Ultrahigh electron mobility induced by strain engineering in directly semiconducting monolayer Bi₂TeSe₂ Supplementary

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I. 2D STRUCTURE PREDICTION USING THE PARTICLE SWARM OPTIMIZATION METHOD

We use the particle swarm optimization (PSO) method within the evolutionary algorithm implemented in the Crystal structure AnaLYsis by Particle Swarm Optimization (CALYPSO) code^{1,2}, to conduct the global structure search for the global energy-minimum structure of the stoichiometric monolayer Bi₂TeSe₂. The unit cell containing two Bi atoms, one Te atom and two Se atoms is considered, and three sublayers with a vacuum gap of 30Å are set as the parameters for 2D structure prediction. In the first step, 20 random structures with certain crystal symmetry are constructed, in which the atomic coordinates are generated by the crystallographic symmetry. Then, we use VASP code to carefully optimize the random structures until the Gibbs free energy changes are smaller than 1×10^{-5} eV. After finishing the structural optimization of all 20 random structures, 60% of them with lower Gibbs free energies are selected to partially construct the next-generation of PSO. The rest 40% new structures of the next-generation PSO are generated based on the so-called structure fingerprinting technique of bond characterization matrix to avoid identical structures². In the case of monolayer Bi₂TeSe₂, the structure searching simulations stop at 10 generations and 200 structures are generated and under investigations. The information about the ten structures with lowest formation enthalpies is listed as Table S1.

TABLE S1: Ten structures with lowest formation enthalpies predicted by the PSO method.

No.	1	2	3	4	5	6	7	8	9	10
Symmetry(Group)	Pmm2(25)	P-3m1(164)	R3m(160)	P-3m1(164)	R3m(160)	R3m(160)	Fm2m(42)	P3m1(156)	P1(1)	P1(1)
Enthalpy (eV)	-4.32352	-3.88476	-3.88332	-3.88113	-3.88065	-3.87994	-3.87106	-3.83183	-3.74489	-3.73646

The ten structures with lowest formation enthalpies are shown in Table S1. The structure with the symmetry of Pmm2 as shown in Fig. S1(a) and the most lowest formation enthalpy predicted by CALYPSO as shown in Table S1, composes of three isolated monolayers with large distances between each other, so it is invalid and we neglect this structure. The second structure possesses the crystal symmetry of P-3m1, and it is shown in Fig. S1(b). By comparison, we find that, the second structure shown in Fig. S1(b) is the monolayer Bi_2TeSe_2 we predicted by the atomic transmutation method in the present work.



FIG. S1: Two predicted structures with lowest formation enthalpies, corresponding to (a) No. 1 and (b) No. 2 in Table S1.

II. VDW CORRECTIONS TO THE BAND STRUCTURES, OPTICAL PROPERTIES AND CAR-RIER MOBILITIES



FIG. S2: Electronic band structures of monolayer Bi_2TeSe_2 calculated at the (a) PBE (b) PBE+SOC levels with/without vdW interaction.



FIG. S3: Calculations of (a) dielectric functions for in-plane polarization ($E\perp c$) and out-of-plane polarization (E//c), (b) absorption coefficients, (c) reflectivities of monolayer Bi₂TeSe₂ with HSE+vdW method, respectively.

Bi ₂ TeSe ₂	Strain	Direction	$C^{2D}(N/m)$	$D_l(e)$	$D_l(h)$	$m_e^*(m_0)$	$m_h^*(m_0)$	$\mu_e(\text{cm}^2\text{V}^{-1}\text{s}^{-1})$	$\mu_h(\mathrm{cm}^2\mathrm{V}^{-1}\mathrm{s}^{-1})$
	-	x	59.94	7.17	7.75	0.0812	0.1037	2510	1317
w- vdW	-	x	61.22	7.32	8.28	0.0871	0.1029	2133	1198
	-	У	54.65	7.32	8.33	0.0812	0.1037	2195	1039
w- vdW	-	У	56.03	7.73	8.27	0.0871	0.1029	1930	1099
	-6%	x	87.23	7.13	11.61	0.0338	0.0332	21353	8316
w- vdW	-6%	x	83.54	7.15	11.54	0.0335	0.0330	20678	8181

TABLE S2: Calculated elastic modulus (C^{2D}), deformation potential constant (D_l), effective mass (m_0), and mobility (μ) in x and y directions of monolayer Bi₂TeSe₂ with/without vdW at 300 K.

III. ANLYSIS OF THE BAND STRUCTURES FOR MONOLAYER BI2 TESE2

We employ the QUANTUM EXPRESSO code with both fully relativistic and scalar relativistic pseudopotentials to analyze the band structures for monolayer Bi₂TeSe₂ with and without SOC effects^{3,4}, as shown in Fig. S4. The irreducible representations (irreps) for energy levels near Fermi level at Γ point are obtained by the QUANTUM EXPRESSO code as well and shown in Fig. S4. The small group for Γ point is D_{3d} . The irreps for maximum of valence band and minimum of conduction band without SOC effects at Γ point are two-dimensional (2D) Γ_3^- with odd parity and one-dimensional (1D) Γ_1^+ with even parity, as shown in Fig. S4(a). The SOC effect splits the 2D Γ_3^- into degenerate 1D Γ_6^- and 1D Γ_5^- , and 2D Γ_4^- in the double-group representations, as shown in Fig. S4(b). The SOC effect also transfers the 1D Γ_1^+ into 2D Γ_4^+ . Therefore, the irreps for VBM and CBM in the double-group representations are degenerate $\Gamma_5^- \& \Gamma_6^- and \Gamma_4^+$ respectively, as shown in Fig. S4(b). Since the order of parities does not change when considering SOC effects, no band inversion takes place at Γ point for monolayer Bi₂TeSe₂ by using the state-of-art software SymTopo⁵, and the calculated symmetry-based indicator is (0,0,0,0), which confirms that monolayer Bi₂TeSe₂ is a trivial insulator.



