Supporting Information

Highly Conjugated Poly(N-heteroacene) Nanofibers for Reversible Na storage with Ultra-High-capacity and Long cycle life

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1. XRD patterns of the pristine PAN and PAN thermal treated at different temperatures for 6

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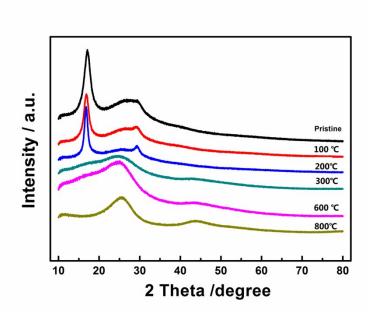


Figure S1. XRD patterns of the pristine PAN and PAN thermal treated at different temperatures (100°C - 800°C)

The products of PAN treated at different temperatures (100~800°C) are investigated by X-ray diffraction (XRD). For the pristine PAN, the typical sharp peaks at 16.88° and 29.38° can be ascribed

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to (100) and (010) diffraction, while the peak at 25.88 is speculated to the diffraction of amorphous molecular chain. When temperature increases to 300°C, the peaks at 25.88° and 29.38° disappear. When further increasing to 800 °C, the peak at 16.88 disappears. Two sharp peaks at 25.18 and 43.68 assigned to (002) and (10) diffraction of pseudo-graphite appear, implying the deposition of the organic polymer structure and carbonization.

2. Raman spectrum

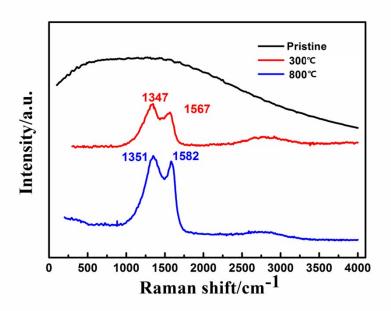


Figure S2. Raman spectrum of the pristine PAN and PAN treated at 300 °C and 800 °C

The pristine PAN shows a strong fluorescence in Raman spectrum. When treated at 300°C, two bands centered at 1347 cm⁻¹ and 1567 cm⁻¹ corresponding to the D and G-band. In general, the D band is attributed to defects and disorder in the hexagonal graphitic layers, while the G band is ascribed to the vibration of sp² carbon atoms in a 2D hexagonal lattice. As the temperature increase to 800 °C, the I_D/I_G value decreases from 1.14 to 1.06, indicating higher orientation and higher graphitic order.

3. FT-IR spectrum of the pristine PAN and PAN thermal treated at different temperatures

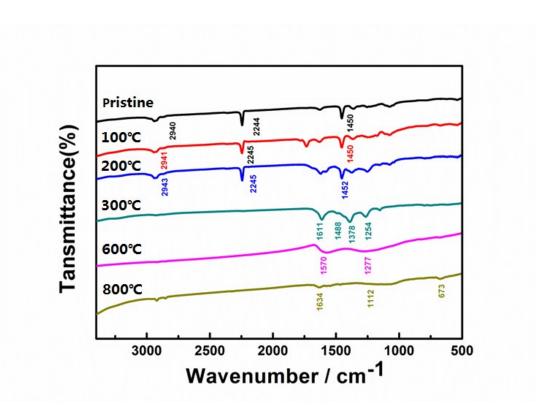


Figure S3. FT-IR spectrum of the pristine PAN and PAN thermal treated at different temperatures (100°C - 800°C)

For the pristine PAN, the single peak at 2240 cm⁻¹ should be ascribed to the typical absorption of C≡N groups. For the sample treated at 300°C, the C≡N bands disappear and the broad absorption band around 1611 cm⁻¹ should be ascribed to the absorption of mixed double bonds [C=C and C=N], implying the cyclization of cyano-group (C≡N) and generation of heterocyclic ring (C=N and C=C groups). As the temperature gets higher, the poly(heteroacene) structure transform to N-doped carbon fibers.

