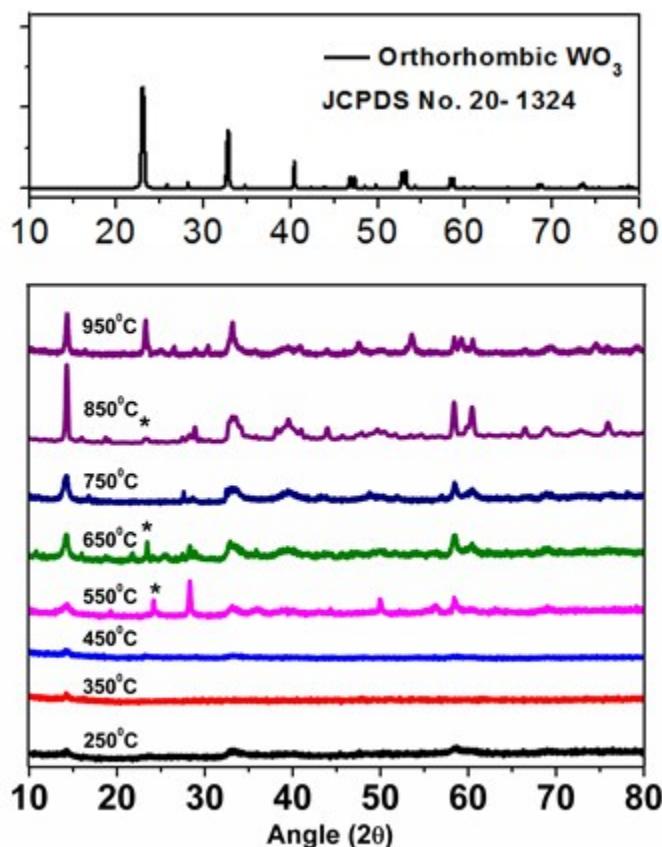


## Electronic Supplementary Information

### Calcination of $\text{WS}_3$ at different temperatures

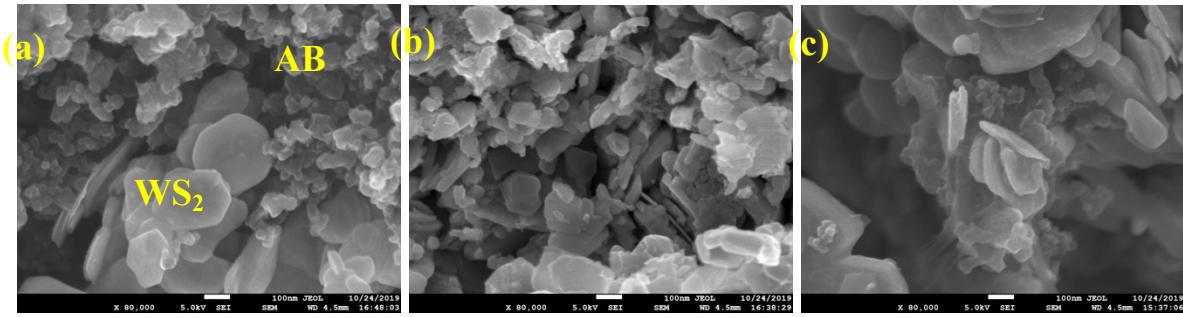
$\text{WS}_3$  was calcined at different temperatures to produce  $\text{WS}_2$ . Fig. S1 shows the complete XRD of the calcined products. From the figure, one can observe that with increasing calcination temperature, the (002) reflections became more crystalline and resolved. At 850°C, we have found the best crystalline  $\text{WS}_2$  as seen from the intensity of (002) reflection. We also observe a slight impurity of orthorhombic  $\text{WO}_3$  at calcining temperatures 550°C, 650°C and 850°C as shown in Fig.S1. When the temperature is increased to 950°C, the (002) reflections broaden and become less intense along with a higher intensity peak of  $\text{WO}_3$ . The  $\text{WS}_2$  material at 850°C was used for electrochemical characterization based on its purity (< 1wt%  $\text{WO}_3$  phase) and crystallinity.



