

Electronic Supplementary Information

Rational design of oxide/carbon composite to achieve superior rate-capability *via* enhanced lithium-ion transport across carbon to oxide

Jun Hui Jeong^a, Myeong-Seong Kim^a, Yeon Jun Choi^a, Geon-Woo Lee^a, Byung Hoon Park^a,
Suk-Woo Lee^a, Kwang Chul Roh^{b,*} and Kwang-Bum Kim^{a,*}

^a Department of Materials Science and Engineering, Yonsei University, 50 Yonsei-ro,
Seodaemun-gu, Seoul 03722, Republic of Korea

^b Energy and Environmental Division, Korea Institute of Ceramic Engineering and
Technology, 101 Soho-ro, Jinju-si, Gyeongsangnam-do, 52851, Republic of Korea.

* Corresponding authors.

E-mail addresses: rkc@kicet.re.kr (K. C. Roh), kbkim@yonsei.ac.kr (K. B. Kim)

Table S1. Brunauer–Emmett–Teller (BET) specific surface areas and pore volumes of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites

	BET surface area (m² g⁻¹)	Pore Volume (cm³ g⁻¹)
LTO/rGO	25.4	0.208
LTO/L-NPG	43.7	0.311
LTO/M-NPG	44.1	0.327
LTO/H-NPG	41.9	0.364

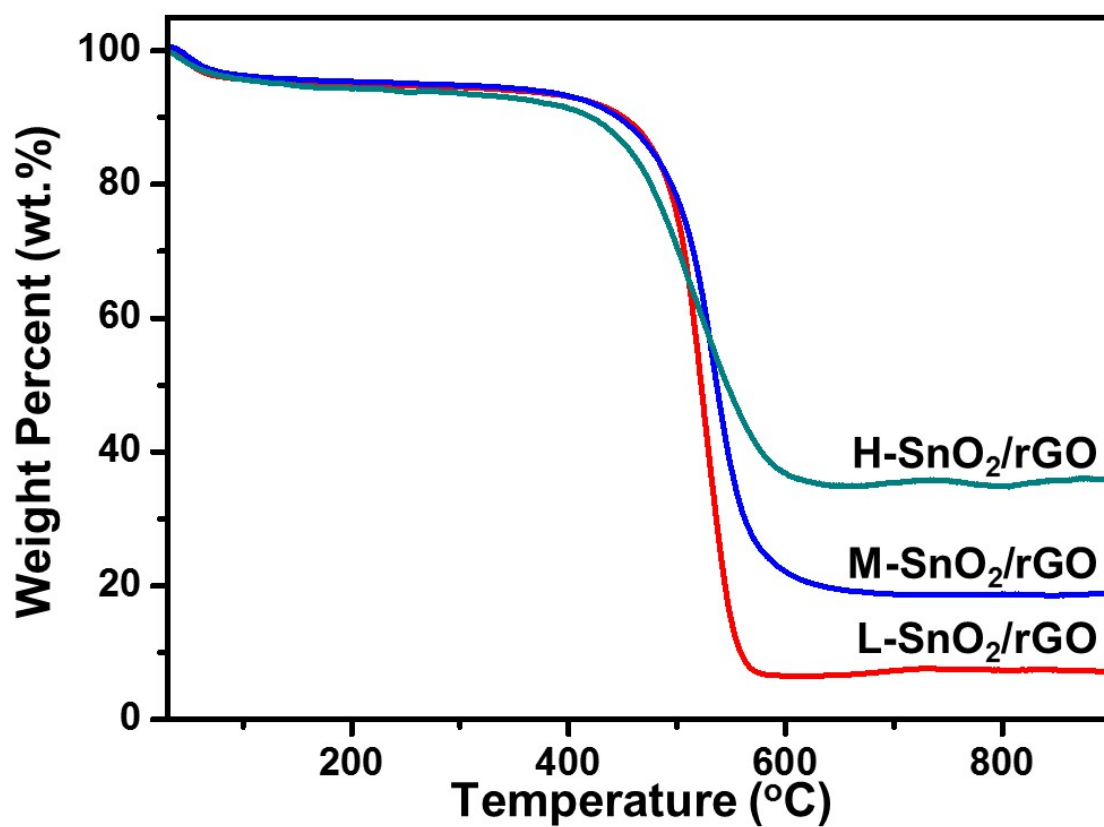


Fig. S1. TGA profiles of SnO₂/rGO composites with different loading amounts of SnO₂.

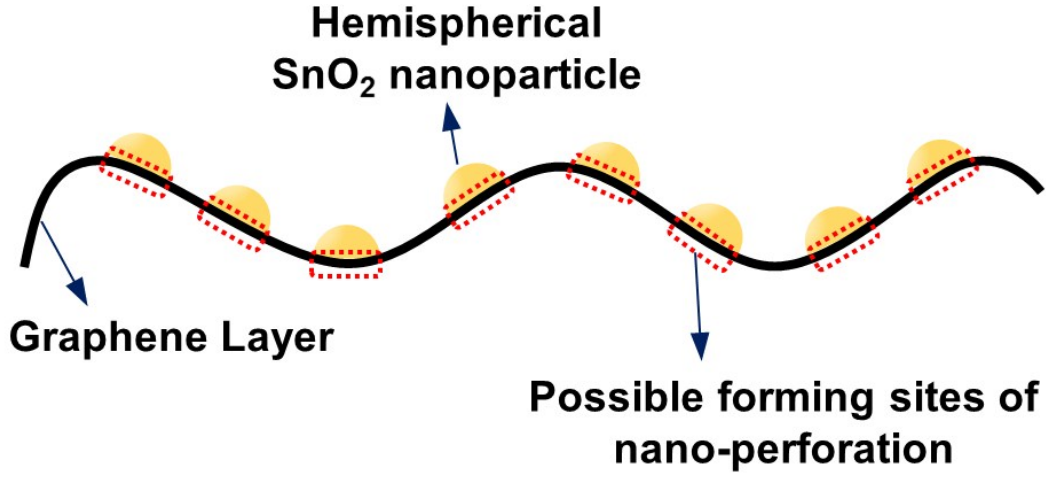


Fig. S2. Schematic diagram of SnO₂/rGO composite illustrating hemispherical SnO₂ nanoparticles deposited on the basal plane of graphene.

As illustrated in Fig. S2, we assumed the hemispherical SnO₂ nanoparticles deposited on the basal plane of graphene in calculating the number of nano-perforations in the NPG. Since nano-perforations were created only at the contact points with the hemispherical SnO₂ nanoparticles, number of nano-perforations in the NPG was determined by the number of hemispherical SnO₂ nanoparticles in the composite. The number of SnO₂ nanoparticles (N), which is equivalent to the number of nano-perforations, in the composites was calculated by dividing the weight of SnO₂ in SnO₂/rGO composites by the weight of a single hemispherical SnO₂ nanoparticle.

$$N = (Wx/100) / \{(4\pi r^3 \rho / 3) / 2\} \quad (\text{Eq. S1})$$

where W is the weight of SnO₂/rGO composite (g) used for TGA, x is the weight percent of SnO₂ nanoparticles in SnO₂/rGO composite (wt%) from TGA, r is the radius of a SnO₂ nanoparticle (3 nm), ρ is the density of SnO₂ (6.95 g cm⁻³).

As the weight percent of SnO₂ nanoparticles in SnO₂/rGO composite increases, the weight percent of rGO in SnO₂/rGO composite decreases. Therefore, the area of rGO available for SnO₂ nanoparticle deposition decreases. Finally, number density of nano-perforations, which is the number of nano-perforations per unit area, in the NPG was calculated as follows: (1) the area of rGO (S , m²) available for SnO₂ nanoparticle deposition in SnO₂/rGO composite was calculated by multiplying the weight of rGO in the composite and the specific surface area of rGO; and (2) the number density of nano-perforations in the NPG (μm^{-2}) was calculated by dividing the number of nano-perforations in the NPG by the area of rGO in SnO₂/rGO composite:

$$S = W\{1-(x/100)\}A \quad (\text{Eq. S2})$$

$$\text{Number Density} = N/S \quad (\text{Eq. S3})$$

where A is the specific surface area of rGO (300 m² g⁻¹) measured in BET analysis.

