# Electronic Supporting Information (ESI) for

# Aqueous Two-Polymer Phase Extraction of Single-Wall Carbon Nanotubes using Surfactants

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This supplemental information contains workflows for various diameter populations of single-wall carbon nanotubes (SWCNT)s.

Certain equipment, instruments or materials are identified in this paper in order to adequately specify the experimental details. Such identification does not imply recommendation by the National Institute of Standards and Technology (NIST) nor does it imply the materials are necessarily the best available for the purpose. Where provided, uncertainty is reported as one standard deviation.

## Outline

- 1. Acronyms
- 2. Methods Information (SWCNT Preparation)
- 3. Description of Mimic Phases used in this Supplemental
- 4. Workflow Schematics for Various Diameter SWCNT Populations
  - A. Cobalt-Molybdenum Catalyst (CoMoCat) Synthesis
  - B. High-Pressure CO Disproportionation (HiPco) Synthesis (and similar)
  - C. Plasma Torch (PT) Synthesis or Laser Vaporization (LV) Synthesis
  - D. Electric Arc (EA) Synthesis
  - E. Floating Catalytic Decomposition (FCCD) Synthesis
- 5. Multistage Data Record Sheet Example

## Part 1. Acronyms

aqueous two-phase extraction (ATPE).<sup>1</sup> polyethylene glycol (PEG) dextran (DEX) sodium cholate (SC) sodium deoxycholate (DOC) sodium dodecyl sulfate (SDS) Sodium hypochlorite (NaClO) Semiconducting-Metallic separation (S/M) SDS-DOC ATPE separated top phase (T) SDS-DOC ATPE separated bottom phase (B) SDS-SC ATPE separated top phase (M<sub>T</sub>)

### Part 2. Methods Information

SWCNT powders from which previously unpublished results are shown or described were as-produced and purified electric arc synthesized SWCNT powders, lots AP-A218 and P2-02-A011, from Carbon Solutions of Riverside CA. Sodium deoxycholate (DOC, BioXtra >98 %, Sigma), sodium cholate hydrate (SC, BioXtra >99 %, Sigma), sodium dodecyl sulfate (SDS, BioXtra 99 %, Sigma), eicosane (C<sub>20</sub>H<sub>42</sub>, 99%, Aldrich), iodixanol (sold as Opti-Prep, 60 % w/v solution, Sigma), deuterium oxide (D<sub>2</sub>O, 99.8 %, Cambridge Isotopes), dextran 40, dextran 70, dextran 250 (TCI, Tokyo, Japan), polyethylene glycol PEG (6 kDa, Alfa Aesar) were purchased and used as received.

Dispersions of SWCNT populations, and either rate-zonal purification or separation of empty from water-filled SWCNTs, were performed as previously reported. Briefly, dispersion of as-received, purified or alkane-filled, SWCNT soot was *via* tip sonication (generally 45 min,  $\approx 0.9$  W/mL), with the vial in an ice-water bath, at a nominal concentration of 1 mg/mL of SWCNTs in 20 g/L DOC solution. Initial purification to remove large aggregates and non-nanotube components was performed *via* centrifugation in a J2-21 high-speed centrifuge (JA-20 rotor, 1885 rad/s (18 kRPM), 2 h) and collecting the resulting supernatant. This sonicated-centrifuged dispersion is subsequently layered ( $\approx 8.6$  mL) above a dense race layer (28 mL) comprised of either 10 % (w/v) or 12 % (w/v) iodixanol and 10.0 g/L DOC in 36.2 mL OptiSeal tubes for rate-zonal purification (VTi 50 rotor, 5236 rad/s (50 kRPM), 2 h 45 min, 20 °C) in a Beckman L80-XP ultracentrifuge to remove small bundles and reduce the content of morphologically impure SWCNTs.

Stirred ultrafiltration cells (Millipore) were used to concentrate SWCNT dispersions, reduce iodixanol or ATPE polymer concentrations to negligible (<< 1 ug/mL) levels, and to reset the DOC concentration to 10.0 g/L before ATPE, and for finished samples. Different sizes of the ultrafiltration cells from 76 mm to < 25 mm diameter (the membrane area) were utilized depending on the volume of sample. Regenerated cellulose membranes of 30 kDa or 100 kDa pore size were primarily utilized.

