Electronic Supplementary Information (ESI) for

A High-Performance Highly-Bendable Quasi-Solid-State Zinc-Organic

Battery Enabled by Intelligent Proton-Self-Buffering Copolymer

Cathodes

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Preparation of carbon cloth-porous carbon (CC-PC) conductive scaffold: Polystyrene (PS) spheres as the template was prepared according to previous reports.¹ Typically, a 500-mL of two-necked flask was charged with 0.1 g of ammonium persulfate, 0.12 g of Na₂CO₃, 5 mL of styrene, 0.5 mL of acrylic acid and 100 mL of deionized water (DI) under inert N₂ protection. The reaction was kept at 70 °C for 7 h to generate white suspension, followed by the centrifugation at 8000 rpm for 5 mins and subsequent extensive washing with DI water. The obtained white solid was ultrasonically dispersed in 100 mL of DI water. Afterwards, a piece of carbon cloth was subjected to electrochemical treatment successively in 1 M H₂SO₄ for 3 h at a constant voltage of 2 V, followed by repeated rinsing with ethanol and DI water and vacuum drying at 60 °C. The treated carbon cloth was cut into pieces of 1 x 5 cm⁻² and placed vertically into a vial containing 9 mL of PS dispersion. Afterwards, the loaded vial was put in a temperature- and humidity-controlled chamber with the temperature of 65 °C and the relative humidity of 75 % for 2-3 days for generating the opal structure assembled on CC. The CC anchored with the opal structure was impregnated in an aqueous solution of cube sugar (0.06 M) and H_2SO_4 (0.025 M). After subjecting to heat-treatment at 100 °C for 6 h, the CC turned brown and was subsequently transferred to an empty glass bottle in an oven with the temperature at 150 °C for 6 h. Finally, the CC-PC was obtained by the calcination of the above CC at 650 °C for 2 h, and 800 °C for another 2 h in a N₂ steam.

Preparation of CC-PC@PANAC Cathode: The poly (aniline-co-Azure C) (PANAC) was conformally assembled onto the CC-PC scaffold by the electrochemical copolymerization using a CHI 760e electrochemical workstation with CC-PC as the working electrode, the Pt foil as the counter electrode and the saturated calomel electrode (SCE) as the reference electrode. Cyclic voltammetry (CV) was conducted in an aqueous solution consisting of 0.2 M aniline, 2.5 mM Azure C (AC) and 0.5 M H_2SO_4 in a potential range of -0.20 to 1.10 V at a scan rate of 60 mV s⁻¹ for 50 cycles. After the copolymerization, the PANAC films were grown on the CC-PC substrate. The obtained CC-PC@PANAC was rinsed with DI water and dried at 60 °C in

vacuum. The mass loading of PANAC on CC-PC can be regulated by varying polymerization time. The optimal mass loading of PANAC was 3.6 mg cm⁻² for achieving the best electroactivity.

Preparation of CC-PC@PANI Cathode: The poly (aniline) (PANI) was conformally assembled onto the CC-PC scaffold by the electrochemical copolymerization using a CHI 760e electrochemical workstation with CC-PC as the working electrode, the Pt foil as the counter electrode and SCE as the reference electrode. CV was conducted in an aqueous solution containing 0.2 M aniline and 0.5 M H_2SO_4 in a potential range of -0.20 to 1.10 V at a scan rate of 60 mV s⁻¹ for 50 cycles. The mass loading of PANI on CC-PC can be regulated by varying polymerization time.

Preparation of CC@Zn NP Anode: Zn electrodeposition was performed on a CHI 760e workstation with a piece of CC (1 cm \times 2 cm) as the working electrode, the Pt foil as the counter electrode and SCE as the reference electrode. Typically, cathode pulse (-40 mA cm⁻², 10 s on, 6 s off, 25 mins) deposition method was applied in 100 mL of aqueous solution containing 12.5 g of zinc sulfate (ZnSO₄ ·7H₂O), 12.5 g of sodium sulfate (Na₂SO₄) and 2 g of boric acid (H₃BO₃) at room temperature. The mass loading of CC@Zn NP can be readily adjusted by controlling the electrodeposition time. The optimal Zn mass loading on CC is 16.35 mg cm⁻².