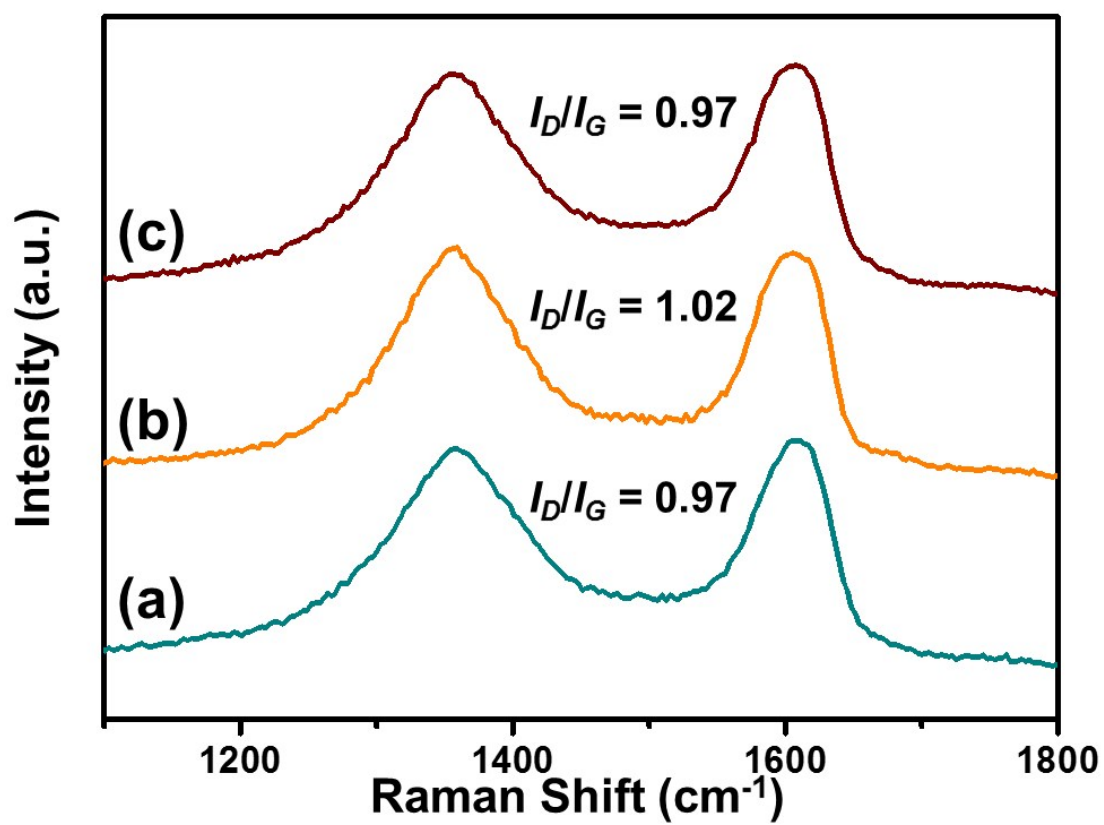


Fig. S3. Raman spectra of (a) NPG, (b) acid-treated NPG (a-NPG), and (c) heat-treated a-NPG.

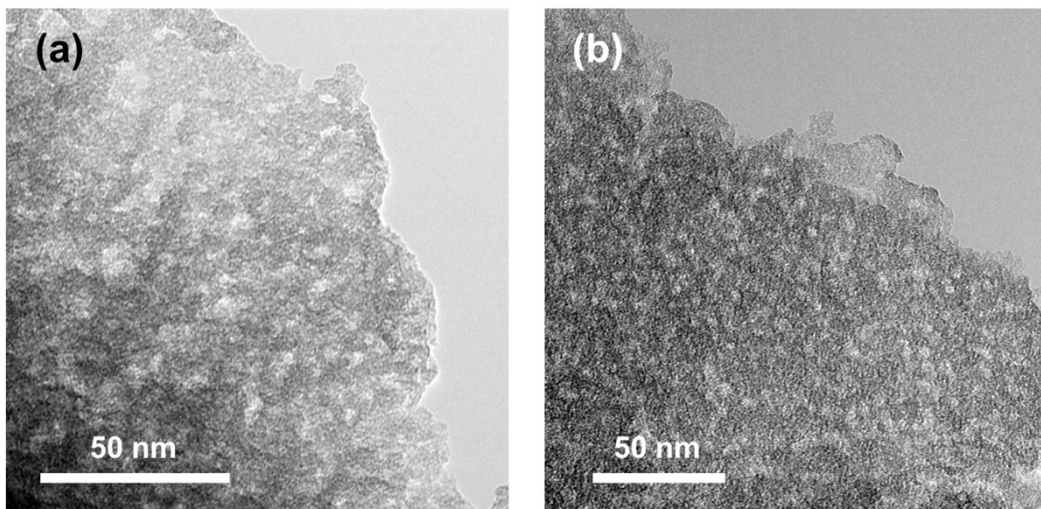


Fig. S4. TEM images of (a) a-NPG and (b) heat-treated a-NPG.

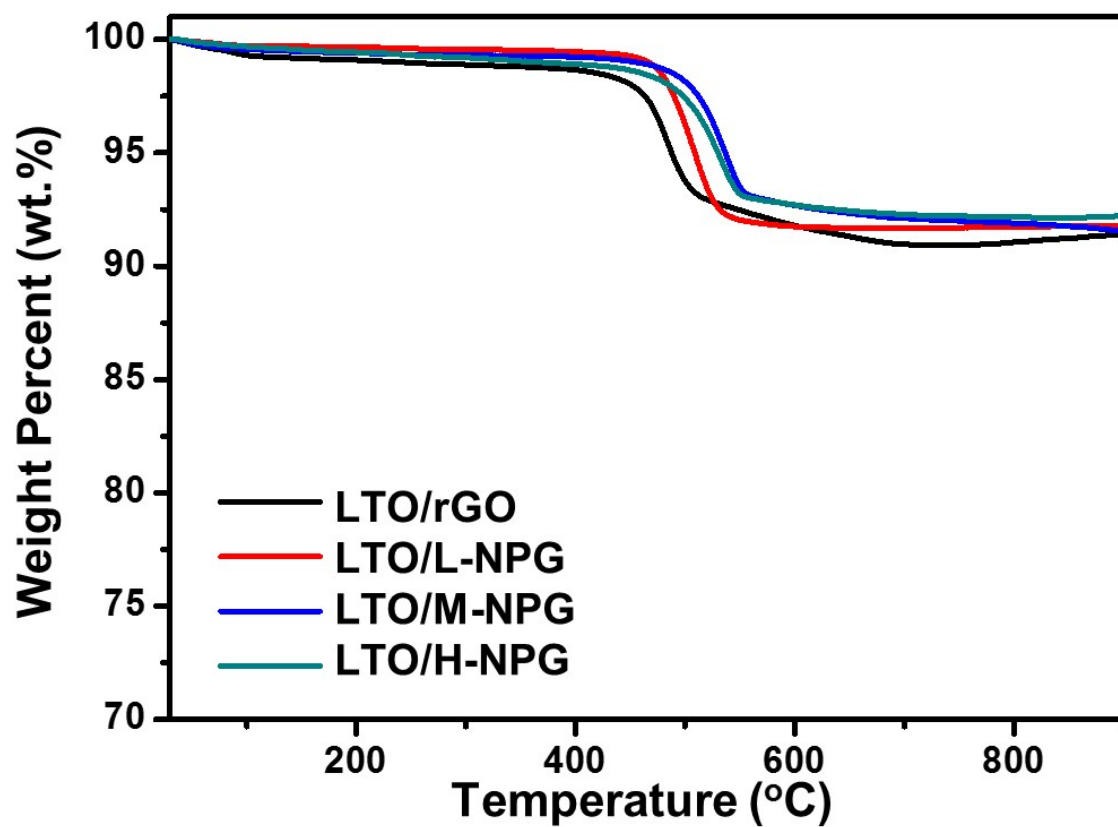


Fig. S5. TGA profiles of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites.

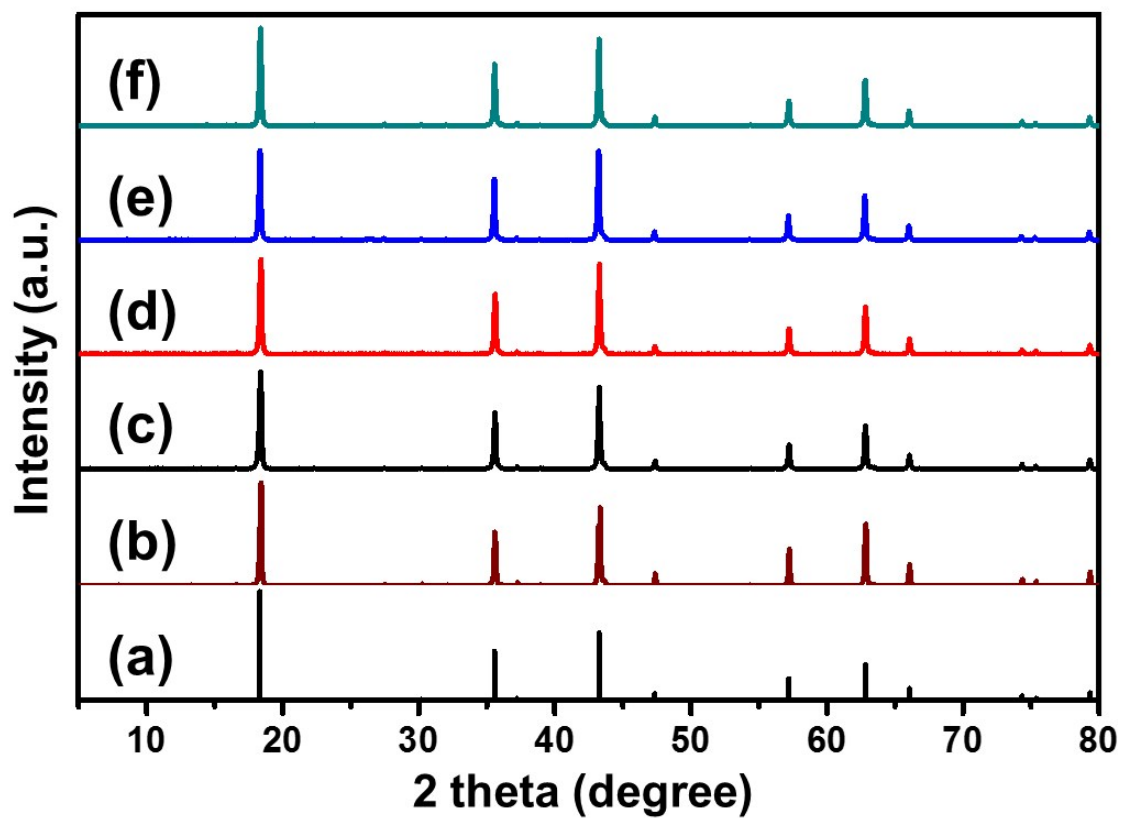


Fig. S6. XRD patterns of (a) JCPDS card No. 490207, (b) bare LTO, (c) LTO/rGO composite, (d) LTO/L-NPG composite, (e) LTO/M-NPG composite, and (f) LTO/H-NPG composite.

