

## Supplementary Information

### Influence of $\text{Ti}^{3+}$ Defect-type on Heterogeneous Photocatalytic $\text{H}_2$ Evolution

#### Activity of $\text{TiO}_2$

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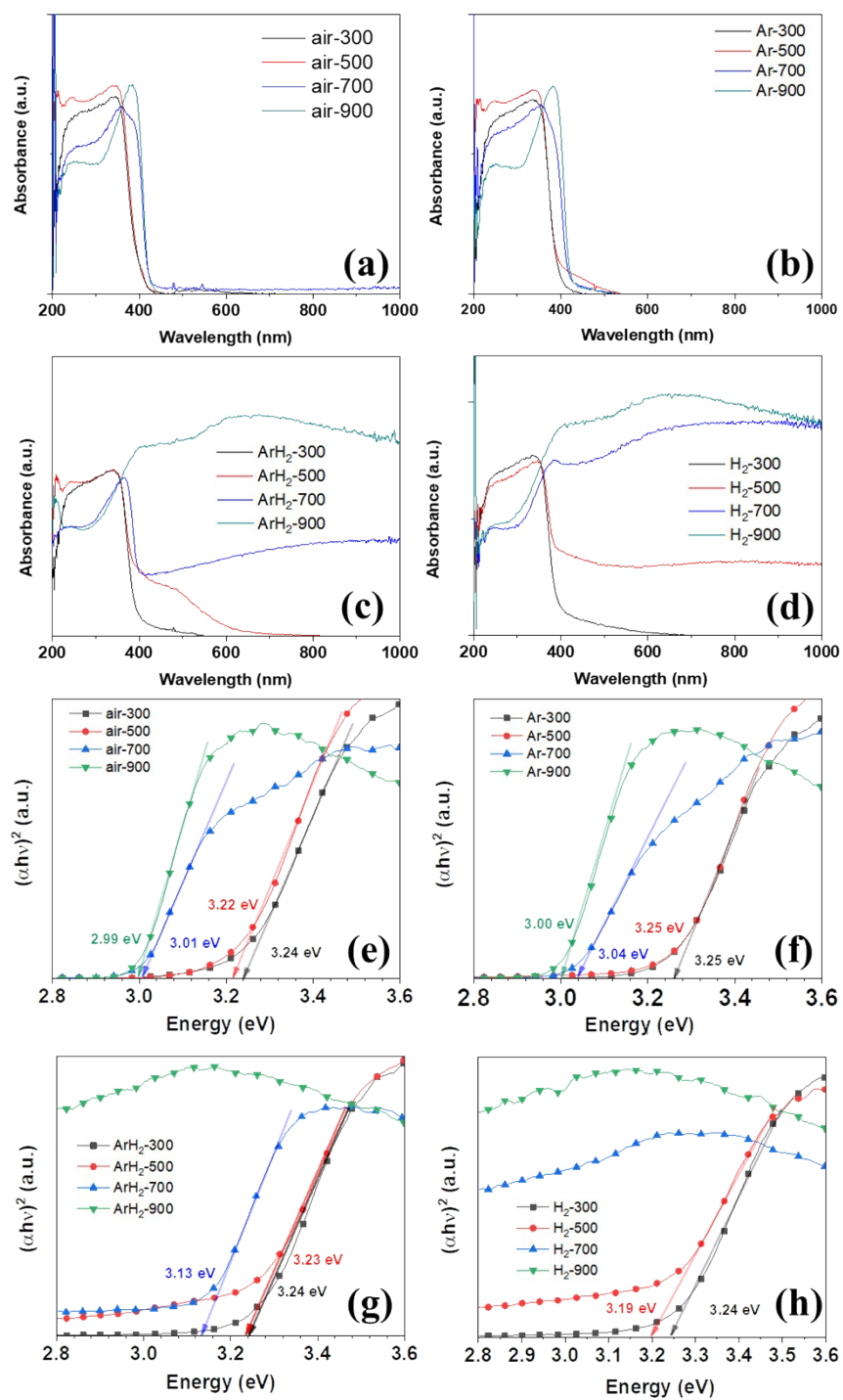
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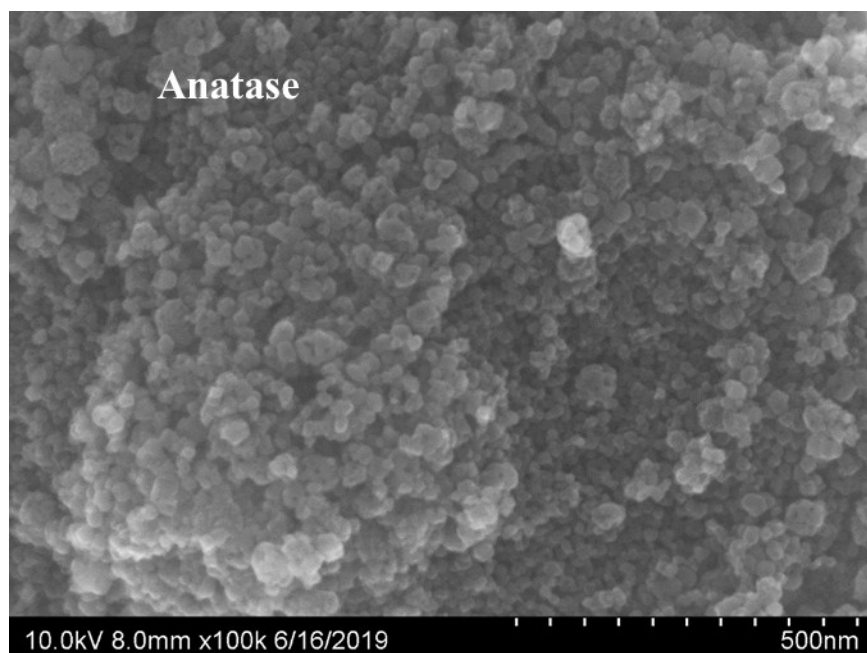
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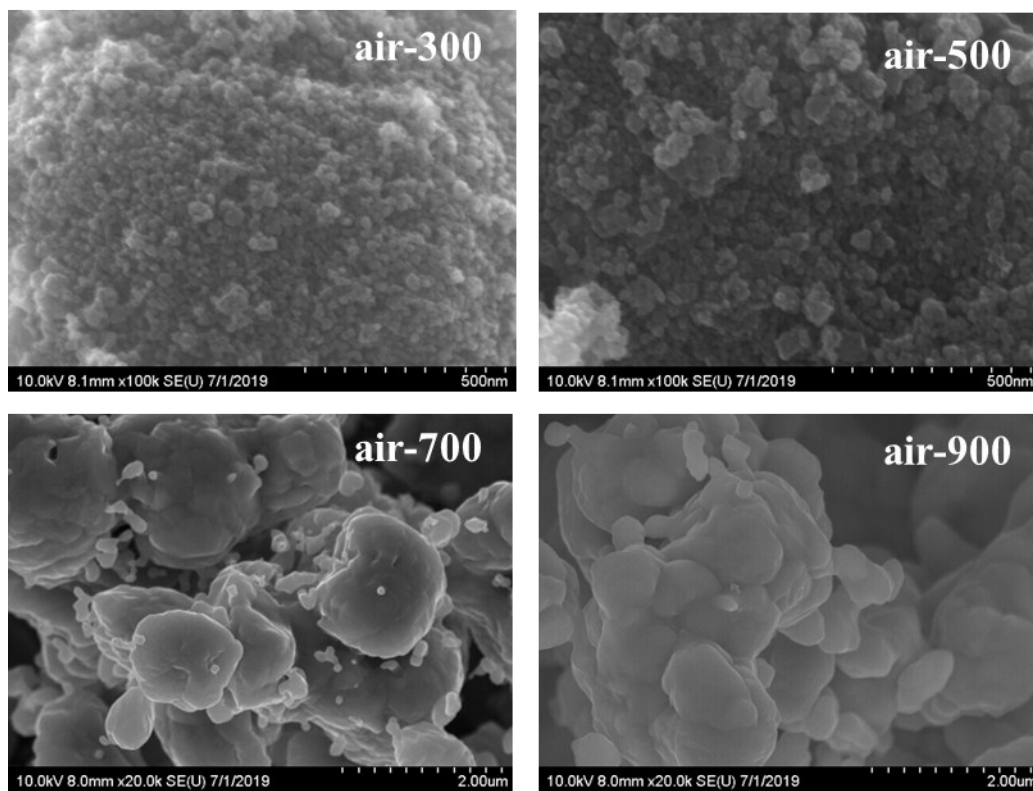
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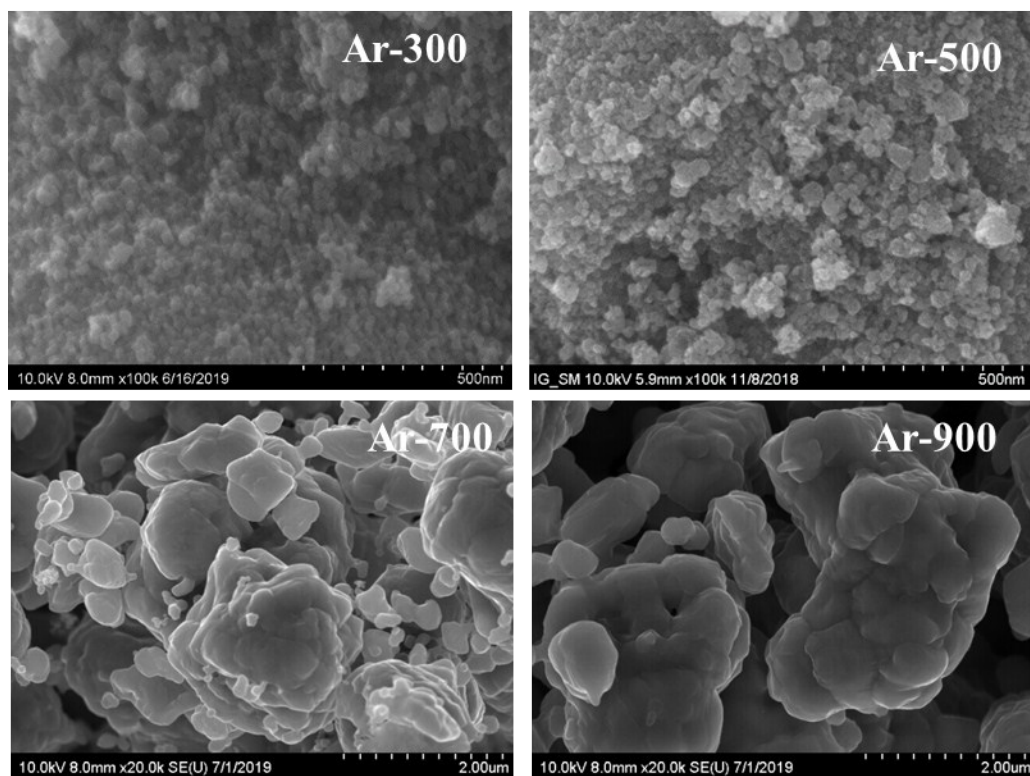
**Figure S1.** (a-d) absorbance spectra and (e-h) Tauc plots of the powders treated at different temperatures in air, Ar, H<sub>2</sub> and Ar/H<sub>2</sub>.