Materials Characterization: The morphology and microstructure of samples were examined on scanning electron microscope (SEM; Hitachi, S-4800), scanning electron microscopy-energy dispersive X-ray spectroscopy (SEM-EDX) and transmission electron microscopy (TEM, FEI Tecnai G² F30). The crystal structures of samples were assessed on X-ray power diffractometer (XRD, Rigaku D/Max2550, Cu K α radiation) at room temperature. The Fourier transform infrared (FT-IR) spectra were recorded on a Thermal-Nicolet Nexus 670 spectrophotometer. UV-visible spectra of samples were carried out using an UV-2501PC spectrometer. The chemical state and compositions of samples were recorded on X-ray photoelectron

spectroscopy (XPS, ESCALab250).

Fabrication of Quasi-Solid-State Zn-PANAC Battery: The quasi-solid-state Zn-PANAC battery was assembled with the CC-PC@PANAC cathode, the NKK separator, (Nippon Kodoshi Corporation), the CC@Zn NP anode and the PVA/NH₄Cl-ZnCl₂ gel electrolyte. PVA/NH₄Cl-ZnCl₂ gel electrolyte was produced as follows: 3 M NH₄Cl, 2 M ZnCl₂ and PVA (8.5 g) was mixed in 100 mL of DI water, and the mixture solution was heated to 85 °C under vigorous stirring until it became clear. Two electrodes (1 cm²) were immersed in the gel electrolyte for 5 min and then dried until the gel solidification under ambient conditions. The mechanically robust quasi-solid-state Zn-conductive polymer cell was packed for subsequent tests.

Electrochemical Measurements: The electrochemical measurements including CV, glvanostatic charge-discharge (GCD) curves, and electrochemical impedance spectroscopy were conducted on a CHI 760E electrochemical workstation with the sample as the working electrode, the Pt foil as the counter electrode and SCE as the reference electrode. The electrochemical characterization of two-electrode aqueous Zn-PANAC batteries was carried out at room temperature in an aqueous solution containing 3 M NH₄Cl and 2 M ZnCl₂ (pH = 5). According to previous reports,² The specific capacity (C_m , mAh g⁻¹) was calculated according to the following formula:

$$C_m = \frac{\int_{0}^{\Delta t} I \times dt}{m} \tag{1}$$

where C_m is the specific capacity of the Zn-PANAC battery, Δt (h) is the discharge time, I (mA) is the discharge current during GCD and the m (g) is the mass of the active material of PANAC cathode.

The areal capacity (C_s , mAh cm⁻²) was calculated from the discharge curve using the following equations:

$$C_{s} = \frac{\int_{0}^{\Delta t} I \times dt}{s}$$
(2)

where I (mA) is the applied discharging current, Δt (h) is the discharging time and S (cm²) is the area of cell (1 cm²).

The volume capacity (C_v , mAh cm⁻³) was calculated from the discharge curve using the following equations:

$$C_V = \frac{\int_0^{\Delta t} I \times dt}{V}$$
(3)

where I (mA) is the applied discharging current, Δt (h) is the discharging time and V (cm³) is the volume of the single cathode (0.036 cm³) and the whole cell (0.083 cm³). The specific energy density E_m (Wh kg⁻¹), E_v (Wh cm⁻³) and specific power density P_m (kW kg⁻¹) of the Zn-PANAC batteries

$$E_m = C_m \times \Delta V \tag{4}$$
$$E_V = C_V \times \Delta V \tag{5}$$

$$P = \frac{C_m \times \Delta V}{1000 \times \Delta t} \tag{6}$$

Where the C_m and C_v are specific capacity and volume capacity, ΔV is the average discharge voltage and Δt (h) is the discharge time of the Zn-PANAC battery.



Fig. S1. SEM images of the CC@Zn NP electrode with different deposition time. (a) 1 min, (b) 5 min, (c) 10 min and (d) 30 min.



