## **Supporting Information**

# Nitrogen-Doped Graphdiyne Nanowalls Stabilized Dendrite-Free Lithium Metal Anodes

Tao Jiang<sup>a+</sup>, Ke Chen<sup>b+\*</sup>, Jingjing Wang<sup>c+</sup>, Zhongli Hu<sup>a</sup>, Gulian Wang<sup>a</sup>, Xu-Dong Chen<sup>c\*</sup>, Pengfei Sun<sup>a</sup>, Qiaobao Zhang<sup>d</sup>, Chenglin Yan and Li Zhang<sup>a\*</sup>

<sup>a.</sup>College of Energy, Soochow Institute for Energy and Materials InnovationS, Key Laboratory of Advanced Carbon Materials and Wearable Energy Technologies of Jiangsu Province, Soochow University, Suzhou 215006, China. E-mail: zhangli81@suda.edu.cn

<sup>b.</sup>Institute of Micro/Nano Photonic Materials and Applications, School of Physics and Electronics, Henan University, Kaifeng 475004, China. E-mail: kchen@henu.edu.cn

<sup>c</sup> Institute for New Energy Materials and Low Carbon Technologies, School of Materials Science and Engineering, Tianjin University of Technology, Tianjin 300384, China. E-mail: chenxd@email.tjut.edu.cn

<sup>d.</sup>Department of Materials Science and Engineering, College of Materials, Xiamen University, Xiamen 361005, Fujian, China

<sup>+</sup>*These authors contributed equally to this work.* 

#### Supporting Information Notes

#### Supplementary note 1

Electrochemistry: The nucleation overpotential and Coulombic efficiency was performed on a LAND CT2001A battery testing system. For the Li plating/stripping cycling efficiency measurement, the batteries were first cycled in a potential range of 0-1 V at 50  $\mu$ A for five cycles to stablize the SEI and remove surface contaminations. 1 mAh·cm<sup>-2</sup> (or 3 mAh·cm<sup>-2</sup>, 5 mAh·cm<sup>-2</sup>) of Li was then deposited onto the working electrodes (GDY and N-GDY) and charged to 0.5 V at various current densities from 1 to 5 mA·cm<sup>-2</sup> for each cycle. The Coulombic efficiency was calculated based on the ratio of Li stripping to plating. Rate capability of N-GDY electrodes were measured at current densities ranging from 0.5 to 6 mA·cm<sup>-2</sup> for 1 h in both the stripping/plating processes of each cycle. The long-term cycling stability and voltage hysteresis were investigated through a symmetric Li/Li test, GDY and N-GDY hosts were firstly plated by Li metal with a pre-stored capacity of 2 mAh·cm<sup>-2</sup> (or 5 mAh·cm<sup>-2</sup>, 8 mAh·cm<sup>-2</sup>), then cycled at various current densities with a constant areal capacity of 1 mAh·cm<sup>-2</sup> (or 3 mAh·cm<sup>-2</sup>, 5 mAh·cm<sup>-2</sup>). Electrochemical impedance spectroscopy (EIS) was measured after different electrochemical cycles by applying an alternating voltage of 5 mV over the frequency ranging from 10<sup>-2</sup> to 10<sup>5</sup> Hz.

Full cells were constructed with a LiNi<sub>0.5</sub>Co<sub>0.2</sub>Mn<sub>0.3</sub>O<sub>2</sub> (NCM) cathode and a N-GDY@Li anode with a pre-stored capacity of of 5 mAh·cm<sup>-2</sup> (or pure Li metal). The NCM electrodes were punched into circle discs of *ca*. 0.33 cm<sup>2</sup> and the areal mass loading of NCM was ~3 mg cm<sup>-2</sup>. Cyclic voltammetry (CV) measurements were performed in a potential range of 2.8-4.3 V at a scan rate of 0.1 mV s<sup>-1</sup>. Galvanostatic charge-discharge cycling tests of N-GDY@Li||NCM and Li||NCM full cells were carried out in a potential range of 2.8-4.3 V at 0.2 C, corresponding to a current density of 0.096 mA·cm<sup>-2</sup>.

