Electronic Supplementary Information

Diffusion controlled porous WO₃ thin film photoanodes for efficient solar-driven photoelectrochemical permanganic acid production

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Phase purity and crystal structure of the porous WO₃ thin film photoanode

The XRD pattern of WO₃ thin film prepared on the FTO substrate was shown in Fig. S1. The multiple diffraction peaks at around 22.5–25.0° were evaluated to be 3 peaks indicating a monoclinic $P2_1/c$ space group, and the 200 peak was very intense. All of the reflection positions were almost consistent with the reported values (JCPDS 20-1324¹). Based on this analysis, we concluded that the obtained WO₃ thin film can be assigned to be a pure γ -phase of WO₃ with the preferential 100 orientation.



Fig. S1: (a) XRD patterns of the WO₃ thin film photoanode and the γ -WO₃ powder. The F represents the peaks assigned to the FTO. (b) The diffraction pattern for the WO₃ thin film was fitted as 2 and 3 peaks at around 22.5–25.0°. The multiple peak deconvolution analysis revealed that the 3 peak fitting associated with the monoclinic space group was very reasonable.

PEC measurement setup and spectrum of illuminated solar simulator

The simulated sunlight at 1 Sun was illuminated from the backside of WO_3 film photoanode to avoid the excess absorption of illuminated light by the colored electrolytes contained Mn ions.





Determination of Mn⁷⁺ and S₂O₈²⁻ quantity in the solutions

The production of MnQ_4^{-1} ions (Mn^{7+}) was evaluated by colorimetry by using the UV-VIS spectroscopy. Firstly, the absorption spectra of reference solutions for Mn^{2+} ($MnSO_4$) and Mn^{7+} ($KMnO_4$) in the aqueous sulfuric acid were measured (Fig. S3), and we confirmed that the Mn^{2+} was transparent in the range of 400–800 nm, and the spectral shape was almost unchanged in each sulfuric acid concentration and the intensity was just varied by the Mn ion concentrations. By using the reference $KMnO_4$ spectra, the MnO_4^{-1} ions (Mn^{7+}) concentration was determined by the absorption peak intensity at 545 nm. The colorimetry for quantitative analysis of the MnO_4^{-1} ions (Mn^{7+}) concentration is a very reliable way in this study, since the other methods such as an observation of oxidation of reactants by the evolved Mn^{7+} would have a possibility that the reaction is affected by other high valence manganese ions. The production of $S_2O_8^{2-}$ was also evaluated by colorimetry. After the consumption of Mn^{7+} by the successive reaction with the Mn^{2+} residue, the obtained solution specimen was mixed with equivalent amount of 0.01 M FeSO₄ and 1 M H₂SO₄ solution. The optical absorbance spectra were measured after the stirring of mixed solutions, and the $S_2O_8^{2-}$ quantity was evaluated from the absorption at 310 nm.





Thickness and morphological effect of WO₃ photoanodes on the PEC property

We prepared WO₃ film photoanodes by several processes to check the film thickness and morphology effects on the PEC property. In the nanoparticle dispersion coating process,² the obtained films have 1–7 μ m of thickness and nanopores in the layer (Fig. S5a). Additionally, to vary the porosity we prepared samples by using dispersions with and without polyethylene glycol (PEG) 300 and by the inter-particle necking which was carried out by a 2-step coating (0.3 M tungsten phenoxide solution was deposited after the coating of nanoparticle dispersion). The solution deposition means the preparation process described in the main text. To vary the morphology, we also prepared by the samples without the addition of ethylcellulose, resulting in the dense films (Fig. S5b). The samples obtained by anodization process which was carried out by using W metal in H₃PO₄ electrolyte at 100 °C under 8 V of bias voltage, the dense oxide layer was found under the anodized layer. We compared the PEC properties for the obtained samples, and found a clear tendency: porous thick films gave the large photocurrent density while the $\eta(Mn^{7+})$ was reduced with the increasing thickness. On the contrary, the photocurrent density was not high in the thin films, however, the η (Mn⁷⁺) was enhanced. Less porosity of layer was favorable to obtain the $\eta(Mn^{7+})$, linking to the degradation of film surface by the successive reaction of Mn⁷⁺ with the Mn²⁺ residue described in the main text. On the basis of these results, we employed three WO₃ film photoanodes as the main and reference samples, which are expressed as the red and blue colored bars in Fig. S4. The detailed preparation procedures for pn-, dn-, pk-, pk2- and an-WO₃ photoanodes are summarized in Table S1.