4. FT-IR spectrum of the polyacene (PAc) and cPAN.

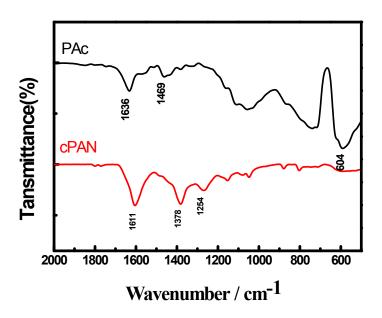


Figure S4. FT-IR spectrum of the PAc and cPAN

For the polyacene, the peaks in the region of 1400 - 1600cm⁻¹ are corresponding to the conjugated vibration, and the peak at 604 cm⁻¹ belongs to the aromatic ring.

5. Thermogravmetric analysis (TGA)

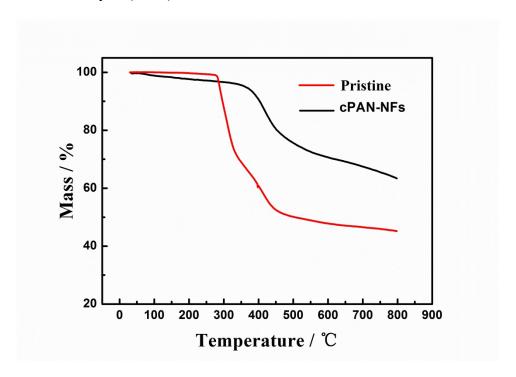


Figure S5. TG curves of pristine PAN and as-prepared cPAN-NFs

The TG curve of pristine PAN exhibits two main stages of weight loss. The first occur in the temperature range of 250–300°C, which could be ascribed to the cyclization process of PAN with little weight loss. The great weight loss of pristine PAN in the temperature range of 280–450°C, which corresponds to the carbonization with the emission of H, C, N and oligomers. [1]

6. ¹³C solid-state NMR spectrum

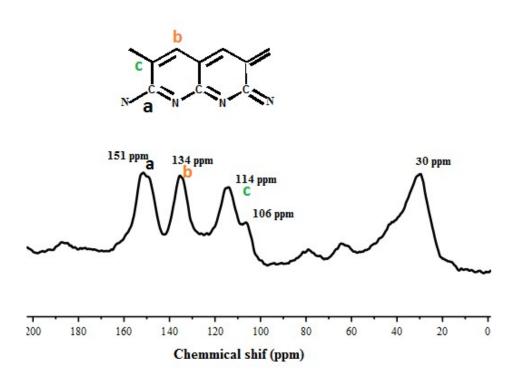


Figure S6. The ¹³C solid state DP/MAS NMR spectra of cPAN-NFs

7. XPS characterization of cPAN

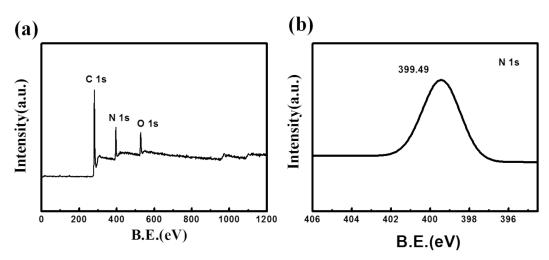


Figure S7. (a) X-ray photoelectron spectra of the as-prepared cPAN, (b) high resolution image of N1s bands

The binding energy for N 1s electrons appeared at 399.49 eV which is attributed to the aromatic nitrogen, indicating that the nitrogen incorporation into the polyacene backbone.

