## **Supporting Information**

## Prussian blue analogue-derived Ni and Co bimetallic oxide nanoplate arrays block-built from porous and hollow nanocubes for the efficient oxygen evolution reaction

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## **Computational details:**

Density functional theory (DFT) calculations are done using the projector-augmented wave method and a plane-wave basis set as implemented in the Vienna Ab initio Simulation Package (VASP).<sup>[1,2]</sup> The valence configurations are treated as 1s<sup>1</sup> for H, 2s<sup>2</sup>2p<sup>4</sup> for O, 3d<sup>8</sup>4s<sup>1</sup> for Co, 3d<sup>9</sup>4s<sup>1</sup> for Ni, and 4d<sup>7</sup>5s<sup>1</sup> for Ru. The Bayesian Error Estimation Functional with van der Waals correlation (BEEF-vdW) is employed.<sup>[3]</sup> The cutoff energy for plane-wave basis functions is 550 eV. By substituting half Co atoms with Ni atoms within the inverse spinel structure of NiCo<sub>2</sub>O<sub>4</sub> (JCPDS No. 20-0781), the bulk structure of Ni<sub>2</sub>CoO<sub>4</sub> is generated via the Special Quasirandom Structures (SQS) method, which has been widely used to determine the atomic distributions in solid solutions.<sup>[4]</sup> The bulk lattice parameters of NiO and RuO<sub>2</sub> are fully optimized based on experimental data (NiO JCPDS No. 47-1049, RuO2 JCPDS No. 40-1290). Based on the optimized structural parameters, we construct periodic surface slabs with six to eight Co/Ni or Ru layers separated by at least 16 Å of vacuum for Ni<sub>2</sub>CoO<sub>4</sub> (001), NiO (100) and RuO<sub>2</sub> (110). Atomic positions within the top three layers of the slabs are allowed to relax in OH\*, O\* and OOH\* binding energy calculations. All calculations are done in Γ-centered Monkhorst-Pack k-point meshes with a reciprocal-space resolution of 0.15 Å<sup>-1</sup>. The energy convergence is  $10^{-5}$ eV and the force convergence 0.01 eV/Å.

As is known, catalytic activity of the material is determined by the binding energies of the reaction intermediates to the active sites of the catalyst. In the oxygen evolution reaction, OH\*, O\* and OOH\* intermediates are involved. To estimate the adsorption free energies  $\Delta G$  of different intermediate at zero potential and pH = 0, we calculate the binding energies  $\Delta E$  of each individual intermediate and corrected them with zero point energy (ZPE) and entropy (TS) using  $\Delta G = \Delta E + \Delta ZPE - T\Delta S$ .<sup>[5]</sup> Here, we use the computational hydrogen electrode (CHE) model, which exploits that the chemical potential of a proton-electron pair is equal to gas-phase H<sub>2</sub> at standard conditions. As the ground state of the O<sub>2</sub> is poorly described in DFT calculations we use gas phase  $H_2O$  and  $H_2$  as reference states as they are readily treated in the DFT calculations. The entropy for  $H_2O$  is calculated at 0.035 bar which is the equilibrium pressure of  $H_2O$  at 300 K. The free energy of this state is therefore equal to that of liquid water.<sup>[5]</sup>

The theoretical overpotential  $\eta$  is defined as the difference between the limiting potential and equilibrium potential. In the oxygen evolution reaction, the limiting potential is related to the highest free energy step  $\Delta G^{OER} = Max[(\Delta G_{O*}-\Delta G_{OH*}), (\Delta G_{OOH*}-\Delta G_{O*})]$ , and the equilibrium potential is 1.23 V. Thus, we get  $\eta = (\Delta G^{OER}/e)$  –1.23 V.<sup>[6]</sup>



Fig. S1 XRD patterns of (a)  $Ni_xCo_{3-x}O_4/NF$  and (b) NiO/NF (peaks at 44°, 52° and 76° ascribed to Ni substrate).



Fig. S2 TEM image of  $Ni_xCo_{3-x}O_4/NF$ .



Fig. S3 (a) and (b) FESEM images of NiO/NF.



Fig. S4 (a) TEM and (b) HRTEM images of NiO/NF.



Fig. S5 SEM-EDX (a) elemental mapping and (b) spectrum of  $Ni_xCo_{3-x}O_4$ . (Trace amount of K element was from the residual potassium cations embedded at the interstitial sites of the frameworks.)



Fig. S6 Nitrogen adsorption-desorption isotherms of  $Ni_xCo_{3-x}O_4/NF$  (inset corresponding to pore size distribution).



Fig. S7 Low-magnified FESEM images of (a) bare NF and (b)  $Ni_xCo_{3-x}O_4/NF$ .



**Fig. S8** (a) Full XPS spectrum of NiO and high-resolution spectra of (b) Ni 2p and (c) O 1s.



**Fig. S9** CV curves of the (a)  $Ni_xCo_{3-x}O_4/NF$ , (b) NiO/NF, (c) Ni-Co PBA/NF, (d)  $Ni(OH)_2/NF$ , (e) RuO<sub>2</sub>, (f) NF and (g)  $Ni_xCo_{3-x}O_4$  P measured in 1.0 M KOH solution at scan rates from 20 to 80 mV s<sup>-1</sup>.



Fig. S10 FESEM images of (a) Ni-Co PBA P and (b)  $Ni_xCo_{3-x}O_4$  P.



