

Supporting information

Porous Si@C Ball-in-Ball Hollow Spheres for Lithium-ion Capacitor with improved energy and power density

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Experimental

All Chemical reagents were purchased from Shanghai Chemical Co. Ltd., and they were directly used without further purification.

Materials Preparations

MHSiO₂: The preparation of MHSiO₂ was following a procedure reported in the literature. Typically, cetyltrimethylammonium bromide (CTAB, 0.3 g) was dissolved in ethanol aqueous solution (60 mL C₂H₅OH/100 mL H₂O) containing concentrated ammonia aqueous solution (2 mL, 25 wt%). Then, tetraethoxysilane (TEOS, 2 mL) was rapidly added to the mixture under vigorous stirring. After stirring at 35 °C for 24 h, the white precipitation was collected by centrifugation and washed with ethanol. Then, the product was dispersed in water and kept at 90 °C for 2 h. The product collected by centrifugation was dispersed into an ethanol solution (240 mL) containing concentrated HCl (480 μL, 37%) and stirred at 60 °C for 3 h, generating the template-free MHSiO₂.

MHSiO₂@Al₂O₃: MHSiO₂@Al₂O₃ nanospheres were prepared by a facile synthetic protocol in a buffer solution. In a typical synthesis, 0.63 g of ammonium formate was dispersed in 200 ml deionized water, and then formic acid was added till the pH value to 4.4. Then 0.1 g of MHSiO₂ and 0.3 g aluminum sulfate were added and dispersed by ultrasound treatment for 30 min. The solution was kept at 70 °C for 2 h under vigorously stirring. The resulted core-shell NPs were collected by centrifugation, washed with water and ethanol, and dried at 80 °C for 4 h.

MHSiO₂@Al₂O₃@C: Carbon coating of MHSiO₂@Al₂O₃ was carried out by thermal decomposition of acetylene gas at 800 °C for 10min in a quartz furnace. The mixture of acetylene and high-purity argon (argon: acetylene = 9: 1 by volume) was introduced at a flow rate of 150 sccm.

SiO₂@C ball-in-ball hollow spheres: The obtained MHSiO₂@Al₂O₃@C was then immersed in a 1 M HCl solution for 6 h to remove Al₂O₃. Finally, the resulted powders were collected by filtration and washed with water for several times and vacuum-dried at 70 °C for 6 h.

Porous Si@C ball-in-ball hollow spheres: Magnesium powder (0.3 g) was mixed with MHSiO₂@void@C (0.3 g) by grinding. The mixture was then heated in a tube furnace at 650 °C for 5 h under an argon atmosphere containing 5 vol% H₂. The ramp rate was kept at 1 °C min⁻¹. The obtained powder was then immersed in a 2 M HCl solution for 6 h to remove MgO and. Finally, the resulted powders were collected by filtration and washed with water for several times and vacuum-dried at 70 °C for 4 h.

MHSi@SiC@C: The preparation process of Si@SiC@C is similar to the method of Si@void@C, except for the addition of void space between SiO₂ and carbon.

RAC: The RHs were firstly leached with a 10% HCl solution to remove the metal ions impurities. The leached RHs were heated to 700 °C, and then maintained at 700 °C for 2 h in a tube furnace under the protection of nitrogen to obtain carbonized RHs (CRHs). CRHs were mixed with KOH in a 1:5 mass ratio. The mixture was then transferred to ceramic boats and heated to 700 °C at a ramp rate of 2 °C min⁻¹, and then maintained at 700 °C for 1 h in a tube furnace under nitrogen. The resultant activation products were filtered and washed with water until the filtrate became neutral. Finally, the RACs were dried at 90 °C for 10h.