8. Elemental composition of the pristine PAN, as-prepared cPAN-NFs

Table S1. Elemental composition information of pristine PAN and cPAN-NFs

Sample	C(wt%)	N(wt%)	H(wt%)
Pristine PAN	67.92	26.42	5.660
cPAN-NFs	65.67	22.23	4.462

9. The morphology characteristics of the as-prepared PAc, cPAN-PANs and bulk cPAN

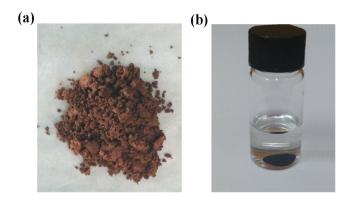


Figure S8. Digital image of (a) cPAN-NFs powders (b) cPAN-NFs electrode steeped in NaPF₆/DME after 36 h

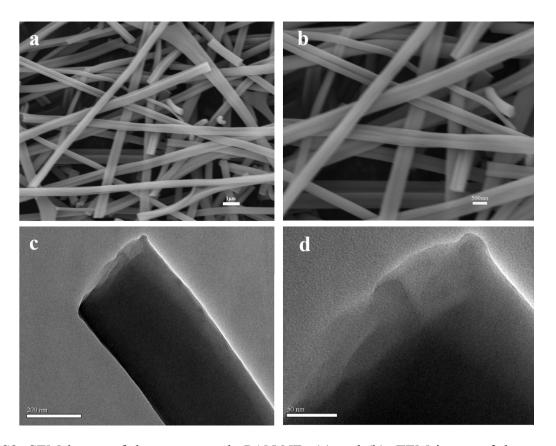


Figure S9. SEM image of the as-prepared cPAN-NFs (a) and (b); TEM image of the as-prepared cPAN-NFs (c) and (d).

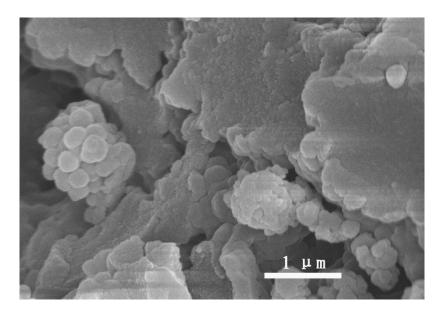


Figure S10. SEM image of the bulk cPAN

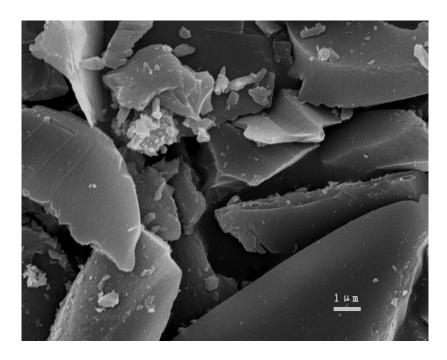


Figure S11. SEM image of the polyacene

For comparison, polyacene (PAc) show a flake-like architecture with particle size of ~10 um.

10. Nitrogen adsorption/desorption isotherms of the as-prepared cPAN-NFs

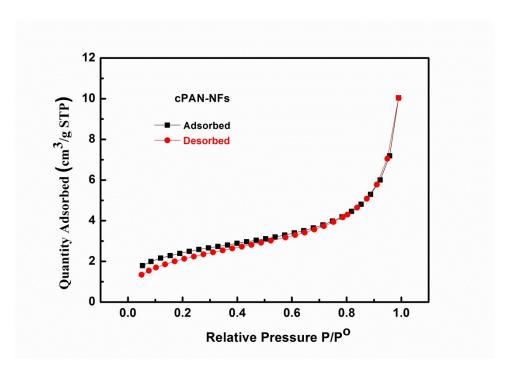


Figure S12. Nitrogen adsorption/desorption isotherms of as-prepared cPAN-NFs

11. The electronic structure of cPAN and other conventional conducting polymers (PAc, PANI, PPy, PTh)

Table S2. Elemental composition information of PAN and cPAN-NFs

Polymer		cPAN	PAc	PANI	PPy	PTh	
НОМО(е	V)	-5.44	-4.08	-3.96	-4.04	-4.66	
LUMO (e	eV)	-4.65	-3.02	-0.23	-0.85	-2.93	
$E_g(eV)$	(НОМО-	0.79	1.06	3.73	3.19	1.73	
LUMO gaps)							

