

Electronic Supplementary Information

A Facile Random Copolymer Strategy to Achieve Highly Conductive, Polymer Gel Electrolytes for Electrochemical Applications

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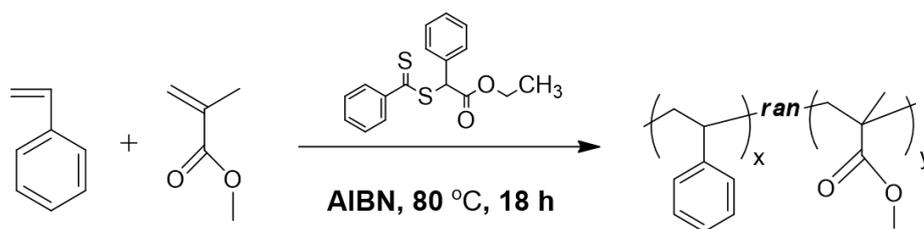
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Synthesis of poly(styrene-*ran*-methyl methacrylate)

Poly(styrene-*ran*-methyl methacrylate) (PS-*r*-PMMA) copolymers in this work were prepared using one-pot reversible addition-fragmentation chain transfer (RAFT) polymerization with a RAFT agent of ethyl 2-(phenylcarbonothioylthio)-2-phenylacetate (Fig. S1a). A brief synthetic procedure is described for PS-*r*-PMMA-H. Both monomers (i.e., styrene and methyl methacrylate) were purified by passing through a column filled with basic alumina. Then, styrene (7.6 g, 72.9 mmol), methyl methacrylate (29.2 g, 291.8mmol), benzodithioate (0.0048 g, 0.015 mmol), and AIBN (0.0005 g, 0.00304 mmol) were mixed in a dried two-neck flask. The reaction mixture was purged with Ar gas at room temperature for 1 h. Then, the polymerization was carried out at 80 °C for 18 h, followed by quenching with liquid nitrogen for termination. The solution was poured into a large amount of methanol for precipitation, which process was conducted three times. The resulting PS-*r*-PMMA was collected and dried at 50 °C under reduced pressure. The adjustment of molecular weight of PS-*r*-PMMA was achieved by controlling the polymerization time and feed ratio of monomer mixtures and the RAFT agent. For example, PS-*r*-PMMA-M was polymerized at a molar ratio of RAFT agent : monomer mixture ~ 5 : 60000 for 6 h. To further decrease molecular weight (namely, for PS-*r*-PMMA-L), the amount of monomer mixtures was reduced by half (i.e. mole ratio of RAFT agent : monomer mixture ~ 5: 30000), whereas the polymerization time (6 h) was fixed. It is noted that a molar ratio of AIBN : RAFT agent was 1: 5 for all cases. The control of PS-*r*-PMMA compositions was achieved by varying the feed monomer composition.

(a)



(b)

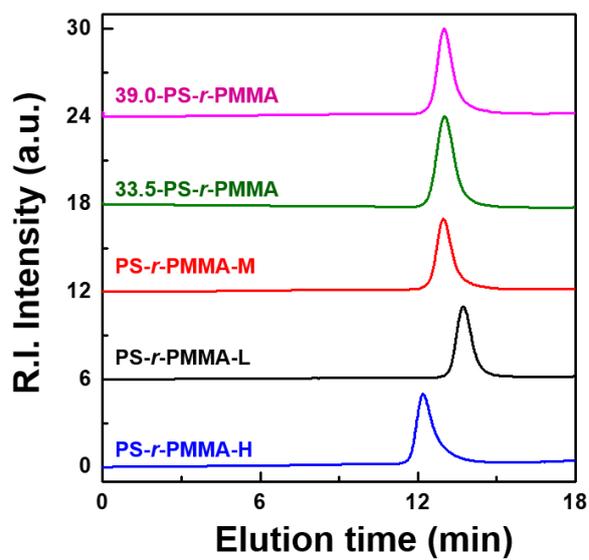


Fig. S1 (a) Synthetic route for PS-*r*-PMMA. (b) SEC chromatograms of PS-*r*-PMMA employed in this study.