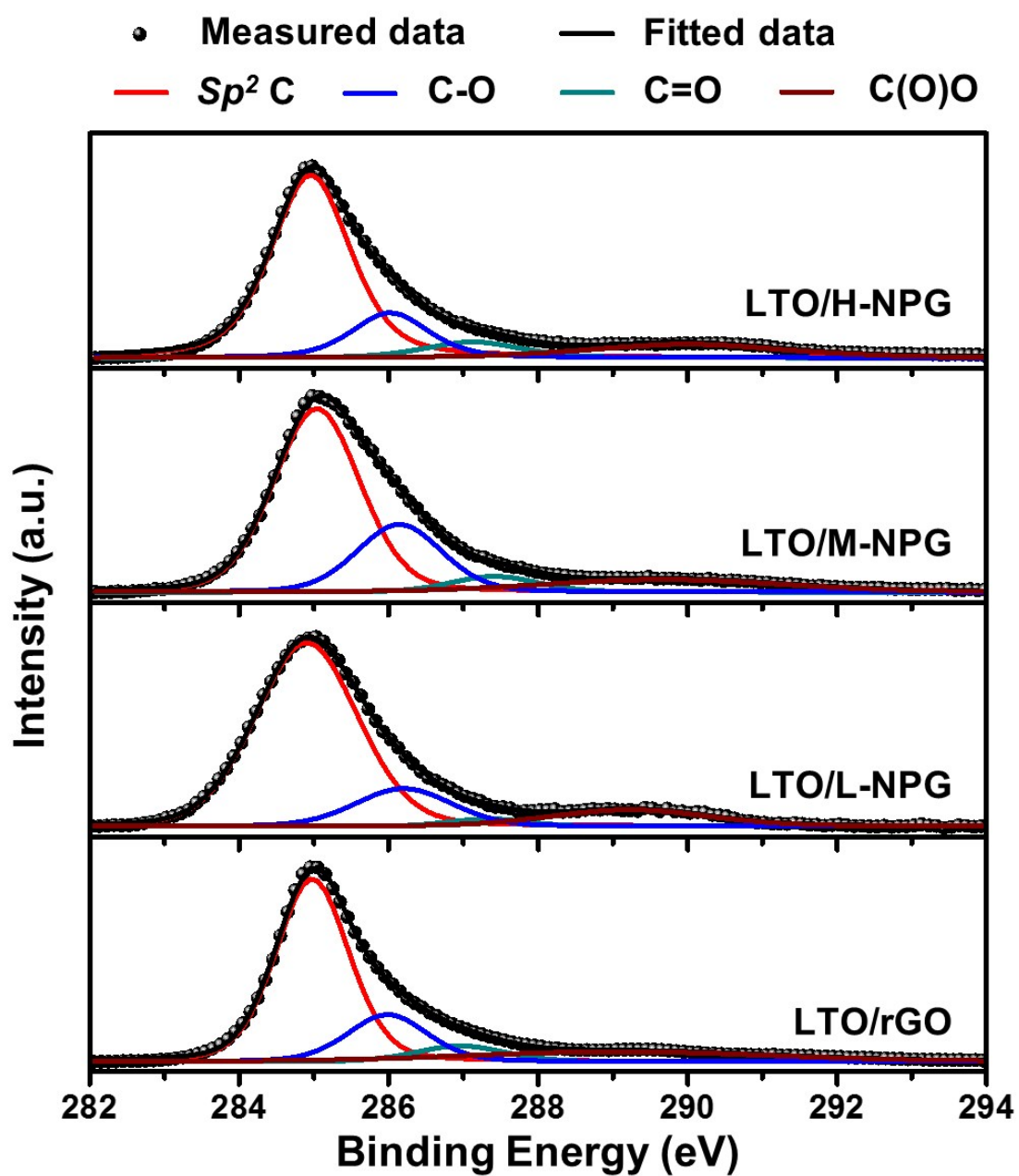


Fig. S7. XPS $C1s$ spectra of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites.

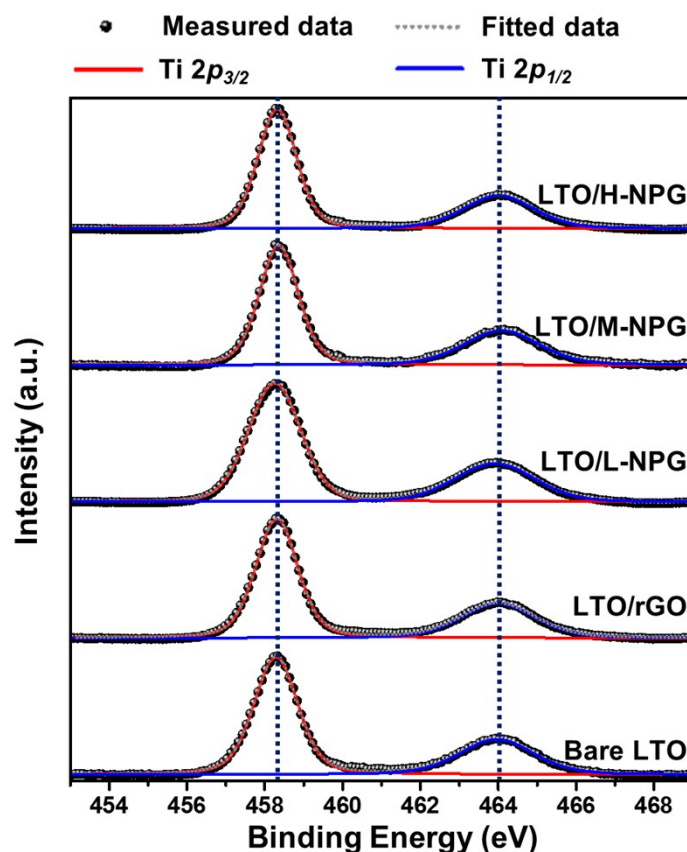


Fig. S8. XPS Ti 2 p spectra of bare LTO and LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites.

The Ti2 p XPS spectra of bare LTO and LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites were analyzed to investigate the oxidation state of Ti in the composites (Fig. S8[†]). All the LTO composites and bare LTO could be deconvoluted into two different peaks centered at 458.3 and 464.0 eV, corresponding to Ti2 $p_{2/3}$ and Ti2 $p_{1/2}$, respectively, which was the characteristic peaks of Ti⁴⁺. In addition, the binding energy difference of 5.7 eV between the two peaks was observed for all the LTO composites and bare LTO, indicating that the oxidation state of Ti was Ti⁴⁺.

