, we can see the sub-peaks associated with P-O (13.4.2 eV), P-C (133.2 eV). The peaks at 130.4 eV should be ascribed to P-Mo or P-Co. Based on PMo<sub>12</sub> structure, the P is surround by Mo, thus it can easily combine with Mo in the products due to close distance of Mo and P.



Figure S11. XRD patterns of PMo/ZIF-67-6-6N after etching away of Co with IM  $H_2SO_4$ .

After etching in 1M H<sub>2</sub>SO<sub>4</sub>, most of Co can be removed and Co<sub>6</sub>Mo<sub>6</sub>C@Carbon can be obtained.The peaks located at  $2\theta = 43.1^{\circ}$ ,  $40.5^{\circ}$ ,  $35.8^{\circ}$ ,  $32.8^{\circ}$  and  $47.2^{\circ}$  and  $73.7^{\circ}$  can be indexed to (511), (422), (331), (400), (440) and (822) planes of Co<sub>6</sub>Mo<sub>6</sub>C. No obvious peaks about metal Co can be observed.



Figure S12. (a) EDX spectrum, (b-d) TEM, (e) STEM image and corresponding to EDS mapping of Mo, Co, C, N and P for PMo/ZIF-67-6-6N-E (The E presents the etching away of Co NPs in H<sub>2</sub>SO<sub>4</sub>).

EDX spectrum shows that the ratio of Co/Mo from TEM-EDX is about 1.64. TEM test of Co<sub>6</sub>Mo6C@C shows the uniform distribution of small particles encapsulated in carbon polyhedrons, being similar with that of PMo/ZIF-67-6-6N. The HRTEM of one particles shows the interplanar distance of 0.21 nm, which is indexed to (511) plane of Mo<sub>6</sub>Co<sub>6</sub>C. The EDS mapping confirms the uniform distribution of Co, Mo, C, P, N (Figure S8) through the polyhedrons.



**Figure S13.** (a) XPS survey spectra and high-resolution spectra of (b) Co 2p, (c) Mo 3d, of PMo/ZIF-67-6-6N after etching with H<sub>2</sub>SO<sub>4</sub> (PMo/ZIF-67-6-6N-E).

After removing Co NPs, the XPS spectra of Co2p and Mo3d are consistent with that in Mo<sub>6</sub>Co<sub>6</sub>C. For Mo3d, a sub-peak corresponding to Mo<sup>4+</sup> can be observed after deconvolution of curve, which should be due to slight oxidation of Mo species in etching process by H<sub>2</sub>SO<sub>4</sub> (Figure S13b). In Co2p XPS, the sub-peaks indexed to Co<sup>2+</sup>, Co<sup>3+</sup> and Co<sup>0</sup> can be observed. The presence of Co<sup>0</sup> shows the presence of Co NPs still after etching due to protecting role of carbon layers. However, the ratio of Co<sup>0</sup> for PMo/ZIF-67-6-6N-E has much lower than that for PMo/ZIF-67-6-6N, implying the lower content of Co<sup>0</sup> in former than later. The combination of XRD and XPS indicates the removement of most of Co NPs from PMo/ZIF-67-6-6N by acid etching.



**Figure S14.** (a) XRD patterns of PMo/ZIF-67-6-5N, PMo/ZIF-67-6-7N, PMo/ZIF-67-6-8N and ZIF-67-7N; (b) the SEM image of PMo/ZIF-67-6-7N.