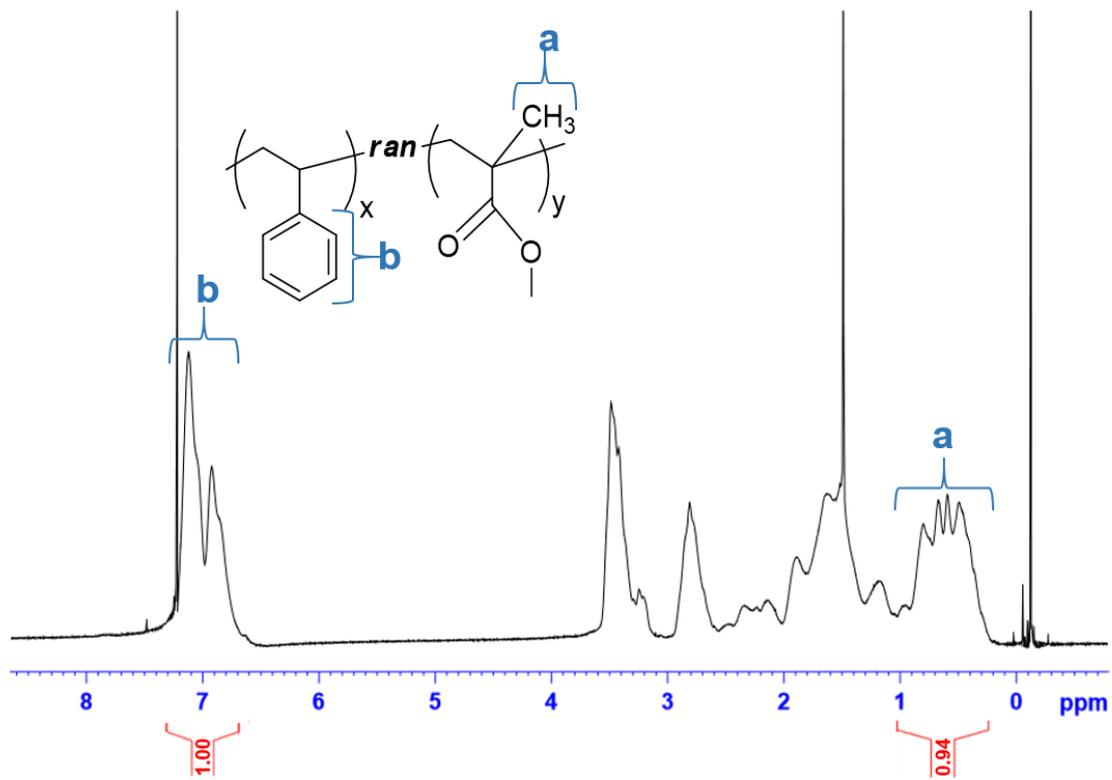


Fig. S2 ^1H NMR spectrum of 39.0-PS-*r*-PMMA.

