

Supplementary Information for "Intrinsic Interaction between In-Plane Ferroelectric Polarization and Surface Adsorption"

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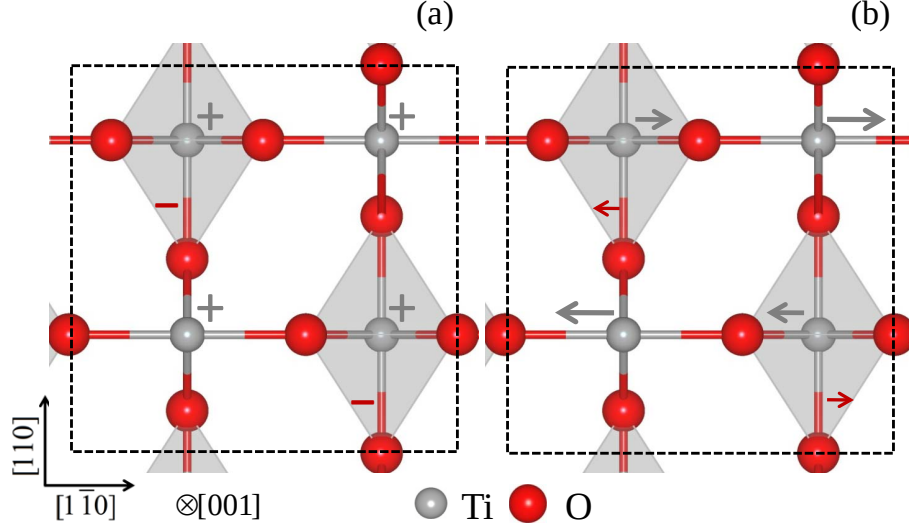


Figure S1: Schematic show of the spontaneous polarization of bulk rutile TiO_2 under tensile strain. The $(\sqrt{2} \times \sqrt{2} \times 1)$ supercells are marked by the dashed lines. The oxygen octahedrons can be divided into two groups, depending on whether the equatorial planes normal to $[1\bar{1}0]$ (OC_n) or in parallel to $[1\bar{1}0]$ (OC_p). Shaded areas mark the octahedrons of OC_p to guide the eyes. (a) When the strain is applied along $[001]$ direction, both groups are equivalent to each other, and the six neighboring O atoms displace in the opposite directions of the Ti atom, as marked by red negative signs for O and grey positive signs for Ti, respectively. (b) When the strain is applied along $[1\bar{1}0]$ direction, OC_n and OC_p become inequivalent. As shown by the red arrows for O and grey arrows for Ti atoms, respectively, the six neighboring O atoms displace oppositely relative to the Ti atom in OC_p , while in OC_n the four O atoms in the equatorial planes displace in the same direction as the Ti atom while the two apical O atoms displace in the opposite direction.

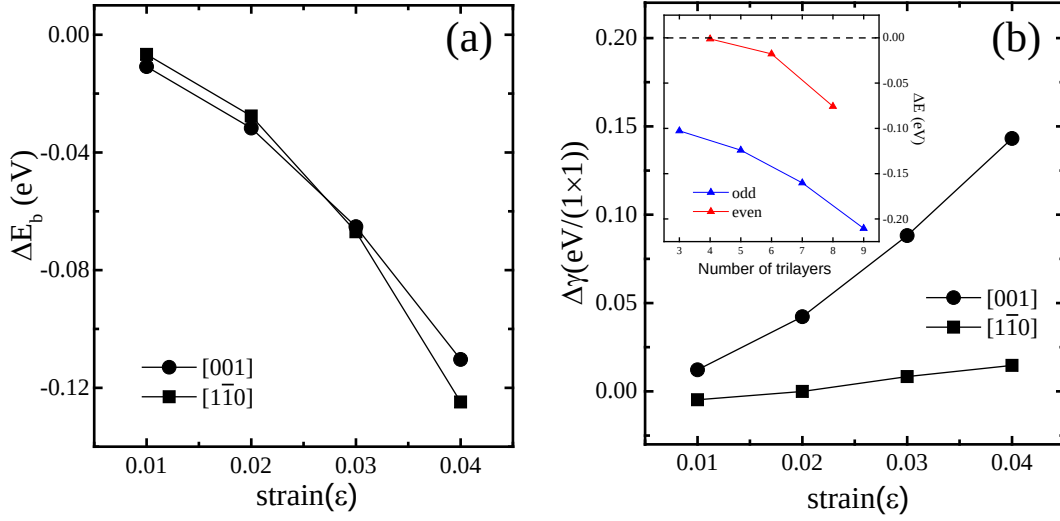


Figure S2: The changes of (a) the total energy of bulk TiO₂ and (b) the surface energy of the TiO₂(110) induced by the in-plane polarization. The bulk rutile TiO₂ is modeled by using a $\sqrt{2} \times \sqrt{2} \times 1$ supercell. The TiO₂(110) surface is modeled as a (1×1) slab containing nine O-Ti-O trilayers and a vacuum with thickness of 15 Å. All the atoms are relaxed without constraints until the forces were converged to 0.01 eV/Å. The inset of (b) shows the polarization-induced change of the total energy of the slab as a function of the slab thickness (n) when $\epsilon_{[001]} = 0.04$. It shows that the energy difference appears an odd-even oscillation behavior with n . Slabs with odd n are more favorable to polarization than even n .