**Figure S1.** PXRD of  $WS_3$  calcined at different temperatures

**FESEM images of  $WS_2$  electrodes before and after cycling**

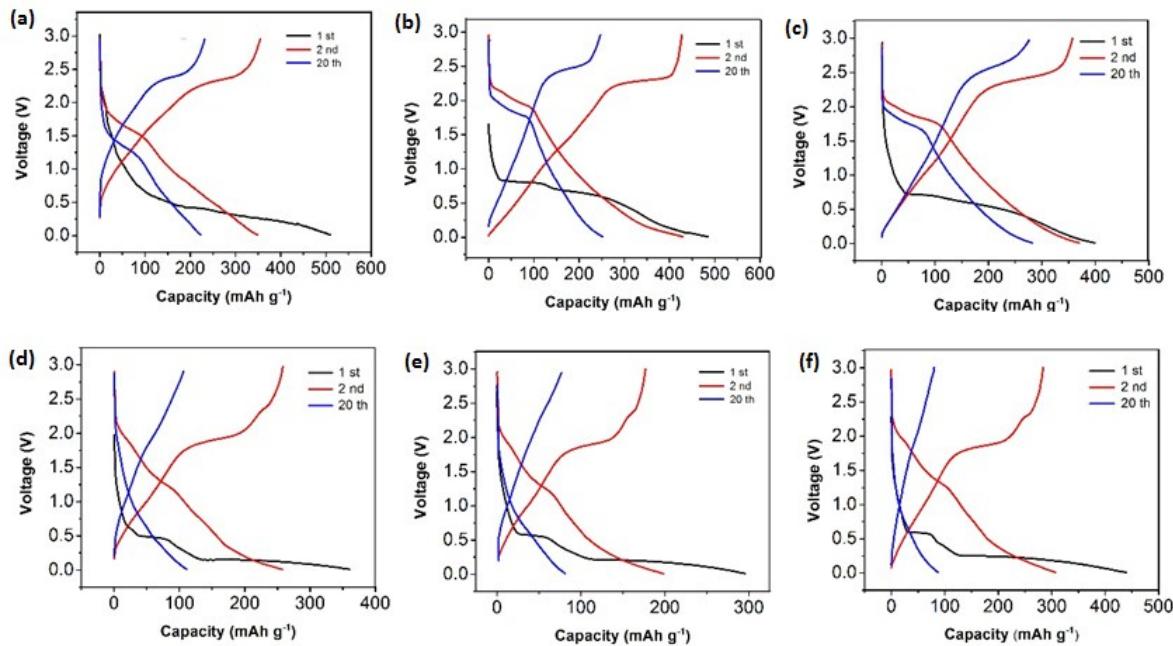
FESEM images of the electrodes were recorded with JSM-7610F, JEOL, Japan, with an accelerating voltage of 20 kV. The electrode images before and after 20 cycles are shown in Fig. S2. The morphology of  $WS_2$  has changed due to repeated insertion/conversion reactions during charging and discharging. The  $WS_2$  platelets/sheets and the acetylene black (AB) added as a conducting agent in the electrode are marked.



**Figure S2.** FESEM of the  $WS_2$  electrode (a) pristine (b) after 20 cycles in LIB and (c) after 20 cycles in NIB.

**Electrochemical performance of calcined materials**

The calcined materials at different temperatures, namely,  $WS_2$  synthesized at 650°C (denoted as  $WS_2$ -650), 750°C (denoted as  $WS_2$ -750) and 850°C (denoted as  $WS_2$ -850) were tested as anodes for both LIB's and NIB's. The charge-discharge curves for the 1<sup>st</sup>, 2<sup>nd</sup> and 20<sup>th</sup> cycles are shown in Fig. S3 (a) to (f). Table S1 shows the reversible capacities obtained after cycle 1 and 20 of all the three samples in LIB and NIB.



**Figure S3.** Battery cycling performance of (a) WS<sub>2</sub>-650, (b) WS<sub>2</sub>-750, (c) WS<sub>2</sub>-850 in LIB and (d) WS<sub>2</sub>-650, (e) WS<sub>2</sub>-750 and (f) WS<sub>2</sub>-850 in NIB.