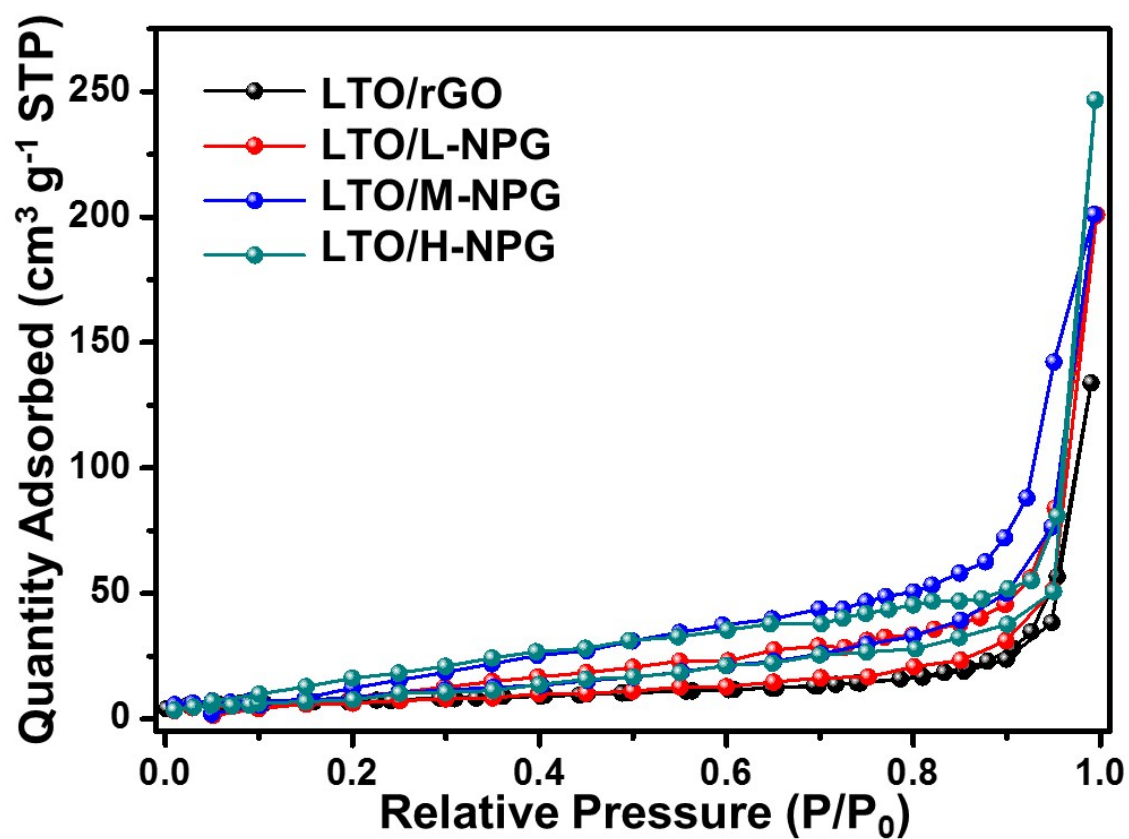


Fig. S9. N₂ gas adsorption/desorption isotherms of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites.

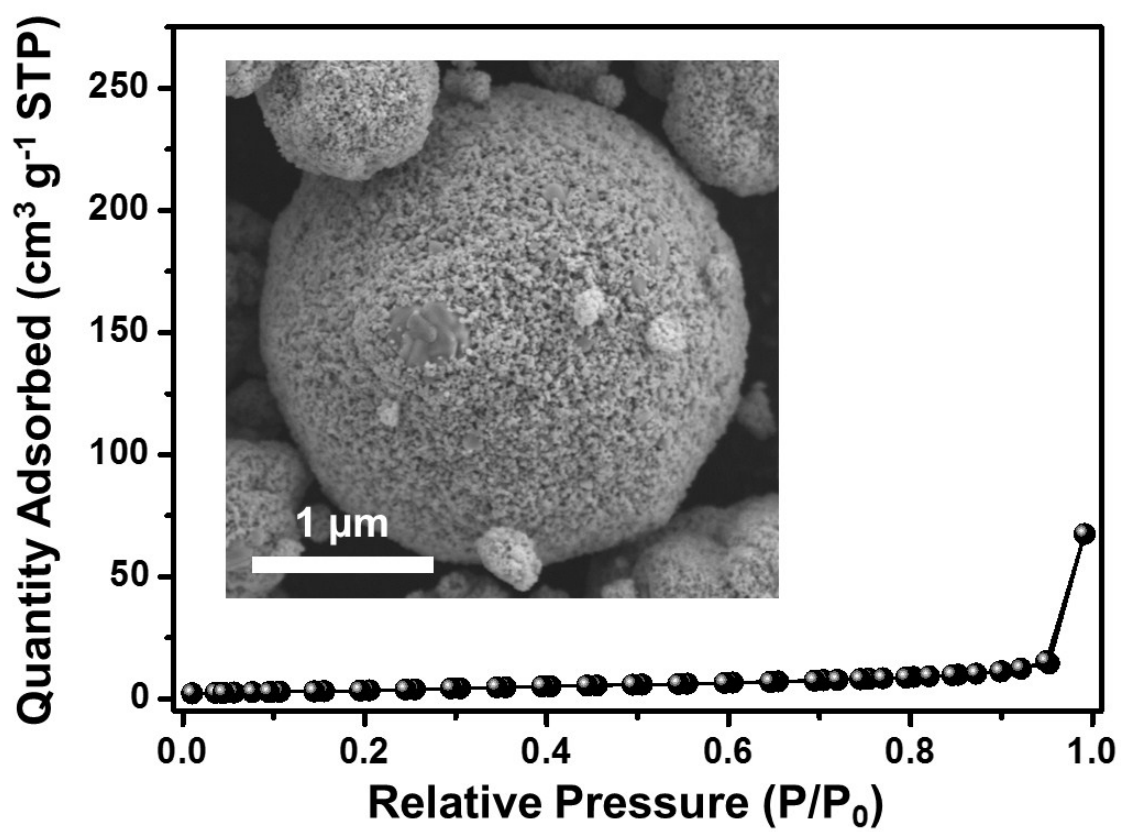


Fig. S10. SEM image (inset) and N₂ adsorption/desorption isotherms of pure LTO microspheres without NPG.

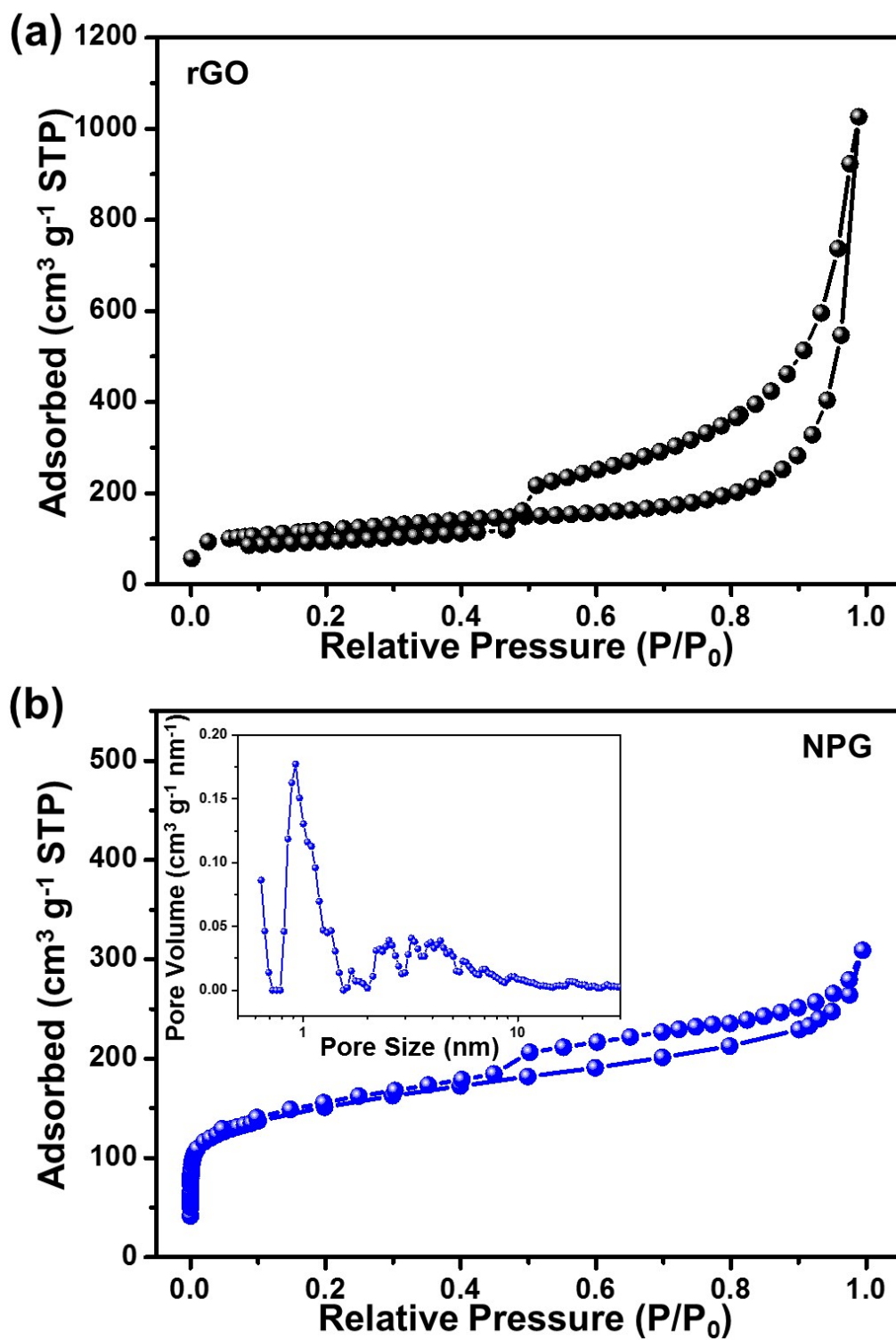


Fig. S11. N_2 adsorption/desorption isotherms of (a) rGO and (b) NPG.