Figure S14a shows XRD patterns of PMo/ZIF-67-6-5N, PMo/ZIF-67-6-7N, PMo/ZIF-67-6-8N and ZIF-67-7N. We can observe Co peaks for PMo/ZIF-67-6-5N from calcinating PMo/ZIF-6 at 500°C, and there are no obvious peaks relative with carbides. The sample from the calcination of PMo/ZIF-67-6 at 600 °C is composed of Mo<sub>6</sub>Co<sub>6</sub>C with little amount of Co metal (Figure S5, 6). When increasing the calcination temperature to 700°C, the intensive peaks corresponding to Co<sub>3</sub>Mo<sub>3</sub>C can be seen, in company with little amount of Co. No peaks about mono-metal carbides can be observed, such as cobalt carbide and molybdenum carbide in PMo/ZIF-67-6-7N. The PMo/ZIF-67-6-8N shows similar XRD pattern with

PMo/ZIF-67-6-7N, but with more marrow peaks due to the formation of large particles induced by high temperature. In addition, only peak of metal Co can be observed for the sample of ZIF-67-7N. The results indicate that 1) the higher temperature are favourable to the formation of the bi-metal carbide, and 2) the components and size of materials can be tuned by changing the synthetic parameters. The SEM image of PMo/ZIF-67-6-7N shows the presence of polyhedrons with relative uniform size and morphology. There are many partices on polyhedrons, ascribing to the formation of BTMCs. The particle sizes are larger than that on PMo/ZIF-67-6-6N polyhedrons due to high calcination temperature.



**Figure S15.** SEM images of (a) PMo/ZIF-67-2, (b) PMo/ZIF-67-6 and (c) PMo/ZIF-67-10. The TEM images of d) PMo/ZIF-67-2, (e) PMo/ZIF-67-6 and (f) PMo/ZIF-67-10. The g-j is the TG curves of PMo/ZIF-67-2, PMo/ZIF-67-6 and PMo/ZIF-67-10 under air atmosphere.

Figure S15 shows SEM images of PMo/ZIF-67-2, PMo/ZIF-67-6 and PMo/ZIF-67-10. It is obvious that the presence of regular polyhedrons in PMo/ZIF-67-2 (a, d), PMo/ZIF-67-6 samples (b, e). While for PMo/ZIF-67-10, there are presence of the particles with non-polyhedron morphology besides regular polyhedrons. The result indicates that catching too much of PMo<sub>12</sub> can result in the damage of the structure of ZIF-67, just like that the excessive gas can make the balloon explosion. Thus, the control of the trapping time is very important to obtain

PMo/ZIF-67-x with regular morphology of ZIF-67.

Figure S15 g-j gives TG curves of PMo/ZIF-67-2, PMo/ZIF-67-6 and PMo/ZIF-67-10 under air atmosphere. It is shown that the loss weight is about 53% for PMo/ZIF-67-2 in the range of 30 to 700 °C. Under the same temperature range, the mass lose of PMo/ZIF-67-6 and PMo/ZIF-67-10 are about 47% and 37.5% of its initial mass respectively. We can see that the mass loss is decreased in order of PMo/ZIF-67-2 > PMo/ZIF-67-6 > PMo/ZIF-67-10. The little loss weight implies the high amount of PMo<sub>12</sub> in PMo/ZIF-67 samples. The result indicates the that the amount of PMo<sub>12</sub> in final solid increase with the increase of the solve-thermal time.



**Figure S16.** SEM image of Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6 prepared with Na<sub>2</sub>MoO<sub>4</sub> as Mo source.

Figure S16 shows SEM image of Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6 prepared with Na<sub>2</sub>MoO<sub>4</sub> as Mo source. We can see the presence of many irregular particles, implying the damage of the morphology of ZIF-67 in this case.



**Figure S17.** XRD patterns of PMo/ZIF-67-6-7N and Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6-7N prepared in the presence of  $H_3PMo_{12}O_{40}$  and Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O as Mo source, respectively.

We further studied the impact of the differrent kinds Mo source on the final products. To this, the H<sub>3</sub>PMo<sub>12</sub>O<sub>40</sub> was instead by Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O in the solvothermal Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6. After process to obtain calcination under  $N_{2}$ , Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6-7N was obtained. Figure S14 shows the XRD patterns of PMo/ZIF-67-6-7N and Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6-7N. It is worth noting that Mo<sub>6</sub>Co<sub>6</sub>C (JCPDS 65-8115) can be obtained in the case of using PMo<sub>12</sub> as Mo source (PMo/ZIF-67-6-7N). For Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6-7N which obtained from the use of Na<sub>2</sub>MoO<sub>4</sub> as Mo source, a mixture of MoC (JCPDS 06-0546) and Co<sub>2</sub>C (JCPDS 50-1371) can be formed. The formation of mixture is relative to the weak interaction of Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O with 2-MIM and small molecular size, which is not favorable for the stabilization of Mo source in cage and their contact with Co source in close distance.