Fig. S2. (a) The high-resolution TEM image and (b) corresponding selected area electron diffraction of Zn nanoplates in CC@Zn NP anode. (c) XRD profile of the CC@Zn NP anode.



Fig. S3. (a, c) Low- and (b, d) high-magnification SEM images of PS-COOH spheres.



Fig. S4. (a) Nitrogen adsorption-desorption isotherms and (b) the corresponding pore size distribution curves of CC and PC.

We performed N_2 sorption measurements to investigate the texture property of CC-PC. As shown in Fig. S4, PC yields a specific surface area (SAA) of 948.9 m² g⁻¹, which is much larger than CC (1.4 m² g⁻¹). PC possesses predominant micropores and a few mesopores. In addition, there are plentiful periodic macropores with an average pore size of 300 nm in PC (see Fig. 1c in the main text). Therefore, for CC-PC composite, CC mainly serves as a flexible substrate while the coated PC functions as excellent conductive scaffold with hierarchical porous structure and high surface area which is beneficial to the ion diffusion and exposure of active sites.



Fig. S5. (a-b) SEM images of CC and individual carbon fiber. (c-d) SEM images of CC-PC@PANAC.



Fig. S6. SEM image and EDX spectrum of CC-PC@PANAC electrode.



Fig. S7. FT-IR spectra of PAC, PANI and PANAC.

The PANAC cathode was characterized by Fourier transform infrared spectra (FTIR) with the parent PANI as a reference (Fig. S7). Clearly, the FTIR plot of the PANAC copolymer is analogous to that of the parent PANI to some extent and contains multiple characteristic bands,³ e. g. the signals at 3224, 1584, 1494 cm⁻¹ corresponding to the N-H stretching vibration of secondary amine, the C=N stretching of quinoid (Q) ring and the C-N stretching of benzenoid (B) ring, respectively. Noteworthy, the peak shift from 3224 cm⁻¹ of PANI to 3203 cm⁻¹ of PANAC is observed, manifesting the copolymer growth through the amino groups.³ Compared to PANI, PANAC exhibits several new bands at 2900 (asymmetric C-H stretching of methyl groups in AC units), 2832 (symmetric C-H stretching of methyl groups), 1527 (N-H bending in secondary amine and C-N stretching in Q=N-B units), 1445 (C=N stretching of phenothiazine ring), 1162 cm⁻¹ (the C-H in-plane binding vibration of phenothiazine ring), respectively.³ Furthermore, another three new peaks at 1079, 1052 and 1003 cm⁻¹ (in-plane C-H bending) are attributed to the incorporation of phenothiazine ring into the copolymer chain, while five peaks at 6601000 cm⁻¹ are assigned to the out-of-plane C–H bending.⁴ In particularly, the new peak at 876 cm⁻¹ in PANAC represents the AC unit are ortho-coupled, because the AC monomer both exist at least one primary amino group and at least non-substituted ortho position which is similar to that of polythionine^{5,6} and poly(o-phenylenediamine)⁷. The above results confirm the successful copolymerization of aniline and AC. It is noted that the appearance of the 1120 cm⁻¹ band for PANAC comes from the sulfate dopant as counter ions to neutralize the positive charge,⁸ since the copolymerization of aniline and AC was performed in H₂SO₄. The FT-IR spectrum of PANAC is different from PAC (Fig. S7). From the above results, we can see that the shape of the FTIR spectrum of PANAC is not a simple superposition of the shape of the spectrum of PANAC is a copolymer and the aniline units dominant the chain of PANAC rather than the AC units.⁵⁻⁷



Fig. S8. XPS S 2p (a) and Cl 2p (b) spectra of CC-PC@PANAC cathode at the original, state.



Fig. S9. Schematic representation of the possible chemical structure of PANAC.



Fig. S10. CV curves of CC-PC@PANAC electrode at different aniline/AC ratios in aqueous Zn-PANAC rechargeable battery at the scan rate of 1 mV s⁻¹.



Fig. S11. CV curves of the CC-PC@PANAC electrode at different scan rates in aqueous Zn-PANAC rechargeable battery.