Fig. S4: The preparation process and thickness dependences of η (Mn⁷⁺) for the WO₃ film photoanodes in 1.0 mM MnSO₄ and 1.0 M H₂SO₄ electrolyte at 1.3 V_{RHE} for 1 h. The wP, woP, wEC and woEC represent the solution/dispersion "with PEG300", "without PEG300", "with ethylcellulose" and "without ethylcellulose", respectively. The colored bars indicate the samples which are introduced in the main text as pn-, pk-, pk2-, an- and dn-WO₃ photoanodes.

	Table S1	Sample	preparation	conditions
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Photoanode	Process	Procedure
pn-WO ₃	Solution deposition	A volume of 2.0 mL of 0.3 M tungsten phenoxide in toluene was mixed with 0.05 g of ethylcellulose, and the resulting solution was spin coated at 1000 rpm for 15 s onto FTO glass substrate. The film produced was preheated at 500 °C for 5 min. The coating and preheating processes were repeated four times, and the preheated film was then fired at 550 °C for 60 min in air.
dn-WO ₃	Solution deposition	0.3 M tungsten phenoxide in toluene was spin coated at 1000 rpm for 15 s onto FTO glass substrate. The film produced was preheated at 500 °C for 5 min. The coating and preheating processes were repeated four timess, and the preheated film was then fired at 550 °C for 60 min in air.
pk-WO ₃	Nanoparticle deposition	WO_3 nanocrystal powder was ball milled in isopropanol for 6 h with a toluene solution of tungsten phenoxide. PEG300 was added to the WO_3 dispersion in a 1:1 volume ratio. The WO_3 dispersion was spin coated at 2000 rpm for 10 s onto FTO glass substrate. The film was preheated at 500 °C for 5 min. The coating and preheating processes were repeated 13 times, and the preheated film was then fired at 550 °C for 30 min in air.
pk2-WO ₃	Nanoparticle deposition	The same WO ₃ dispersion for the pk-WO ₃ was spin coated at 2000 rpm for 10 s onto FTO glass substrates. The film was preheated at 500 °C for 5 min. The coating and preheating processes were repeated 2 times. 0.3 M tungsten phenoxide solution was additionally spin coated at 1000 rpm for 15 s. The film was preheated at 500 °C for 5 min, and then was finally fired at 550 °C for 60 min in air.
an-WO $_3$	Anodization	The anodization process was carried out by using W metal in H_3PO_4 electrolyte at 100 °C under 8 V of bias voltage for 120 min. The obtained precursor was fired at 550 °C for 30 min in air.

Figure S5f shows the xrd patterns for the pn-, dn-, pk-, pk2- and an-WO₃ photoanodes. The crystal orientation property along the substrate normal was different from each other, however, the crystallites/crystal grains in the WO₃ nanostructures had contact with the electrolyte in random crystal face directions as shown in Figs. 7 and S5a-e. Moreover, the calculated crystallite size for the all photoanodes was 29–38 nm, indicating the very similar crystallinity. (The crystallite size was evaluated by using Scherrer's equation, $D = (K \cdot \lambda / \beta \cos \theta)$, where K is the Scherrer constant (K = 0.94), λ is the wavelength of the incident X-ray (λ = 0.15401 nm), β is the full width at half-maximum intensity of the 002 peak, and θ is the Bragg diffraction angle for the 002 reflection.) These results suggest that the crystal property among the tested photoanodes would not have a considerable effect on the PEC property.