Scheme S1. The formation mechanism of PMo/ZIF-67.

When ZIF-67 was heated in methanol, partial of 2-MIM and  $Co^{2+}$  of ZIF-67 dissolved in the solvent. The disassembly process can make the formation of some defects, thus opening the window of ZIF-67 and allowing the access of PMo<sub>12</sub> into ZIF-67 cage. The cage can accommodate PMo<sub>12</sub> after opening the window due to the suitable size. Also, the residual 2-MIM in cages can coordinate with PMo<sub>12</sub>, making the stabilization of PMo<sub>12</sub> in cages. The reassembly of  $Co^{2+}$  and 2-MIM on ZIF-67 (especially under cooling process) and the coordination between PMo<sub>12</sub> and residual 2-MIM in cages can make the stabilization and encapsulation of PMo<sub>12</sub> in cages. The flexibility of the trapping process makes easy tune of the trapping amount of PMo<sub>12</sub> in ZIF-67 by tuning reaction time. The amount of PMo<sub>12</sub> in final PMo/ZIF-67 increase with the increase of solve-thermal time. Also, the trapping of too much PMo<sub>12</sub> can result in the damage of the structure of ZIF-67, just like the blast of balloon as filling too much gases.



**Figure S18.** Cyclic voltammograms (CV) of PMo/ZIF-67-6-6N (a) and ZIF-67-6N (c) taken under different rates from 40 to 220 mVs<sup>-1</sup>. The corresponding capacitive current at 0.15 V as a function of scan rate for PMo/ZIF-67-6-6N (b) and ZIF-67-6N (d).

In order to evaluate the electrochemical active surface areas, the cyclic voltammetry (CV) method was used to measured the electrochemical double layer capacitance (EDLC, Cdll) of the studied catalysts. The Cdll values of PMo/ZIF-6-6N is 27.32 mF cm<sup>-2</sup>, much higher than that of ZIF-67-6N (15.58 mF cm<sup>-2</sup>),which is associated with its large exchange current density (0.67mA cm<sup>-2</sup>).



**Figure S19.** (a) Polarization curves of the PMo/ZIF-67-6 calcinated at 500, 600, 700 and 800°C under  $N_2$  in 1M KOH at the scan rate of 5 mVs<sup>-1</sup> for HER. (b) Polarization curves of PMo/ZIF-67-2-6N, PMo/ZIF-67-6-6N and PMo/ZIF-67-10-6N in 1M KOH at the scan rate of 5 mVs<sup>-1</sup> for HER. (c) Polarization curves of Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6-6N and PMo/ZIF-67-6-6N under the same condition for HER.

We calcinate PMo/ZIF-67-6 at different temperature and find that the catalyst treated at 600 °C shows the best HER performance (Figure S19 a). When we adjust the the solvothermal time of the precursor, we can acknowledge that there is little difference between PMo/ZIF-67-6-6N and PMo/ZIF-67-10-6N (Figure S19 b). But the morphology for PMo/ZIF-67-6-6N is much more regular. We also use Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O as Mo source instead of H<sub>3</sub>PMo<sub>12</sub>O<sub>40</sub>, the obtained material displays poorer HER catalytic activities than the catalyst prepared with H<sub>3</sub>PMo<sub>12</sub>O<sub>40</sub> (Figure S19 c).



Figure S20. N<sub>2</sub> adsorption-desorption isotherms of PMo/ZIF-67-6-6N.

The N<sub>2</sub> adsorption-desorption isotherms of PMo/ZIF-67-6-6N revealed that the sample has Type I isotherm. The Brunauer–Emmett–Teller (BET) surface area of PMo/ZIF-67-6-6N is about 240  $m^2/g$ .