Separation of the semiconducting SWCNTs from the metallic SWCNTs in each population was achieved *via* ATPE<sup>1</sup> using the redox adjustment advance of Gui *et al.*<sup>3</sup> as described in workflow of Figure 4B or 4C, as described in the main text.

Absorbance spectroscopy was performed on a Cary 5000 UV-visible-near infrared (UV-vis-NIR) absorbance spectrometer through 1 mm path length cuvettes at room temperature. Samples and reference solutions were measured in separate runs and the reference spectra subtracted during data analysis. Measurements were conducted within subranges of 185 nm to 2500 nm depending on the SWCNT sample, but always with a 1 nm step, a 2 nm bandpass, and an integration time of 0.1 s/step.

Presented ESI work flows are additionally based on results from a number of SWCNT sources including: lots AP-A218 and P2-02-A011, from Carbon Solutions of Riverside CA; CoMoCat SG65i SWCNT powder, lot SG65i-L55, from Chasm Nanotechnologies, Norman, OK; as-produced SWCNT powder, lot May-18, from NoPo Technologies of Bangalore, India; Tuball SWCNT powder, lot 4-1803 2014 from OCSiAl, Russia.

Some SWCNT dispersions were made with SWCNT populations comprised of alkane-filled SWCNTs.<sup>2</sup> Briefly, to generate an alkane-filled population for larger diameter SWCNTs, the alkane and an open-ended nanotube soot are first incubated together above the melting point of the alkane for >12 h. During this time the alkane is ingested into the endohedral volume of the nanotubes, which have open ends from their purification process, resulting in filling of the entire population of nanotubes with the alkane. After incubation, the alkane@SWCNT powder was rinsed with heptane, filtered against a membrane (VVLP, 0.1 µm, Millipore), rinsed with heptane against the membrane, and allowed to fully dry at room temperature.

### Part 3. Description of Mimic Phases used in this Supplemental

Except in some special cases, for surfactant ATPE we generate the so-called mimic phases at compositions approximating the phase boundary points, but without including the minor component polymer. Calculating and plotting the effects of such a choice on the polymer:polymer phase diagram in Fig. 1 in the main text, it is clear both that the resulting phase compositions are not precisely on the same tie-line, but also that the degree of drift from the tie-line is minimal even over multistage separation. The reasons behind this choice are discussed in the main text.

To simplify recording of mimic phase compositions a shortened notation is used. Written in brackets, we report, in order  $[C_{DEX}, C_{PEG}, C_{DOC}, C_{SDS}, C_{SC}]$ , in which  $C_i$  is the concentration of the denoted component. An example of the type of worksheet we use to rapidly record concentrations, volumes, temperature and fraction numbers is included at the end of this document. An electronic version as a separate, editable, file is available from the author on request.

Common Stock Solutions we use are:

Polymers: (25 and 40) % (mass/mass) PEG; 20 % (mass/mass) DEX

Surfactants: (1, 2, or 4) % (mass/vol) DOC; (4.5 and 10) % (mass/vol) SC; (4.5 and 10) % (mass/vol) SDS Oxidant:  $1/100^{\text{th}}$  dilution with 18 M $\Omega$  water of (10-15) % chlorine content NaClO (value reported by manufacturer, new dilution of oxidant prepared fresh daily)

Stock solutions of polymers of greater concentration exhibit increased viscosity that makes accurate pipetting difficult. Stock solutions of SDS and SC of 5 % would also work, and allow for easy dilutions, 4.5 % is used in our labs for historical reasons, and use is continued because formation of 45 mL volumes of mimic phases is straightforward.

Common Mimic phase used are ([C<sub>DEX</sub>, C<sub>PEG</sub>, C<sub>DOC</sub>, C<sub>SDS</sub>, C<sub>SC</sub>]) in nominal percent concentration:

- a. Semiconducting-Metallic Separation<sup>3</sup>
  - a. top phase [0, 11.1, 0.0, 0.7, 0.9] note, 11.1 % is 20 mL of 25 % PEG + 25 mL other components
  - b. bottom phase [0, 15, 0, 0.7, 0.9]

note, for separating (6,5), (6,4) and (7,3) to the top phase [0, 11.1, 0.0, 1.15, 0.9] is used.