12. Charge-discharge profiles of the PAN treated at 300 °C for different time.

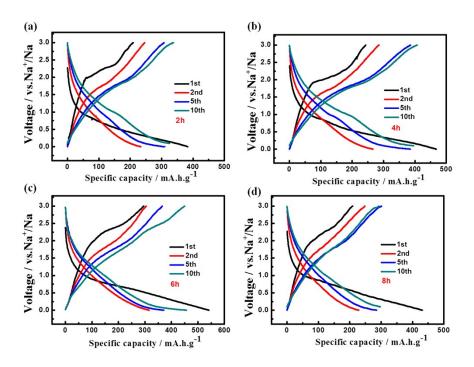


Figure S13. Charge-discharge profiles of cPAN-NFs electrode thermal treated at 300 °C for (a) 2 h; (b) 4 h; (c) 6 h; (d) 8 h in NaPF₆/DME in the voltage range of from 0.001 to 3 V. Current density: 50 mA g⁻¹

To evaluate the effect of thermal treating time on the electrochemical performances of cPAN-NFs, the Na-storage performance of the cPAN-NF thermal treated at 300 °C for various periods of time is measured. As shown in Figure S14, the cPAN-NFs electrode deliver reversible capacities of 320, 400, 450 and 300 mA h g⁻¹ for the samples treated for 2, 4, 6, 8 h, respectively.

The cPAN-NFs samples treated for 6 h exhibit the best Na-storage performances due to the well-formed pseudo-graphite microstructure. Extending the heating time is beneficial to generate conjugated planes with high conjugated extent and large size of pseudo-graphite crystal [2-6].

13. Electrochemical performances of the carbon addictive (KB and MWCNT)

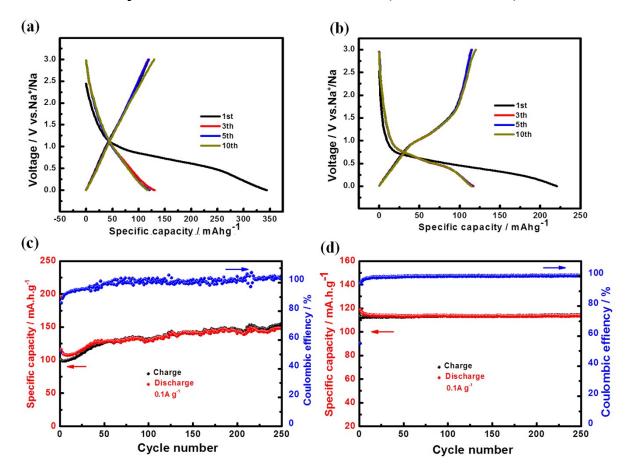


Figure S14. The charge-discharge profiles of Ketchen black (a) and MWCNT (b) at 0.05A g⁻¹; Cycling performance of Ketchen black (c) and MWCNT (d) in 1.0 mol L⁻¹ NaPF₆/DME at 0.1 A g⁻¹.

As shown in Figure S15, the carbon addictive KB and MWCNT can deliver a reversible capacity of 130, 115 mA h $\rm g^{-1}$, respectively. As the weight ratio of cPAN/KB/MWCNT = 6/2/1, the capacities contribution of KB and MWCNT are 26 and 11.5 mA h $\rm g^{-1}$, which are negligible compared to the high capacity of cPAN (527 mA h $\rm g^{-1}$).