Fig. S12. (a) CV curves of the CC-PC@PANI cathode at different scan rates, comparative CV (b) of aqueous rechargeable zinc batteries using CC-PC@PANAC and CC-PC@PANI as the cathodes. (c-d) Galvanostatic charge-discharge curves of aqueous rechargeable zinc batteries using CC-PC@PANI (c) and CC@PANI cathode (d), respectively.



Fig. S13. Comparative CV curves of CC-PC and CC-PC@PANAC cathodes.



Fig. S14. Specific capacity as a function of current density of aqueous Zn–PAC batteries at pH=5.



Fig. S15. Experimental and fitting Nyquist plots of aqueous rechargeable zinc batteries with CC-PC@PANI and CC-PC@PANAC cathodes. The inset is equivalent circuit diagram for two kinds of batteries.

Fig. S15 exhibits the Nyquist plots of various electrodes. Apparently, the PANAC electrode shows similar series resistance but markedly lower charge transfer resistance ($R_{ct} = 0.8 \Omega$) with respect to PANI ($R_{ct} = 4.1 \Omega$), implying drastically-enhanced charge-transfer rate and much higher redox activity of PANAC.



Fig. S16. (a) The charge-discharge curves of Zn-PANAC batteries at a current density of 1 mA cm⁻². XPS N 1s spectrum of CC-PC@PANAC cathode at different charged and discharged states: (b), (c) and (d) correspond to point b, d and e in (a), respectively.



Fig. S17. XPS Cl 2p and S 2p spectra of CC-PC@PANAC cathode at the original, charged, and discharged states. Upon the discharge, the PANAC was reduced along with the significant reduction of the Cl⁻ dopant from 16.9 at % to 3.2 at %.



Fig. S18. (a) CV curves of CC-PC@PANAC electrode at different scan rates. (b) Current response at the cathode peak vs. the scan rate (y = 7.254x-0.04159, $R^2 = 0.9999$) of CC-PC@PANAC electrode. (c) CV curves of Pt plate@PANAC electrode at different scan rates. (d) Current response at the cathode peak vs. the square root of scan rate (y = 3.027x-0.509, $R^2 = 0.9994$) of Pt plate@PANAC electrode.

We recorded CV curves of two electrodes at different scan rates. As seen in Fig.

S18, the cathodic peak current for CC-PC@PANAC shows a linear increase with

increasing scan rate. In contrast, the peak current of Pt-plate@PANAC increases linearly with an increase in the square root of scan rate. The contributions of surface reactions at different voltages can be assessed on the basis of the equation of $i = av^b$, where b ranges from 0.5 to 1, representing the reactions from bulk to the surface and/or near-surface.⁹ CC-PC@PANAC gives a much higher b value (b = 1) than that of Pt-plate@PANAC (b = 0.5), indicating a dominant fraction of contributions from surface reactions during the charge-discharge process, leading to the rapid kinetics of cells.



Fig. S19. Plots of specific capacity of aqueous Zn-PANAC battery *versus* the different mass loading. It is noted that the optimal performance corresponds to a mass loading of \sim 3.6 mg cm⁻².



Fig. S20. Experimental and fitting Nyquist plots of aqueous and quasi-solid-state Zn-PANAC batteries. The inset is used equivalent circuit diagram for quasi-solid-state Zn-PANAC batteries. The used equivalent circuit diagram for aqueous Zn-PANAC batteries is presented in inset of Fig. S15.

The Nyquist plot of quasi-solid-state Zn-PANAC batteries consists of two semicircles in high and medium frequency regions respectively, and a straight slopping line at the low frequency region. The first semicircle at the high frequency region is probably derived from the interfacial contact resistance (R_{int}) between the gel electrolyte and solid electrodes, while CPE1 represent its related double layer capacitance.¹⁰⁻¹³ The second circle at the medium-frequency region represents the charge transfer resistance (R_{ct}) and its related double layer capacitance (CPE2).¹⁰⁻¹³ The straight slopping line is attributed to the diffusion process in electrodes.



Fig. S21. SEM images of CC@Zn NP anode: (a-b) before cycling test. (c-d) after cycling test.