Characterization

X-ray diffraction (XRD) patterns were recorded on a D/max 2550VL/PC X-ray diffractometer (Rigaku, Japan) equipped with Cu K α radiation ($\lambda = 1.5418 \text{ \AA}$, 40 kV, 30 mA). Raman spectra were acquired using an inVia-reflex micro-Raman spectrometer (Renishaw, UK) with a 532 nm wavelength incident laser. X-ray photoelectron spectroscopy (XPS) was performed on an AXIS Ultra DLD spectrometer (Kratos, Japan) with Al K α radiation ($h\nu = 1486.6 \text{ eV}$). Thermogravimetric analysis (TGA) was performed on an SDT Q600 thermoanalyzer (DSC-TGA, TA, USA) in air. The specific surface area and pore size distribution of materials were measured using a NOVA2200e analyzer (Quantachrome, USA). The morphology of samples was observed using a NOVA NanoSEM 230 field-emission scanning electron microscope

(FESEM, FEI, USA). The microstructure of samples was characterized using a JEM-2100F transmission electron microscope (TEM, JEOL, Japan) operating at 200 kV.

Electrochemical measurements

The electrochemical experiments were performed using CR2016 coin cells, which were assembled in a glove box filled with ultra-high purity argon. For half-cell testing, Si/C and RAC electrodes were used as working electrodes and Li metal as the reference electrode. The Si/C electrodes were prepared by casting a slurry consisting of 70 wt% of active material, 15 wt% poly(acrylic acid) (PAA) binder, and 15 wt% of Super-P carbon black on Cu foil. The RAC electrodes were prepared by mixing 80 wt% of active material, 10 wt% Cellulose Sodium (CMC) binder, and 10 wt% of Super-P carbon black on Al foil. The typical mass loading of the active materials in cathode and anode are 2 mg and 1 mg, respectively. 1 M LiPF_6 in a mixture of ethylene carbonate, dimethyl carbonate (EC/DEC, 1:1 by vol%) and 10 wt% fluoroethylene carbonate (FEC) was used as the electrolyte. The Si/C||RAC LICs were also assembly in coin cells with Si/C anode (pre-cycled for 3 cycles at 0.4 A g^{-1}) and RAC cathode in the same electrolyte, and the optimized mass ratio of anode and cathode was 1:7. The galvanostatic charge and discharge experiment were performed on a battery tester LAND-CT2001A at room temperature. The voltage range of Si/C electrode was 0.01-1.5 V, and the cell is conducted at a low current density of 0.4 A g^{-1} in the first cycle to activate the anode materials. The RAC electrodes and the Si/C||RAC LICs were measured at the same voltage range of 2.0-4.0 V. Cyclicvoltammetry (CV) was conducted on an electrochemical workstation. Electrochemical impedance spectroscopy (EIS)

measurements were performed on a ZAHNER Zennium electrochemical workstation.

The energy density (E) and power density (P) were calculated from the galvanostatic charge/discharge curve by the equation

$$E = \int_{t_2}^{t_1} IV dt$$

(I)

Where E (W h kg⁻¹) is energy density, I is the constant current density (A g⁻¹), V is the voltage, and t_i, t₂ is the start time and end time in the discharge process, respectively.

$$P = \frac{E}{\Delta t}$$

(II)

Where P (W g⁻¹) is power density, Δt is discharge time (s). The mass includes both anode and cathode active materials.