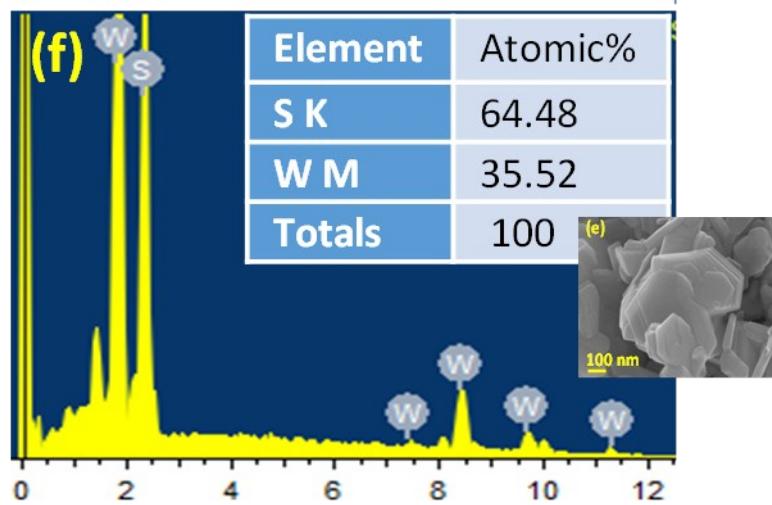
System	Compound	Reversible capacity Cycle 1 (mAh g <sup>-1</sup> )	Reversible capacity Cycle 20 (mAh g <sup>-1</sup> )
Li ion battery	WS <sub>2</sub> -650	338.27	216.96
	WS <sub>2</sub> -750	479.80	241.45
	WS <sub>2</sub> -850	358.50	281.02
Na ion battery	WS <sub>2</sub> -650	259.15	110.39
	WS <sub>2</sub> -750	176.93	72.79
	WS <sub>2</sub> -850	366.36	75.83

**Table S1.** Reversible capacities obtained after the first and twentieth cycles of WS<sub>2</sub> 650, WS<sub>2</sub> 750 and WS<sub>2</sub> 850 in LIB and NIB.

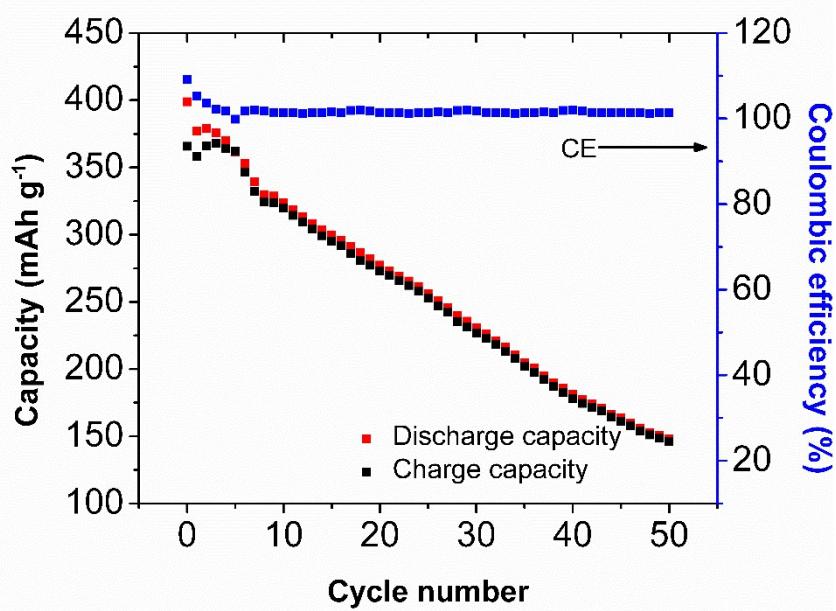
### Synthesis of WS<sub>2</sub>/C nanosheets

Citric acid (99 wt%, Loba Chemie) is used as the carbon precursor to synthesize WS<sub>2</sub>/C composite. The desired stoichiometric amount of citric acid (8 wt% of C) is added into a beaker with ethanol (99.9%) as the solvent. After complete dissolution under sonication, WS<sub>2</sub> powder was dispersed into this solution at a concentration of 20 mg mL<sup>-1</sup> and sonicated again for 3h. The solution is filtered, washed, and dried in an oven at 80°C. The powder was carbonized in a tube furnace at