#### Supplementary note 2

**DFT simulations**: Density functional theory (DFT) simulations were performed using the Vienna Ab-initio Simulation Package (VASP) package. The generalized gradient approximation (GGA) with the Perdew-Burke-Ernzerhof (PBE) functional was employed to describe the electronic exchange and correlation effects. Uniform G-centered k-points meshes with a resolution of  $2\pi$ \*0.03 Å<sup>-1</sup> and Methfessel-Paxton electronic smearing were adopted for the integration in the Brillouin zone for geometric optimization. The simulation was run with a cutoff energy of 500 eV throughout the computations. These settings ensure convergence of the total energies to within 1 meV per atom. Structure relaxation proceeded until all forces on atoms are less than 1 meV Å<sup>-1</sup> and the total stress tensor is within 0.01 GPa of the target value. The adsorption energy ( $\Delta E_{ads}$ ) was estimated by the following formula:  $\Delta E(ads) = [E(surface+n*Li)-E(surface)-n*E(Li)]/n$ 

where n is the number of adsorbed Li atoms, E(surface+n\*Li) is the total energy of the surface adsorbed n Li atoms, E(surface) is the total energy of the Cu (111) or carbon-based surfaces, and E(Li) is the total energy of one Li atom.

## > Supporting Information Figures



Fig. S1. Schematic structure of a single-layer graphdiyne.



Fig. S2. Schematic of the synthetic procedure of the N-GDY host.



Fig. S3. Digital photographs of the pristine Cu foam, GDY host after the low-temperature wet-chemistry synthesis, and N-GDY host after the thermal nitridation process.



Fig. S4. SEM images of the pristine Cu foam at different magnifications.



Fig. S5. SEM images of the as-obtained GDY foam at different magnifications.



Fig. S6.  $N_2$  adsorption-desorption profiles of N-GDY powders that are stripped from the 3D Cu foam.



Fig. S7. HRTEM image of the as-obtained N-GDY nanowalls.



Fig. S8. Full-scan XPS spectra of GDY and N-GDY hosts displaying C, O and N signals.



Fig. S9. High-resolution C 1s spectrum of the GDY foam.







**Fig. S11.** The initial nucleation potentials and nucleation overpotentials ( $\mu_n$ ) of the Cu foam, Cu@DGY and N-DGY electrodes.



**Fig. S12.** Cross-section SEM images of a, b) the N-GDY and c, d) Cu foam substrates after depositing Li under two different conditions of 1 mAh·cm<sup>-2</sup>/1 mA·cm<sup>-2</sup> and 3 mAh·cm<sup>-2</sup>/3 mA·cm<sup>-2</sup> at the same 20<sup>th</sup> cycle.



**Fig. S13.** Schematic diagram of cycling half cells for measurement of the Li plating/ stripping Coulombic efficiency during long-term cycling.



**Fig. S14.** Schematic diagram of cycling symmetric cells for measurement of the cycling stability and voltage hysteresis during long-term cycling.



**Fig. S15.** SEM images of the GDY@Li and N-GDY@Li electrodes after the 25<sup>th</sup> Li stripping/plating (3 mA·cm<sup>-2</sup> and 3 mAh·cm<sup>-2</sup>) in their charged states with a residue of 2 mAh·cm<sup>-2</sup> Li.



Fig. S16. SEM images of the commercial NCM powders at different magnifications.

## > Supporting Information Tables

Table S1.	The	calculated	binding	energies	of	Li	atom	with	Cu,	VG,	un-doped	GDY	and
different functional groups in N-GDY.													