14. Na storage performances of the bulk cPAN electrode

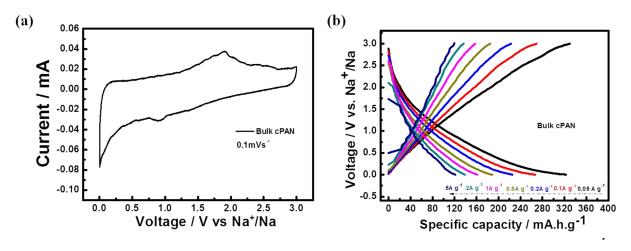


Figure S15. Na storage performance of the bulk cPAN electrode in NaPF₆/DME in the voltage range of from 0.001 to 3 V: (a) Cyclic voltammetry curves scanned at a rate of 0.1 mVs⁻¹ (b) charge-discharge profiles at different current densities.

15. Capacity retention and charge-discharge profiles of cPAN-NFs at 5Ag-1

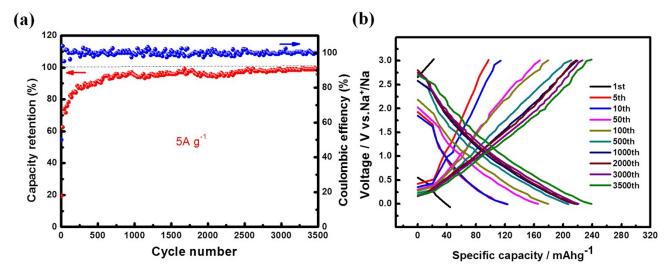


Figure S16. Capacity retention (a) and charge-discharge profiles (b) of cPAN-NFs at 5Ag⁻¹.

As shown in Fig 3c and FigS19 a, the discharge and charge capacities of the PAN-NFs in the first cycle are 44.4 and 22.2 mA h g⁻¹, corresponding to an initial columbic efficiency of 50%. Then the

capacity increased up to 200 mA h g⁻¹ and came to be stable. After 3500 cycles, a capacity retention of 99.4% could be obtained with stable coulombic efficiency up to 99.9%. As shown in FigS19 b, the charge/discharge curves exhibit no obvious change except for the lower polarization.

16. Comparison of the Na –storage performances of cPAN-NFs in this study with other Nastorage anode materials previously reported in the literature

Table S3. Comparison of Na storage performances of cPAN-NFs with other organic anode materials

Anode	Capacity/Current density	Cycle number/capacity retention	Reference
cPAN-NFs	527 mA h g ⁻¹ /0.1 A g ⁻¹	3500/~100%	This work
	195.7 mA h g ⁻¹ /5 A g ⁻¹	3300/~100%	
Na ₂ TP	295 mA h g ⁻¹ /0.03 A g ⁻¹	90/90%	[7]
	$100 \text{ mA h g}^{-1}/3 \text{ A g}^{-1}$	30/30/0	[/]
Na ₂ bpdc	200 mA h g ⁻¹ /0.04A g ⁻¹	150/90%	[8]
	100 mA h g ⁻¹ /3.74 A g ⁻¹	150/7070	
Na ₂ C ₆ H ₂ O ₄ /CNT	259 mA h g ⁻¹ /0.029A g ⁻¹	30/92%	
	142 mA h g ⁻¹ /2.03 A g ⁻¹	33.7270	[9]
Juglone/RGO	280 mA h g ⁻¹ /0.1 A g ⁻¹	100/91.8%	[10]
	210 mA h g ⁻¹ /0.4A g ⁻¹		[]
SSDC	222 mA h g ⁻¹ /0.05 A g ⁻¹	400/70%	[11]
	$72\text{mA}\ \text{h}\ \text{g}^{\text{-}1}/\ 10 \text{A}\ \text{g}^{\text{-}1}$		
O-PDA-2	508 mA h g ⁻¹ /0.05A g ⁻¹	1024/~ 100%	[12]
	122 mA h g ⁻¹ /3.2A g ⁻¹		
Na ₂ NC/G	207 mA h g ⁻¹ /0.02Ag ⁻¹	100/92%	[13]
	$88 \text{ mA h g}^{-1}/2 \text{ A g}^{-1}$		
Na ₂ PDC	270 mA h g ⁻¹ /0.015Ag ⁻¹	100/83%	[14]
	135 mA h g ⁻¹ /1.57Ag ⁻¹		