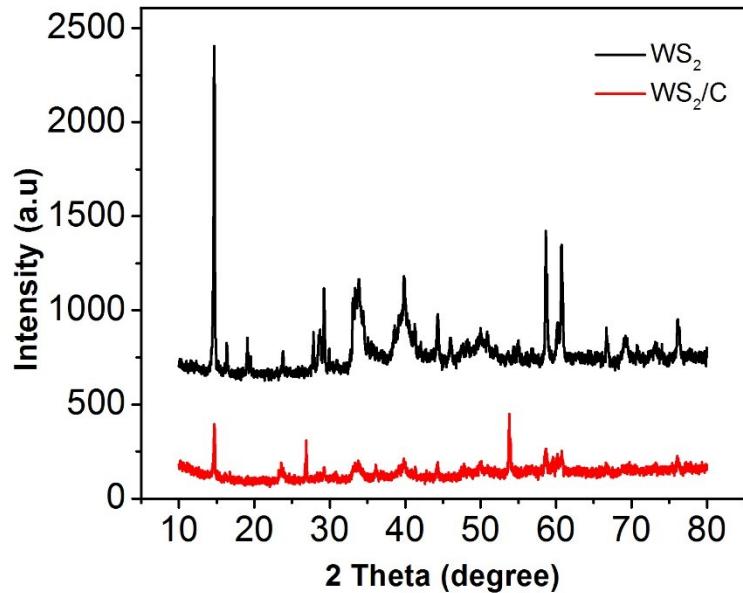
450°C for 1h at a heating rate of 4°C/min under Ar (99.999%). The obtained  $\text{WS}_2/\text{C}$  is characterized using CHNS analysis and powder XRD. PXRD shows phase pure  $\text{WS}_2$  after carbonization treatment (Fig. S6). The final carbon wt% in  $\text{WS}_2$  from CHNS analysis was found to be 3.84%. Fig. S8 shows the FESEM images of (a) pristine  $\text{WS}_2$  nanosheets and (b) carbon-coated  $\text{WS}_2/\text{C}$  nanosheets.



**Figure S4.** EDX analysis of pristine  $\text{WS}_2$  nanosheets.



**Figure S5.** Charge-discharge cycles and CE of pristine  $WS_2$  (vs.  $Li/Li^+$ ) at 1C in the voltage range 0.01 V – 3.0 V for 50 cycles.



**Figure S6.** The PXRD of pristine  $WS_2$  and  $WS_2/C$ .

Material morphology	C-rate	Sp. Capacity (mAhg <sup>-1</sup> )/Cycle number	References
$WS_2$ - rGO/SWCNT	2.5C	688/1000	1
$WS_2$ /CNF	1C	545/800	2
$WS_2$ /NF/rGO	2.5C	250/200	3
$WS_2$ /NT/graphene	2.5C	320/500	4
$WS_2$ -double shell C-coating	1C	486/200	5
$WS_2$ -triple shell C-coating	2.5C	784/1000	6
$WS_2/C$ nanosheets	1C	400/700	This work

**Table S2.** The capacity retention of  $WS_2@C$  reported until now at a rate 1C or more and the synthesized  $WS_2/C$  nanosheets cycled at 1C rate without any electrolyte additives (FEC, VC).