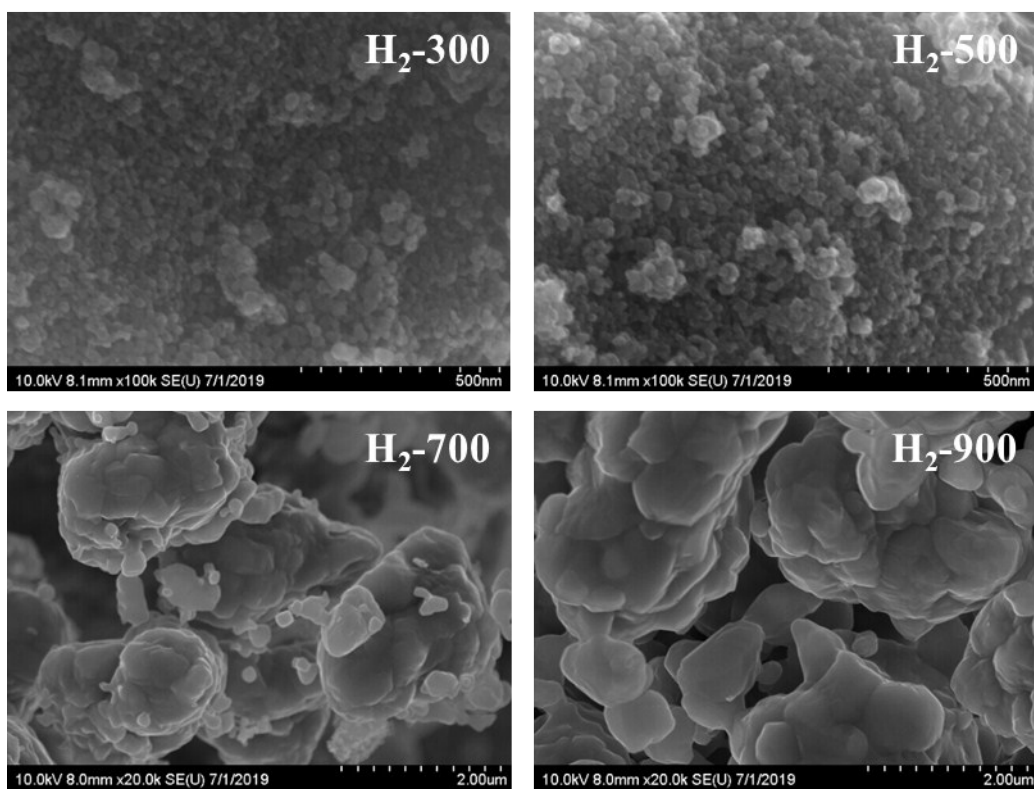
**Figure S2.** SEM images of the pristine  $\text{TiO}_2$  anatase nanopowders.



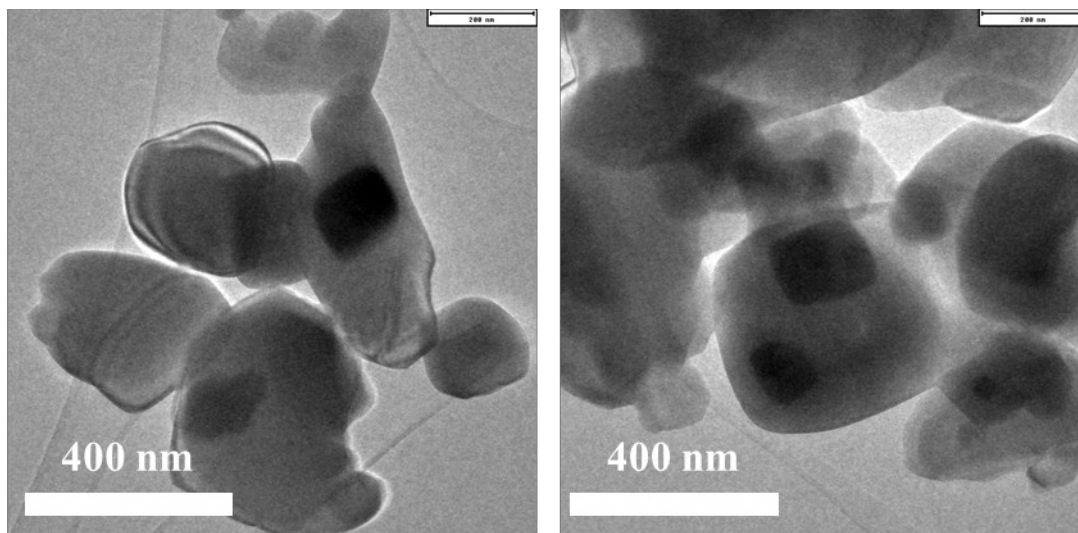
**Figure S3.** SEM images of  $\text{TiO}_2$  powder treated in air at 300° C, 500° , 700° C, and 900° C.