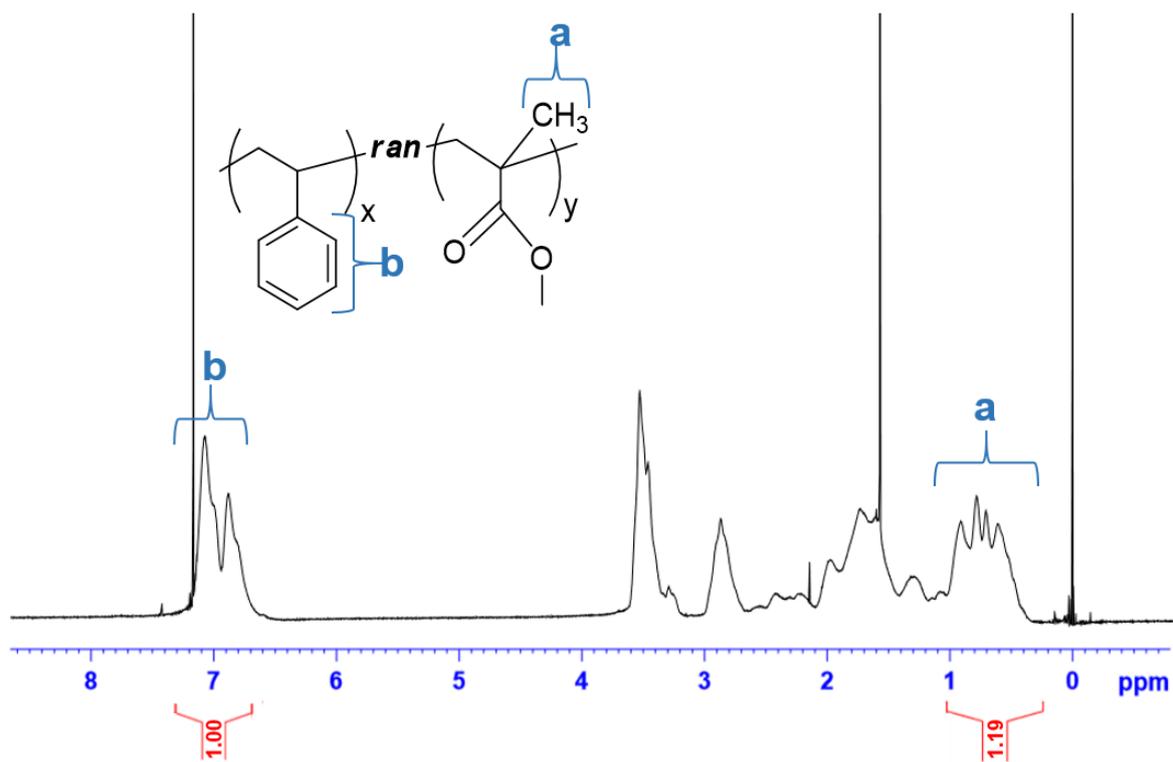


Fig. S3 ^1H NMR spectrum of 33.5-PS-*r*-PMMA.

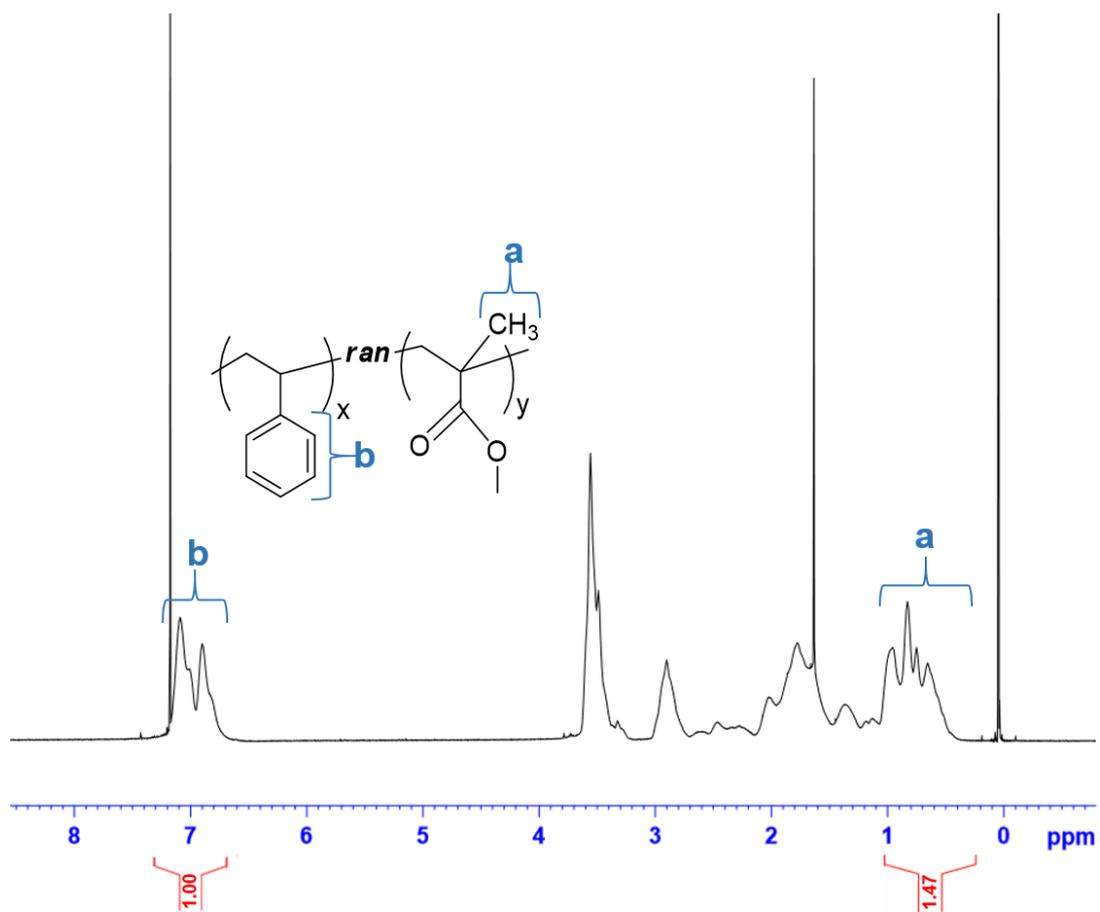


Fig. S4 ^1H NMR spectrum of PS-*r*-PMMA-M.