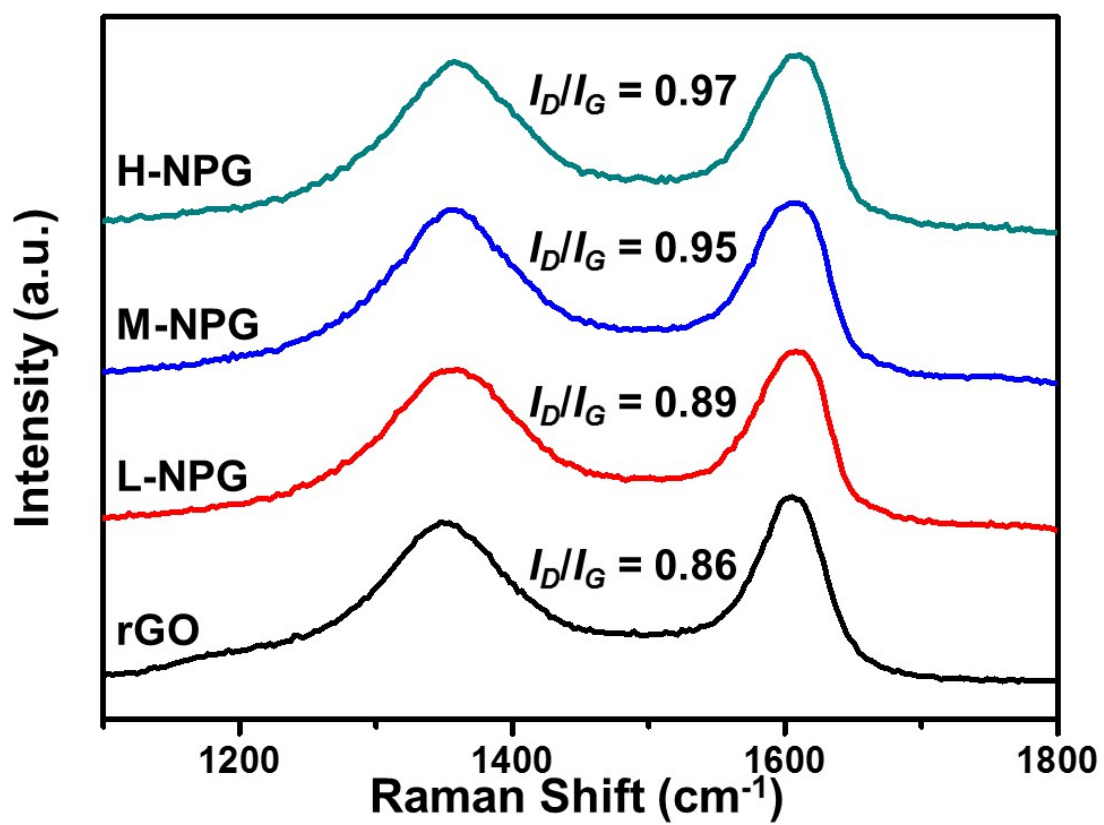


Fig. S12. Raman spectra of rGO, L-NPG, M-NPG, and H-NPG.

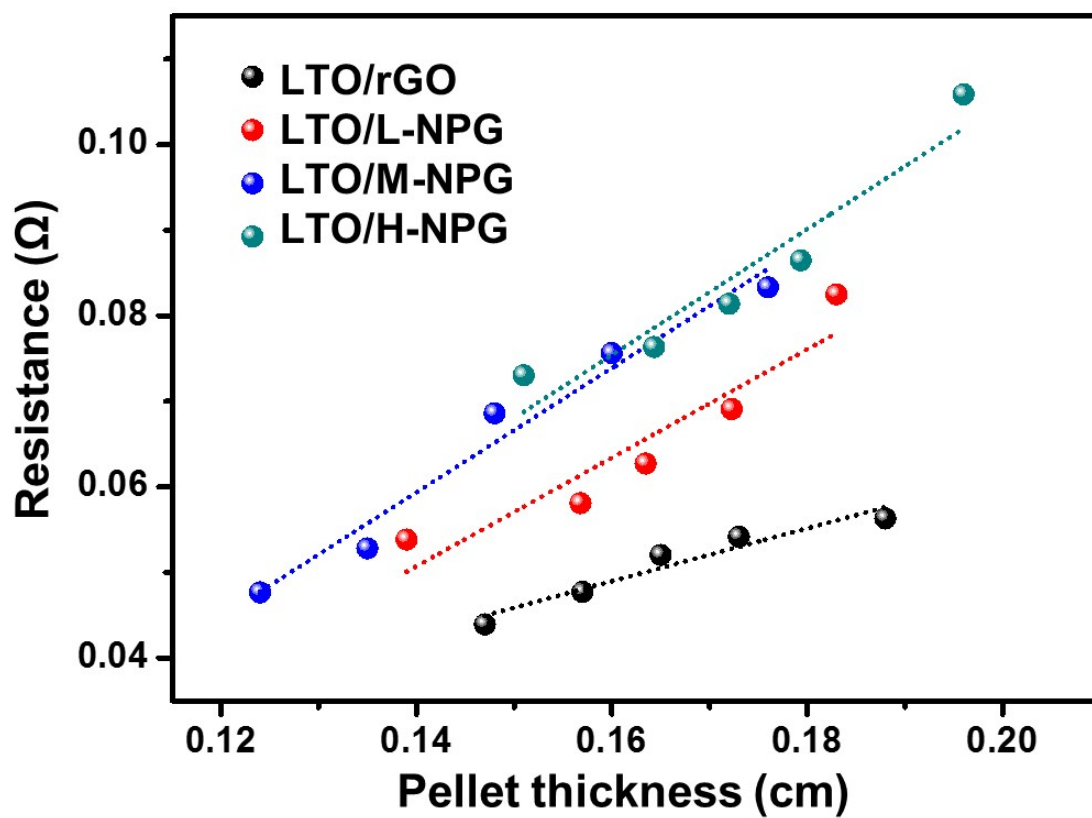


Fig. S13. Linear plot of the resistance as a function of the pellet thickness for calculating the electronic conductivity of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites.

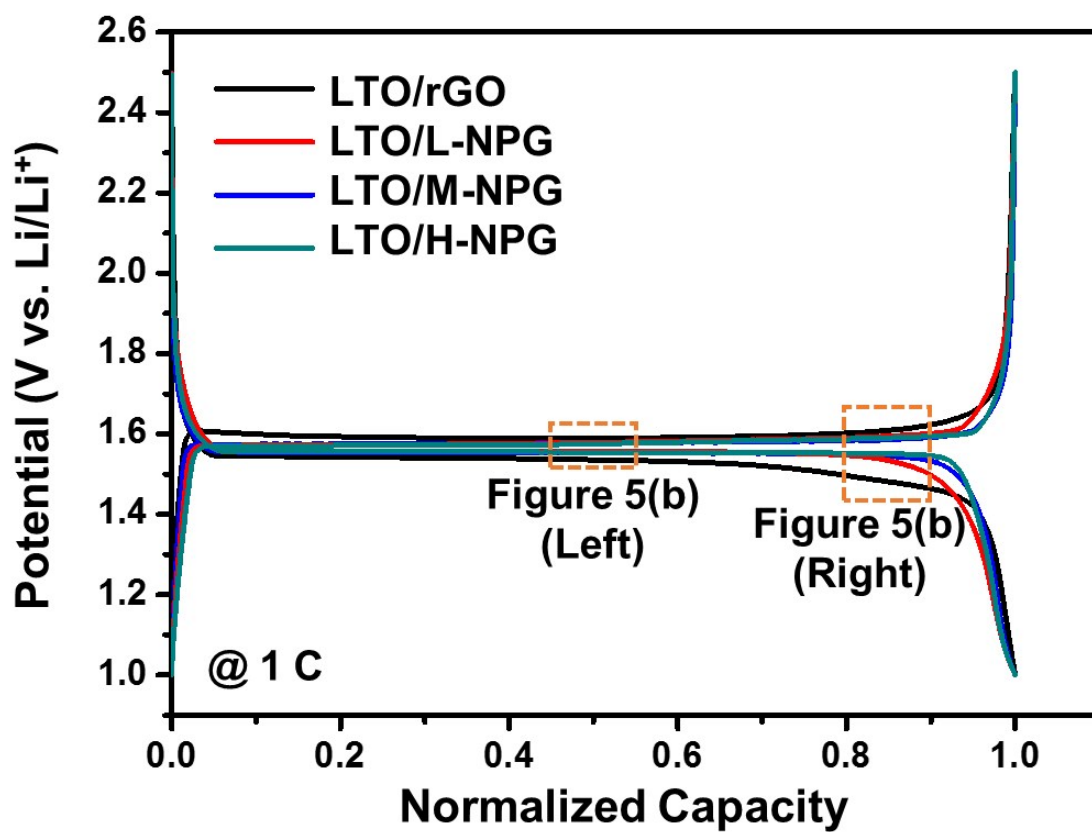


Fig. S14. GCD profiles of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites normalized with respect to the specific discharge capacities at 1 C.