**Figure S21.** Work function (WF) drawings of (a) PMo/ZIF-67-6-6N and (b) Pt/C and (c) ZIF-67-6N.

The surface work function (WF) represents the ability of catalysts to trap electrons, which is measured by scanning kelvin probe test (SKP). The Scanning Kelvin Probe (SKP) technique is a non-destructive test method that can detect differences across the surface of a sample. The high WF value of Pt leads to it's remarkable performance for HER. The PMo/ZIF-67-6-6N shows work function (WF) of 5.68 eV (Figure S20a). The WF of ZIF-67-6N is about 5.78 eV (Figure S20c). The PMo/ZIF-67-6-6N give work function more close to Pt (Figure S20b), implies the enhanced trapping ability to electrons after surface activation.



**Figure S22.** (a) Polarization curves of PMo/ZIF-67-6 calcinated at 500, 600, 700 and 800°C under N<sub>2</sub> in 1M KOH at the scan rate of 5 mVs<sup>-1</sup> for OER. (b) Polarization curves of PMo/ZIF-67-2-7N, PMo/ZIF-67-6-7N and PMo/ZIF-67-10-7N in 1M KOH at the scan rate of 5 mVs<sup>-1</sup> for OER. (c) Polarization curves of Na<sub>2</sub>MoO<sub>4</sub>/ZIF-67-6-7N and PMo/ZIF-67-6-7N under the same condition for OER.

From Figure S22, we can see that PMo/ZIF-67-7N exhibits highest activity for OER than the catalysts obtained from other preparation conditions. At the same time, the catalyst prepared by using PMo<sub>12</sub> as Mo source shows better activity than that prepared by using Na<sub>2</sub>MoO<sub>4</sub> as Mo source for OER. Above results indicate the important effect of the synthetic parameters on the performance.



**Figure S23.** Cyclic voltammograms (CV) of (a) PMo/ZIF-67-6-7N and (c) ZIF-67-7N taken under different rates from 40 to 220 mVs<sup>-1</sup>. The corresponding capacitive currents at 1.25 V as a function of scan rate for (b) PMo/ZIF-67-6-7N and (d) ZIF-67-7N.

The cyclic voltammetry (CV) method was used to study details on charge storage capabilities of the electrocatalysts, which can be used to evaluate the electrochemical active surface areas of the catalysts. The Cdl values of PMo/ZIF-67-6-7N and ZIF-67-7N are 47.4 and 9.0 mF cm<sup>-2</sup>, respectively. It is obviously that PMo/ZIF-67-6-7N has a much higher electrochemically active surface area than ZIF-67-7N.



**Figure S24.** The experimentally determined and theoretically calculated amounts of  $H_2$  on PMo/ZIF-67-6-6N and  $O_2$  on PMo/ZIF-67-6-7N. The Faradic efficiency (FE) is close to 100% for HER and 93% for OER at test of 60 min.



**Figure S25.** HER performance of PMo/ZIF-67-6-6N and PMo/ZIF-67-6-6N-E (E represented etching) in 1M KOH. The performance have no obvious change before and after etching of Co NPs.



Figure S26. XRD pattern of PMo/ZIF-67-6-7N after OER test (40 CV cycles).

Figure S26 shows XRD pattern of PMo/ZIF-67-6-7N after OER test (40 CV cycles). The XRD patterns show the existence of intensive peaks of CoOH in company with the slight low peaks of Co metal, indicating the oxidation of Co in PMo/ZIF-67-6-7N. The peaks of carbides can not be obviously seen.



Figure S27. XPS of Co2p of PMo/ZIF-67-6-7N after OER test (40 CV cycles).

The high-resolution Co 2p spectrum can be deconvoluted into six peaks. The two peaks at 780.2 and 795.0 eV corresponding to the  $2p_{3/2}$  and  $2p_{5/2}$  of Co<sup>3+</sup>. The peaks related to Co<sup>2+</sup>  $2p_{3/2}$  and  $2p_{5/2}$  are located at 782.1 and 797.5eV. The XPS of Co<sup>2+</sup> and Co<sup>3+</sup> can be ascribed to Co(OH)<sub>2</sub>, being consistent with XRD result.