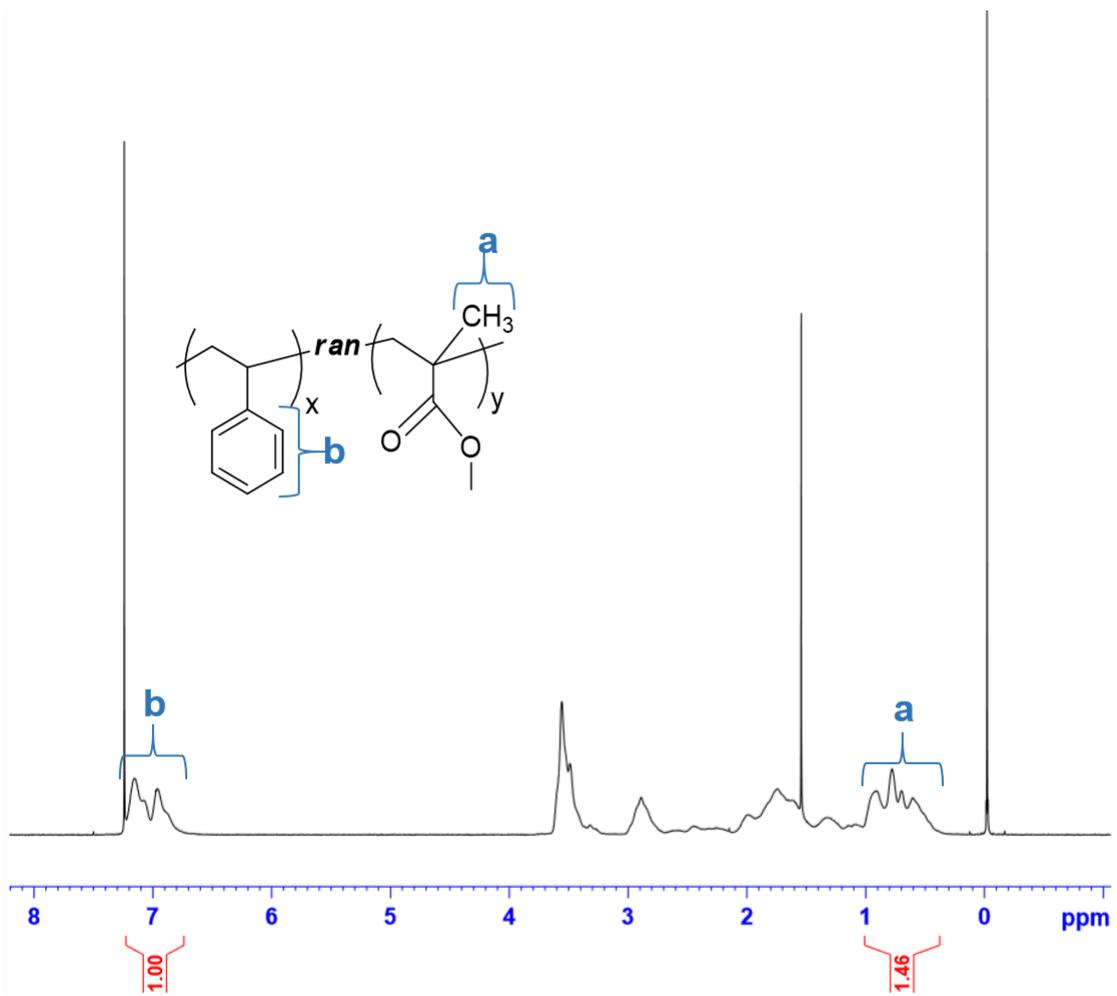


Fig. S5 ^1H NMR spectrum of PS-*r*-PMMA-L.