Table S4. Comparison of Na storage performances of cPAN-NFs with other carbon-based materials

Anode	Capacity/Current density	Cycle number/capacity retention	Reference
cPAN-NFs	527 mA h g ⁻¹ /0.05 A g ⁻¹	3500/~100%	This work
	195.7 mA h g ⁻¹ /5 A g ⁻¹	3300/~10076	THIS WOLK
DC-S	561 mA h g ⁻¹ /0.02 A g ⁻¹	1000/85.9%	[15]
	$275 \text{ mA h g}^{-1}/1 \text{ A g}^{-1}$	1000/83.970	
3D PCFs	356.1 mA h g ⁻¹ /0.1A g ⁻¹	5000/80%	[16]
	187.6 mA h g ⁻¹ /2.5 A g ⁻¹	3000/8076	
3D OC-500	483 mA h g ⁻¹ /0.1Ag ⁻¹	1200/81.4%	[1 <i>7</i>]
	299 mA h g ⁻¹ /1 A g ⁻¹	1200/81.470	[17]
BN-CNFs	691 mA h $g^{-1}/0.1A g^{-1}$	1000/84.1%	[1 0]
	314 mA h g ⁻¹ /10A g ⁻¹	1000/04.170	[18]
P-CNSs	$321 \text{ mA h g}^{-1}/0.1 \text{A g}^{-1}$	5000/marky 1000/	[19]
	149mA h g ⁻¹ /5A g ⁻¹	5000/ nearly 100%	
G/C	432.3 mA h g ⁻¹ /0.1A g ⁻¹	1000/94.691%	[20]
	330 mA h g ⁻¹ /2A g ⁻¹	1000/74.071/0	[20]

17. Structure and morphology changes of pristine electrodes and electrodes cycled after 1700 cycles.

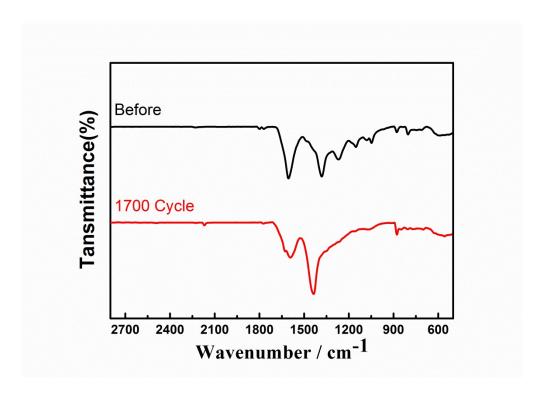


Figure S17. FT-IR spectrum of the pristine and cycled electrodes (after 1700 cycles) at 5 A g⁻¹

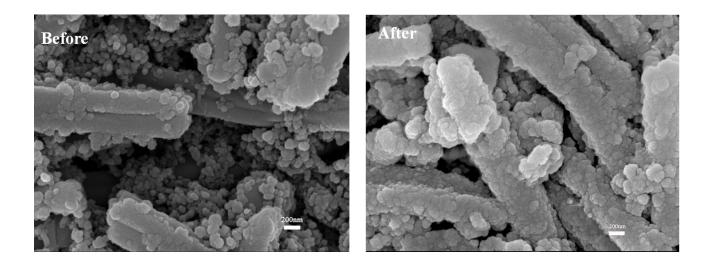


Figure S18. SEM images of the pristine and cycled electrodes (after 1700 cycles) at 5 A g⁻¹

18. Geometries and frontier molecular orbitals.

DFT calculations were done using Gaussian 09 package at the computational level of B3LYP/6-31G(D) with PCM solvation model.