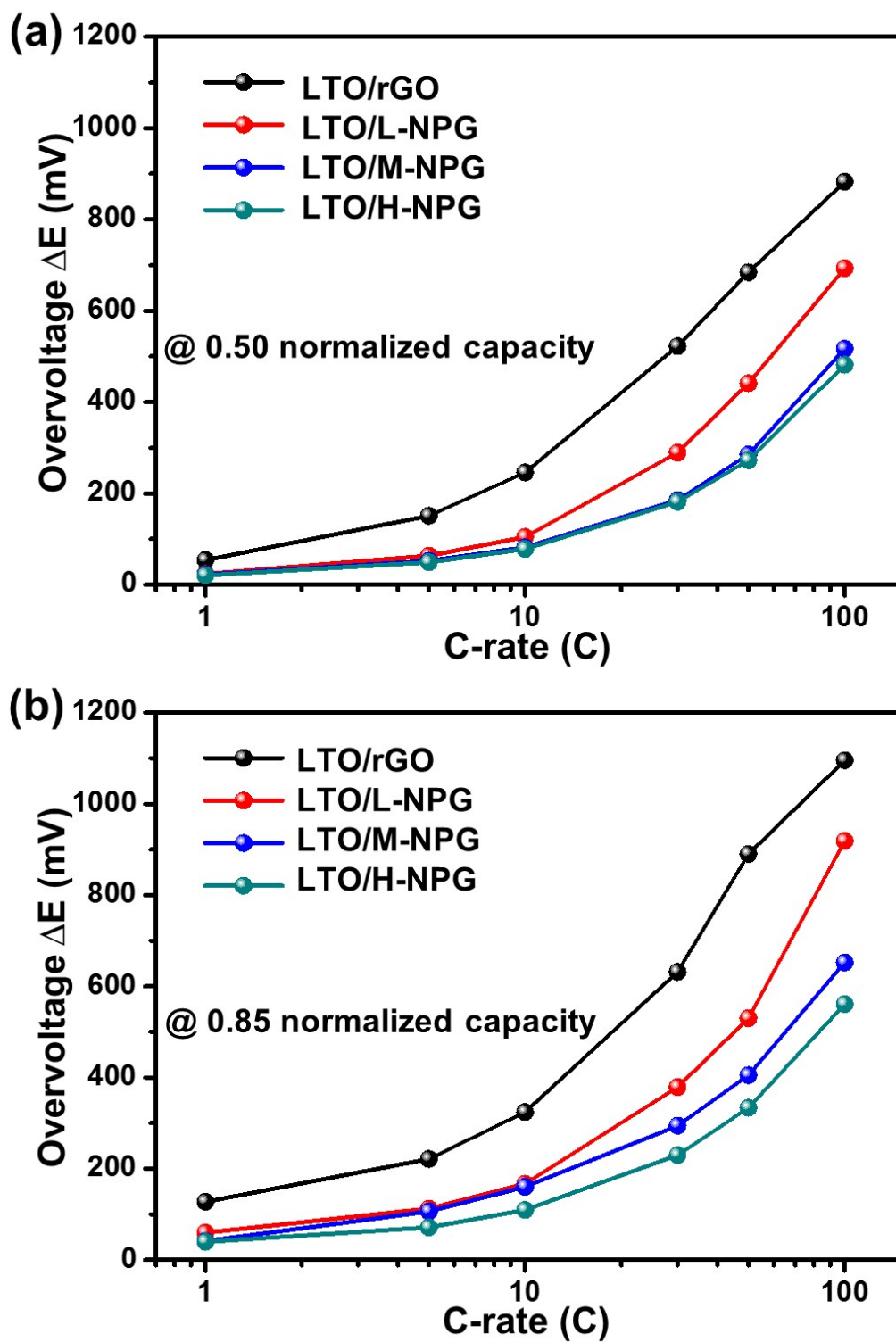


Fig. S15. Overvoltage plots of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites at normalized capacities of (a) 0.5 and (b) 0.85.

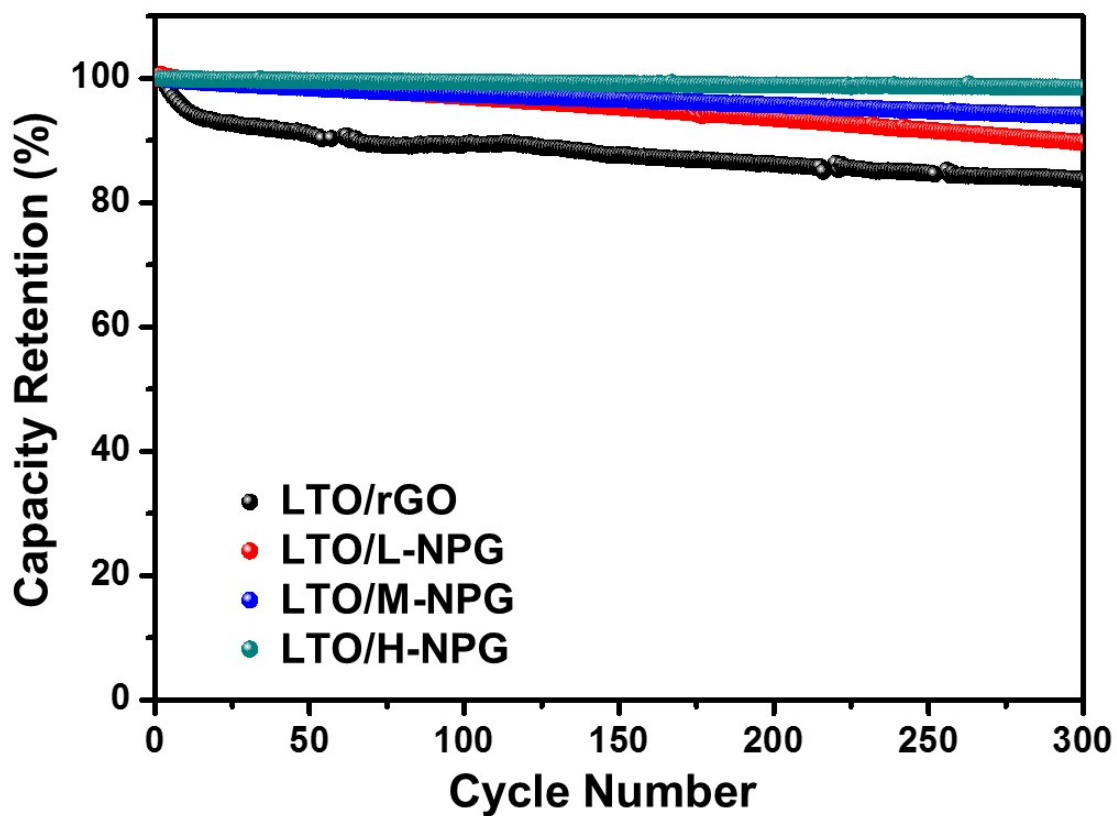


Fig. S16. Cycle performance of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites.

Fig. S16[†] compares the cyclabilities of LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites over 300 cycles at 10 C. The LTO/NPG composites showed better cycling stabilities than LTO/rGO composite. After 300 cycles, LTO/rGO, LTO/L-NPG, LTO/M-NPG, and LTO/H-NPG composites could retain 83.7, 89.7, 95.4, and 98.6% of their initial discharge capacities, respectively.