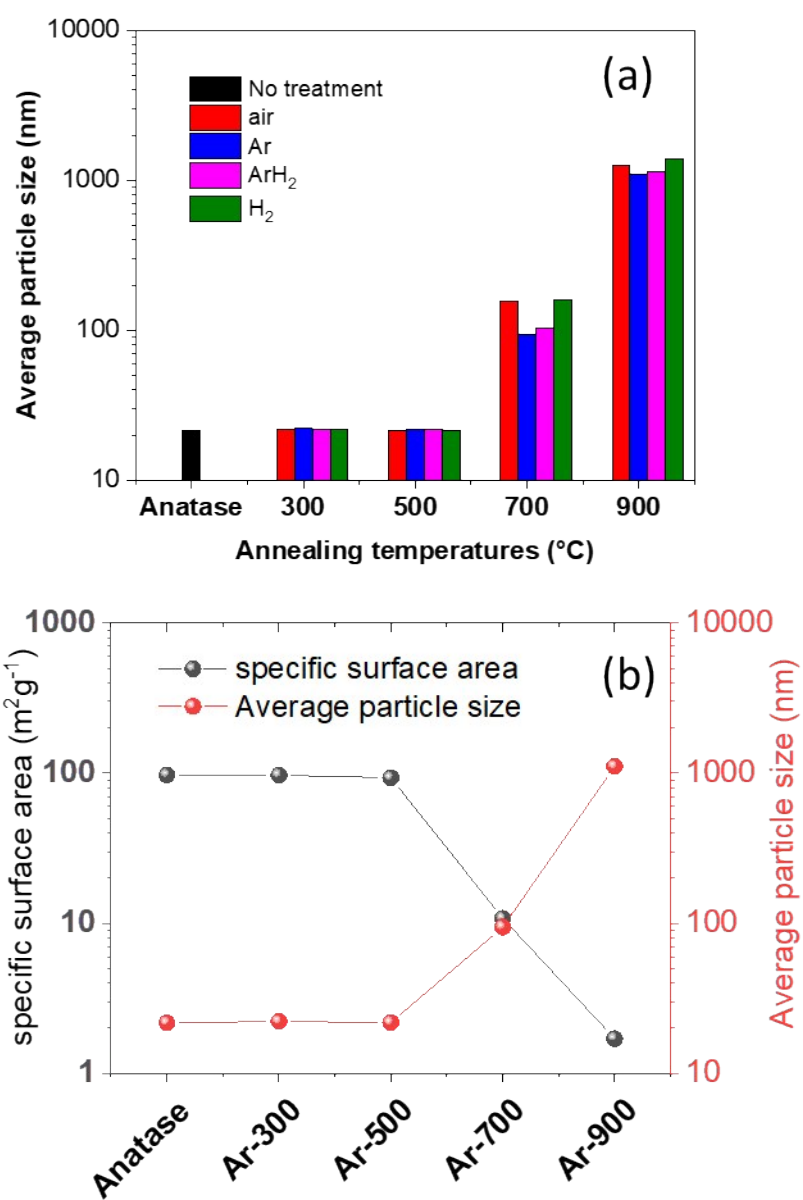
**Figure S4.** SEM images of  $\text{TiO}_2$  powder treated in Ar at 300 °C, 500°, 700° C, and 900° C.



**Figure S5.** SEM images of  $\text{TiO}_2$  powder treated in  $\text{H}_2$  at 300 °C, 500°, 700° C, and 900° C.

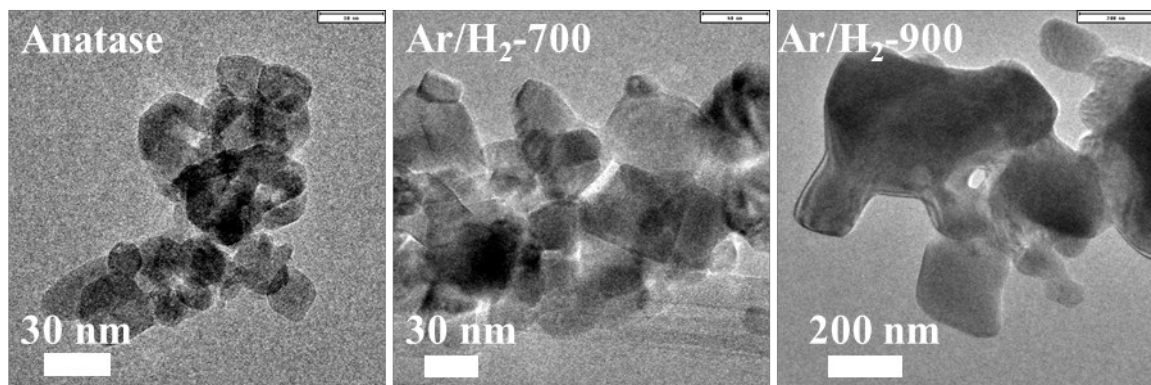


**Figure S6.** TEM images of Ar/H<sub>2</sub> treated powder at 900 °C indicating different sizes of the powders.

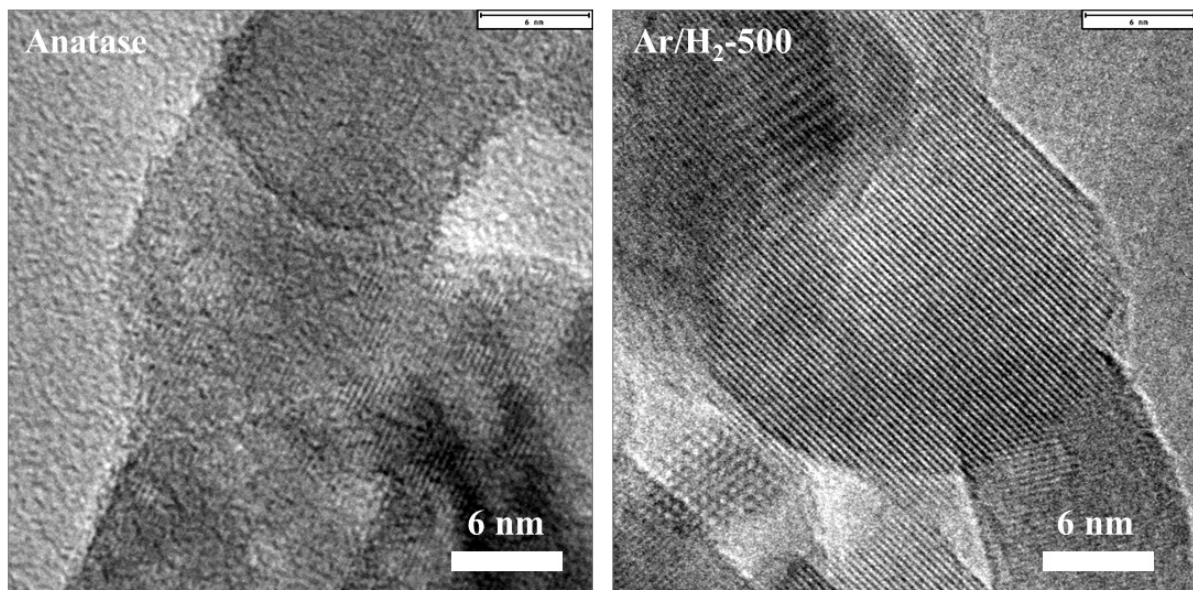


**Figure S7.** (a) Average size of TiO<sub>2</sub> powder treated in different environments and temperatures; (b) comparison of Ar treated TiO<sub>2</sub> powders specific surface area and average particle size in different temperatures.