Figure S28. XPS of Mo3d of PMo/ZIF-67-6-7N after OER test (40 CV cycles).



**Figure S29.** The OER performance of PMo/ZIF-67-6-7N and PMo/ZIF-67-6-7N-E (E represented etching of Co NPs in  $H_2SO_4$ ) in 1M KOH. The  $\eta 10$  and  $\eta 100$  for PMo/ZIF-67-6-7N are 295 and 370mV respectively, lower than 320 and 410mV for PMo/ZIF-67-6-7N-E. The result indicate the contribution of Co NPs in PMo/ZIF-67-6-7N on OER.



**Figure S30.** The photograph for overall water splitting reaction in a two-electrode configuration. The inset is an optical photograph showing the generation of  $H_2$  and  $O_2$  bubbles for PMo/ZIF-67-6-6N||PMo/ZIF-67-6-7N on Ni foam.



Figure S31. SEM image of S-PMo/ZIF-67-6N.

Figure S31 shows SEM image of S-PMo/ZIF-67-6N. We can observe the formation of spheres with crumpled surface. The morphology is obviously different from the regular polyhedron morphology of PMo/ZIF-67-6-6N.



Figure S32. XRD patterns of S-PMo/ZIF-67-6N.

The XRD patterns of sample (S-PMo/ZIF-67-6N) prepared based on simultaneous route (like POMOF-based route) is mainly composed of Co with no obvious peaks corresponding to  $Mo_xCo_xC$ . In contrast, the sample (PMo/ZIF-67-6-6N) obtained based on "trapping" route is composed of  $Mo_6Co_6C$  and Co. It is obvious that the "trapping" route is effective to give  $Mo_xCo_xC$ .



**Figure S33.** Polarization curves of PMo/ZIF-67-6-6N and S-PMo/ZIF-67-6N in 1M KOH at the scan rate of 5 mVs<sup>-1</sup> for HER.

Figure S33 shows the polarization curves of PMo/ZIF-67-6-6N and S-PMo/ZIF-67-6N in 1M KOH at the scan rate of 5 mVs<sup>-1</sup> for HER. In order to reach the current density of 10 mA cm<sup>-2</sup>, S-PMo/ZIF-67-6N needs an overpotential of 114 mV, whereas PMo/ZIF-67-6-6N only requires an overpotential of 83 mV. Thus, the catalyst prepared based on "trapping" route exhibits better activity than that of the catalyst obtained based on simultaneous route (like POMOF-based route) for HER.



**Figure S34.** Polarization curves of PMo/ZIF-67-6-7N and S-PMo/ZIF-67-6N in 1M KOH at the scan rate of 5 mVs<sup>-1</sup> for OER.

Figure S34 shows the polarization curves of PMo/ZIF-67-6-7N and S-PMo/ZIF-67-6N in 1M KOH at the scan rate of 5 mVs<sup>-1</sup> for OER. As for PMo/ZIF-67-6-7N, an overpotential of 295 mV is needed to reach the current density of 10 mA cm<sup>-2</sup>, while S-PMo/ZIF-67-6N needs the overpotential of 330 mV. In addition, the overpotential of 430mV and 370mV are needed to reach the current density of 100 mA cm<sup>-2</sup> for S-PMo/ZIF-67-6N and PMo/ZIF-67-6-7N respectively. It is obviously that PMo/ZIF-67-6-7N prepared based on "trapping" route shows superior performance for OER to the catalyst obtained based on simultaneous route.

**Table S2.** Comparison of HER performance of PMo/ZIF-67-6-6N with othernon-noble metal HER electrocatalysts in alkaline condition.