Fig. S22. The XRD pattern of the CC@Zn NP anode before and after cycling test as well as CC@PVA/ZnCl₂-NH₄Cl

It is worthy to note that the cycle stability of our quasi-solid-state Zn-PANAC battery is tested in the PVA/ZnCl₂-NH₄Cl gel electrolyte. In order to maintain the structure of the Zn nanosheets of CC@Zn NP anode after cycling test, we directly measured the dried post-test electrode without any other treatment. As can be seen from Fig. S22, the XRD peaks for the CC@Zn NP anode prior to cycling tests are attributed to the Zn nanoplates. After cycling tests in gel electrolytes, the XRD peaks

of the Zn nanoplate still exist and the newly appeared XRD peaks come from the PVA/ZnCl₂-NH₄Cl electrolyte (Fig. S22). Of note, no other new XRD peaks are observed after cycling tests (Fig. S22), suggesting the absence of Zn-related compounds (e.g. ZnO et al) at the cycled CC@Zn NP anode. In addition, SEM images shown in Fig. S21 indicate that the Zn nanoplate still maintains sheet-like morphology. Similar with some previous reports,^{14, 15} the observed wrinkles could be mainly attributed to the coating of the PVA polymer, since the cycling stability test is performed in PVA/ZnCl₂-NH₄Cl gel electrolyte.

cumences			
Cathode	Electrolyte	Capacity (mAh g ⁻¹)	Reference
PANAC	2 M ZnCl ₂ , 3 M NH ₄ Cl	306.3	This work
	et nH 5	(1 mA cm^{-2})	
calf dapad	1 M 7 n S O	(1 mA cm) 194	Angon Cham Int Ed
sen-doped	$1 \text{ M} \Sigma \text{IISO}_4$		Angew. Chem. Int. Ed.
PANI		(0.2 A g^{-1})	2018, 130, 16597.
PANI	Zn(CF ₃ SO ₃) ₂	200	Adv. Funct. Mater. 2018 , 28, 1804975
PANI	1 M ZnCl ₂	162	ACS Nano
		(1 C)	2018 , <i>DOI</i> :
			10.1021/acsnano.8b027 44.
PANI with	1 M ZnSO ₄ , 0.3 M	183.28	J. Power Sources 2015,
4,4-	$(NH_4)_2SO_4$	(2.5 mA cm^{-2})	283, 125.
diaminobiph enyl	, <i>"_</i> .	× ,	
PANI	1 M NH ₄ Cl, 0.5 M	139.4-170.1	Electrochim. Acta 2009,
	ZnCl ₂	$(0.02 \text{ mA cm}^{-2})$	54, 2941.
self-doped	$1 \text{ M ZnCl}_2 0.5 \text{ M}$	146.4	J. Power Sources 2002
PANI	NH ₄ Cl	1.000	110 229
poly(aniline-	$2 \text{ M ZnCl}_2 3 \text{ M NH}_4\text{Cl}$	146 3	Electrochim Acta 2016
co-N-		(1 mA cm^{-2})	190 240
methylthioni		(1 mil em)	170, 240.
ne)			
DANI/graphit	$1 M Z_{p} C = 0.5 M$	142.2	I Dower Sources 2007
r Alvi/graphit	$\frac{1}{1} \text{ Im } \sum_{i=1}^{n} C_{12}^{i}, 0.3 \text{ Im}$	$(0.6 \text{ m} \text{ A} \text{ am}^{-2})$	J. I Ower Sources 2007,
	$\frac{1}{2} M \frac{1}{2} M \frac{1}$	(0.0 IIIA CIII -)	170,135. Sundhadia Mad 2009
PANI	$2 \text{ M ZnCl}_2, 3 \text{ M NH}_4\text{Cl}$	141.2	Synthetic Met. 2008,
1 ('1'		(0.5 mA cm^{-2})	<i>138, 242.</i>
poly(aniline-	$2 \text{ M ZnCl}_2, 3 \text{ M NH}_4 \text{Cl}$	137.5	J. Power Sources 2006,
co-m-		(0.5 mA cm^{-2})	161, 685.
aminophenol			
)			
PANI	0.2 M ZnCl ₂ , 0.50 M	130	J. Power Sources 2006,
	NH ₄ Cl with 0.33 M Na-	$(0.25 \text{ mA cm}^{-2})$	160, 1447.
	citrate		
PANI	0.5 M NH ₄ Cl, 0.2 M	109.5	Int. J. Electrochem. Sci.
	$ZnCl_2$	(0.05 A g ⁻¹)	2016 , 11, 8571.
poly(aniline-	2.5 M ZnCl ₂ , 3 M	103.0	Synthetic Met. 2014,
со-о-	NH ₄ Cl	(0.5 mA cm^{-2})	143, 269.