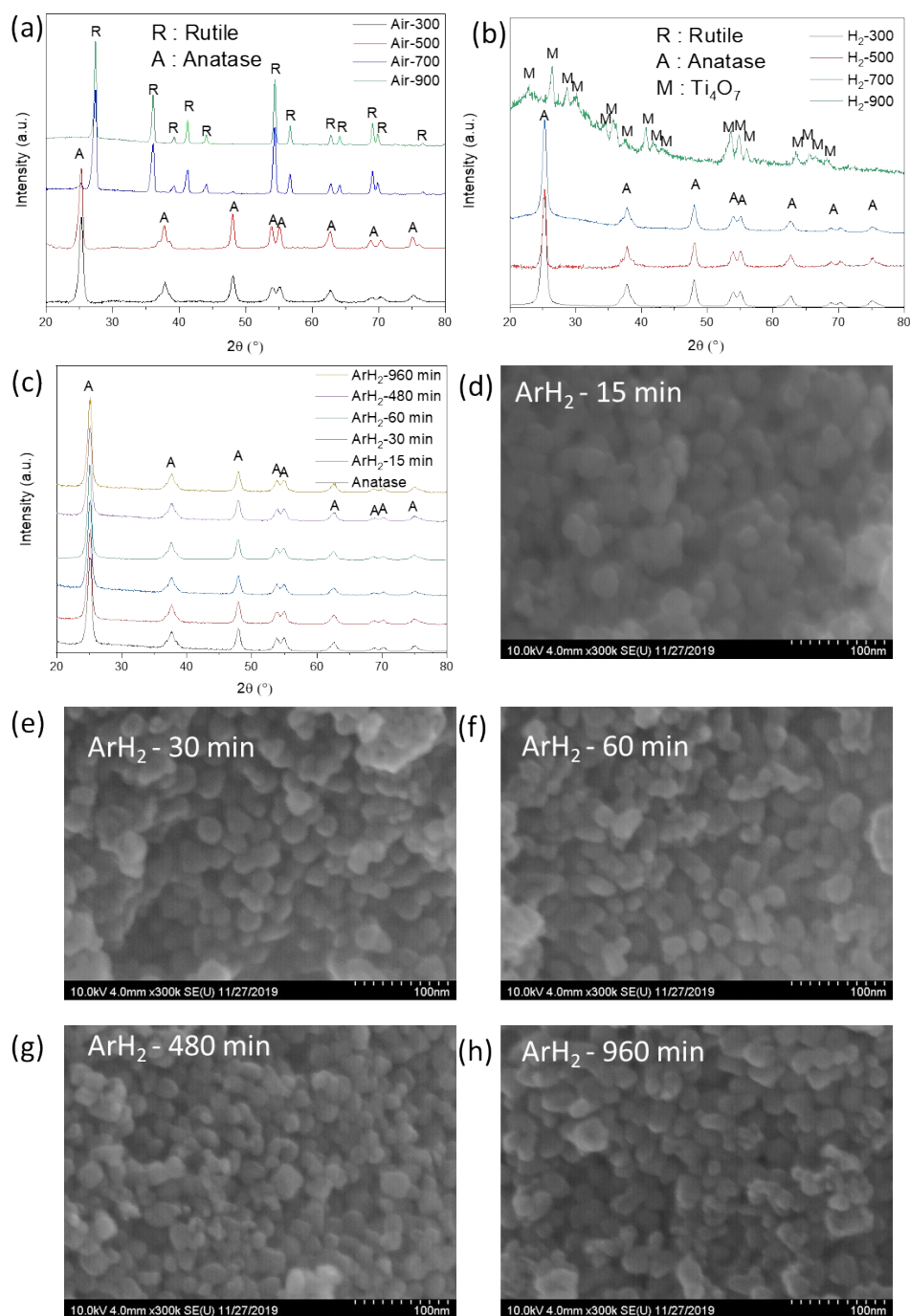




**Figure S8.** TEM images of Ar/H<sub>2</sub> treated powder at 700°C and 900°C compared to pristine nanopowder.



**Figure S9.** HR-TEM images of the Ar/H<sub>2</sub> treated powder at 500°C compared to pristine nanopowder.



**Figure S10** XRD patterns of (a) air, (b) H<sub>2</sub> treated powder at different temperatures; (c) XRD spectra and (d-h) SEM images of TiO<sub>2</sub> powders treated at 500°C in Ar/H<sub>2</sub> in different annealing time.

Experimental Evaluation of the spin density ( $S_T$ ) using the EPR technique. Spin density ( $S_T$ ) was evaluated by using the  $\text{CuSO}_4 \times 5\text{H}_2\text{O}$  (99.999%, CAS Number: 7758-99-8) as  $S=1/2$  standard and used in powder form by comparing the double integrated intensities of the various  $\text{TiO}_2$  samples against the spin standard through Eq. (1).

Standard:  $\text{CuSO}_4 \times 5\text{H}_2\text{O}$ , MW = 249.69

$$N_A = 6.02214 \times 10^{23}$$

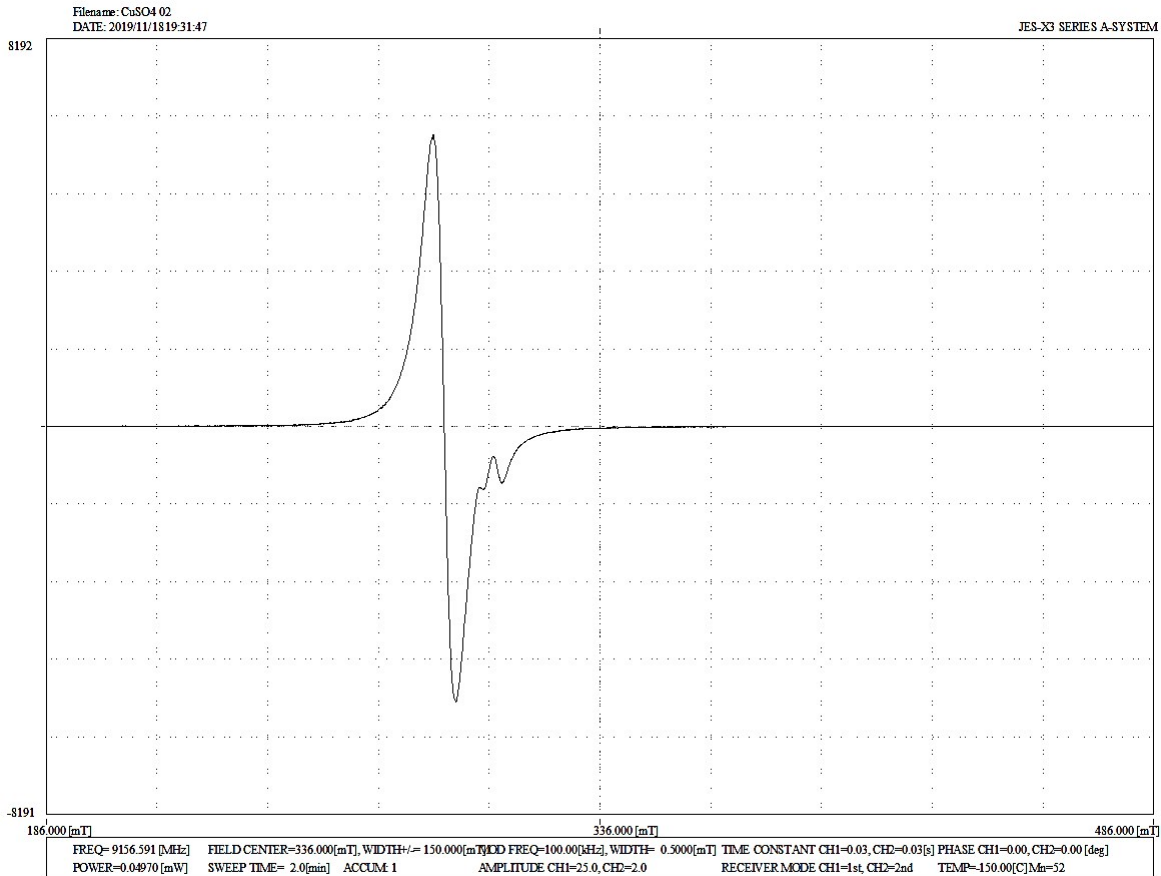
$$N_A (\text{standard}) = 2.41 \times 10^{21} (\text{per molecule}), \text{ or } 9.48 \times 10^{21} (\text{per } \text{Cu}^{2+} \text{ cation})$$

