# **Supporting Information**

#### FeP<sub>3</sub> Monolayer as High-Efficiency Catalysts for Hydrogen

#### **Evolution Reaction**

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### **Computational Details**

The particle swarm optimization (PSO) method within the evolutionary algorithm as implemented in the Crystal structure AnaLYsis by Particle Swarm Optimization (CALYPSO) code<sup>1,2</sup> was employed to find the lowest energy structures of FeP<sub>x</sub> (x =1-4) monolayers. Unit cells containing 1, 2, and 4 formula units (f.u.), including buckled and planar structures, were considered. In the first step, random structures with certain symmetry are constructed in which atomic coordinates are generated by the crystallographic symmetry operations. Local optimizations using the VASP code<sup>3</sup> were done with the conjugate gradients method and stopped when Gibbs free energy changes became smaller than  $1 \times 10^{-6}$  eV per cell. After processing the first generation structures, 60% of them with lower enthalpies are selected to construct the next generation structures by Particle Swarm Optimization (PSO). 40% of the structures in the new generation are randomly generated. A structure fingerprinting technique of bond characterization matrix is applied to the generated structures, so that identical structures are strictly forbidden. These procedures significantly enhance the diversity of the structures, which is crucial for structural global search efficiency. In most cases, structural searching simulations for each calculation were stopped after generating  $1000 \sim 1200$  structures (e.g., about  $20 \sim 30$  generations).

The local structural relaxations and electronic properties calculations were performed in the framework of the density functional theory (DFT)<sup>4</sup> within the generalized gradient approximation (GGA)<sup>5</sup> as implemented in the VASP code. The  $3d^74s^1$ , and  $3s^23p^3$  atomic orbitals were treated as valence states for Fe, and P respectively. The cut-off energy for the expansion of wavefunctions into plane waves is set to 450 eV in all calculations. Phonon calculations were performed by using a supercell approach with the finite displacement method<sup>6</sup> as implemented in the Phonopy code.<sup>7</sup> Dependent on specific structures of stable monolayers, different supercells are used:  $3 \times 3 \times 1$  for FeP with *P4/nmm* symmetry,  $1 \times 4 \times 1$  for FeP<sub>2</sub> with *C2/m* symmetry,  $2 \times 2 \times 1$  for FeP<sub>3</sub> with *C2/m* symmetry, and  $3 \times 2 \times 1$  for FeP<sub>4</sub> with *Pm* symmetry. Among the predicted stable structures, the FeP with *P4/nmm* 

symmetry is the most stable. First-principles molecular dynamics (MD)<sup>8</sup> simulations for a large supercell were performed at different temperatures of 500 K. The supercells adopt  $5 \times 5 \times 1$  for FeP with *P4/nmm* symmetry,  $1 \times 7 \times 1$  for FeP<sub>2</sub> with *C2/m* symmetry,  $2 \times 3 \times 1$  for FeP<sub>3</sub> with *C2/m* symmetry, and  $4 \times 4 \times 1$  for FeP<sub>4</sub> with *Pm* symmetry. MD simulation in NVT ensemble lasted for 10 ps with a time step of 1.0 fs. The temperature was controlled by using the Nosé-Hoover method.

## **Cohesive energy**

Cohesive energy is widely used to ascertain the feasibility for experimental synthesis of the predicted 2D materials. Here, the cohesive energy  $E_{coh}$  is calculated based on the equation of  $E_{coh} = (E_{Fe} + x \times E_P - E_{FeP_x})/(x+1)$ , where  $E_{Fe}$ ,  $E_P$ , and  $E_{FeP_x}$  are the energies of Fe, P atom, and 2D FeP<sub>x</sub> unit cell, respectively.

# **Supporting Figures**



**Figure. S1** ELF map of the (a) FeP monolayer, (b)  $FeP_2$  monolayer, (c) - (d)  $FeP_3$  monolayer, and (e) - (f)  $FeP_4$  monolayer.



Figure. S2 The  $P_2$  and  $P_4$  units are alternate arrangement through Fe-P bonding of FeP<sub>3</sub> monolayer.



**Figure. S3** Relative formation energy of  $\text{FeP}_x$  monolayers with respect to Fe, P atoms and the Fe<sub>3</sub>P monolayer.<sup>9</sup>



**Figure. S4** Phonon dispersive curves and projected phonon density of states (PHDOS) of the (a) FeP monolayer, (b)  $FeP_2$  monolayer, (c)  $FeP_3$  monolayer, and (d)  $FeP_4$  monolayer.



**Figure. S5** Snapshots of the final frame of (a) FeP, (b)  $\text{FeP}_2$ , (c)  $\text{FeP}_3$ , and (d)  $\text{FeP}_4$  monolayer at time of 10 ps during AIMD simulations under the temperatures of 500 K.



**Figure. S6** Snapshots of the final frame of FeP monolayer at time of 10 ps during AIMD simulations under the temperatures of 1000 K.



Figure. S7 Projected density of states (PDOS) for (a) FeP monolayer, (b)  $FeP_2$  monolayer, (c) FeP<sub>3</sub> monolayer, and (d) FeP<sub>4</sub> monolayer.