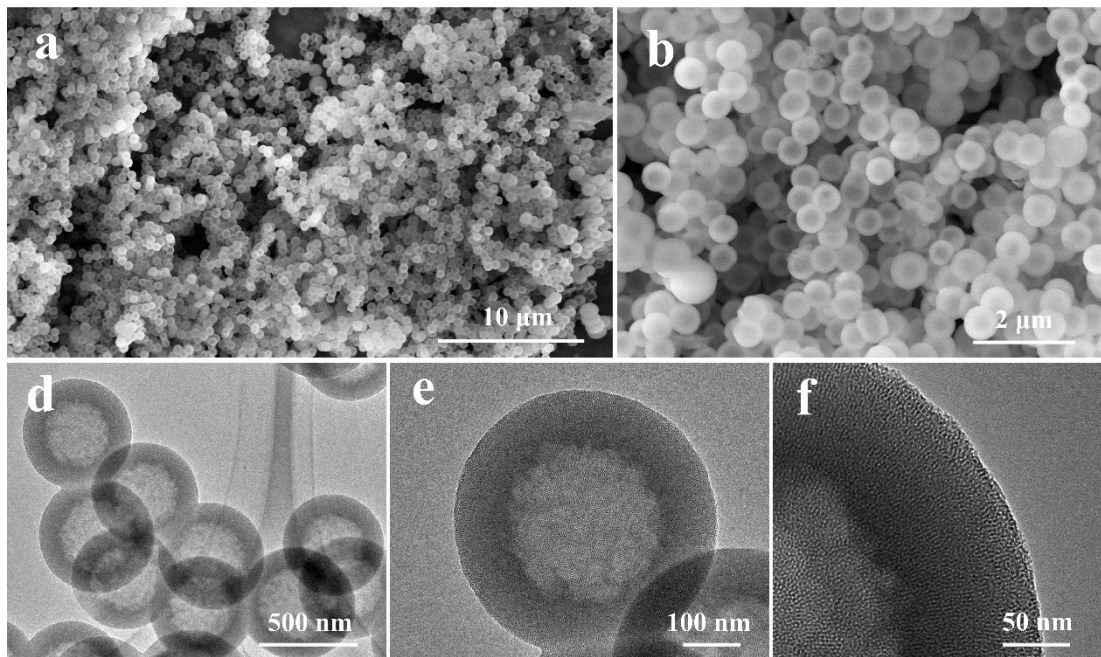


Figure S1. SEM images of (a), (b) TEM images of (d), (e) and (f) SiO₂

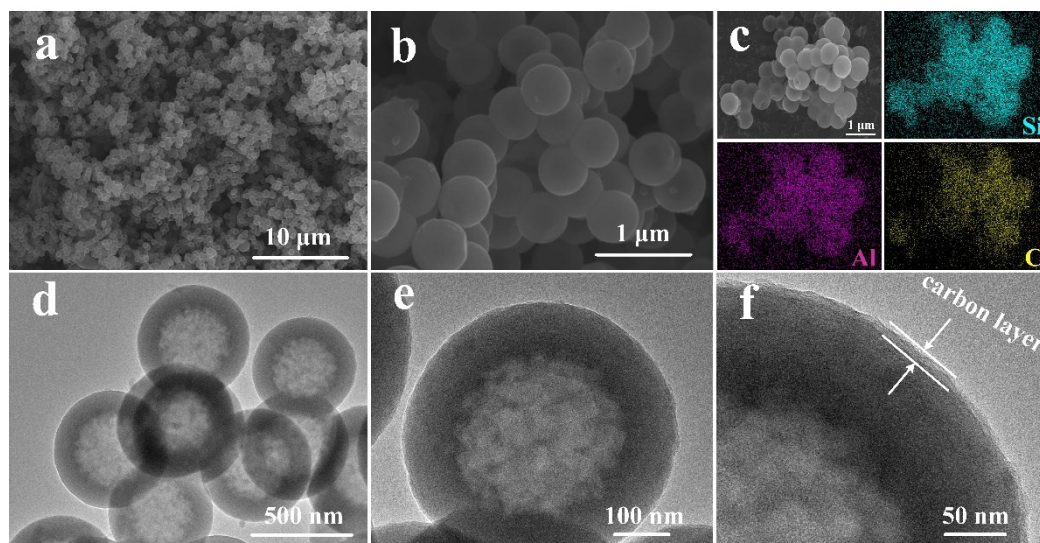


Figure S2. SEM images of SiO₂@Al₂O₃@C (a) and (b). (c) SEM image of SiO₂@Al₂O₃@C and corresponding EDX elemental mapping images of Si (blue), Al (purple) and C (yellow). TEM images of SiO₂@Al₂O₃@C (d), (e) and (f) Magnified TEM images of SiO₂@Al₂O₃@C.