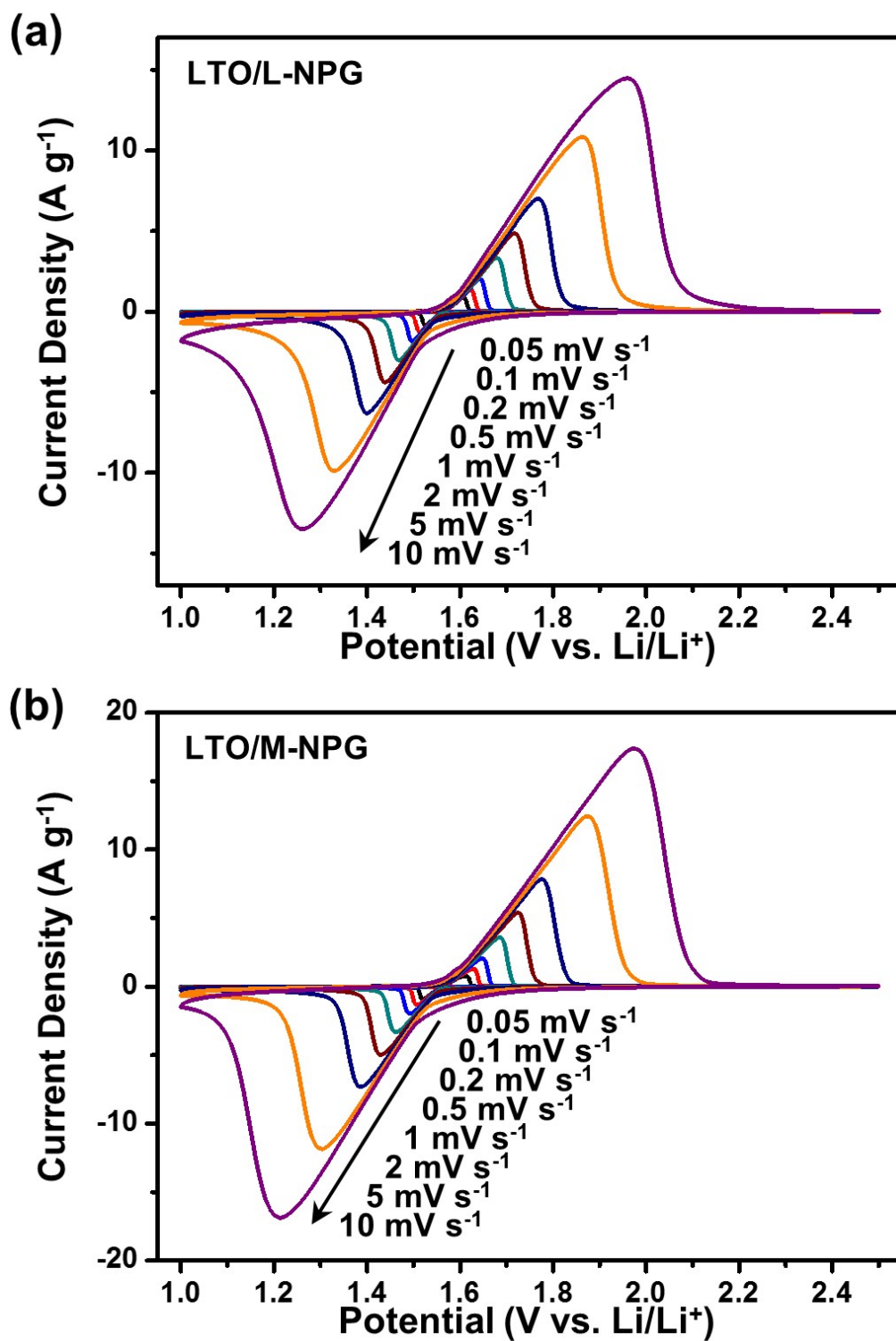


Fig. S17. CV profiles of (a) LTO/L-NPG and (b) LTO/M-NPG composites at increasing scan rates from 0.05 to 10 mV s^{-1} .

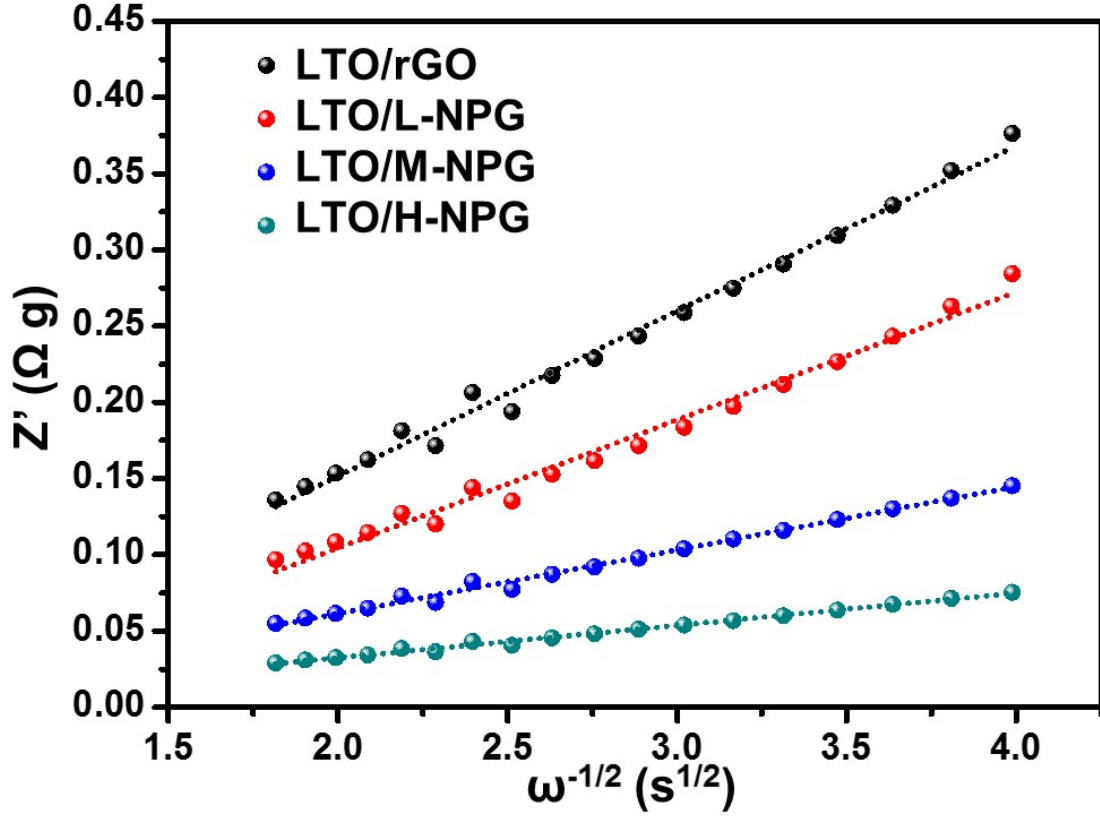


Fig. S18. Plot of real part of impedance vs. $\omega^{-1/2}$ (angular frequency^{-1/2}) for LTO/rGO, LTO/L-NPG, LTO/M-NPG and LTO/H-NPG composites.

To further analyze the EIS data of the LTO composites, the real part of the impedance in the low-frequency region (Z') was plotted against the reciprocal square root of the angular frequency ($\omega^{-1/2}$) at low frequencies in Fig. S18[†]. The linearity in Fig. S18[†] is associated with solid state diffusion of lithium ions in the bulk LTO. Equations S4 and S5 show the relationship between D_{app} and the Warburg factor (σ).

$$Z' = R_s + R_l + R_{ct} + \sigma\omega^{-1/2} \quad (\text{Eq. S4})$$

$$D_{app} = (R^2T^2)/(2A^2n^4F^4C_{Li}^2\sigma^2) \quad (\text{Eq. S5})$$

where D_{app} is the apparent diffusion coefficient, R is the gas constant, T is the absolute temperature, A is the specific surface area of the electrochemically active sites ($\text{m}^2\cdot\text{g}^{-1}$), n is the number of electrons per molecule during the reaction, F is the Faraday constant, C_{Li} is the concentration of the reactant, and σ is the Warburg factor calculated from the slope of the linear part of the EIS profiles between the real axis Z' and $\omega^{-1/2}$ using Equation S4.

In this EIS analysis, we again set D_{app} a constant ($10^{-12} \text{ cm}^2\cdot\text{s}^{-1}$) for LTO in the LTO composites according to the assumptions used for the CV data analysis using Randles–Sevcik equation Fig. 6(c). Finally, Equations S4 and S5 were used to calculate A for LTO/H-NPG, LTO/M-NPG, LTO/L-NPG, and LTO/rGO composites. A was calculated to be 4.327, 2.202, 1.092, and 0.847 $\text{m}^2\cdot\text{g}^{-1}$ for LTO/H-NPG, LTO/M-NPG, LTO/L-NPG and LTO/rGO composites, respectively.

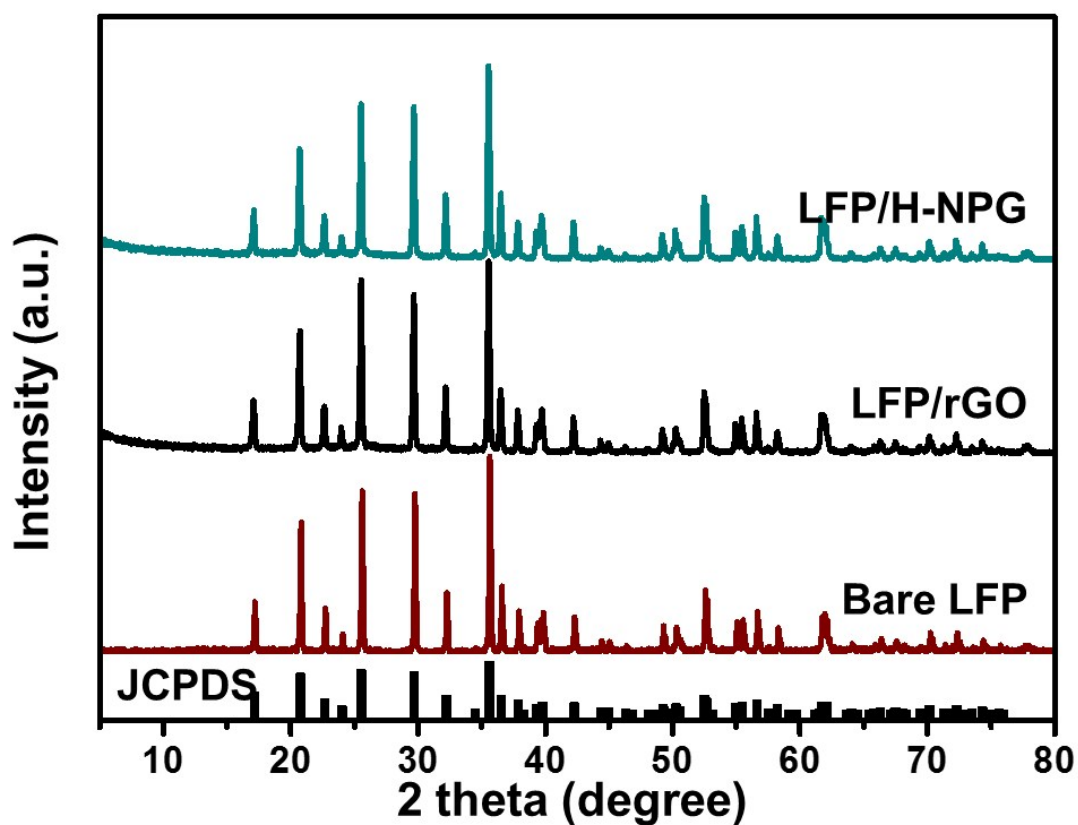


Fig. S19. XRD patterns of bare LFP, LFP/rGO composite, and LFP/H-NPG composite.