- b. Diameter-Structure Separation<sup>4,5</sup>
  - a. Small diameter nanotube populations top phase [0, 11.1, 0.04, 1.2, 0]
  - b. Small diameter nanotube populations bottom phase [0, 15, 0.05, 0, 0]
  - c. Large diameter nanotube populations top phase [0, 11.1, 0.1X, 0.7, 0]
  - d. Large diameter nanotube populations bottom phase [15, 0, 0.1X, 0, 0]
    X depends on the nanotube diameter.

## Part 4. Workflow Schematics for Various Diameter SWCNT Populations

All SWCNT populations start at 10 g/L (nominally 1 %) DOC concentration in water. To reach separation conditions for ATPE the DOC concentration must be reduced, although to varying degrees depending on the nanotube type.

Note that within the workflow schematics the concentrations specified [% DOC,% SDS, % SC], are global concentrations. Arrows are intended to represent approximately one stage of separation, although generally the fractions labelled with specific (n,m) species may need more processing steps to reach very high purities. Where they occur, asterisks are used to save space when specifying the separation occurs at the "\*" specified concentration.

#### **Preconcentration:**

For each of the following workflows, one or two "preconcentration" steps are applied before the first separation step.

2 parts 20 % DEX + 2 parts SWCNT dispersion + 1 part 40 % PEG.  $\rightarrow$  [8, 8, 0.4, 0, 0] global composition. All SWCNTs will partition to the bottom phase. Top is discarded.

1 part Bottom Phase (PC #1) + 1 part 12.5 % PEG  $\rightarrow$  0.2 % DOC, while staying  $\approx$  on the tie-line

All SWCNTs will partition to the bottom phase. Top is discarded. Move onto separations.

#### Remember to end separations in the PEG-rich Phase!

In almost all cases it is preferable to end in the upper, PEG-rich, phase in the PEG-DEX ATPE system. If your final sample is in a bottom phase, do one more ATPE step with a reduced DOC % and/or increased SDS % to shift the sample into the upper phase (the sample can often be successfully concentrated by partitioning into a smaller upper phase volume in this step as well). More comments are in the main text.

#### A. Cobalt Molybdenum Catalyst (CoMoCat) SWCNTs

CoMoCat synthesis method SWCNTs, the most common of which are SG65i type, typically includes a narrow distribution of SWCNTs  $\approx$  0.7 nm to 0.85 nm (carbon center to carbon center diameter definition). The SG65i population is strongly biased to the (6,5) structure for semiconducting (n,m)s, and in our experience to the (7,4) for the metallic population.





#### B. High Pressure CO Disproportionation (HiPco) type

HiPco-type SWCNT populations are significantly more complex than CoMoCat materials to sort via ATPE as the diameter range, and consequently the number of (n,m)s is substantially larger. Additionally, the HiPco diameter range includes both outlier partitioning behavior SWCNT (n,m)s such as the (10,5), and (8,8), as well as small quantities of the (6,5), (6,4) and similar SWCNT species that added difficulty to the SG65i workflow. Significant spectral data for processing of HiPco SWCNTs *via* ATPE are presented in the SOI of Fagan *et al.*<sup>4</sup>



## C. Plasma Torch (PT) or Laser Vaporization (LV)

These SWCNT populations tend to have a larger average diameter and narrower distribution than most HiPco populations that we have worked with, but a smaller average diameter than most electric arc (EA) synthesis populations. Substantial spectroscopy and other results were demonstrated in Fagan *et al.*<sup>4</sup>