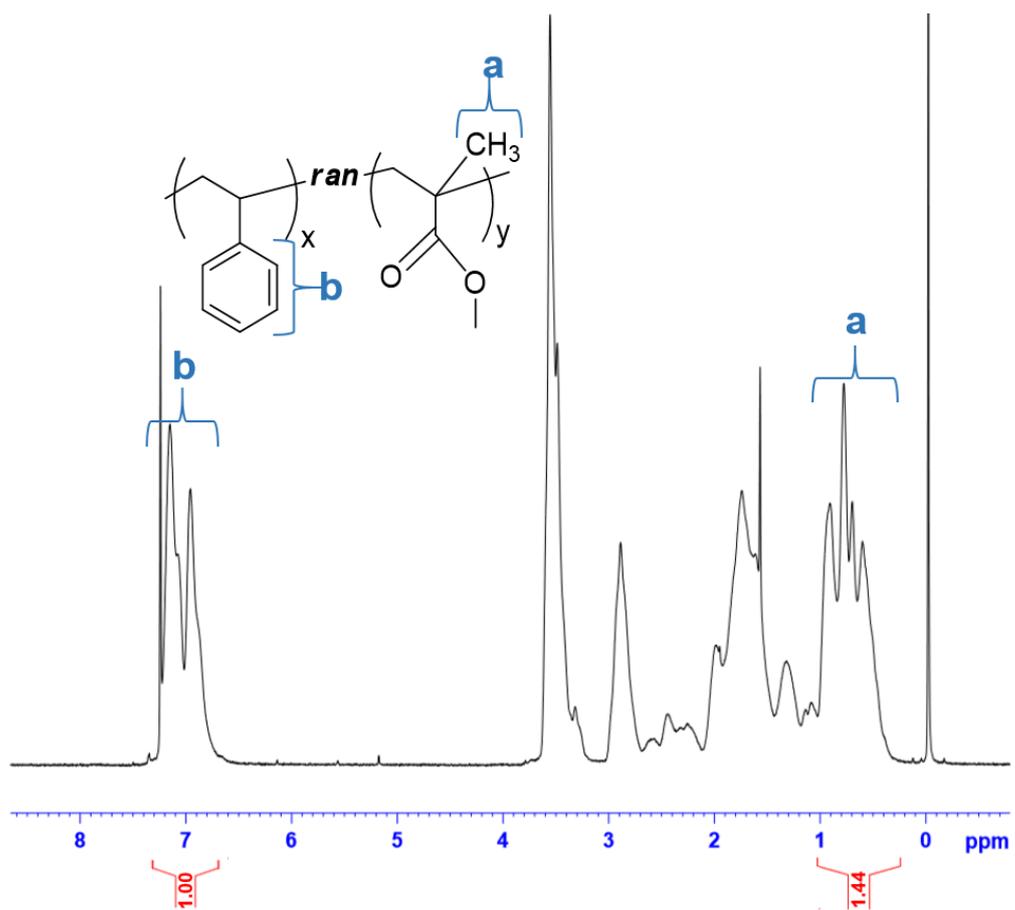


Fig. S6 ¹H NMR spectrum of PS-*r*-PMMA-H.