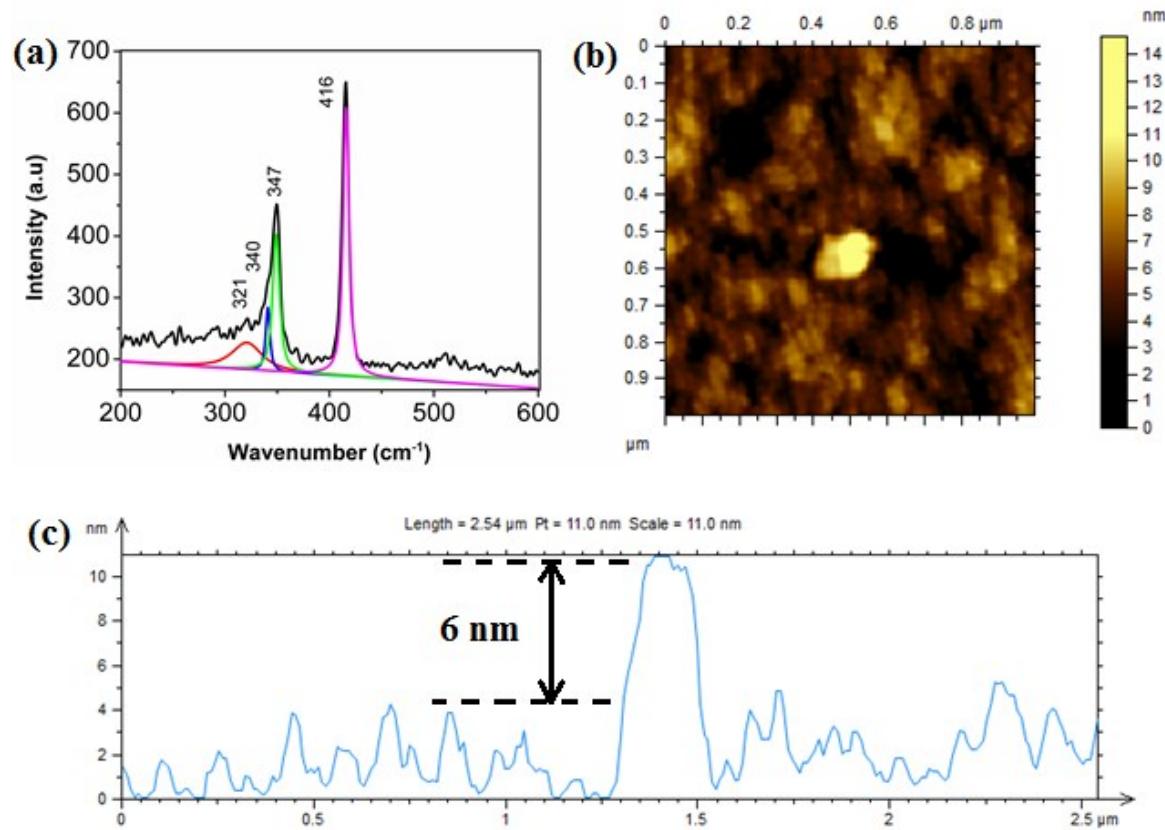
### Raman spectroscopy

**Figure S7(a)** shows the Raman spectra of  $WS_2$  nanosheets with an excitation wavelength of 514.5 nm. The Raman spectra at  $\lambda_{\text{excitation}} = 514.5$  nm shows additional second-order peaks, especially when the number of  $WS_2$  layers in the sample is less than three. The first-order optical modes show the two most prominent peaks seen at  $347\text{ cm}^{-1}$  and  $416\text{ cm}^{-1}$  correspond to the  $E_{1g}$  and  $A_{1g}$  modes

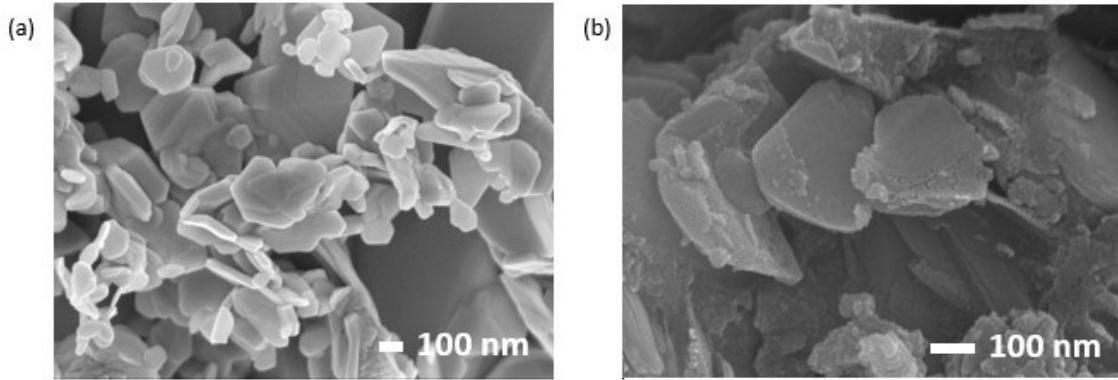
of vibration. The second-order peaks are observed at  $340\text{ cm}^{-1}$  and  $321\text{ cm}^{-1}$ . We performed multi-peak Lorentzian fitting on the spectra to deconvolute the observed peaks. The peak at  $340\text{ cm}^{-1}$  represents the 2LA (M) mode of vibration. In order to figure out the number of layers in the WS<sub>2</sub> sample, we compared the intensity ratio of the peaks. The ratio of peak intensities of  $E^{1g}$  and  $A_{1g}$  modes ( $I_{E^{1g}}/I_{A_{1g}}$ ) is around 0.69, and that of 2LA (M) and  $A_{1g}$  ( $I_{2LA(M)}/I_{A_{1g}}$ ) modes is 0.51. Comparing with the reported values, the number of layers of WS<sub>2</sub> present in the nanosheets is more than three.<sup>7</sup> The difference between the peak positions of  $E^{1g}$  and  $A_{1g}$  modes is  $69\text{ cm}^{-1}$ , also confirms the presence of more than three layers of WS<sub>2</sub>.<sup>8,9</sup> The AFM height profile (Figure S7 (c)) shows the thickness is approximately 6 nm, which corresponds to the presence of 5-6 layers of WS<sub>2</sub>, reaffirming the results of Raman spectroscopy.

