Supporting Information

Surface Chemistry Imposing Selective Reduction of CO₂ to CO over

Ta₃N₅/LaTiO₂N Photocatalyst

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Fig. S1. XRD patterns for as-prepared (a) $La_2Ti_2O_7$ and $LaTiO_2N$, and (b) $KTaO_3$ and Ta_3N_5 . SEM image of (c) $La_2Ti_2O_7$ and (d) $KTaO_3$.



Fig. S2. (a) XRD patterns and (b) UV-vis absorption spectra for $LaTiO_2N$ and Ta-doped $LaTiO_2N$. (c) Enlarged XRD patterns for $xTa_3N_5/LaTiO_2N$ (x = 0.1, 0.3, and 0.5).



Fig. S3. (a) UV-vis absorption spectra for $0.3Ta_3N_5/LaTiO_2N$ and mixtures of Ta_3N_5 and $LaTiO_2N$ with the same mole ratio. (b) La 3d XPS spectra for $LaTiO_2N$ and $0.3Ta_3N_5/LaTiO_2N$.



Fig. S4. Mott-Schottky plots for the (a) Ta_3N_5 and (b) $LaTiO_2N$. Electrolyte: 0.5 M Na_2SO_4 solution (pH 7). (c) The relative band positions for $Ta_3N_5/LaTiO_2N$.

The relative band positions of Ta₃N₅ and LaTiO₂N were analyzed by combining their Mott-Schottky (M-S) plots and UV-vis results. As shown in Fig. S4a, the evaluated flat band potential of the as-prepared Ta₃N₅ was about -0.018 V vs reversible hydrogen electrode (RHE), and the measured flat band potential of the LaTiO₂N was -0.132 V vs RHE (Fig. 4b). Normally, the CB bottom for an n-type semiconductor is about 0.2 V above the flat band potential.^[1,2] Then, the corresponding CB positions of the n-type Ta_3N_5 and $LaTiO_2N$ were, respectively, estimated to be -0.218 and -0.332 eV. Based on their band gaps calculated from the UV-vis results (Fig. 3a), the relative VB positions of the Ta₃N₅ and LaTiO₂N were deduced and given in Fig. S4c. Obviously, the composite materials showed a typical type- I heterojunction structure, in which the photogenerated electrons were expected to transfer from the CB of LaTiO₂N to Ta₃N₅, while the holes were followed by an opposite way. This indicated that the CO₂ reduction would actually take place on Ta₃N₅ surface. Here, it was noted that the measured CB and VB levels of LaTiO₂N were in consistent with the reported value,^[3] while the corresponding CB and VB energies of Ta₃N₅ were both lower than the previous reported value.^[4] This was possibly resulted from the high content of TaO_x species (13.1%) in the as-prepared Ta_3N_5 (Fig. 3e). Indeed, both the previous theoretical calculations and experimental results confirmed that the oxygen impurity contributed to the downshift of band position due to the lower energy of O 2p orbital than that of N 2p orbital, which composed the valance band of Ta₃N₅.^[5,6]



Fig. S5. (a) PL patterns and (b) according CO_2 reduction products for LaTiO₂N with different annealing time under air.



Fig. S6. (a) Ta 4f XPS peaks for Ta₃N₅ by nitriding KTaO₃ at 1223 K for 8 h under 500 mL/min NH₃. (b) CO₂ reduction products for Ta₃N₅ samples from KTaO₃ with different nitridation process. (c) N₂ adsorption-desorption isotherms for LaTiO₂N, Ta₃N₅, and $0.3Ta_3N_5/LaTiO_2N$ samples. (d) The CO₂ reduction products over KOH modified (020) Ta₃N₅ under the visible light irradiation for 8 h. The inset shows the FT-IR spectrum.



Fig. S7. (a) The adsorbed energy for CO on Ta_3N_5 (023) facet. The calculated energy for Ta_3N_5 (023) facet was -3528.7 eV. (b) CO₂ reduction products over polycrystalline Ta_3N_5 prepared by direct nitriding of Ta_2O_5 at 1123 K for 8 h under 500 mL/min NH₃. The inset in Fig. S7a shows the TEM image of polycrystalline Ta_3N_5 .

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