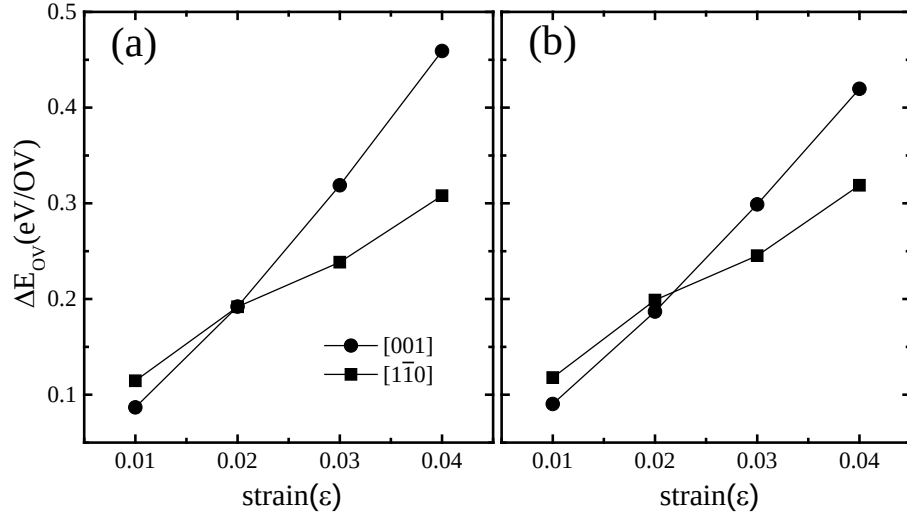


Figure S3: The changes of the formation energy of a bridging oxygen vacancy on $\text{TiO}_2(110)$ induced by in-plane polarization. The surface is modeled as a slab of (1×4) supercell containing five O-Ti-O trilayers and a vacuum with thickness of $15 \text{ \AA}$, (a) The atoms in the bottom trilayer are fixed as the same as in the corresponding bulk phase, and the other atoms were relaxed until the forces were converged to $0.01 \text{ eV/\AA}$. (b) The surface with in-plane polarization is modeled as in (a), while the non-polarized surface is modeled by simply scaling the zero-strained non-polarized surface according to the specific strain.

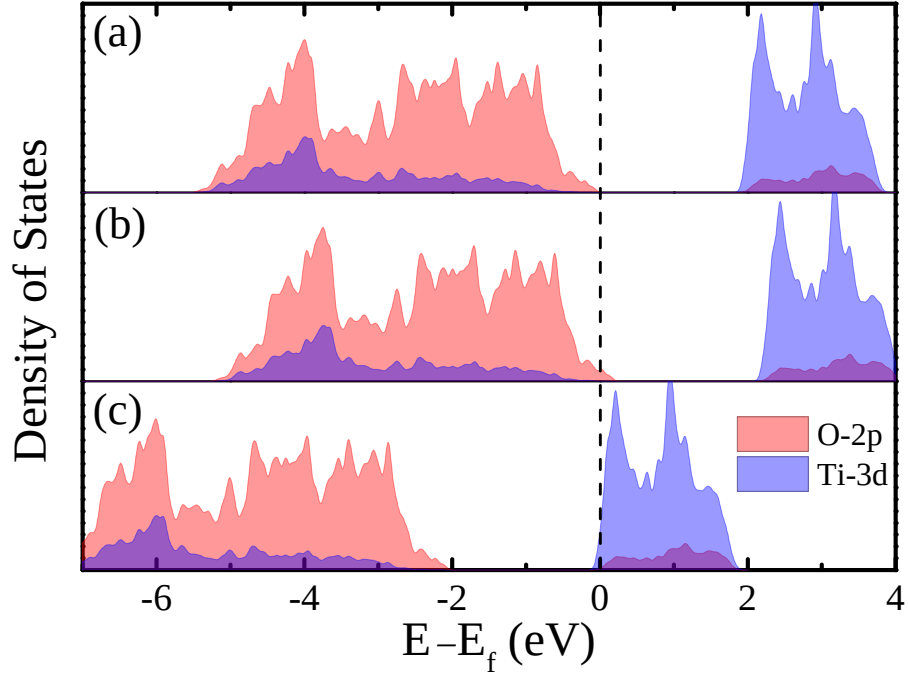


Figure S4: The partial density of states of bulk TiO_2 with doping concentration of (a) 0, (b) -0.075 and (c) 0.075 electron per unit cell when $\varepsilon_{[001]} = 0.03$. It suggests that the doping electrons occupy the Ti-3d orbitals while the doping holes are in the O-2p orbitals.