	E(surface+n*Li)	E(surface)	<i>E</i> (Li)	Δ <i>E</i> (ads)
Cu	-128.03115268	-125.56868228	-0.07691098	-2.38556
VG	-664.90602816	-661.65694673	-0.07691098	-3.1721705
GDY	-1245.5995159	-1243.1136841	-0.07691098	-2.4089208
Ν(α)	-1244.7034133	-1242.1785241	-0.07691098	-2.4479782
N( <i>6</i> )	-1244.2592898	-1241.8330269	-0.07691098	-2.3493519
Ν(γ)	-1244.3618429	-1241.1065557	-0.07691098	-3.1783762
Ν(δ)	-1243.2989976	-1239.6315907	-0.07691098	-3.5904959
Ν(ε)	-1239.9926123	-1238.3202938	-0.07691098	-1.5954075

**Table S2**. Comparison of the electrochemical performances between the N-GDY hybrid host and other porous Cu and carbon allotropes hosts previously reported.

3D porous matrix	Current density	Cycling capacity	Stable cycling	Ref.	
host	(mA·cm-2)	(mAh∙cm-2)	numbers		
	1	1	>720		
N-GDY host	5	1	>260	This	
	1	5	>30	work	
	3	3	>175		
3D Porous Cu	1	1	~300	1	
3D Cu	0.2	0.5	>120	2	
3D hollow tubular	1	2	>60	2	
carbon fiber				5	
3D porous Ni core-	3	1	>315	4	
shell					
Hollow carbon	1	1	>75	5	
nanospheres				5	
3DNG	0.5	1	~150	6	

Graphitized carbon fibers	1	1	>500	7
Nitrogen-doped graphene	1	1	>727	8
G-C <sub>3</sub> N <sub>4</sub>	1	1	>450	9
Unstacked	2	0.1	>800	
graphene				10
drum				
Cu@VG foam	1	1	>160	11
	3	3	>50	

#### **Reference for Table S2**

[1] Q. Li, S. Zhu and Y. Lu, Adv. Funct. Mater., 2017, 27, 1606422.

[2] C. P. Yang, Y. X. Yin, S. F. Zhang, N. W. Li and Y. G. Guo, Nat. Commun., 2015, 6, 8058.

[3] L. Liu, Y. X. Yin, J. Y. Li, N. W. Li, X. X. Zeng, H. Ye, Y. G. Guo and L. J. Wan, Joule, 2017, 1, 563-575.

[4] L. L. Lu, Y. Zhang, Z. Pan, H. B. Yao, F. Zhou and S. H. Yu, *Energy Storage Mater.*, 2017, 9, 31-38.

[5] G. Zheng, S. W. Lee, Z. Liang, H. W. Lee, K. Yan, H. Yao, H. Wang, W. Li, S. Chu and Y. Cui, *Nat. Nanotechnol.*, 2014, **9**, 618-623.

[6] R. Zhang, S. Wen, N. Wang, K. Qin, E. Liu, C. Shi and N. Zhao, *Adv. Energy Mater.*, 2018, **8**, 1800914.

[7] T. T. Zuo, X. W. Wu, C. P. Yang, Y.X. Yin, H. Ye, N. W. Li and Y. G. Guo, Adv. Mater., 2017, 29, 1700389.

[8] G. Huang, J. Han, F. Zhang, Z. Wang, H. Kashani, K. Watanabe and M. Chen, *Adv. Mater.*, 2019, **31**, 1805334.

[9] Z. Lu, Q. Liang, B. Wang, Y. Tao, Y. Zhao, W. Lv, D. Liu, C. Zhang, Z. Weng, J. Liang, H. Li and Q. H. Yang, *Adv. Energy Mater.*, 2019, **9**, 1803186.

[10] R. Zhang, X. B. Cheng, C. Z. Zhao, H. J. Peng, J. L. Shi, J. Q. Huang, J. Wang, F. Wei and Q. Zhang, Adv. Mater., 2016, 28, 2155-2162.

[11] Z. Hu, Z. Li, Z. Xia, T. Jiang, G. Wang, J. Sun, P. Sun, C. Yan and L. Zhang, *Energy Storage Mater.*, 2019, https://doi.org/10.1016/j.ensm.2018.12.020.