Supporting information

Amplified spontaneous emission and random lasing using CsPbBr₃ quantum

dots glass through controlling crystallization

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1. Experimental

1.1 Materials

All the raw materials were from Aladdin and used without further purification. They include the following: SiO₂ (AR), B₂O₃ (99.9%), ZnO (99%), Cs₂CO₃ (99%), PbBr₂ (99%), and NaBr (AR, 99%).

1.2 Fabrication method for the CsPbBr₃ QDs glass

CsPbBr₃ QDs based on a borosilicate glass (B-Zn-Si) was synthesized via melt-quenching and glass crystallization methods. First, the weighed material was placed in a mortar and mixed well. Second, the mixed powder was added to an alumina crucible, placed in a muffle furnace and heated up to 1220 °C at a heating rate of 5 °C/min. This temperature was maintained 10 min. Third, the melt was placed into a mold and the solidified glasses were quickly moved into a 410 °C annealing furnace for 150 min. Finally, the as-prepared glasses were cut into the pieces, and the CsPbBr₃ QDs were obtained via glass crystallization at different heat treatment (440~530 °C) temperatures for 5 h. And all the prepared CsPbBr₃ QDs glasses were cut into the thickness of 0.8

nm pieces for further characterization.

1.3 Measurements and Characterization

PL spectra were used to characterize the optical properties of the as-prepared CsPbBr₃ QDs glass and were obtained using a spectrofluorometer (FLS920) with pulsed xenon lamps. The UV-Vis absorption spectra of the CsPbBr₃ QDs glass were obtained using a UV-Vis spectrometer (PerkinElmer Lambda 750). The PL decay time of the CsPbBr₃ QDs glasses were monitored via time-correlated single-photon counting (TCSPC) with a nanosecond-pulsed diode laser (nanoLED-370). XRD patterns within the range of 10° ~80° were obtained to identify the crystal structure of the glass samples using XRD (XRD, D8 Advance, Bruker, Germany) with Cu K α radiation operating at 40 kV and 40 mA. An FEI Tecnai F20 instrument operating at 200 kV was used to obtain the TEM, HR-TEM, and SAED. To detect the valence states of the elements, an AXI ULTRA DLD spectrometer with a monochromator and Al K α as the X-ray source was used to obtain the XPS spectra. A QE-2100 spectrophotometer from Otsuka Japan Photal Electronics were used to measured PLQY.

The UC lasing emission of the CsPbBr₃ QDs glasses at different annealing temperature was recorded under 800 nm femtosecond (fs) pulsed laser excitation. Here, a Ti: sapphire laser (coherent Libra) was used as the excitation source and used to generate the 800 nm fs pulses (1kHz). A laser beam with an 8 mm diameter was focused by a convex lens with a 10 cm focal lens on the surface of the QDs glasses, which was located inside a low temperature chamber (Linkam DSC 600 temperature-controlled stage) purged with N₂ gas. The emission from the CsPbBr₃ QDs glass surface was recorded by an optical fiber (600 µm in diameter). The optical fiber was coupled to a monochromator (Princeton Spectra Pro 2750 integrated with a Pro EM EMCCD camera with a spectral resolution of less than 0.1 nm) for spectral analysis.

The optical gain was measured using the variable stripe length (VSL) method. The 800 nm fs pulsed laser with the diameter of 8 mm was focused by a cylindrical lens to a line on the CsPbBr₃ QDs glasses surface, which was located inside a low temperature chamber (Linkam DSC 600 temperature-controlled stage) purged with N₂ gas. An adjustable slit was placed between the cylindrical lens and the CsPbBr₃ QDs glasses surface to accurately modulate the length of the stripe. By changing the length of the strips on the CsPbBr₃ QDs glasses surface, the corresponding emission from the edge of the sample was collected by an optical fiber (600 µm in diameter). The optical fiber was coupled to a spectrometer (Ocean Optics USB 2000+VIS-NIR).

2. Supporting Tables

Sample	t ₁ (ns)	t ₂ (ns)	A ₁	A ₂	τ _{avg} (ns)
440 °C	11.28	45.11	0.2	0.8	43.12
470 °C	19.45	77.79	0.2	0.8	74.36
500 °C	25.26	101.05	0.2	0.8	96.59
530 °C	34.88	139.54	0.2	0.8	133.39

Table S1 Detailed information for calculating the decay time with the biexponential fit.

Pump density (mJ/cm ²)	g (cm ⁻¹)/ 440 °C	g (cm ⁻¹)/ 470 °C	g (cm ⁻¹)/ 500 °C	g (cm ⁻¹)/ 530 °C
0.017	-12.10	1.66	5.18	9.71
0.053	-3.40	8.31	12.58	10.90
0.088	-0.35	12.47	17.67	20.04
0.124	0.66	19.60	26.29	36.11

Table S2 Optical gain versus excitation power for the CsPbBr₃ QDs glasses at different heat treatment temperatures at ambient pressure and a temperature of -180 °C. The excitation wavelength was set to 800 nm.

3. Supporting Figures



Fig. S1 HR-TEM image of the CsPbBr₃ QDs glass.



Fig. S2 (a) The survey XPS spectra of the CsPbBr₃ QDs glass. The high resolution XPS spectrum of **(b)** Cs 3d, **(c)** Pb 4f, **(d)** Br 3d.



Fig. S3 The line graph shows the relationship between the τ_{avg} and the heat treatment temperature.



Fig. S4 The relationship between the integral emission intensities and thermal cycle of CsPbBr₃ QDs glasses heat-treated at **(a)** 440 °C, **(b)** 470 °C, **(c)** 500 °C and **(d)** 530 °C. The emission intensity of all the samples were maintained above 80% of the original intensity, although they have

experienced a 30 \sim 210 °C heating and cooling cycles process.



Fig. S5 Photographs of various CsPbBr₃ QDs glasses with different heat treatment temperature under of 800 nm fs pulsed laser excitation. (a) 440 °C, (b) 470 °C, (c) 500 °C, and (d) 530 °C.



Fig. S6 The UC emission for the CsPbBr₃ QDs glasses heat-treated at (a) 440 °C, (b) 470 °C, (c) 500 °C, and (d) 530 °C. The results show that a higher heat treatment temperature makes it easier to achieve random lasing. The random lasing spectra shows a shark peak.



Fig. S7 Stripe-length dependent emission spectra with a pump intensity of 0.017 mJ/cm², 0.053 mJ/cm², 0.088 mJ/cm², and 0.124 mJ/cm² for CsPbBr₃ QDs glasses heat-treated at different temperatures. (a) 440 °C, (b) 470 °C, (c) 500 °C, and (d) 530 °C. The experiments were carried out at ambient pressure and at a temperature of -180 °C.