Fig. S5: Cross-sectional views of FESEM images for reference WO₃ photoanodes; (a) the pk-WO₃ and (b) the dn-WO₃. The enlarged view of grain morphology for (c) the pk-WO₃, (d) the dn-WO₃ and (e) the pn-WO₃. (f) XRD patterns and (g) peak fittings for all WO₃ film photoanodes. The F represents the peaks assigned to the FTO substrate. (h) The calculated crystallite size for the WO₃ thin film photoanodes.

The PEC property of pk-WO₃ photoanodes

Figure S6 shows the illumination time dependence of the PEC reaction in the electrolyte containing 1.0 mM Mn²⁺ and 1.0 M H₂SO₄ using the pk-WO₃ photoanode at 1.30 V_{RHE}. As with the case of pn-WO₃ photoanodes (Fig. 3b), the chromatic change of the solution from colorless to pink during the illumination was observed (Fig. S6). The absorption peaks are also assigned to the MnO₄⁻ ions (Mn⁷⁺). The evolved Mn⁷⁺ for 30 min (4.8 C) at this PEC condition was 28 μ M, and the η (Mn⁷⁺) was 5.2%.



Fig. S6: Illumination-time dependence of absorption spectra for the pk-WO₃ photoanode in the electrolyte containing 1.0 mM Mn^{2+} and 1.0 M H₂SO₄ under simulated sunlight at 1.3 V_{RHE}.

The concentration dependences of Mn^{2+} and H_2SO_4 in the PEC reaction for the porous WO_3 thick film photoanode also showed similar behavior with the porous thin film photoanodes (Figs. S6–S8). The evolution of Mn^{7+} increased with the increasing H_2SO_4 concentration up to 1.0 M, and the Mn^{2+} concentration dependence had a maximum at 1.0 mM. Above 10 mM of $MnSO_4$, the broad absorption peak at 490 nm assigned to the Mn^{3+} ions was also observed.



Fig. S7: H_2SO_4 concentration dependence of the evolved Mn^{7+} for the pk-WO₃ photoanode in the electrolyte containing 1.0 mM Mn^{2+} under simulated sunlight at 1.3 V_{RHE} .



Fig. S8: $MnSO_4$ concentration dependence of absorption spectra for the pk- WO_3 photoanode in the electrolyte containing 1.0 M H_2SO_4 under simulated sunlight at 1.3 V_{RHE} .

The Mn^{2+} concentration dependence of *I-t* curves for the pk-WO₃ photoanode showed a gradual decrease of photocurrent (Fig. S9). Above 10 mM of Mn^{2+} , the photocurrent was very small from the initial state, and the thick brownish deposit was observed at the film surface.



Fig. S9: The Mn²⁺ concentration dependence of *I-t* curves for the pk-WO₃ photoanode in 1.0 M H₂SO₄ electrolyte at 1.3 V_{RHE}.

The EC property of FTO anodes

Figure S10 shows the LSV curve for the electrolyte containing 1.0 mM Mn^{2+} and 1.0 M H_2SO_4 (pH \approx 0.0) using the FTO glass anode under the dark condition. The onset potential of the photocurrent was about 1.8 V_{RHE} , and the J gradually increased up to 0.3 mA·cm⁻² at 2.5 V_{RHE} , followed by the rapid rise above 2.7 V_{RHE} .



Fig. S10: J-V curves for EC reaction in the FTO glass anode. The electrolyte was 1.0 mM Mn²⁺ and 1.0 M H₂SO₄.

Figure S11 shows the illumination time dependence of the PEC reaction in the electrolyte containing 1.0 mM Mn²⁺ and 1.0 M H₂SO₄ using the FTO glass anode at 2.8 V_{RHE}. The chromatic change of the solution from colorless to pink during the application of bias voltage was observed. The absorption peaks are assigned to the MnO₄⁻ ions (Mn⁷⁺) as well as the PEC reactions (Figs. 3b and S6). The evolved Mn⁷⁺ for 30 min (4.1 C) at this EC condition was 11 μ M, and the η (Mn⁷⁺) was 3.9%.



Fig. S11: Time dependence of absorption spectra for the FTO anode in the electrolyte containing 1.0 mM Mn^{2+} and 1.0 M H_2SO_4 under dark condition at 2.8 V_{RHE} .