### **Atomic Force Microscopy studies on pristine WS<sub>2</sub>**

Synthesized WS<sub>2</sub> nanosheets were characterized by AFM. Fig. S7 (b) and (c) show the AFM image and the particle thickness distribution, respectively. Particle thickness was found to be approximately 6 nm, considering a substrate roughness of 4 nm. This value matches well with the crystallite size value of 7.1 nm obtained from the XRD data.<sup>7,12,13</sup>



**Figure S7.** (a) Raman spectra of  $\text{WS}_2$  nanosheets with 514.5 nm laser excitation along with multi-peak Lorentzian fitting. (b) AFM of  $\text{WS}_2$  nanosheets and (c) particle thickness distribution



**Figure S8.** FESEM images of (a) pristine  $\text{WS}_2$  nanosheets (b) carbon coated  $\text{WS}_2/\text{C}$  nanosheets

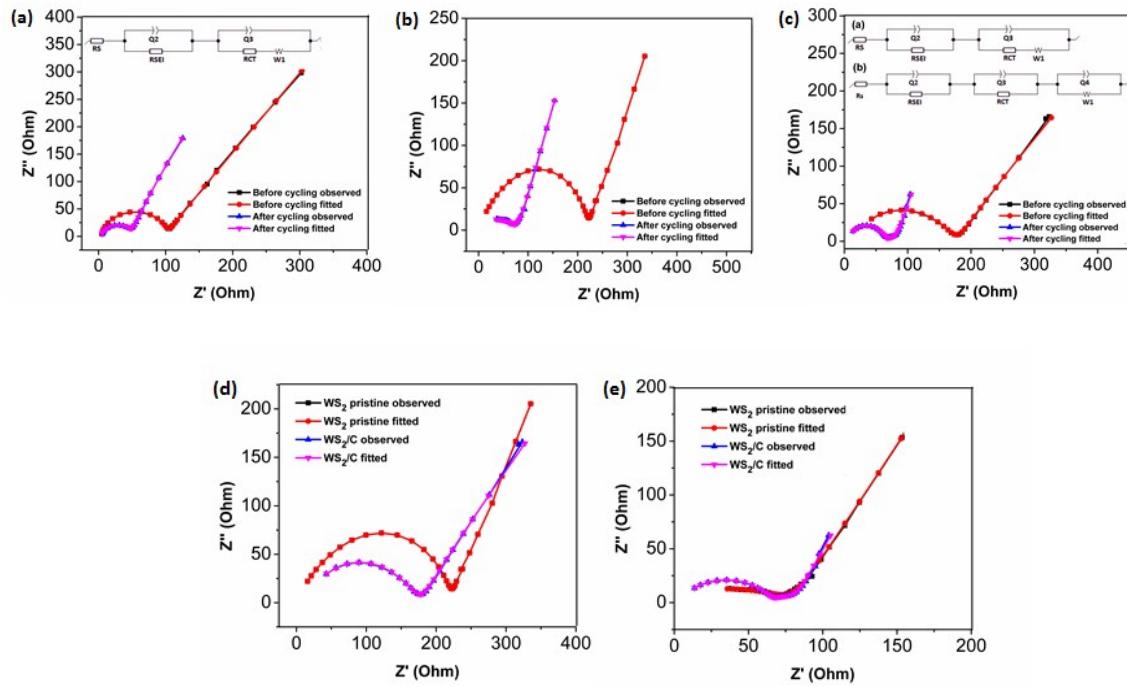
### Electrochemical Impedance Spectroscopy measurements

EIS measurements were carried out on pristine  $WS_2$  and  $WS_2/C$  electrodes in LIB and NIB. The Nyquist plots obtained are shown in Fig. S9 (a) to (e). The circuit is shown in Fig. S9 (a) was used to fit the EIS data for pristine  $WS_2$  in LIB and NIB. The amorphous carbon in  $WS_2/C$  composite, shows a capacitive effect during cycling. This changes the overall interface behavior of the material during cycling. So, for  $WS_2/C$  electrode, the circuit in Fig. S10(a) was used for fitting the before cycling data and Fig. S10 (b) for after cycling data.  $R_S$ ,  $R_{CT}$  and  $R_{SEI}$  refer to the resistances of the solution, charge transfer resistance at the interface and the SEI layer resistance respectively.  $Q1$  and  $Q2$  are the constant phase elements associated.  $W1$  refers to the Warburg impedance related to ion diffusion in the bulk of  $WS_2$ . EIS was carried out on the cells in the frequency range 10 kHz to 10 mHz before and after cycling them for 20 cycles each at 1C rate from 0.01V – 3.0V.