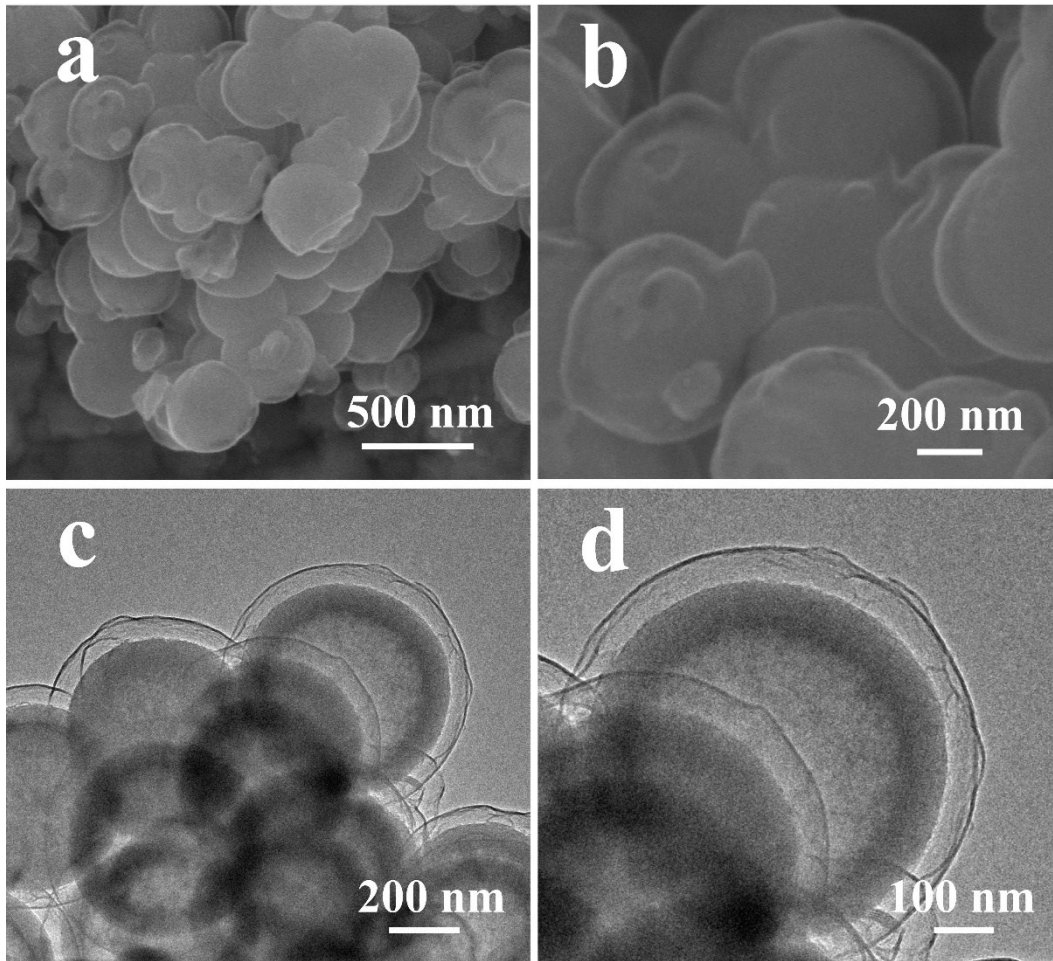


Figure S3. SEM images (a) and (b), and TEM images of (c) and (d) of SiO₂@C ball-in-ball hollow spheres.