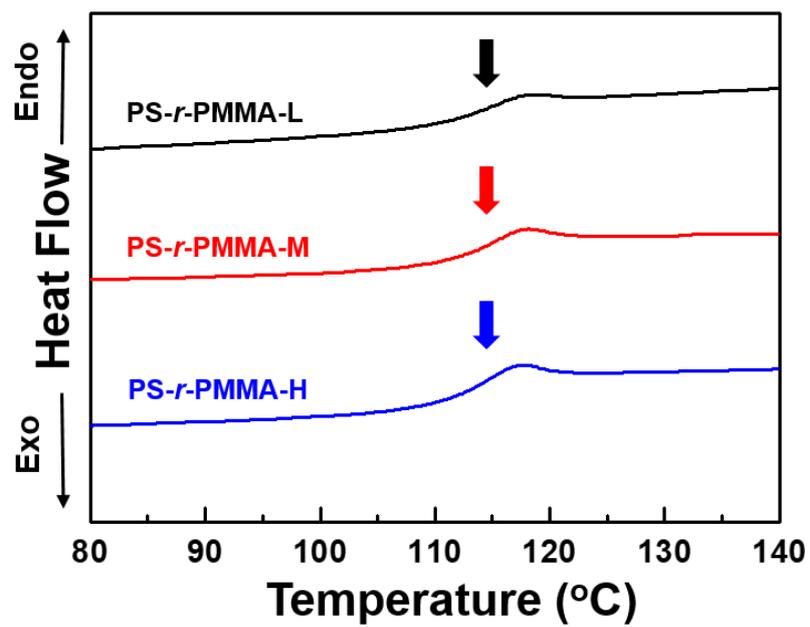


Fig. S7 DSC thermograms of PS-*r*-PMMA-L,-M,-H obtained during the second run at a heating rate of 10 °C/min.

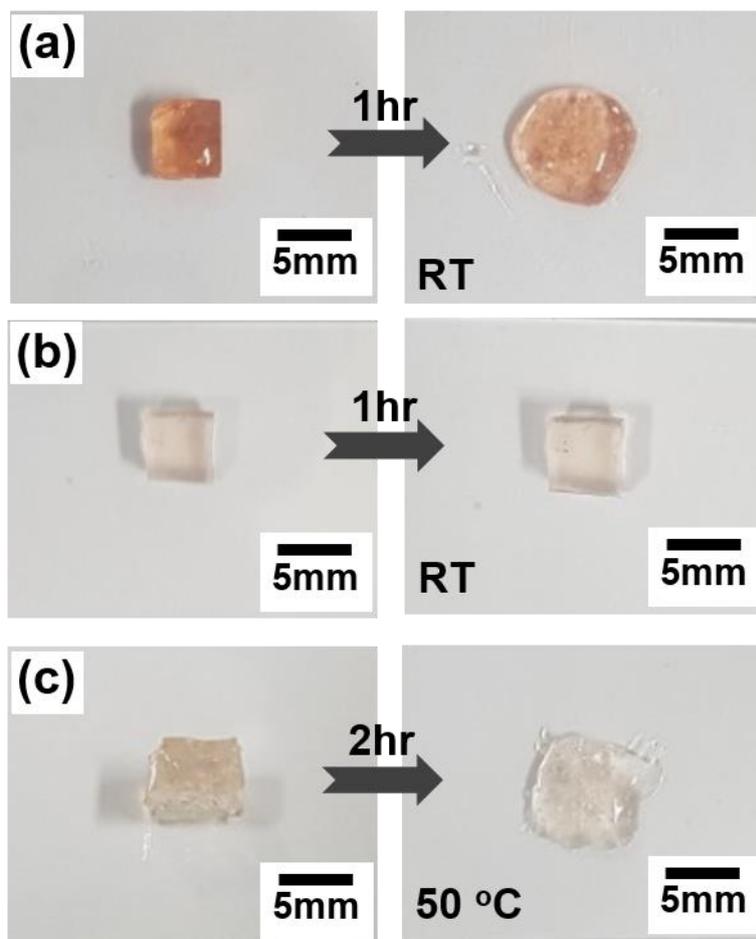


Fig. S8 Changes in a shape of PS-*r*-PMMA gels after 1 h: (a) PS-*r*-PMMA-L gel at RT, and PS-*r*-PMMA-M gel (b) at RT, and (c) 50 °C, in which The 70 wt% ionic liquid ([EMI][TFSI]) was included in all gels. The shape of PS-*r*-PMMA-L gel having elastic modulus of $\sim 2.3 \times 10^4$ Pa was deformed even after 1 h standing at RT, whereas the gel based on PS-*r*-PMMA-M having $\sim 5.0 \times 10^4$ Pa at RT was maintained its original shape. However, the improvement of mechanical robustness is still necessary for higher dimensional stability even at elevated temperature (e.g. 50 °C), so we adjusted the total molecular weight of PS-*r*-PMMA.