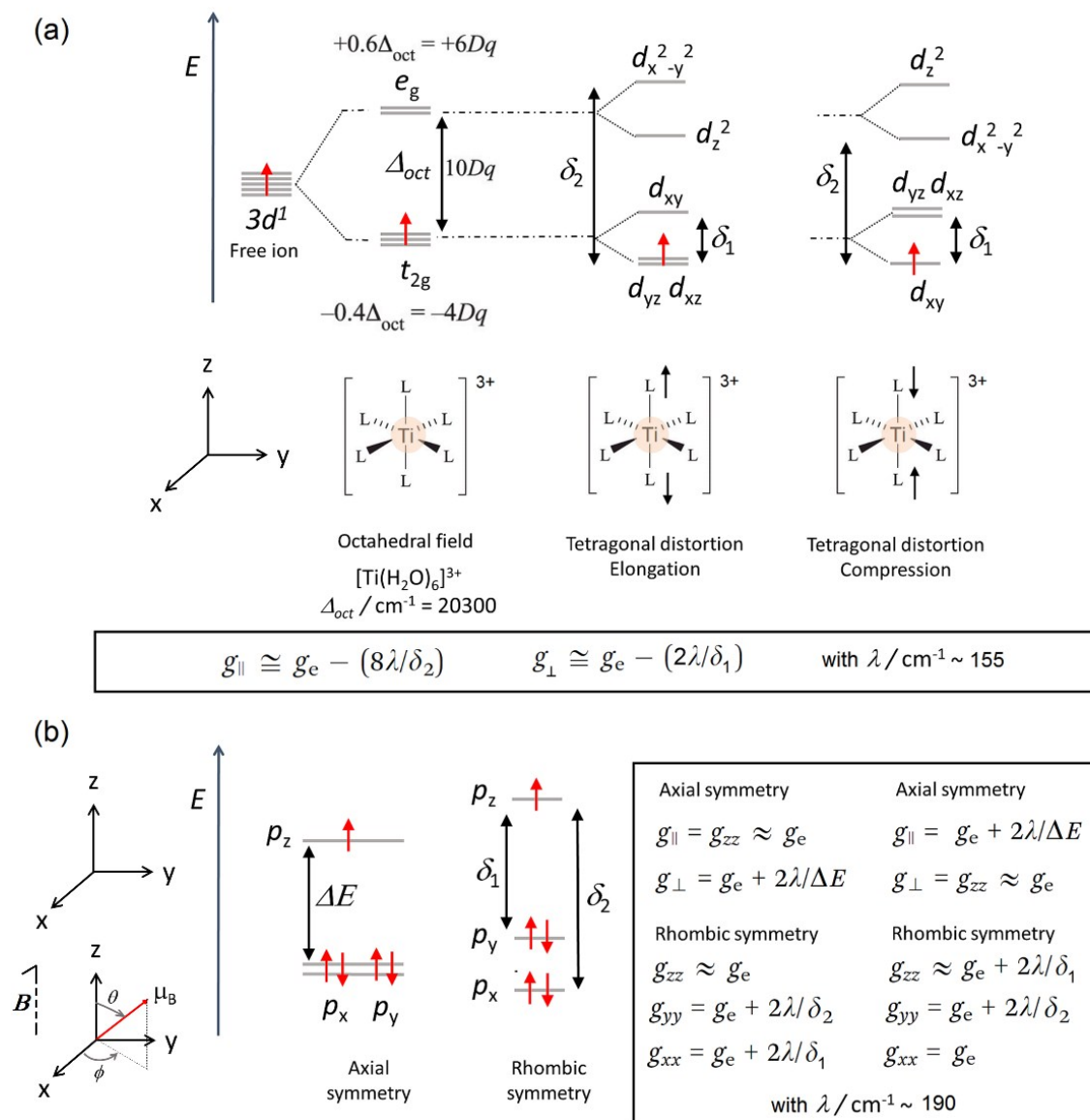
$$D_s (S_T) = \left[ \alpha \times \frac{\iint \text{Signal Intensity } \text{TiO}_2}{\iint \text{Signal Intensity } \text{CuSO}_4} \right] \times \frac{g_{\text{CuSO}_4}}{g_{\text{TiO}_2}} \times N_{A, \text{CuSO}_4} \quad (\text{Eq. 1})$$

$g_{\text{CuSO}_4}$  = Grams of copper sulphate pentahydrate standard,  $g_{\text{TiO}_2}$  = Grams of the titanium dioxide materials

$\alpha$  =  $g$  values ratio,  $g(\text{CuSO}_4)/g(\text{TiO}_2)$

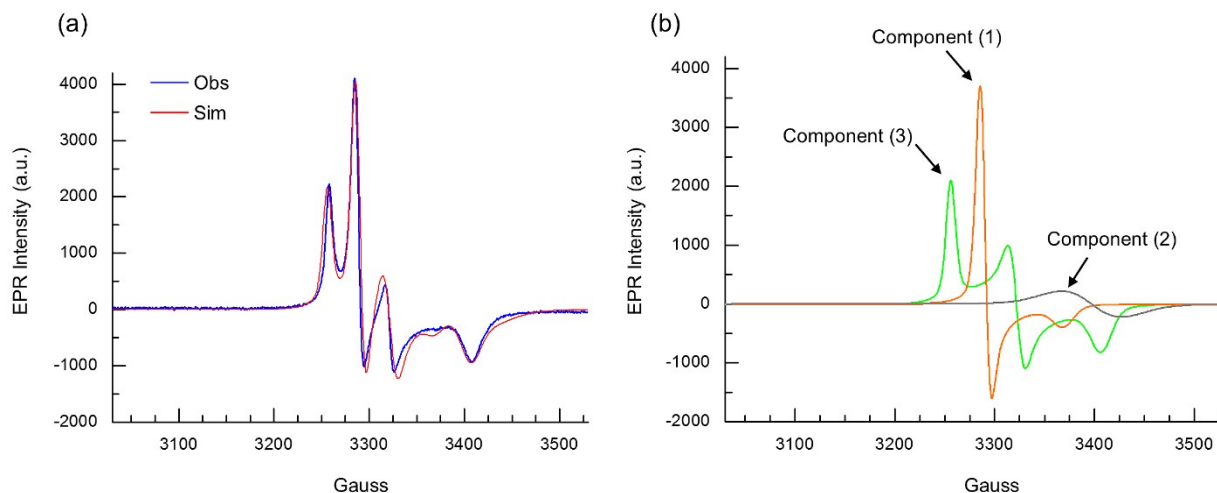
$$g_{\text{avg}} (\text{CuSO}_4 \times 5\text{H}_2\text{O}) = 2.180$$





**Fig**

**re S11. (a)** Energy level of the  $\text{Ti}^{3+}$  ( $d^1$ ,  $S=1/2$ )  $d$ -orbitals in a tetragonal crystal field and the theoretically derived  $g$ -tensor values.  $\Delta$  and  $\delta$  indicate the energy difference of specific sets of  $d$ -orbitals and  $\lambda$  is the spin-orbit coupling constant. **(b)** Energy levels of a hole center ( $\text{O}^\bullet$  radical anion,  $d^5$ ,  $S=1/2$ ) under axial or rhombic-field symmetry and the theoretically derived  $g$ -tensor values.<sup>1,2</sup> In panel **(b)** the theoretically derived  $g$ -tensor components can assume diverse combination of values, depending on the selected molecular axis coordinates (polar coordinates depicted on the left, see equations on the right where  $g_{\parallel} > g_{\perp}$ ,  $g_{zz} > g_{yy} > g_{xx}$ ).



**Figure S12. (a)** X-band EPR spectrum (9163.299 MHz) of the air-700 sample (60 min annealing time) recorded at  $T = 123$  K and 0.2 mW applied power (blue line, experimental) and its computer simulation (solid red-line). The simulated EPR envelope results from the sum of three components (1, 2 and 3) and is shown in panel **(b)**.