**Figure. S8** Hydrogen adsorption and calculated Gibbs free energies at different sites for the (a) FeP monolayer, (b) FeP<sub>2</sub> monolayer, (c) FeP<sub>3</sub> monolayer, and (d) FeP<sub>4</sub> monolayer. The FeP<sub>2</sub> monolayer destroy when hydrogen adsorption at 2 site.



**Figure. S9** Top and side views of the structure of the FeP monolayer with hydrogen adsorption. The different H coverages can be simulated with an increment of 1/9.



**Figure. S10** Top and side views of the structure of the  $FeP_2$  monolayer with hydrogen adsorption. The different H coverages can be simulated with an increment of 1/8.





Figure. S11 Top and side views of the structure of the  $FeP_3$  monolayer with hydrogen adsorption. The different H coverages can be simulated with an increment of 1/16.



**Figure. S12** Top and side views of the structure of the  $FeP_4$  monolayer with hydrogen adsorption. The different H coverages can be simulated with an increment of 1/9.



**Figure. S13** The Gibbs free energies under biaxial strain for FeP (a)  $a^{\Delta G_H}$ , (b)  $d^{\Delta G_H}$ ; for FeP<sub>2</sub> (c)  $a^{\Delta G_H}$ , (d)  $d^{\Delta G_H}$ ; for FeP<sub>4</sub> (e)  $a^{\Delta G_H}$ , (f)  $d^{\Delta G_H}$ .

# **Supporting Tables**

Phase	Space	Lattice	Wyckoff Positions			
	Group	Parameters	(fractional)			
		(Å, °)	Atoms	x	у	Z
FeP	P4/nmm	<i>a</i> = 3.73020	Fe(2b)	1.00000	0.00000	0.50000
		<i>b</i> = 3.73020	P(2c)	0.50000	0.00000	0.45450
		<i>c</i> = 23.90970				
		$\alpha = \beta = \gamma = 90.000$				
FeP <sub>2</sub>	C2/m	<i>a</i> = 12.13140	Fe(4i)	0.89830	0.50000	0.49280
		<i>b</i> = 2.62750	P(4i)	0.73576	0.50000	0.54650
		<i>c</i> = 22.85930	P(4i)	0.01034	0.00000	0.43575
		$\alpha = \gamma = 90.000$				
		$\beta = 91.5487$				
FeP <sub>3</sub>	C2/m	<i>a</i> = 7.91300	Fe(4f)	0.25000	0.25000	0.50000
		<i>b</i> = 7.52210	P(4i)	0.33197	0.50000	0.54798
		<i>c</i> = 22.69210	P(4i)	0.36340	0.50000	0.45138
		$\alpha = \gamma = 90.000$	P(4h)	0.50000	0.13999	0.50000
		$\beta = 83.340$				
FeP <sub>4</sub>	Pm	<i>a</i> = 3.74680	Fe(1b)	0.34831	0.50000	0.47196
		<i>b</i> = 4.20240	P(1b)	0.83283	0.50000	0.42797
		c = 22.83070	P(1b)	0.86791	0.50000	0.52770
		$\alpha = \gamma = 90.000$	P(1a)	0.88318	0.00000	0.57110
		β = 93.3094	P(1a)	0.36214	0.00000	0.51128

**Table S1** Detailed structural information of the predicted stable  $FeP_x$  monolayers.

Phase	Atom	Charge
FeP	Fe	0.30
	Р	-0.30
FeP <sub>2</sub>	Fe	0.20
	P1	-0.12
	P2	0.08
FeP <sub>3</sub>	Fe	0.33
	P1	-0.08
	P2	-0.10
	P3	-0.15
FeP <sub>4</sub>	Fe	0.24
	P1	-0.02
	P2	-0.02
	Р3	-0.16
	P4	-0.04

**Table S2** Bader charge analysis of the  $FeP_x$  monolayers.

## References

- 1 Y. Wang, J. Lv, L. Zhu and Y. Ma, *Comput. Phys. Commun.*, 2012, **183**, 2063–2070.
- 2 Y. Wang, J. Lv, L. Zhu and Y. Ma, *Phys. Rev. B Condens. Matter Mater. Phys.*, 2010, **82**, 094116.
- 3 G. Kresse and J. Furthmu, *Phys. Rev. B.*, 1996, **54**, 11169–11186.
- 4 W. KOHN and L. J. SHAM, *Phys. Rev*, 1965, **140**, A1133–A1138.
- 5 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 6 K. Parlinski, Z. Q. Li and Y. Kawazoe, *Phys. Rev. Lett.*, 1997, 78, 4063–4066.
- A. Togo, F. Oba and I. Tanaka, *Phys. Rev. B Condens. Matter Mater. Phys.*, 2008, **78**, 134106.
- 8 G. J. Martyna, M. L. Klein and M. Tuckerman, J. Chem. Phys., 1992, 97, 2635–2643.
- 9 S. Zheng, C. Huang, T. Yu, M. Xu, S. Zhang, H. Xu, Y. Liu, E. Kan, Y. Wang and G. Yang, *J. Phys. Chem. Lett.*, 2019, **10**, 2733–2738.