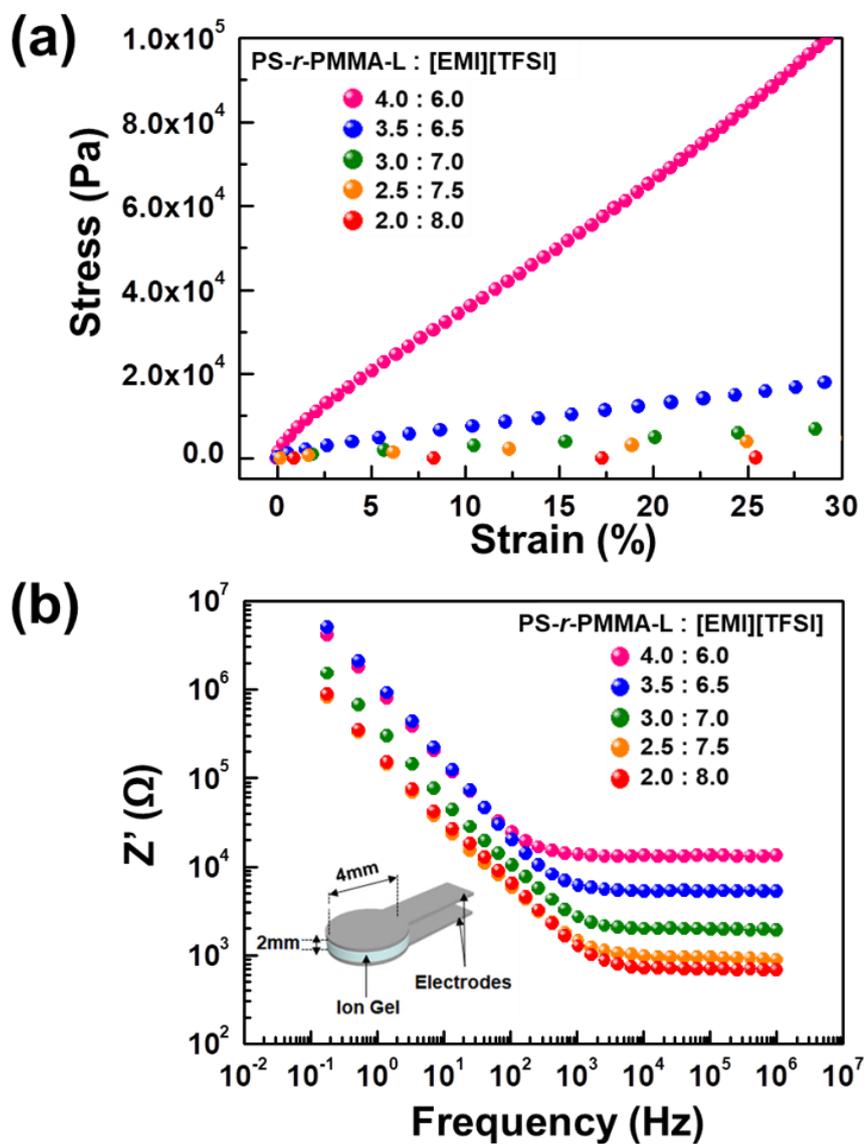


Fig. S9 (a) Stress-strain curves, and (b) frequency dependence of the resistance (Z') for ion gels, in which five different weight ratios of PS-*r*-PMMA-L and [EMI][TFSI] were tested.