As shown in Fig. S19[†], the XRD patterns of the LFP composites exhibited all the standard peaks observed for olivine LFP with the P_{nma} space group (JCPDS card No. 83-2092). No impurity peaks were observed in the XRD patterns of the LFP composites, indicating that the spray-drying process and heat treatment did not cause any structural changes of the olivine LFP.

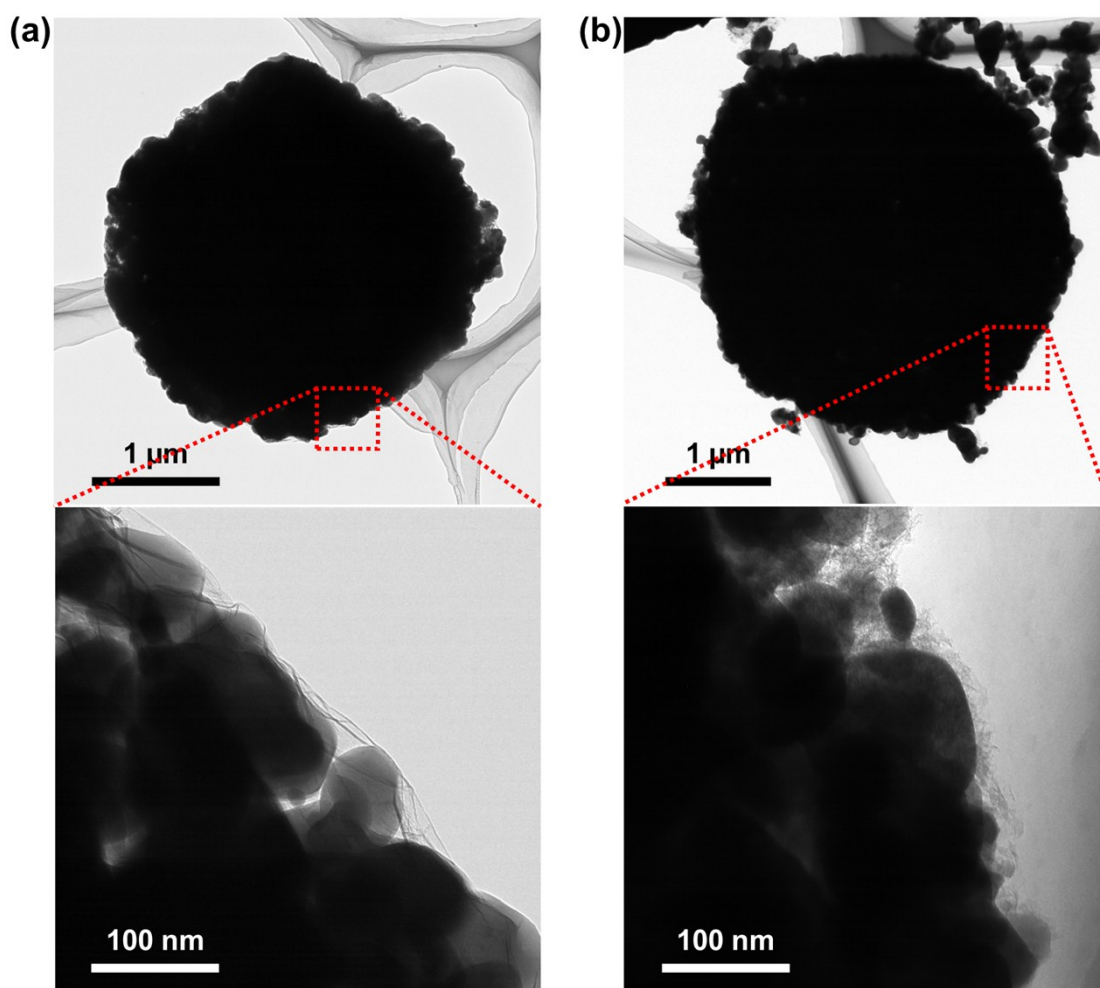


Fig. S20. TEM images of (a) LFP/rGO and (b) LFP/H-NPG composites.

Fig. S20[†] shows TEM images of LFP/rGO and LFP/H-NPG composites. A typical low-magnification TEM image reveals that the primary LFP nanoparticles were densely packed, consisting of secondary spherical micro-particles with diameters of $\sim 3 \mu\text{m}$. In addition, the appearance of the edge of an individual composite particle at higher magnification reveals that the LFP particles were wrapped by a thin layer of NPG (or rGO).

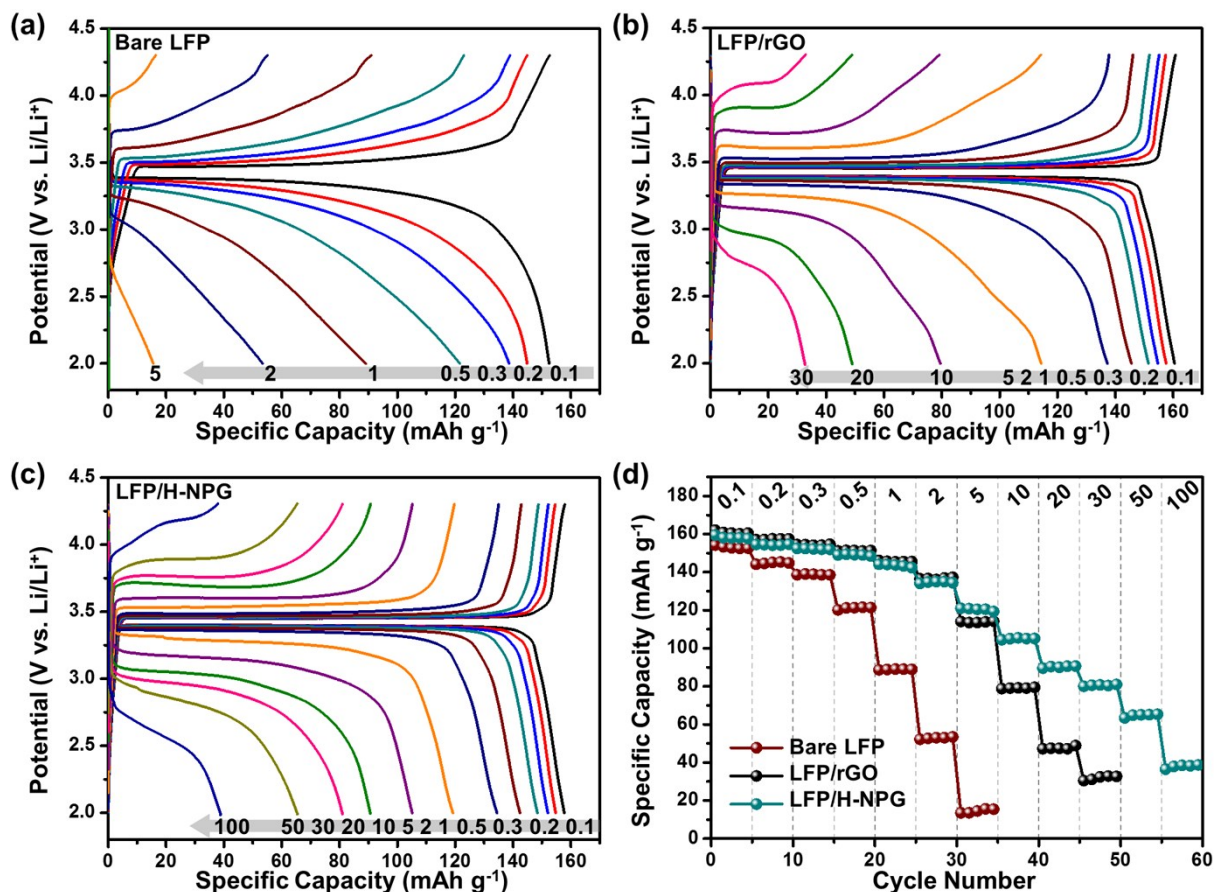


Fig. S21. Electrochemical properties of bare LFP, LFP/rGO composite and LFP/H-NPG composite in 1 M LiPF₆ 1:1 EC/DMC electrolyte in the 2.0–4.3 V potential window. The GCD profiles of (a) bare LFP, (b) LFP/rGO composite and (c) LFP/H-NPG composite at increasing C-rates from 0.1 to 100 C. (d) Plots of the rate capability of bare LFP, LFP/rGO composite, and LFP/H-NPG composite.