Catalyst	Loading density (mg/cm²)	Current density (j) (mA cm <sup>-2</sup> )	Overpotential at corresponding J (mV)	IR compensatio n	Reference
Mo <sub>2</sub> C micro- particles	0.8	10	190	$\checkmark$	Angew. Chem. Int. Ed. <b>2012</b> , 51,
Bulk MoB	2.3	1	150		12703.
Mo <sub>2</sub> C Nanotube	0.75	10	112	$\checkmark$	Angew. Chem. Int. Ed. <b>2015</b> , 54, 15395
MoxC		10	128	$\checkmark$	Nanoscale, <b>2017</b> ,9, 7260–7267
Mo <sub>2</sub> C nanorods	0.43	10	130	$\checkmark$	<i>Appl. Catal. B:</i> <i>Environ.</i> , <b>2014</b> , <i>154</i> , 232-237.
MoP/FTO	0.1	10	190	$\checkmark$	<i>J.Mater. Chem.A,</i> <b>2015</b> , <i>3</i> , 4368–4373
Mo <sub>2</sub> C@NC	~0.28	10	60	$\checkmark$	Angew.Chem.Int.Ed . <b>2015,</b> 54, 10752 -10757
Mo <sub>2</sub> C	0.102	10	176		J. Mater. Chem. A
Mo <sub>2</sub> N	0.102	10	353	-	<b>2015</b> , <i>3</i> , 8361.
MoP	0.86	10	150		Energy Environ. Sci. <b>2014</b> ,7, 2624
MoC/C			138		J.Mater. Chem.A.
Mo <sub>2</sub> C/C		10	96		<b>2017</b> , <i>5</i> , 4879–4885
MoCx	0.8	10	151		<i>Nat.</i> <i>Commun.</i> , <b>2015</b> ,6, 6512-6519.
Mo <sub>2</sub> C-C		10	149		Nano Enerov
MoP-C	0.84	10	169		<b>2017</b> 32 511–519
MoS <sub>2</sub> -C	0.01	10	203		2017,52 511 517
MoS <sub>2</sub> @Mo P	0.35	10	119		Nanoscale, <b>2016</b> ,8, 11052–11059
MoP	0.86	10	~140		Energy Environ. Sci., <b>2014</b> ,7, 2624–2629

MoxC-Ni@ NCV	1.1	10	126		J.Am.Chem. Soc. <b>2015</b> , 137, 15753 - 15759
Mo <sub>2</sub> C embedded N-doped carbon nanotubes	~ 3	10	257		J.Mater. Chem. A, <b>2015</b> ,3, 5783
Co-P film		10	94	$\checkmark$	Angew. Chem.Int. Ed. <b>2015</b> , 54, 6251 –6254
CoP/CC	0.92	10	209		J.Am.Chem. Soc. <b>2014</b> , 136, 7587–7590
Co-NRCN	0.28	1	160		Angew Chem Int
Ts		10	370		<i>Ed.</i> <b>2014</b> , <i>53</i> ,4372
	0.283	10	191		
		20	212		
Co D/NC		100	277		<i>Chem. Mater.</i> <b>2015</b> ,
CO-F/INC		10	154		27, 7636–7642
	1.0	20	173		
		100	234		
CoMoS	0.84	10	98		ACS Appl. Mater. Interfaces 2017, 9, 5288–5294
PMo/ZIF-6 7-6-6N	0.708	10	83	$\checkmark$	This work

onset  $\eta_{10} \,(mV)$  $\eta_{100} (mV)$ Catalyst overpotential (mV) PMo/ZIF-67-6-5N 69 119 249 PMo/ZIF-67-6-6N 83 202 36 PMo/ZIF-67-6-7N 69 118 256 PMo/ZIF-67-6-8N 109 142 284

 Table S3. The HER performance of catalysts from the calcination of PMo/ZIF-67-6 at different temperature.

 Table S4. The HER performance of catalysts prepared by the calcination of

	onset	η <sub>10</sub> (mV)	η <sub>100</sub> (mV)
Catalyst	overpotential		
	(mV)		
PMo/ZIF-67-2-6N	53	125	248
PMo/ZIF-67-6-6N	36	83	202
PMo/ZIF-67-10-6N	35	83	190

PMo/ZIF-67 from different solvothermal time.