FIG. S4: Band structures for monolayer Bi₂TeSe₂ (a) without and (b) with SOC effects.



FIG. S5: Dependence of the bandgaps of Bi₂Se₃, Bi₂TeSe₂, Bi₂Te₂Se, Bi₂Se₃ on SOC strength.

IV. SOC EFFECTS ON BANDGAPS AND MOBILITIES

Since Te is below Se in the periodic table, the SOC effect of Te atom is stronger than Se atoms. As shown in Table 1, the calculated SOC strength for Te atom is 140 meV smaller than Se atom, which means that the SOC effect of heavier Te atoms is stronger than Se atoms. To further investigate influences of the SOC effects on band structures and carrier mobilities, we change the ratios of Te and Se in monolayer Bi_2TeSe_2 with fixed crystal structures by replacing Se atoms with Te atoms, and monolayers of Bi_2Se_3 , Bi_2TeSe_2 , Bi_2Te_2Se and Bi_2Se_3 with identical crystal structures are generated. By using first-principle method, the calculated bandgaps for the four monolayers are shown in Fig. S5, which shows that, when the Te concentration increases, the value of bandgap decreases from 1.08 eV to 0.29 eV. The decreased bandgap is resulted from the stronger SOC strength of Te atoms, considering similar chemical properties of Te and Se atoms and identical crystal structures. According to the simple two-band kp model shown as Eq. (12), the value of m^* increases when the value of bandgap increases. Thus, when the value of bandgap decreases the value of carrier mobilities, according to the deformation-potential theory shown as Eq. (7).

V. CLEAVAGE ENERGY FOR BI₂TE₂SE

Fig. S6 shows the resulting cleavage energy by determining the total energy difference (per unit cell) with respect to the ground state of the fractured structure with four quintuple layers (QLs) slab as a function of distance. And the calculated cleavage energy of Bi_2TeSe_2 is 0.37 J/m², which confirms that monolayer Bi_2TeSe_2 could be feasible to be experimentally exfoliated from its bulk single crystals considering the weak vdW interactions between adjacent QLs.



FIG. S6: Cleavage energy as a function of the distance between a top QL and the bottom three QLs of Bi₂TeSe₂. Inserted is the schematic view of the cleavage process of monolayer from the four-QLs slab of Bi₂TeSe₂.

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- ¹ Y. Wang, J. Lv, L. Zhu and Y. Ma, *Phys. Rev. B*, 2010, **82**, 094116.
- ² H. Wang, Y. Wang, J. Lv, Q. Li, L. Zhang and Y. Ma, *Comp. Mater. Sci.*, 2016, **112**, 406 415.
- ³ P. Giannozzi, S. Baroni, N. Bonini, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, G. L. Chiarotti, M. Cococcioni, I. Dabo, A. D. Corso, S. de Gironcoli, S. Fabris, G. Fratesi, R. Gebauer, U. Gerstmann, C. Gougoussis, A. Kokalj, M. Lazzeri, L. Martin-Samos, N. Marzari, F. Mauri, R. Mazzarello, S. Paolini, A. Pasquarello, L. Paulatto, C. Sbraccia, S. Scandolo, G. Sclauzero, A. P. Seitsonen, A. Smogunov,
 - P. Umari and R. M. Wentzcovitch, Journal of Physics: Condensed Matter, 2009, 21, 395502.

- ⁴ P. Giannozzi, O. Andreussi, T. Brumme, O. Bunau, M. B. Nardelli, M. Calandra, R. Car, C. Cavazzoni, D. Ceresoli, M. Cococcioni, N. Colonna, I. Carnimeo, A. D. Corso, S. de Gironcoli, P. Delugas, R. A. DiStasio, A. Ferretti, A. Floris, G. Fratesi, G. Fugallo, R. Gebauer, U. Gerstmann, F. Giustino, T. Gorni, J. Jia, M. Kawamura, H.-Y. Ko, A. Kokalj, E. Küçükbenli, M. Lazzeri, M. Marsili, N. Marzari, F. Mauri, N. L. Nguyen, H.-V. Nguyen, A. O. de-la Roza, L. Paulatto, S. Poncé, D. Rocca, R. Sabatini, B. Santra, M. Schlipf, A. P. Seitsonen, A. Smogunov, I. Timrov, T. Thonhauser, P. Umari, N. Vast, X. Wu and S. Baroni, *Journal of Physics: Condensed Matter*, 2017, **29**, 465901.
- ⁵ T. Z. H. H. C. F. Z. J. Yuqing He, Yi Jiang, *Chinese Phys. B*, 2019, **28**, 87102.