**Comment.** Component (1) represents the  $\text{Ti}^{3+}$  centres residing in regular lattice position with moderate tetragonal distortion. The simulated  $g$ -tensor values accounts for  $g_x = 1.994$ ,  $g_y = 1.994$  and  $g_z = 1.944$ , line width tensor  $L_w(x,y,z)$  of 1.0, 1.0, 1.0 mT, Lorentzian/Gaussian ratio of 0.63.

Component (2) arises from those  $\text{Ti}^{3+}$  centres that are surface exposed, and highly disordered. The signal is simulated using an isotropic  $g$ -tensor of  $g_{x,y,z} = 1.930$  and line width tensor  $L_w(x,y,z)$  of 6.0, 6.0, 6.0 mT, Lorentzian/Gaussian ratio = 0.63.

Component (3) corresponds to a  $S=1/2$  system associated to a hole (O-centre) in the lattice. The simulated spin-Hamiltonian parameters of such system are  $g_1 = 2.010$ ,  $g_2 = 1.970$  and  $g_3 = 1.921$ , line width tensor  $L_w(x,y,z)$  of 2.0, 1.3, 0.9 mT, and Lorentzian/Gaussian ratio = 0.63.

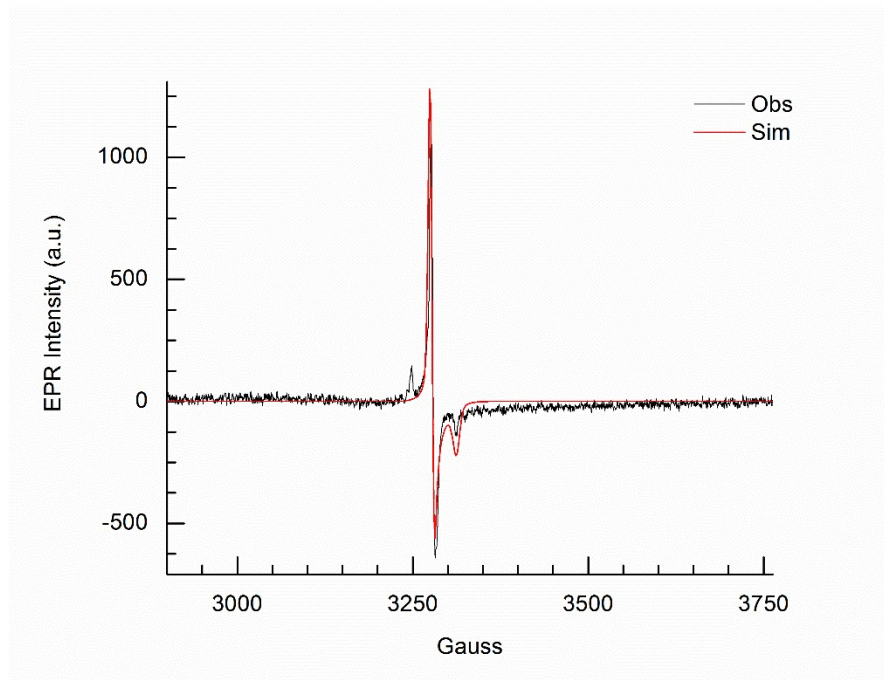
The weighted percentage of the total spin-population for components (1), (2) and (3) is estimated as 39 % ( $\text{Ti}^{3+}$  regular sites), 5% ( $\text{Ti}^{3+}$  surface exposed) and 56 % (holes), respectively.

Simulation of the powder EPR trace was carried out by the WinEPR SimFonia software (V.1.25, EPR Division, Bruker Instruments, Inc., Billerica, USA) using second-order perturbation theory and spherical integration of 200 (theta), 200 (phi), according to the following Hamiltonian:

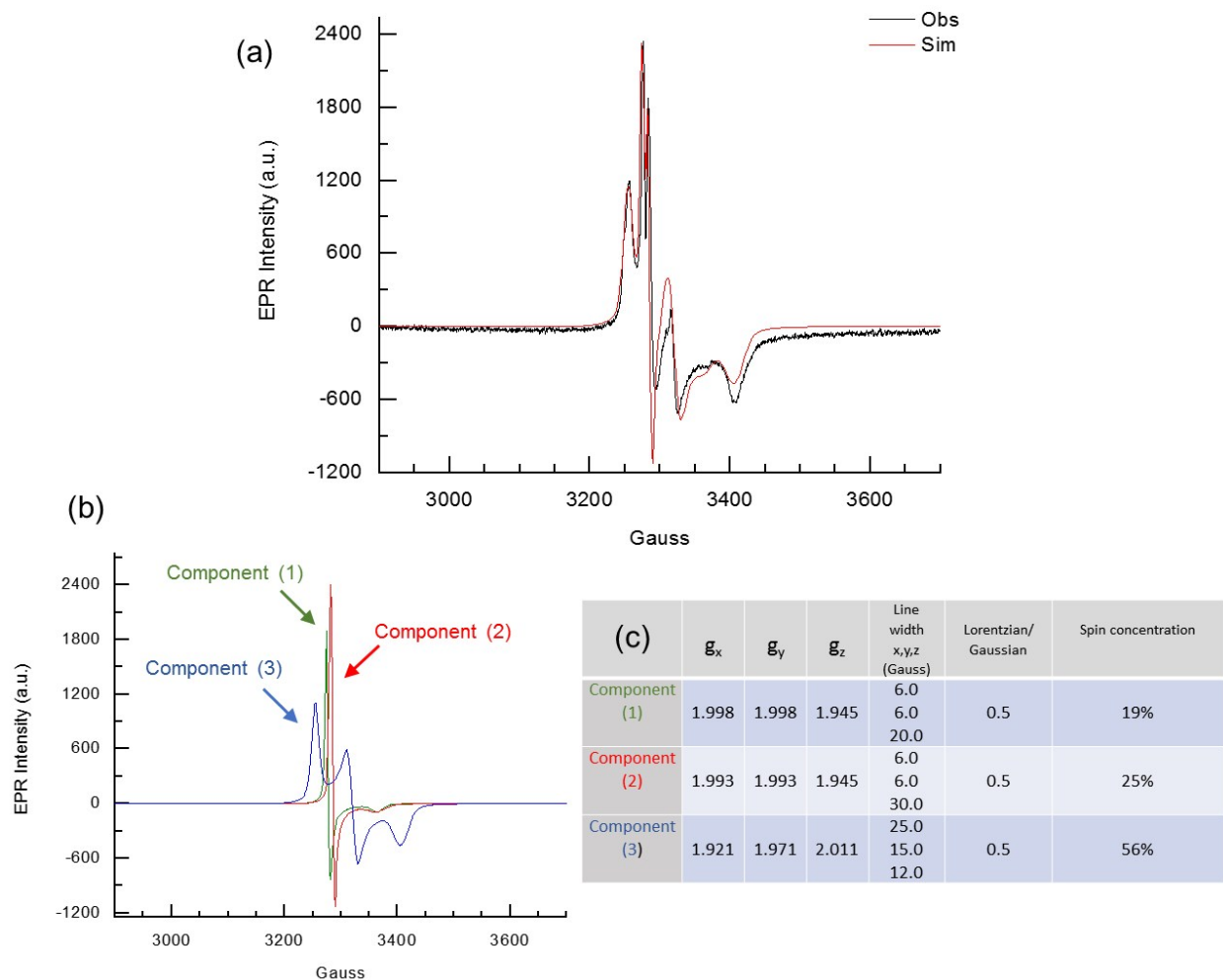
$$\text{Powder spectra: } H = \mu_B B_0 \cdot \mathbf{g} \cdot \mathbf{S} + \mathbf{S} \cdot \mathbf{D} \cdot \mathbf{S} + \mathbf{S} \cdot \mathbf{A} \cdot \mathbf{I} + \mathbf{I} \cdot \mathbf{P} \cdot \mathbf{I} + \mu_N B_0 \cdot \mathbf{g}_N \cdot \mathbf{I}$$