**Table S5.** The HER performance of catalysts from the calcination of the precursorprepared by using Na2MoO4 and PMo12 as Mo source.

Catalyst	onset	η <sub>10</sub> (mV)	η <sub>100</sub> (mV)
	overpotential		
	(mV)		
Na <sub>2</sub> MoO <sub>4</sub> /ZIF-67	102	173	287
-6-6N			
PMo/ZIF-67-6-6N	36	83	202

**Table S6.** The HER performance of catalysts prepared based on "tapping" route and"simultaneous" route (POMOF-based route).

Catalyst	onset overpotential (mV)	η <sub>10</sub> (mV)	η <sub>100</sub> (mV)
S-PMo/ZIF-67-6N	71	114	221
PMo/ZIF-67-6-7N	36	83	202

Catalyst	onset overpotential (mV)	η <sub>10</sub> (mV)	η <sub>100</sub> (mV)
PMo/ZIF-67-6-5N	330	340	410
PMo/ZIF-67-6-6N	320	330	400
PMo/ZIF-67-6-7N	280	295	370
PMo/ZIF-67-6-8N	340	350	420

**Table S7.** The OER performance of catalysts from the calcination of PMo/ZIF-67-6 at different temperature.

 Table S8. The OER performance of catalysts prepared by the calcination of

	onset	η <sub>10</sub> (mV)	η <sub>100</sub> (mV)
Catalyst	overpotential		
	(mV)		
PMo/ZIF-67-2-7N	270	350	420
PMo/ZIF-67-6-7N	280	295	370
PMo/ZIF-67-10-7N	340	290	360

PMo/ZIF-67 from different solvothermal time.

**Table S9.** The OER performance of catalysts from the calcination of the precursorprepared by using Na2MoO4 and PMo12 as Mo source.

Catalyst	onset	η <sub>10</sub> (mV)	η <sub>100</sub> (mV)
	overpotential		
	(mV)		
Na <sub>2</sub> MoO <sub>4</sub> /ZIF-67	330	360	440
-6-7N			
PMo/ZIF-67-6-7N	280	295	370

**Table S10.** The OER performance of catalysts prepared based on "tapping" route and"simultaneous" route (POMOF-based route).

Catalyst	onset overpotential (mV)	η <sub>10</sub> (mV)	η <sub>100</sub> (mV)
S-PMo/ZIF-67-6N	280	330	430
PMo/ZIF-67-6-7N	280	295	370

 Table S11. Comparison of OER performance of PMo/ZIF-67-6-7N with other

Catalyst	Loading density (mg/cm²)	Current density (j) (mA cm <sup>-2</sup> )	Overpotential at corresponding j (mV)	IR compensation	Reference
NiFe-PVP		10	297	$\checkmark$	Adv. Energy Mater. <b>2017</b> , 1700220
Ni–Mo nitride	3.5	10	295	$\checkmark$	J. Mater. Chem. A, <b>2017</b> , 13648
MoNi <sub>4</sub> /Ni foam	~1.09	10	280	$\checkmark$	J.Mater.Chem.A, 2017, 5, 2508
MoS <sub>2</sub> /Ni <sub>3</sub> S <sub>2</sub> / Ni foam	9.7	10	218		Angew.Chem. Int. Ed., <b>2016</b> , 128, 6814
Co-P film		10	345	$\checkmark$	Angew. Chem. Int. Ed., <b>2015</b> , 127, 6349
NiCoP	~1.6	10	280	$\checkmark$	Nano Lett., <b>2016</b> , 16, 7718
NiCo <sub>2</sub> S <sub>4</sub> NW/ NF		10	260	$\checkmark$	Adv. Funct. Mater. 2016, 26, 4661–4672
CP/CTs/Co-S	~0.32	10	306	$\checkmark$	ACS Nano 2016, 10, 2342
NiCo <sub>2</sub> O <sub>4</sub>	~1	10	290	$\checkmark$	Angew. Chem. Int. Ed. 2016, 55, 6290
СоР	3.3	10	281	$\checkmark$	ChemSusChem 2016, 9, 472 – 477
СоР	~0.05	10	360	$\checkmark$	ACS Catal. 2015, 5, 4066