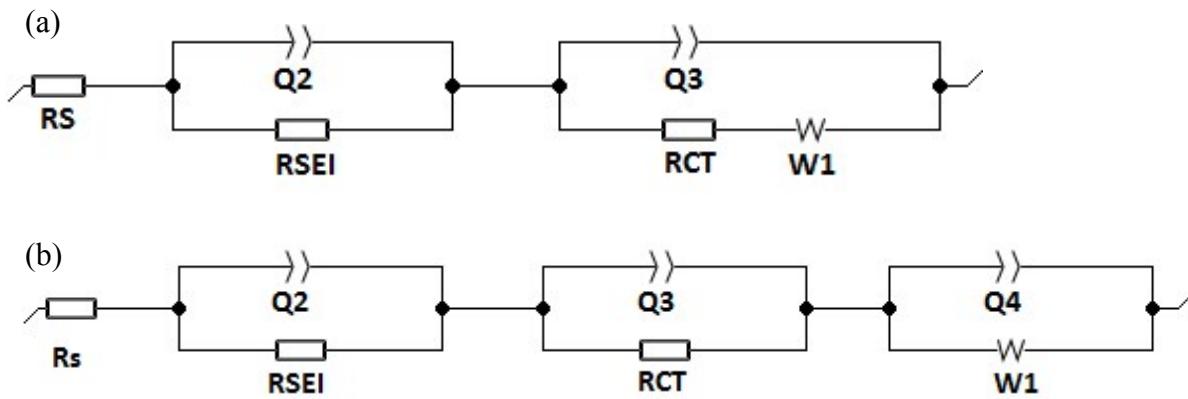
Fig. S9 (a), (b) and (c) show the Nyquist plots of the impedance of  $WS_2$  battery in LIB and NIB and that of  $WS_2/C$  in LIB. From all the three graphs, it is clearly seen that the diameter of the semicircle is reducing after cycling, which indicates the charge transfer (CT) resistance is decreasing. Table S3 shows fitting results obtained from EIS data and confirms the improving CT kinetics with cycling.

During the first discharge process, conversion reaction happens to lead to the formation of metallic tungsten and  $Li_2S$  in the case of LIB and metallic W and  $Na_2S$  in the case of NIB. This improves conductivity, thereby reducing the impedance after 20 cycles.<sup>10</sup>  $R_{SEI}$  increases indicating formation and thickening of the SEI after the first cycle.<sup>2,14-16</sup>

Fig. S9 (d) and (e) show the EIS plots of  $WS_2$  pristine and  $WS_2/C$  in LIB. The impedance of  $WS_2/C$  is much lower than that of the pristine, showing a better CT kinetics from the improved electrical conductivity of the  $WS_2/C$  composite sample.



**Figure S9.** EIS of (a)  $\text{WS}_2$  pristine with Na (b)  $\text{WS}_2$  pristine with Li (c)  $\text{WS}_2/\text{C}$  with Li (d) comparison of  $\text{WS}_2$  pristine and  $\text{WS}_2/\text{C}$  with Li before cycling (e) after cycling.



**Figure S10:** Equivalent circuits (EC) used for fitting the EIS data (a) was used for fitting the EIS data of pristine  $\text{WS}_2$  in LIB and NIB. For  $\text{WS}_2/\text{C}$  (a) was used to fit before cycling data and circuit (b) for after cycling data.

S. No.	Compound	R <sub>s</sub> (Ohm)		R <sub>SEI</sub> (Ohm)		R <sub>CT</sub> (Ohm)	
		Before	After 20 cycles	Before	After 20 cycles	Before	After 20 cycles
1.	WS <sub>2</sub> with Na	1.688	1.795	1.566	4.259	90.65	47.17
2.	WS <sub>2</sub> with Li	2.483	13.44	11.6	14.51	227.1	96.61
3.	WS <sub>2</sub> /C with Li	2.451	2.947	15.2	15.39	185.4	63.04

**Table S3:** Fitting results obtained from EIS data in Fig.S9.

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