Table S1. Comparison of the specific capacity of our Zn-PANAC battery with

 selective recently-reported aqueous Zn batteries with various conductive polymer

 cathodes

aminophenol

Battery	Capacity (mAh g ⁻¹)	Energy density (Wh kg ⁻¹)	Cycle	Reference
Zn-PANAC	306.32	352.3	1100	This work
Zn-ion	300	250	1000	<i>Nat. Energy</i> 2016 , <i>1</i> , 16119.
Zn-PANI	184	*56	2000	Angew. Chem. Int. Ed. 2018 , 130, 16597
Zn-MnO ₂	210	/	100	Angew. Chem. Int. Ed. 2012 , 124, 957
Zn-PANI	200	/	3000	<i>Adv. Funct.</i> <i>Mater.</i> 2018 , 28, 1804975
Zn-PANI	162	/	100	ACS Nano 2018 , 12, 11838
Zn-V ₂ O ₅	372	*144	900	<i>Adv. Mater.</i> 2018 , <i>30</i> , 1703725
Zn-pyrene- 4,5,9,10-tetraone (PTO)	336	186.7	1000	Angew. Chem. Int. Ed. 2018 , 130, 11911.
Ni-Fe	126	141	800	<i>Nat. Commun.</i> 2012 3 917
Ni-Fe	118	*100.7	1000	Nano Lett.
Ni-Fe	81	/	100	2014, 14, 7180. Chem. Commun. 2011, 47, 12473
Ni-Zn	203	355.7	/	<i>Adv. Mater.</i>
Ni-Zn (Zn//Co ₃ O ₄ @Ni O)	182.6	215.51	500	2016 , 28, 8732. Inorg. Chem. Front. 2015 , 2, 184.
Ni-Zn (Zn//NiO)	155	228	500	J. Mater. Chem.
Ni-Zn (Zn//Ni ₃ S ₂)	148	/	100	ACS Appl. Mater. Interfaces 2015,

Table S2. The comparison of the electrochemical performance of our Zn-PANAC cellwith selective recently-reported aqueous batteries.

				7, 26396.
Zn-Co ₃ O ₄	162	241	2000	Adv. Mater.
				2016 , <i>28</i> , 4904.
Ni-Bi	113	92	80	J. Power
				Sources 2015,
				274, 1070.
Ni-Bi	57.2	85.8	90	Adv. Mater.
				2016 , <i>28</i> , 9188.
Li-ion	122.7	/	200	J. Power
$(Na_2V_6O_{16} \cdot 14H_2)$				Sources 2013,
O//LiMn ₂ O ₄)				227, 111.
TiO ₂ //Ni(OH) ₂	68.7	/	100	Energy Environ.
				Sci. 2010, 3,
				1732.

Note: *represents the energy density calculated based on the total mass of two electrodes and electrolytes.

Table S3.	The calculated	contents of	various N	species	of the	CC-PC@PANAC
cathode un	der different cha	rging and dis	scharging sta	ates in Fi	g. S16.	

camode under un	forein enarging a	and discharging st	ates in Fig. 510.		
Samula	Calculated contents of various N species (%)				
Sample	-N=	-NH-	-NH ⁺ -	-NH+=	
a	13.5	57.5	18.8	10.2	
b	14	52	21	13	
с	12.5	46.4	9.8	31.2	
d	12	51	16.1	20.9	
e	4	57	22	17	
f	2	91	7	0	

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