Supporting Information

for

Superior Li-ion storage of VS₄ nanowire anchored on reduced graphene

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Fig. S1 Schematic illustration for the fabrication of VS₄ anchored on reduced graphene.



Fig. S2 SEM images and the corresponding EDS elemental mapping of (a, a1, a2, a3) V, S and V+S for VS₄; and V, S, and C for (b, b1, b2, and b3) VS₄@5rGO; (c, c1, c2, and c3) VS₄@15rGO, and (d, d1, d2, and d3) VS₄@35rGO.

Peaks	Band assignment	Reference
1631	δ(OH)	1
1400	Amorphous carbon	1
1200	C-O-R	2
980	S ²⁻ (V-S-V)	2-4
550	Terminal S stretching	2-4

Table S1 FTIR peaks and their assignments.



Fig. S3 The Nitrogen-adsorption-desorption isotherms of the (a) VS₄, (b) VS₄@5rGO, (c) VS₄@15rGO, and (d) VS₄@35rGO. The insets are their corresponding pore size distribution.



Fig. S4 First three CVs of (a) VS₄, (b) VS₄@5rGO, (c) VS₄@15rGO, and (d) VS₄@35rGO and (e) rGO as well as the (f) EIS before cycle and after 3 cycles of CV scanning with a scan rate of 0.2 mV s⁻¹ in the voltage range of 0.01-3.00 V vs Li/Li⁺.



Fig. S5 First four galvanostatic discharge-charge profiles at a current rate of 0.1 A g⁻¹ in the voltage range of 0.01-3 V vs Li/Li⁺ for: (a) VS₄(a) VS₄(a) 5rGO; (c) VS₄(a) 15rGO; (d) VS₄(a) 35rGO.



Fig. S6 VS₄@15rGO after 100 cycles at a current of 0.1 A g-1 (a3) SEM image in LED mode; (b3) SEM image in TED mode; (c3) TEM image; (d3) HRTEM image showing the d-spacing of 0.153 nm and 0.205 nm in the (352) and (042) plane, respectively (inset: FFT image).



Fig. S7 CV curves of (a) VS₄; (b) VS₄@5rGO; (c) VS₄@15rGO; (d) VS₄@35rGO at different scan rates between 0.01 to 3.00 V;



Fig. S8 (a, a'), (b, b'), and (c, c') are the plots of $v^{1/2}$ vs $i/v^{1/2}$ for VS₄@5rGO, VS₄@15rGO, and VS₄@35rGO at both anodic and cathodic scan. The plots are used for calculating constants k₁ and k₂ at different potentials of the cathodic scan and anodic scan.

The total current at a fixed potential can be described using the following equation: $I(V) = k_1 v + k_2 v^{1/2}$, where $k_1 v$ and $k_2 v^{1/2}$ represent the capacitive contribution and intercalation contribution, respectively. The k_1 and k_2 can be obtained from the slope and y-axis intercept of the straight line: $i(V)/v^{1/2} = k_1 v^{1/2} + k_2$, as shown in Fig. S6. Thus, the contribution from capacitive and intercalation can be calculated, and each component contribution at a scan rate of 0.3 mV/s is shown in Fig. 3f.

XPS survey scanning (Fig. S9) and fine scanning for C, V and S elements (Fig. S10) were conducted for pristine VS₄@15rGO, after discharged at 0.01 V and charged to 3 V. From the survey scan (Fig. S9), it is observed that the main peaks correspond to sulfur, vanadium, carbon, and oxygen in VS₄@15rGO.⁵ After being discharged to 0.01 V, fluorine peaks are apparent and oxygen peaks become stronger. This is reasonable because SEI forms during discharging, where some LiPF₆ electrolyte decompose when contact with electrode materials and form fluoride and oxide based by product.^{6, 7} Also, the vanadium peaks become very weak after discharging, this could be due to the formation of nanocrystal vanadium metal after discharging, which are embedded within the particle and too small difficult to detect. Even 5 mins Ar ion bombarding, the vanadium peaks are still weak. When VS₄@15rGO is further charged to 3 V. the vanadium peak looks more obvious.



Fig. S9 XPS survey spectrum of VS₄@15rGO at pristine, after discharged at 0.01 V and charged to 3 V.

To analyze the process of redox reaction, the C 1s, scanning of V 2p and S 2p spectrum of VS₄@15rGO at pristine, after being discharged to 0.01 V and charged to 3 V (Fig. S10) are carried out by XPS. For the pristine VS₄@15rGO, the C 1s core level spectrum can be deconvoluted into three components at 284.8 eV, 287.1 eV and 289.0 eV (Fig. S10a),

corresponding to the C–C, C–O and O–C=O groups, respectively.^{8,9} It should be noted that the intensity of C-C is higher compared to that of C-O and O-C=O in pristine sample because of C-C bond in rGO. Comparing with the samples at 0.01 V and 3 V (Fig. S10b and c), the intensities of C–O and O–C=O of pristine is lower, which is because no super p amorphous carbon is added in pristine VS₄@15rGO and rGO has lower content of C–O and O–C=O than super p amorphous carbon. Three oxidation states of vanadium species are seen after peak separation of the V 2p core level spectrum, as depicted in Fig. S10a1. The two peaks centered at 517.3 eV and 524.3 eV are correlated with the 2p3/2 and 2p1/2 components of V⁴⁺. Also, the vanadium 2p1/2 peaks located at 522.9 eV with relatively low intensity can be attributed to V^{3+} species, arising from the partial reduction of the vanadium source to V^{3+} in solvothermal reaction.¹⁰ Meanwhile, the two strong peaks at 514 eV and 521.5 eV are ascribed to the 2p3/2 and 2p1/2 components of the V–C bond, similar to previous study.¹⁰ Noticeably, the intensity of the V–C component is even higher than that of V^{4+} , which could be associated with the strong bonding between VS_4 and rGO substrate. With the sample being discharged to 0.01 V, vanadium could be reduced to metal. However, instead of vanadium metal, only V³⁺ is detected by XPS. This could be resulted from the oxidization of vanadium metal nanocrystal with trace of oxygen (Fig. S10b1). After the electrode being charged back to 3 V, V⁴⁺ is detected, which means some VS₄ is reformed (Fig. S10c1). For the sulfur element in pristine VS₄@15rGO (Fig. S10a2), S 2p core level analysis shows two peaks at 162.9 eV and 164.1 eV, in good agreement with sulfur 2p3/2 and 2p1/2 of the S_2^{2-} dimer in VS₄. When being discharged to 0.01 V (Fig. S10b2), the peaks of 159.8 and 161.7 eV appear and are attributed to the S 2p doublet in Li_2S . S_2^{2-} are reformed after charged to 3 V (Fig. S10c2).¹¹



Fig. S10 The C 1s, V 2p and S 2p spectrum of $VS_4@15rGO$ (a, a1, a2) at pristine, after discharged at (b, b1, b2) 0.01 V and charged to (c, c1, c2) 3 V.

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