Electronic Supplementary Information

B₃S Monolayer: Prediction of A High-Performance Anode Material for Lithium–Ion Battery

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Calculation Details:

Structure of the B₃S was generated and optimized using CALYPSO code and VASP respectively. 2 × 2 supercell structure is considered with 30 Å vacuum distance in C direction with 3 × 3 × 1 kpoint grid according to the Monkhorst-Pack scheme. The kinetic energy cut off is 700 eV. The energy and atomic forces convergence criteria are set at 10⁶ eV and 10³ eV/Å for geometry relaxation. *Ab initio* molecular dynamics (AIMD) simulations in NVT ensemble lasted for 5 ps. The force calculated in the Phonopy code using LDA potential. The in-plane Young's modulus and Poisson ratio along different θ direction relative to the positive x-direction can be expressed as

$$E(\theta) = \frac{C_{11}C_{22} - C_{12}^{2}}{C_{11}\sin^{4}\theta + C_{22}\cos^{4}\theta + \left(\frac{C_{11}C_{22} - C_{12}^{2}}{C_{44}} - 2C_{12}\right)\cos^{2}\theta\sin^{2}\theta}$$
(S1)
$$v(\theta) = \frac{\left(C_{11} + C_{22} - \frac{C_{11}C_{22} - C_{12}^{2}}{C_{44}}\right)\cos^{2}\theta\sin^{2}\theta - C_{12}\left(\cos^{4}\theta + \sin^{4}\theta\right)}{C_{11}\sin^{4}\theta + C_{22}\cos^{4}\theta + \left(\frac{C_{11}C_{22} - C_{12}^{2}}{C_{44}} - 2C_{12}\right)\cos^{2}\theta\sin^{2}\theta}$$
(S2)

The calculated $E(\theta)$ and $v(\theta)$ of B₃S monolayer are depicted in the polar diagrams in Fig. S2. From the diagram, it can be said that both Young's modulus and Poisson ratio are anisotropic in nature. Specific capacity is calculated by introducing the Li using CALYPSO code again. To get average open circuit voltage CASM code is used. And Li diffusion barrier computed using the climbing image nudged elastic band (CI-NEB) method interface with VASP.

To understand the environmental effect from other side of the monolayer on the diffusion, we have performed Li and vacancy diffusion in one side when other side is fully lithiated at low and high SOC. The Li diffusion path for the system with other side fully lithiated geometry is $A_1 \rightarrow A_2 \rightarrow$ $A_1 \rightarrow A_1$ (Fig. S5a). Here, Li at A_2 site is very much unstable due to the increment of the distance between the Li atom and monolayer from 1.72 to 1.76 Å at one side and both side lithiated geometry respectively. Therefore, the p_z orbital of the B atom, a responsible factor for the Li atom stability through the formation of non-covalent interaction with Li atom is at the far positon resulting the instability of Li atom at A₂ site. Therefore, no suitable path is observed in between A₁ and A₂ sites, but considerable barrier of 0.33 eV is present between two neighboring A₁ sites. And for high state of charge the diffusion of the vacancy is calculated. The single vacancy at A₂ site is most stable compare to the A₁ sites. Here, stability of the vacant neighboring A₁ sites is different due the puckered geometry in presence of Li at both side of the B₃S monolayer. For the easy understanding, here we have considered two neighboring A1 sites as A1a and A1b (Fig. S5b). If the vacancy is in A_{1a} then the Li atom at A_{1b} is close to the boron plane (1.63 Å) and form stable geometry. Again, the vacancy at A_{1b}, i.e. Li atom at A_{1a} is far away from boron plane (1.79 Å) due to puckered orientation is less stable. Here, the calculated vacancy diffusion barrier for the pathways $A_2 \rightarrow A_1 \rightarrow A_1 \rightarrow A_2$ are 0.53, 0.18, and 0.28 eV respectively.



Fig. S1. Optimized low-lying energy geometries of B₃S monolayer



Fig. S2. The in-plane polar diagrams for (a) Young's modulus, and (b) Poisson ratio along an arbitrary direction θ of the B₃S monolayer.



Fig. S3. Li adsorption energy obtained using PSO algorithm at different concentration of Li atom. Here x represents the Li concentration. The inset structures are of Li₀B₃S, Li₂B₃S, and Li₄B₃S (top and side view).



Fig. S4. Fluctuations of total potential energy of lithiated B₃S monolayer during the AIMD simulations at 300 K. The inset is the structure of lithiated B₃S monolayer at the end of the AIMD simulation (a) full lithiation at one side of the B₃S monolayer (b) full lithiation at both side of the B₃S monolayer.



Fig. S5. (a) Li diffusion energy barrier, where other side of the B₃S monolayer is fully lithiated.(b) Vacancy diffusion barrier, where other side of the B₃S monolayer is fully lithiated.

2D monolayer	Theoretical Capacity (mA h g ⁻¹)
B ₃ S	1662 (this work)
$1T-Ti_3C_2$	320 ¹
Graphite	372 ²
<i>Ψ</i> -Graphene	372 ³
2D-Boron	383 ⁴
Phosphorene	433 ⁵
Orthorhombic Mo ₂ B ₂	444 ⁶
2D-Ti ₂ B ₂	4567
VS_2	466 ⁸
B_2H_2	504 ⁹
1H-Mo ₂ C	526 ¹⁰
Phagraphene	558 11
2D-TiB ₄	588 12
Orthorhombic Fe ₂ B ₂	665 ⁶
MoS_2	670 ¹³
Mn ₂ C	879 14
V ₂ C	940 15
Silicene	954 ¹⁶
TiC ₃	1278 ¹⁷
BP	1283 ¹⁸
Popgraphene	1487 ¹⁹
B_2S	1498 ²⁰
g-SiC ₃	2090 ²¹

Table S1. Summary of theoretical specific capacities (mA h g⁻¹) of some widely investigated promising anode materials for LIBs

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