## Intramolecular Deformation of Zeotypeborogermanate toward Three-dimensional Porous Germanium Anode for High-rate Lithium Storage

Taeseung Yoon,<sup>‡</sup><sup>a</sup> Gyujin Song,<sup>‡</sup><sup>b</sup> Ahmad M. Harzandi,<sup>a</sup> Miran Ha,<sup>c</sup> Sungho Choi,<sup>b</sup> Sahar Shadman,<sup>a</sup> Jaegeon Ryu,<sup>b</sup> Taesoo Bok,<sup>b</sup> Soojin Park,<sup>\*b</sup> and Kwang S. Kim<sup>\*a</sup>

<sup>a</sup>Center for Superfunctional Materials, Department of Chemistry, Ulsan National Institute of Science and Technology (UNIST), 50 UNIST-gil, Ulsan 44919, Republic of Korea. <sup>b</sup>Department of Energy engineering, School of Energy and Chemical Engineering, Ulsan National Institute of Science and Technology (UNIST), 50 UNIST-gil, Ulsan 44919, Republic of Korea. <sup>c</sup>Department of Chemical Engineering, School of Energy and Chemical Engineering, Ulsan National Institute of Science and Technology (UNIST), 50 UNIST-gil, Ulsan 44919, Republic of Korea.

\*E-mail: kimks@unist.ac.kr (K.S.K) or spark@unist.ac.kr (S.P)



**Figure S1.** SEM images for variable KBGO morphology by tuning the reaction time to (a) 1 day, (b) 2 days, (c) 4 days, and (d) 7 days. To study the optimized particles of KBGO as starting material, we synthesized the variable KBGO particles by adjusting the reaction time. The KBGO (1day) observed the unreacted GeO<sub>2</sub> precursor showing nanometer-sized GeO<sub>2</sub> and the KBGO (2day) showed non-uniform particles due to insufficient reaction time. The KBGO (7day) also showed non-uniform and huge particles. Thus, the KBGO (4 day) sample was chosen as a starting material because of uniform size and morphology.

10 µm

2 µm



Figure S2. STEM-HAADF image of KBGO (4days) and its EDX mapping results.



**Figure S3.** Thermogravimetric analysis-differential scanning calorimeter (TGA-DSC) curve for KBGO materials. The TG curve exhibited weigh loss at around 200 °C, which was resulted from the release of absorbed moisture. At temperature above 200 °C, the TG curve indicated no weight loss till 800 °C. However, the DSC diagram showed exothermic peak at 667 °C, which was resulted from the deformation of KBGO. The integration of this peak was 69.14 J g<sup>-1</sup>.



**Figure S4.** (a) The monoclinic unit cell of  $K_2B_2Ge_3O_{10}$  (KBGO). GeO<sub>4</sub> tetrahedrons (grey) and BO<sub>4</sub> tetrahedrons (green) compose elliptical channels in the crystal. K<sup>+</sup> ions are coordinated by O atoms filling these channels. (b) The trigonal unit cell of  $K_2Ge_4O_9$  (KGO). The octaheral-tetrahedral frameworks of crystal are composed of GeO<sub>4</sub> tetrahedrons (grey) and GeO<sub>6</sub> octahedrons (blue). The cavities in this framework are filled with K<sup>+</sup> ions.<sup>1</sup>



**Figure S5.** Raman spectra measured in 3D-pGe. The 3D-pGe showed a single sharp peak with a Raman frequency of ~300 cm<sup>-1</sup>, which was consistent with Ge-Ge mode in metallic Ge. In addition, other Ge in an oxidation state (~440 cm<sup>-1</sup>) was not observed.<sup>2</sup>



Figure S6. Characterizations of Bulk-Ge. (a) SEM image, (b) TEM image, and (c) XRD patterns.



**Figure S7.** Differential charge/discharge capacity of 3D-pGe at featured cycles at C/20 (1st cycle) and C/5 (other cycles).