The polymer system is simplified and modeled by oligomer containing 12 units (C₃NH)

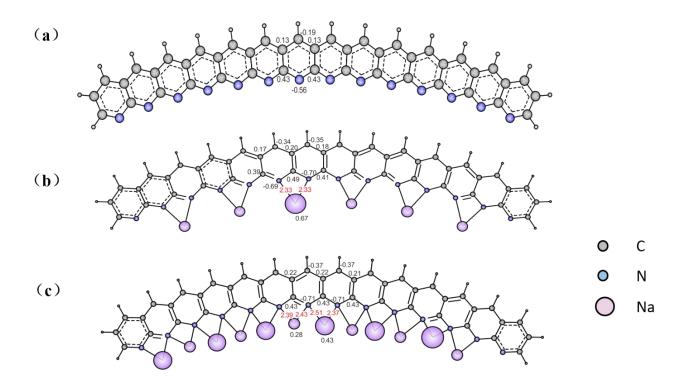


Figure S19. Mulliken charge distribution of (a) cPAN; (b) cPAN-50%Na; (c) cPAN-Na.

Red font: bond distance; Black font: Mulliken charge

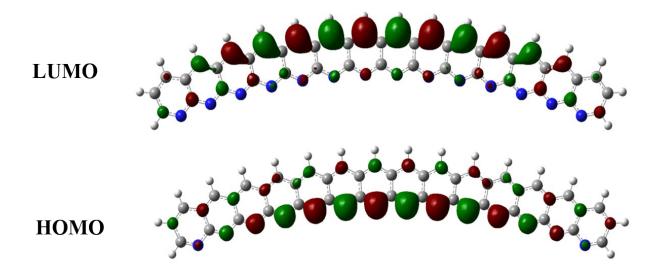


Figure S20. Frontier orbital wavefunctions of **cPAN** in Diethylether with isovalue of 0.02.

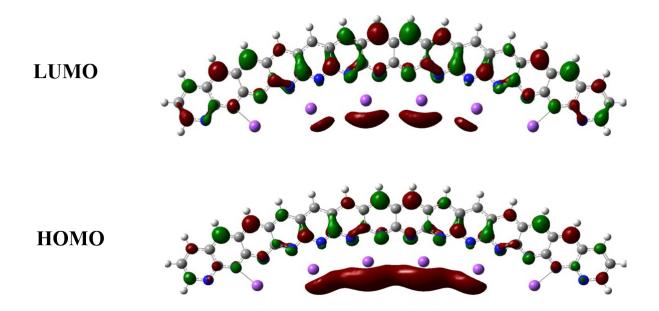


Figure S21. Frontier orbital wavefunctions of cPAN-50%Na in Diethylether with isovalue of 0.02.

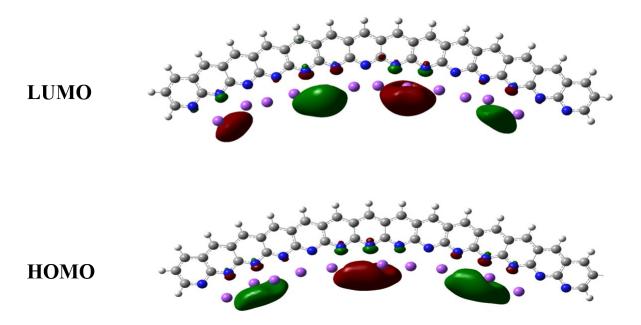


Figure S22. Frontier orbital wavefunctions of **cPAN-Na** in Diethylether with isovalue of 0.02.

Table S5. Sum of electronic and zero-point energies of the optimized structures Na, cPAN, cPAN-50%Na, cPAN-Na

Energy	Hartree	eV
Na	-162.2862551	
cPAN (with 12 units)	-2454.0680976	
cPAN-50%Na	-3428.400376	
(12 units, 6 Na)		
cPAN-Na	-4402.2399948	
(12 units, 12 Na)		
ΔE_I (for the first stage of	-0.1024583	-2.7868667
Na doping)		
ΔE_2 (for the second stage of	-0.0203483	-0.5534747
Na doping)		

The polymer system is simplified and modeled by oligomer containing 12 units of C₃NH.