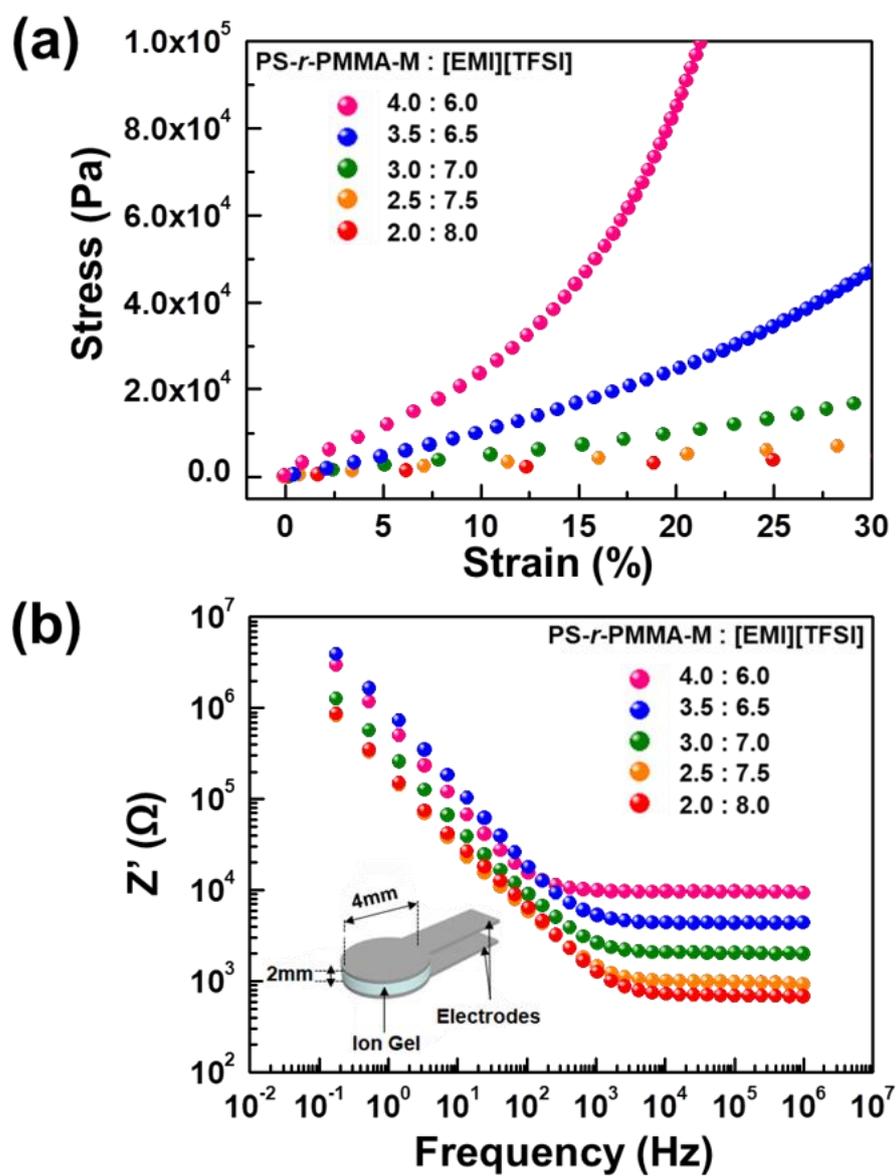


Fig. S10 (a) Stress-strain curves, and (b) frequency dependence of the resistance (Z') for ion gels, in which five different weight ratios of PS-*r*-PMMA-M and [EMI][TFSI] were tested.

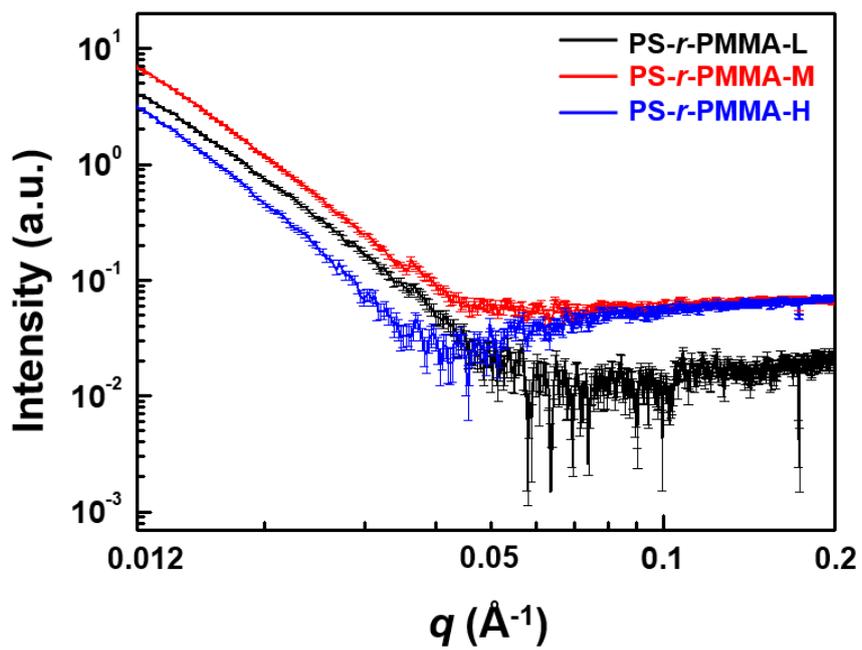


Fig. S11 Small angle x-ray scattering (SAXS) profiles of neat PS-*r*-PMMA, in which any characteristic scatterings were not observed, indicating homogeneous phase of random copolymers.