Fig. S11 (a-c) FESEM images and (d) TEM image of  $Ni_xCo_{3-x}O_4/NF$  after 24 h electrocatalysis.



Fig. S12 High-resolution spectra of (a) Co 2p, (b) O 1s and (c) Ni 2p for  $Ni_xCo_{3-x}O_4/NF$  before and after 24 h electrocatalysis.

Catalyst	Precursor	Structure	η <sup>b</sup> at 10 mA cm <sup>-2</sup> (mV)	Tafel Slope (mV dec <sup>-1</sup> )	Ref.
Ni <sub>x</sub> Co <sub>3-x</sub> O <sub>4</sub> /NF	Ni-Co PBA arrays on nickel foam (NF)	Hierarchical nanoplate arrays composed of porous and hollow nanocubes on NF	287	88	This work
Ni-Co-Fe PBA		Hierarchical hollow nanocuboids	320	49	[7]
NiO/NiCo <sub>2</sub> O <sub>4</sub>	Ni-Co PBA nanocubes	Nanocages consisting of pyramidal walls	380	50	[8]
CoS <sub>4.6</sub> O <sub>0.6</sub>	Co-Fe PBA nanocubes	Amorphous porous nanocubes	290	67	[9]
Ni <sub>5</sub> P <sub>4</sub> /Ni <sub>2</sub> P	Ni-Ni PBA nanoplates	Porous carbon coated nanoplates	300	64	[10]
Co <sub>3</sub> O <sub>4</sub> /Co-Fe oxide	ZIF-67/Co-Fe PBA yolk–shell nanocubes	Double-shelled nanoboxes	297	61	[11]
Co <sub>3</sub> S <sub>4</sub> @MoS <sub>2</sub>	Co-Fe PBA nanocubes	Hollow core-shell cubic heterostructure	280	43	[12]
NiFeSe@NiSe  O/CC	Ni-Fe PBA@ Ni <sub>2</sub> CO <sub>3</sub> (OH) <sub>2</sub> nanosheet arrays on carbon cloth (CC)	Porous and interconnected heterostructures elaborated with defects on CC	270	63.2	[13]

**Table S1**. OER activity<sup>a</sup> of some reported electrocatalysts based on PBAs.

(a) The electrolyte is 1.0 M KOH expect for **Ref. [7]** (0.1 M KOH).

(b)  $\eta$  is overpotential.

Electrocatalyst	$\Delta G_{OH} (eV)$	$\Delta G_{O}(eV)$	$\Delta G_{OOH} (eV)$	$\Delta G^{OER} (eV)$	η (V)
$Ni_{x}Co_{3-x}O_{4}(001)$	1.05	2.84	3.87	1.79	0.56
$RuO_{2}(110)$	0.49	1.45	3.33	1.88	0.65
NiO (100)	0.22	1.52	3.42	1.90	0.67

**Table S2**. Free energies of adsorption for OH\*, O\* and OOH\*,  $\Delta G^{OER}$  and  $\eta$  for oxygen evolution reaction over Ni<sub>x</sub>Co<sub>3-x</sub>O<sub>4</sub> (001), RuO<sub>2</sub> (110) and NiO (100).

## References

- 1 P. E. Blöchl, *Phys. Rev. B*, 1994, **50**, 17953–17979.
- 2 G. Kresse, J. Furthmüller, *Phys. Rev. B*, 1996, **54**, 11169–11186.
- 3 J. Wellendorff, K. T. Lundgaard, A. Møgelhøj, V. Petzold, D. D. Landis, J. K. Nørskov, T. Bligaard, K. W. Jacobsen, *Phys. Rev. B*, 2012, 85, 235149.
- 4 S. -H. Wei, L. G. Ferreira, J. E. Bernard, A. Zunger, *Phys. Rev. B*, 1990, **42**, 9622–9649.
- 5 J. K. Norskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard, H. Jonsson, J. Phys. Chem. B, 2004, 108, 17886-17892.
- 6 M. García-Mota, A. Vojvodic, H. Metiu, I. C. Man, H. Y. Su, J. Rossmeisl, J. K. Nørskov, *ChemCatChem*, 2011, 3, 1607-1611.
- W. Ahn, M. G. Park, D. U. Lee, M. H. Seo, G. Jiang, Z. P. Cano, F. M. Hassan,
   Z. Chen, *Adv. Funct. Mater.*, 2018, 28, 1802129.
- 8 L. Han, X. -Y. Yu, X. W. Lou, Adv. Mater., 2016, 28, 4601-4605.
- 9 P. Cai, J. Huang, J. Chen, Z. Wen, Angew. Chem., 2017, 129, 4936-4939.
- 10 X.-Y. Yu, Y. Feng, B. Guan, X. W. Lou, U. Paik, *Energy Environ. Sci.*, 2016, 9, 1246-1250.
- X. Wang, L. Yu, B. Y. Guan, S. Song, X. W. Lou, Adv. Mater., 2018, 30, 1801211.
- Y. Guo, J. Tang, Z. Wang, Y.-M. Kang, Y. Bando, Y. Yamauchi, *Nano Energy*, 2018, 47, 494-502.
- 13 G. Yilmaz, C. F. Tan, Y.-F. Lim, G. W. Ho, Adv. Energy Mater., 2018, 1802983.