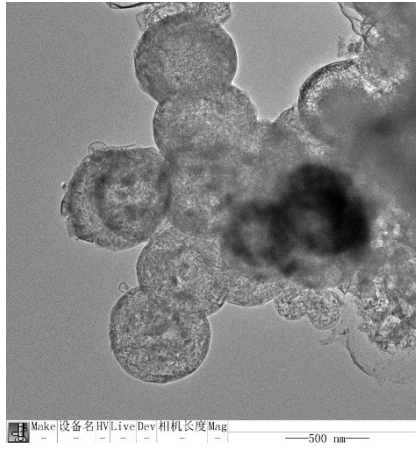


Figure S4. TEM image of Si@SiC@C.

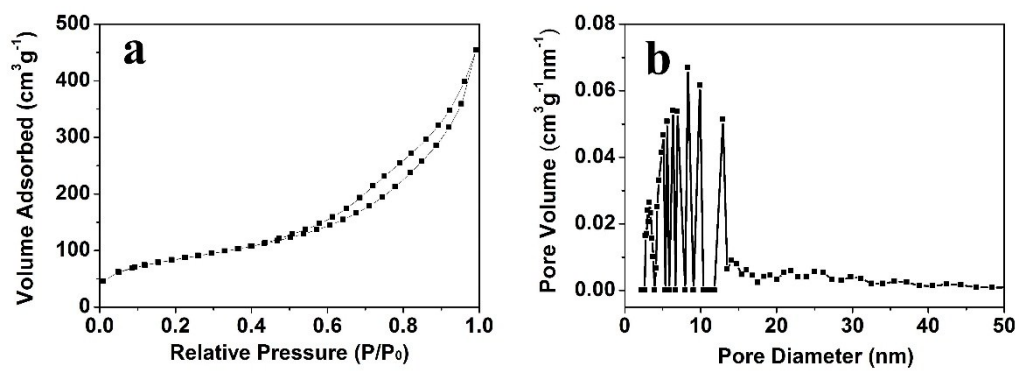


Figure S5. (a) Nitrogen adsorption/desorption isotherms obtained at 77 K and (b) pore size distribution of Si@C ball-in-ball hollow spheres.

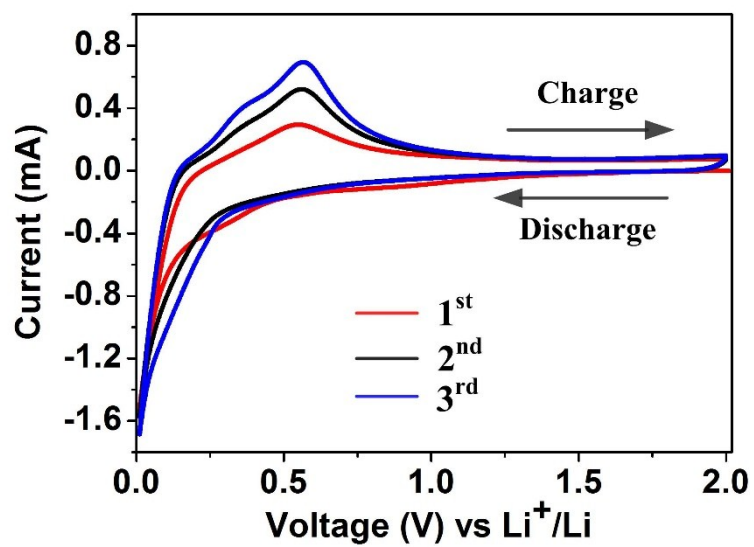


Figure S6. CV curves of Si@C ball-in-ball hollow spheres.