**Note:** 1 mT = 10 Gauss

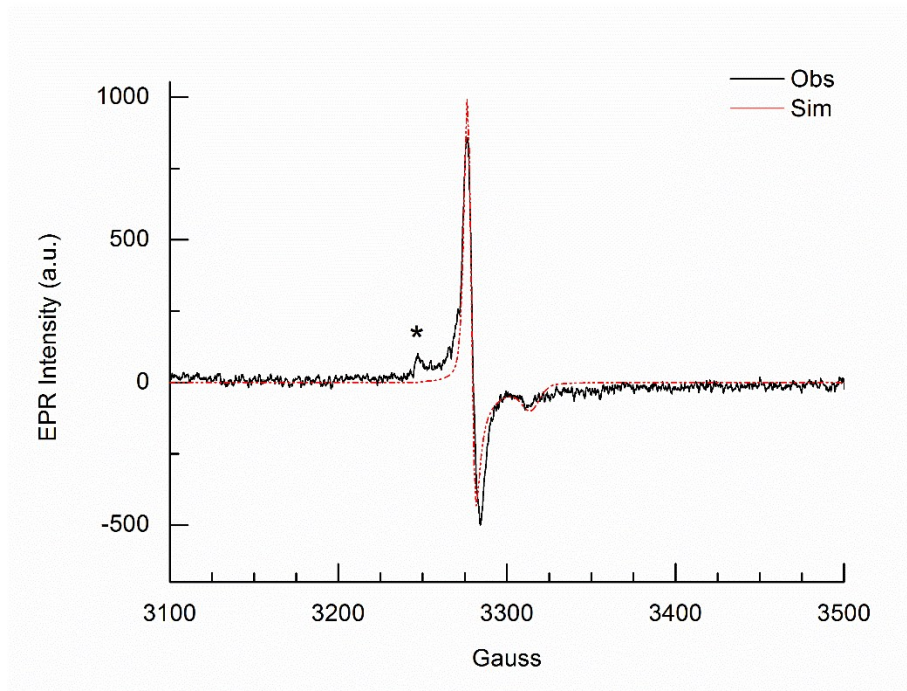




**Figure S13. (a)** X-band EPR spectrum (9161.090 MHz) of the Ar-300 sample (60 min annealing time) recorded at  $T = 123$  K and 0.2 mW applied power (black line, experimental) and its computer simulation (solid red-line). The simulated  $g$ -tensor values account for lattice embedded  $\text{Ti}^{3+}$  centres, with  $g_x = 1.998$ ,  $g_y = 1.998$  and  $g_z = 1.976$ , line width tensor  $L_w(x,y,z)$  of 0.6, 0.6, 0.9 mT, Lorentzian/Gaussian ratio of 0.50. The EPR signal coming from holes ( $h^+$ ) has been neglected.

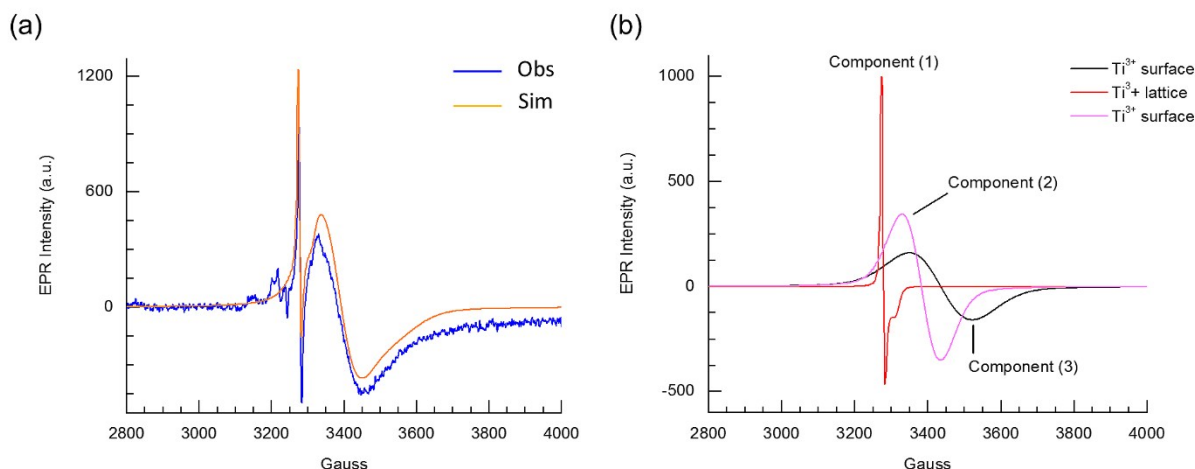


**Figure S14.** (a) X-band EPR spectrum (9160.810 MHz) of the Ar-500 sample (60 min annealing time) recorded at  $T = 123$  K and 0.2 mW applied power (black line, experimental) and its computer simulation (solid red-line). The simulated EPR envelope results from the sum of three components (1, 2 and 3) (see panel b) and simulation parameters are shown in Table (c).



**Figure S15.** X-band EPR spectrum (9162.126 MHz) of the Ar/H<sub>2</sub>-300 sample (60 min annealing time) recorded at  $T = 123$  K and 0.2 mW applied power (black line, experimental) and its computer simulation (dash-dot red-line). The simulated  $g$ -tensor values account for lattice embedded Ti<sup>3+</sup> centres, with  $g_x = 1.997$ ,  $g_y = 1.997$  and  $g_z = 1.976$ , line width tensor  $L_w(x,y,z)$  of 0.5, 0.4, 0.9 mT, Lorentzian/Gaussian ratio of 0.80. The EPR signal coming from holes (marked with an asterisk, (h<sup>+</sup>)) has been neglected.





**F**

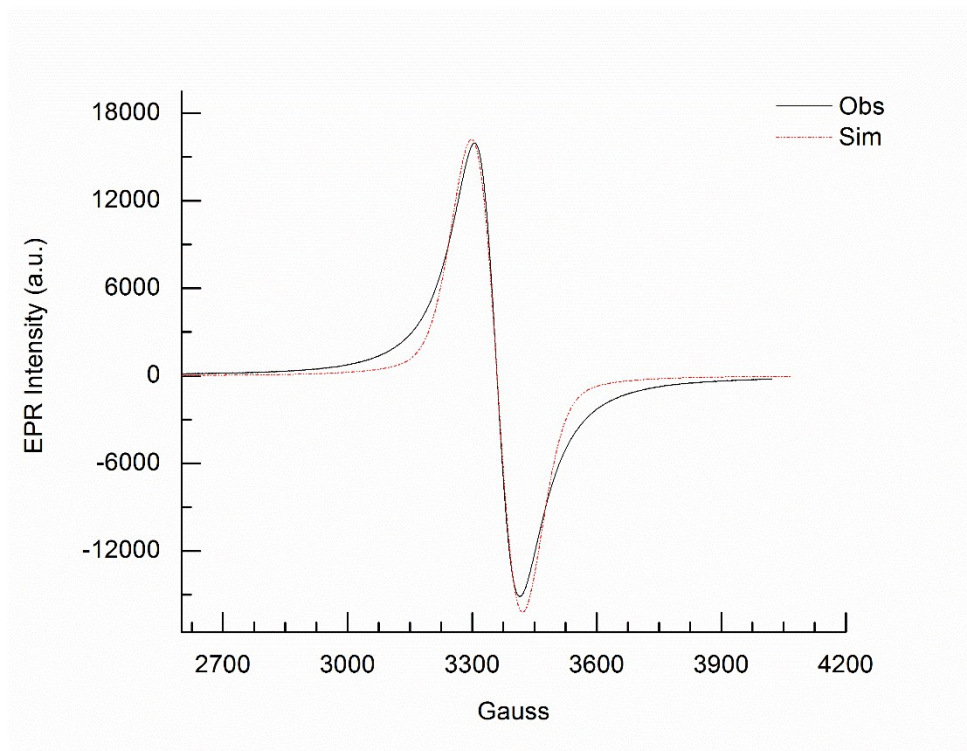
**figure S16. (a)** X-band EPR spectrum (9152.260 MHz) of the Ar/H<sub>2</sub>-500 sample (60 min annealing time) recorded at  $T = 123$  K and 0.6 mW applied power (blue line, experimental) and its computer simulation (solid orange-line). The simulated EPR envelope results from the sum of three components (1, 2 and 3) as shown in panel **(b)**. The EPR signal arising from hole centers ( $h^+$ ) has been neglected.