Normal Processing (S/M then Gradient Strategy)	
Polymers PT or LV Workflow (Simplified)	These SWCNT populations respond with high
[DOC %, SDS %, SC %]	uniformity during S/M separation in our experience.
→ : SDS-DOC ATPE (T/B)	Thus it is highly advantageous to first separate the
$\Rightarrow : SDS-SC ATPE (M_T/M_B)$	semiconducting (n,m)s from the metallic species
PC → : Temp. Swing	before (n,m) separation.
	To improve diameter resolution the use of a
M <sub>T</sub> Discard M <sub>T</sub> BM <sub>T</sub> Large Diam. Semis	somewhat increased DOC concentration during
M <sub>B</sub> Discard	structure-diameter separation, as identified in Fagan
[< 0.02 %, 0.7 %, 0.9 %] Large Diam. Metallics	<i>et al.,</i> <sup>4</sup> is recommended. Ease of isolation of
Add to $M_T$ $\approx 30 \text{ min}$	particular (n,m) structures varies with these SWCNTs,
$2M_B \rightarrow 3M_B \rightarrow Discard$	particularly due to abundance differences, but
[0.05 %, 0.7 %, < 0.9 %]	separation of single (n,m) structures by hand is
	feasible. <sup>4</sup> One challenge, however, is that E22 optical
$M_TBM_T$ OR Metallic $3M_BT$ fraction	transitions are near the edge of human eye sensitivity
<b>I</b> [ 0.08 %, 0.7 %, 0.0 %]	for some nanotubes in the population, making
T → TT → Continued	discrimination of a change more difficult or varying
B TB $\rightarrow$	depending upon a person's individual color
[ 0.08 %, 0.8 %, 0.0 %]	sensitivity.
Smaller diameter semis, continued cascade	A typical oxidant level added during S/M ATPE for
	this material is 5 $\mu$ L/mL to 7 $\mu$ L/mL of the 1/100 <sup>th</sup>
	concentration NaClO.

#### D. Electric Arc (EA) Synthesis

The average diameter of an EA population is in our experience around 1.5 nm, although it can be somewhat smaller or larger depending on the synthesis conditions. This larger average diameter likely includes an increased number of (n,m) structures than smaller average diameter populations simply due to the increased number of combinatorial (n,m) possibilities. This factor, and the spectral congestion of the E33 optical transitions (E22 transitions are outside the range of human eye sensitivity) increase the difficulty in by hand multistage structure-diameter ATPE. The population, however, responds well to S/M separation and is readily separated by that vector. Substantial spectroscopy and other results were demonstrated in Fagan *et al.*<sup>4</sup>



concentration NaClO.

## E. Floating Catalytic Decomposition (FCCD)

SWCNTs produced by this method have a broad diameter distribution centered at greater than 1.7 nm. The diameter distribution is so broad that the transitions add together to generate an absorbance spectrum that can appear to be only a decaying baseline, but it is really comprised of the sum of many transitions occurring across a broad range of wavelengths.





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<sup>1</sup> C. Y. Khripin, J. A. Fagan, M. Zheng Spontaneous Partition of Carbon Nanotubes in Polymer-Modified Aqueous Phases. *J. Am. Chem. Soc.* 2013, **135**, 6822–6825.

<sup>3</sup> H. Gui, J. K. Streit, J. A. Fagan, A. R. H. Walker, C. Zhou, M. Zheng Redox Sorting of Carbon Nanotubes. *Nano Lett.* 2015, **15**, 1642–1646.

<sup>4</sup> J. A. Fagan, C. Y. Khripin, C. Silvera Batista, J. R. Simpson, E. H. Hároz, A. R. Hight Walker, M. Zheng Isolation of Specific Small-Diameter Single-Wall Carbon Nanotube Species via Aqueous Two-Phase Extraction. *Adv. Mater.* 2014, **26**, 2800–2804.

<sup>5</sup> J. A. Fagan, E. H. Hároz, R. Ihly, H. Gui, J. L. Blackburn, J. R. Simpson, S. Lam, A. R. Hight Walker, S. K. Doorn, M. Zheng Isolation of >1 nm Diameter Single-Wall Carbon Nanotube Species Using Aqueous Two-Phase Extraction. *ACS Nano* 2015, **9**, 5377–5390.

<sup>6</sup> J. K. Streit, S. Lam, Y. Piao, A. R. Hight Walker, J. A. Fagan, M. Zheng Separation of Double-Wall Carbon Nanotubes by Electronic Type and Diameter. *Nanoscale* 2017, **9**, 2531–2540.

<sup>&</sup>lt;sup>2</sup> J. Campo, Y. Piao, S. Lam, C. M. Stafford, J. K. Streit, J. R. Simpson, A. R. Hight Walker, J. A. Fagan Enhancing Single-Wall Carbon Nanotube Properties through Controlled Endohedral Filling. *Nanoscale Horizons*, 2016, **1**, 317–324.