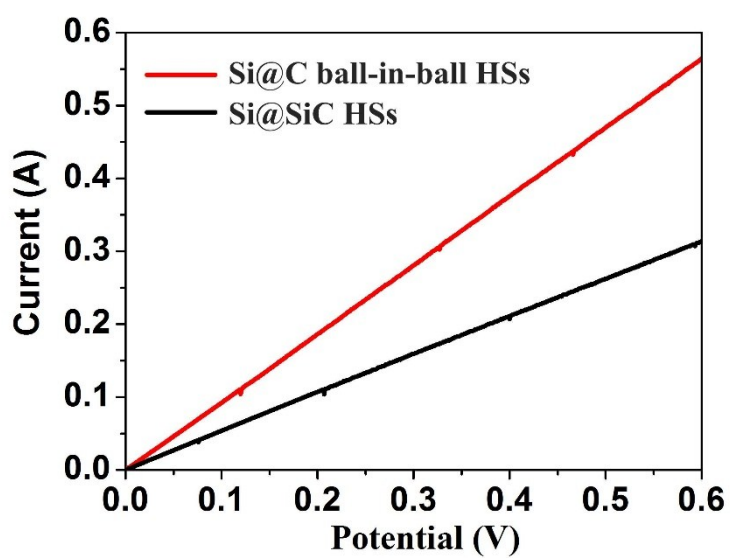


Figure S7. Conductivity of Si@C ball-in-ball hollow spheres and Si@SiC hollow spheres.

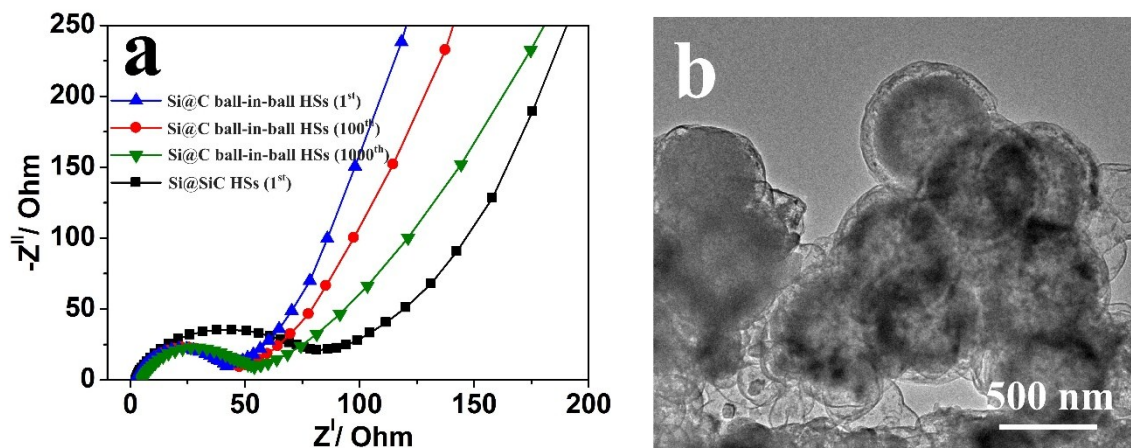


Figure S8. (a) Nyquist plots of Si@C ball-in-ball hollow spheres and Si@SiC@C hollow spheres.

(b) TEM image of Si@C ball-in-ball hollow spheres after 1000 cycles.

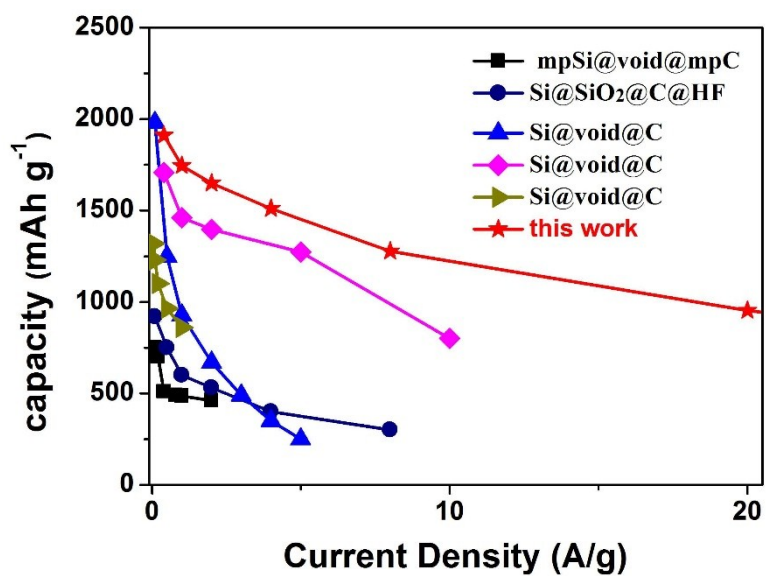


Figure S9. Comparison of capacity at different rates for Si@void@C composite electrode with those of Si/C yolk-shell nanocomposite anodes reported.^[1-5]