At the first stage of Na-doping (from neutral state to a doping level of 50%), 6 Na coordinate with the 12 aromatic nitrogen.

$$\Delta E_I = [E(cPAN-50\%Na) - 6E(Na) - E(cPAN)]/6$$

At the second stage of Na-doping (from doping level of 50% to fully doped), another 6 Na insert into the polymer system

$$\Delta E_2 = [E(cPAN-Na) - 6E(Na) - E(cPAN-50\%Na)]/6$$

Supplementary References

- 1. H. Wang, S. Kim, S. Cho, G. B, H. Choi, Mater. Resear. Bull. 2018, 97, 49-55.
- 2. S. Lei, W. Cao, Z. Fu and L. Xu, J. Appl. Polym. Sci., 2016, 133, 1–7.
- 3. S. Dalton, F. Heatley and P. M. Budd, Polymer., 1999, 40, 5531–5543.
- 4. M. S. A. Rahaman, A. F. Ismail and A. Mustafa, Polym. Degrad. Stab., 2007, 92, 1421–1432.
- 5. Y. Xue, J. Liu and J. Liang, J. Appl. Polym. Sci., 2013, 127, 237–245.
- 6. W. X. Zhang, Y. Z. Wang and C. F. Sun, J. Polym. Res., 2007, 14, 467–474.
- Y. Park, D. S. Shin, S. H. Woo, N. S. Choi, K. H. Shin, S. M. Oh, K. T. Lee, S. Y. Hong, Adv. Mater. 2012, 24, 3562-7.
- **8.** A. Choi, Y. K. Kim, T. K. Kim, M.-S. Kwon, K. T. Lee, H. R. Moon, J. Mater. Chem. A 2014, **2**, 14986-14993.
- **9.** X. Wu, J. Ma, Q. Ma, S. Xu, Y.-S. Hu, Y. Sun, H. Li, L. Chen, X. Huang, J. Mater. Chem. A 2015, **3**, 13193-13197.
- 10.H. Wang, P. Hu, J. Yang, G. Gong, L. Guo, X. Chen, Adv. Mater. 2015, 27, 2348-54.
- **11.**C. Wang, Y. Xu, Y. Fang, M. Zhou, L. Liang, S. Singh, H. Zhao, A. Schober, Y. Lei, J. Am. Chem. Soc. 2015, **137**, 3124-30.
- 12.T. Sun, Z. Li, H. Wang, D. Bao, F. Meng, X. Zhang, Angew. Chem. Int. Ed. 2016, 55, 10662-10666.
- 13. W. Deng, J. Qian, Y. Cao, X. Ai, H. Yang, Small 2016, 12, 583-7.
- 14.H. Padhy, Y. Chen, J. Lüder, S. R. Gajella, S. Manzhos, P. Balaya, Adv. Energy Mater. 2017, 1701572.
- 15. W. Li, M. Zhou, H. Li, K. Wang, S. Cheng, K. Jiang, Energy Environ. Sci. 2015, 8, 2916-2921.
- 16. H. Hou, C. E. Banks, M. Jing, Y. Zhang, X. Ji, Adv. Mater. 2015, 27, 7861-6.
- 17. L. Fan, B. Lu, Small 2016, 12, 2783-91.
- 18. M. Wang, Y. Yang, Z. Yang, L. Gu, Q. Chen, Y. Yu, Adv. Sci. 2017, 4, 1600468.
- 19. H. Hou, L. Shao, Y. Zhang, G. Zou, J. Chen, X. Ji, Adv. Sci. 2017, 4, 1600243.
- 20. Y. Liu, L.-Z. Fan, L. Jiao, J. Mater. Chem. A 2017, 5, 1698-1705.