**Figure S8.** The cross-sectional SEM images to conform electrode swelling properties of (a) bulk-Ge and (b) 3D-pGe; pristine, after 1st cycle, 50th cycle, and 100th cycle from left. (c) The summarized graph of electrodes expansion based on (a) and (b).



**Figure S9.** Formation cycle at C/10 and cyclic performance at 1C of LCO cathode.



Figure S10. The formation cycle of 3D-pGe/LCO full cell at C/10.

Scan rate [mV s <sup>-1</sup> ]	Capacitive current [%]	Faradaic current [%]
0.2	19.0	81.0
0.4	23.4	76.6
0.6	34.2	65.8
0.8	40.4	59.6
1.0	45.0	55.0
2.0	61.2	38.8
3.0	75.4	24.6
4.0	73.3	26.7
5.0	78.8	21.2

**Table S1.** Comparison of capacitive current and faradaic current of Bulk-Ge at various scan rate.

Scan rate [mV s <sup>-1</sup> ]	Capacitive current [%]	Faradaic current [%]
0.2	17.7	82.3
0.4	31.9	68.1
0.6	49.3	50.7
0.8	63.7	36.3
1.0	74.3	25.7
2.0	79.1	20.9
3.0	88.6	11.4
4.0	94.8	5.2
5.0	95.9	4.1

**Table S2.** Comparison of capacitive current and faradaic current of 3D-pGe at various scan rate.

Materials	Energy density (Max.) [Wh kg <sup>-1</sup> ]	Power density (Max.) [W kg <sup>-1</sup> ]	Reference
Mesoporous Ge microparticles	315.8	315.8	3
Hoenycomb-like Ge	470	199	4
3D-pGe	348	365	This work
зл-bge	348	305	I NIS WORK

**Table S3.** Comparison of the energy/power density of Ge-based anodes.

The energy density of 3D-pGe/LCO full cell was calculated using the following equation,

Energy density (Wh kg<sup>-1</sup>) = Average voltage (V) x gravimetric capacity (mAh kg<sup>-1</sup>)

The weight corresponds to the sum of cathode and anode weights in a cell. Average voltage ( $V_{avg.}$ ) can be calculated from the equation as follows,

$$V_{avg.} = \frac{\int_0^x v(x) \, dx}{\int_0^x dx}$$

where v(x) is the curve of voltage profile (Capacity versus V).

Materials	Tap density (g cm⁻³)	Reference
Ge@carbon cubes	1.13	5
Hollow Ge particles	1.16	6
Mirconized Ge <sub>3</sub> N <sub>4</sub> @carbon	1.10	7
3D-pGe	1.50	This work

**Table S4.** Comparison of the tap density of Ge-based anodes.

## References

- 1 G. J. Redhammer and G. Tippelt, *Acta Crystallogr., Sect. C: Cryst. Struct. Commun.,* 2013, *69*, 995.
- 2 J. H. Parker, D. W. Feldman and M. Ashkin, *Phys. Rev.*, 1967, **155**, 712.
- 3 X. Liu, N. Lin, W. Cai, Y. Zhao, J. Zhou, J. Liang, Y. Zhu and Y. Qian, *Dalton Trans.*, 2018, **47**, 7402.
- 4 J. Liang, X. Li, Z. Hou, T. Zhang. Y. Zhu. X. Yan and Y. Qian, *Chem. Mater.*, 2015, **27**, 4156.
- 5 D. Li, H. Wang, H. K. Liu and Z. Guo, *Adv. Energy Mater.*, 2016, **6**, 1501666.
- 6 S. Choi, J. Kim, N.-S. Choi, M. G. Kim and S. Park, *ACS Nano*, 2015, **9**, 2203.
- 7 C. Kim, G. Hwang, J.-W. Jung, S.-H. Cho, J. Y. Cheong, S. Shin, S. Park and I.-D. Kim, *Adv. Funct. Mater.*, 2017, **27**, 1605975.