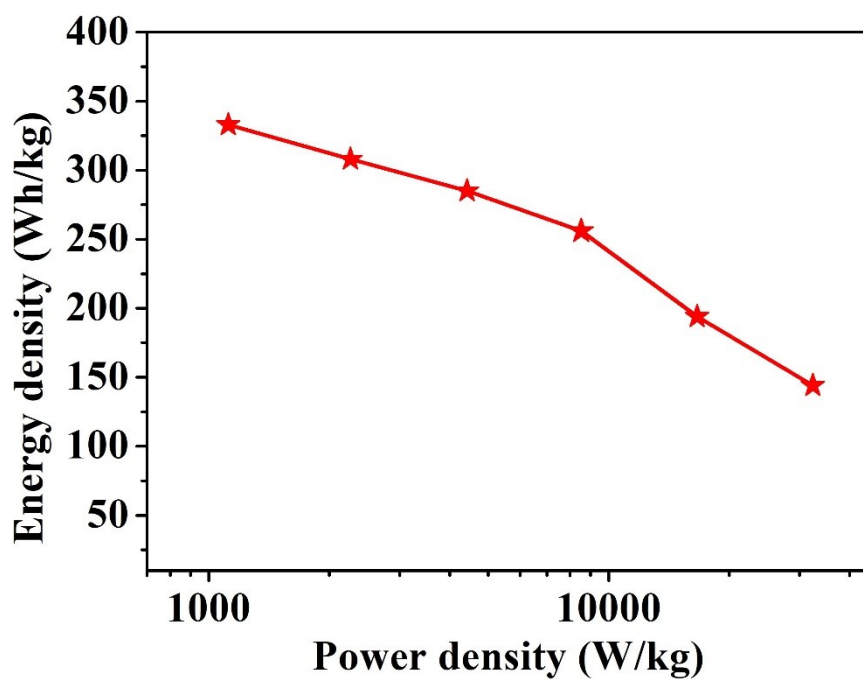


Figure S10. The energy and power density (based on the mass of RAC) of Li || RAC LIC .

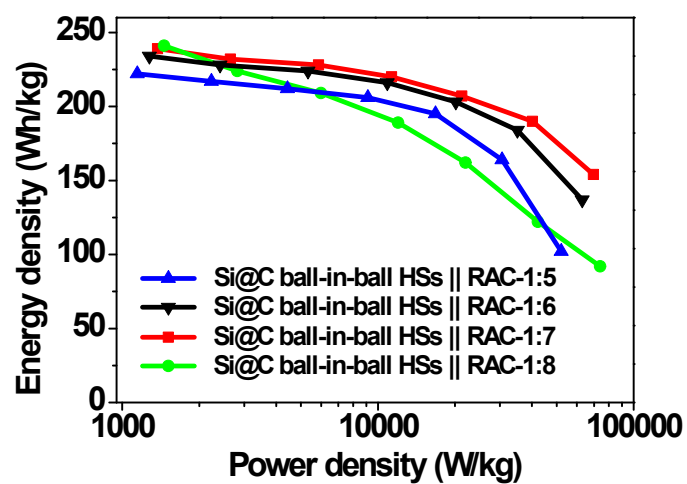


Figure 11. Ragone plot of LICs using Si@C ball-in-ball hollow spheres as anode and RAC as cathode with various mass ratios.

References

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