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PCPTF	0.1	30	330	$\checkmark$	Adv Mater 2015, 27, 3175
NiCo <sub>2</sub> S <sub>4</sub>	4	10	280	~	Nanoscale 2015, 7, 15122
CoFe <sub>2</sub> O <sub>4</sub>	~1.031	10	378		Acs Appl Mater Inter 2015, 7, 17851.
Co <sub>9</sub> S <sub>8</sub>	5	10	320		Adv. Funct. Mater. <b>2017</b> , 1606585
Co/CoP		10	345		Angew Chem Int Edit 2015, 54, 6251
$Co(S_xSe_{1-x})_2$	~1	10	307		Adv. Funct. Mater. <b>2017</b> , 1701008
PMo/ZIF-67- 6-7N	0.708	10	295	$\checkmark$	This work

Catalyst	Overpotential at 10 mA cm <sup>-2</sup> (mV)		Tafel slope (mV dec <sup>-1</sup> )		Overall water voltage at 10 mA cm <sup>-2</sup> (V)	Reference
	HER	OER	HER	OER		
MoS <sub>2</sub> /Ni <sub>3</sub> S <sub>2</sub> / Ni foam	110	218	83	88	1.56	Angew.Che m. Int. Ed., <b>2016</b> , 128, 6814
NiMo-PVP/ NiFe-PVP	130	297	84	48	1.66	Adv. Energy Mater. <b>2017</b> , 1700220
Ni–Mo nitride	89	295	79	94	1.60	J. Mater. Chem. A, <b>2017</b> , 13648
MoNi₄/Ni foam	28	280	36	79	1.58	J. Mater. Chem. A, <b>2017</b> , 5, 2508
Co <sub>9</sub> S <sub>8</sub>	340	320	68	105	1.60	Adv. Funct. Mater. <b>2017</b> , 1606585
Co-P film	94	345	42	47	~1.64	Angew. Chem. Int. Ed., <b>2015</b> , 127, 6349
NiCoP	32	280	37	87	1.58	Nano Lett., <b>2016</b> , 16, 7718
$Co(S_xSe_{1-x})_2$	122	307	85.7	67.5	1.63	Adv. Funct. Mater., <b>2017</b> , 1701008
NiCo <sub>2</sub> S <sub>4</sub> NW/NF	210	260	81.3	40.1	1.63	Adv. Funct. Mater. 2016, 26, 4661–4672

 Table S12. The performance of the recent reported electrocatalysts in alkaline solution for overall water splitting.

						ACS Nano
CP/CTs/Co-S	190	306	131	72	1.743	2016, 10,
						2342-2348
						Angew.
NiCo <sub>2</sub> O <sub>4</sub>	110	290	49.7	53	1.65	Chem. Int.
						Ed. 2016,
						55, 6290
						-6294
СоР	95	281	60	62	1.613	ChemSusC
						hem 2016,
						9, 472 –
						477
						ACS Catal
CoP	220	360		83	1.81	2015, 5,
						4066.
						Angew
Co/CoP	94	345	42	47	1.645	Chem Int
						Edit 2015,
						54, 6251
						Adv Mater
CoxPO <sub>4</sub> /CoP	380	300		65	1.910	2015, 27,
						3175
						Nanoscale
NiCo <sub>2</sub> S <sub>4</sub>	263	280	141	89	1.68	2015, 7,
						15122
						Acs
CoFe <sub>2</sub> O <sub>4</sub>	356	378		73	1.964	Appl Mater
						Inter 2015,
						7, 17851.
Co-CoOx/N						J Am Chem
doped	~134	260	115		1.624	Soc 2015,
Carbon						137, 2688.
PMo/ZIF-67-						
6-7N  PMo/ZI	83	295	44.5	53	1.61	This work
F-67-6-6N						