### **Supplementary Information for**

# **Unique Schrödinger Semimetal State in Ternary**

## **Be<sub>2</sub>P<sub>3</sub>N Honeycomb Lattice**

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#### 1. Computational methods for carrier mobility

Under effective mass approximation, the acoustic-phonon-limited carrier mobility ( $\mu$ ) for 2D materials in the deformation potential (DP) theory [Phys. Rev. 1950, 80, 72] is expressed as:

$$\mu = \frac{2e\hbar^3 C}{3k_B T(m^*)^2 {E_1}^2}$$

where T is the temperature and  $m^*$  is the effective mass of the carrier along the transport direction;  $E_1 = \Delta v / (\Delta l/l_0)$  is the DP constant, where  $\Delta v$  represents the shift of band edges under proper cell compression and dilatation,  $l_0$  is the lattice constant in the transport direction, and  $\Delta l$  is its deformation;  $C = (\partial^2 E / \partial \delta^2) / S_0$  is the elastic modulus, where *E* is the total energy of the cell and  $\delta$  the applied strain while  $S_0$  is the area of the cell. The PBE functional was adopted in the estimations.

### 2. Table S1-S2

	Structure	System (X=Be, Mg, Zn)	Bandgap (eV)
C1		Be <sub>2</sub> -N <sub>3</sub> -P	3.850
		Mg <sub>2</sub> - <b>N</b> <sub>3</sub> -P	2.698
	0 00	Zn <sub>2</sub> -N <sub>3</sub> -P	2.664
C2		Be <sub>2</sub> - <b>P</b> <sub>3</sub> -N	0.000
		Mg <sub>2</sub> - <b>P</b> <sub>3</sub> -N	0.089
		Zn <sub>2</sub> - <b>P</b> <sub>3</sub> -N	0.000
C3	9 9 9	Be <sub>2</sub> - <b>N-P</b> <sub>3</sub>	0.588
		Mg <sub>2</sub> - <b>N-P</b> 3	1.346
	0 0	Zn <sub>2</sub> - <b>N-P</b> <sub>3</sub>	1.00

**Table S1:** Calculated bandgaps of candidate structures with Be, Mg, Zn, P and N in initial material design.

**Table S2.** Predicted deformation potential constant ( $E_1$ ), 2D elastic modulus (C), effective mass ( $m^*$ ) and mobility ( $\mu$ ) for election and hole at 300 K. The vacuum level was set to zero for reference.

Carrier type	$E_1(eV)$	<i>C</i> (N/m)	$m^*(m_e)$	$\mu(\times 10^3 \text{cm}^2/\text{V/s})$
e	-1.20	90	0.31	11.3
h	-1.65	90	-0.92	0.58

## 3. Figure S1-S6



**Figure S1**. Snapshots of the final frame of FPMD of  $Be_2P_3N$  monolayer from 600, 900 and 1200 K (top and side views). The melted local structures at 1200 K are indicated in red-dot circles for clarity.



Figure S2. Band structures computed by HSE06 and PBE functionals. Note that the two functionals predict very similar band dispersions and zero bandgaps at  $\Gamma$  point.



**Figure S3**. The SOC effect on bandgap opening calculated with different samplings of k-point density  $(11 \times 11 \times 1, 13 \times 13 \times 1, 15 \times 15 \times 1, 25 \times 25 \times 1 \text{ and } 45 \times 45 \times 1)$ . Considering the tendency, the ultimate SOC effect on bandgap opening will be further decreased, while may not be null.



Figure S4. Phonon dispersions of  $Be_2P_3N$  monolayer under different biaxial strains.



Figure S5. Band structures of Be<sub>2</sub>P<sub>3</sub>N monolayer under different biaxial strains.



Figure S6. Phonon dispersion (a) and band structure (b) of  $Zn_2P_3N$  monolayer.