Dye absorption/desorption procedure

For the evaluation of surface area, we employed dye absorption/desorption experiments. Ru-based N719 dye (cisbis (isothiocyanato) bis (2,2-bipyridyl 4,4-dicarboxylato) ruthenium (II) bis-tetrabutylammonium) was used. The obtained samples were immersed for 1 day at 40 °C in a mixture of acetonitrile and tert-butyl alcohol (1:1,v/v) containing a 300mM N719 dye (Solaronix SA, Switzerland). After dye-absorption, the samples were rinsed acetonitrile to remove non-chemisorbed dye. Dye desorption measurements were carried out by immersing the samples in 5ml of 10mM KOH for 1 h followed by the concentration measurements by UV-VIS spectrometer at 520 nm. To confirm the reliability of the dye absorption analysis, we also carried out the Brunauer-Emmett-Teller (BET) surface-area measurements by N₂ adsorption-desorption at 77 K using a BELSORP-minill instrument (BEL Japan, Inc.) for the crystallized WO₃ polycrystalline powders. They were obtained by a combustion method using the same solution/dispersion and firing temperature with the pn- and pk-WO₃ (the original solution/dispersion was dried at 250 °C, and then fired at 550 °C for 60 min in air). The samples were degassed under a N₂ flow at 423 K for 60 min prior to data collection. Both of the dye-absorption method and the BET analysis showed similar tendency for the amount of surface absorption, therefore, we concluded that the dye absorption method was available to evaluate relative surface area analysis for the WO₃ film photoanodes.

Table S2 Surface area analysis for the pn- and pk-WO3.				
Photoanode	Dye absorption / mM·cm ^{−3} (Film)	BET surface area / m ² ·g ⁻¹ (Powder sample)		
pn-WO ₃	75.3	27.5		
pk-WO ₃	20.5	13.3		

The PEC property of dn-, pk2 and an-WO₃ photoanodes

Figure S12 shows the illumination time dependence of the PEC reaction in the electrolyte containing 1.0 mM Mn^{2+} and 1.0 M H_2SO_4 using the dn-, pk2 and an-WO₃ photoanodes at 1.30 V_{RHE} . As with the case of pn- and pk-WO₃ photoanodes (Figs. 3b and S6), the chromatic change of the solution from colorless to pink during the illumination was observed. The absorption peaks are also assigned to the MnO_4^{-} ions (Mn^{7+}).



Fig. S12: Illumination-time dependence of absorption spectra for the dn-, pk2 and an-WO₃ photoanodes in the electrolyte containing 1.0 mM Mn^{2+} and 1.0 M H_2SO_4 under simulated sunlight at 1.3 V_{RHE} .

Effective diffusion coefficient (Dp)

The constrictivity δ and tortuosity τ for effective diffusion coefficient (D_p) can be expressed as shown in Fig. S13.



ABPE for the Mn⁷⁺ production



Oxidation experiments: Quantitative evaluation of acetophenone

The production of acetophenone from ethylbenzene using the photoelectrochemically obtained Mn⁷⁺ ions was evaluated by colorimetry by using the UV-VIS spectroscopy. The absorption spectra of reference solutions for the acetophenone in ethylbenzene were measured. Both of ethylbenzene and acetophenone had an intense absorption below 280 nm, and only acetophenone showed two peaks at 290 and 320 nm (Fig. S15). The reference spectra were deconvoluted, and we used the absorption peak at 290 nm for the concentration analysis. Figure S15b shows the fitting result of absorption peaks for the reference solution of 2.0 mM

acetophenone in ethylbenzene. By the deconvolution of the absorption curves, we made a calibration curve by using the obtained absorption intensity at 290 nm at every concentration of acetophenone reference solutions (Fig. S15c).



Fig. S15: (a) The absorption spectra for acetophenone in ethylbenzene and (b) the fitted spectral curve of reference 2.0 mM acetophenone solution. In the inset, three black and red lines represent the deconvoluted curves of raw data. (c) The peak intensity of deconvoluted curve at 290 nm as a function of acetophenone concentration in ethylbenzene.

References

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