**Comment.** Component (1) represents the  $Ti^{3+}$  centres residing in regular lattice position with moderate tetragonal distortion. The simulated  $g$ -tensor values accounts for  $g_x = 1.996$ ,  $g_y = 1.996$  and  $g_z = 1.974$ , line width tensor  $L_w(x,y,z)$  of 0.8, 0.8, 2.0 mT, Lorentzian/Gaussian ratio of 0.80.

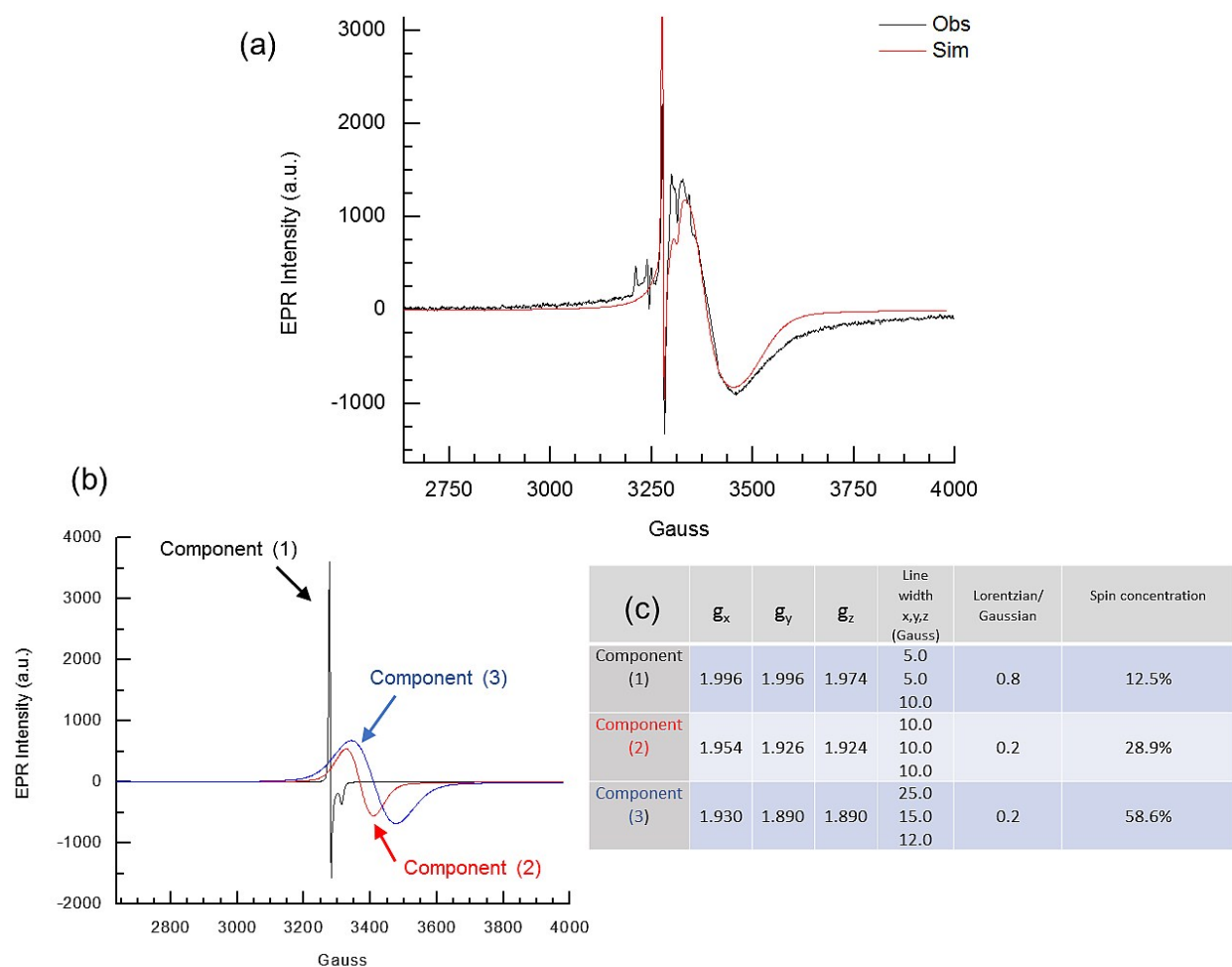
Component (2) arises from a set of those  $Ti^{3+}$  centres that are surface exposed. The signal is simulated using  $g_x = 1.954$ ,  $g_y = 1.924$  and  $g_z = 1.924$  and line width tensor  $L_w(x,y,z)$  of 10.0, 10.0, 10.0 mT, Lorentzian/Gaussian ratio = 0.20.

Component (3) arises from another set of  $Ti^{3+}$  centres that are surface exposed. The signal is simulated using  $g_x = 1.930$ ,  $g_y = 1.890$  and  $g_z = 1.890$  and line width tensor  $L_w(x,y,z)$  of 16.0, 16.0, 16.0 mT, Lorentzian/Gaussian ratio = 0.20.

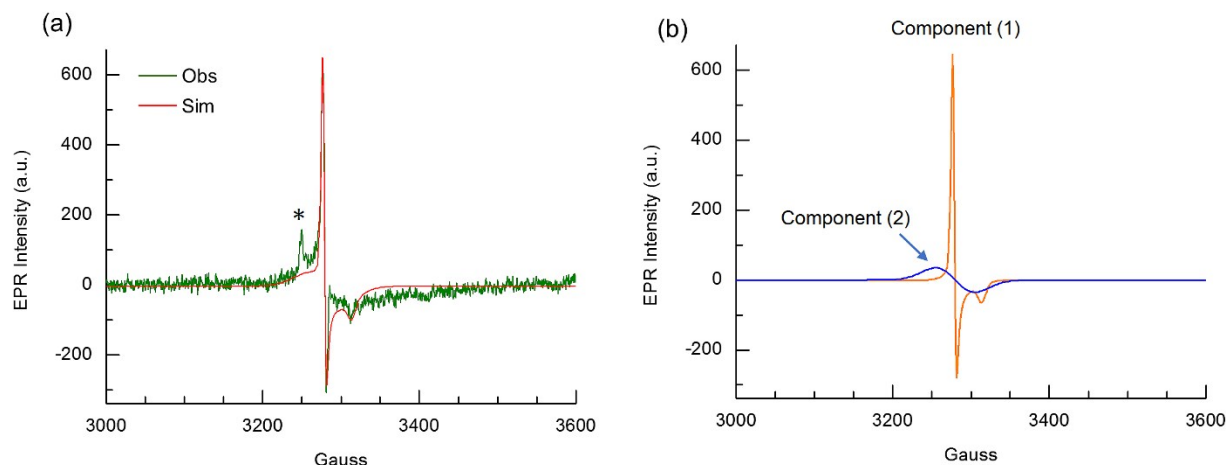
From simulation, the weighted percentage of the total spin-population for components (1), (2) and (3) is estimated as 13.0% ( $Ti^{3+}$  regular sites), 49.7% ( $Ti^{3+}$  surface exposed, component 2) and 37.3% ( $Ti^{3+}$  surface exposed, component 3).



**Figure S17.** X-band EPR spectrum (9163.070 MHz) of the Ar/H<sub>2</sub>-900 sample (60 min annealing time) recorded at  $T = 123$  K and 0.2 mW applied power (black line, experimental) and its computer simulation (solid red-line). The resonance signal is attributed to the dominant presence of surface exposed Ti<sup>3+</sup> centres. Simulation parameters of the resonance line are as follow:  $g_{\text{avg}} = 1.949$ , line width tensor  $L_w(x,y,z)$  of 12.0, 13.0, 12.0 mT, Lorentzian/Gaussian ratio of 0.90.

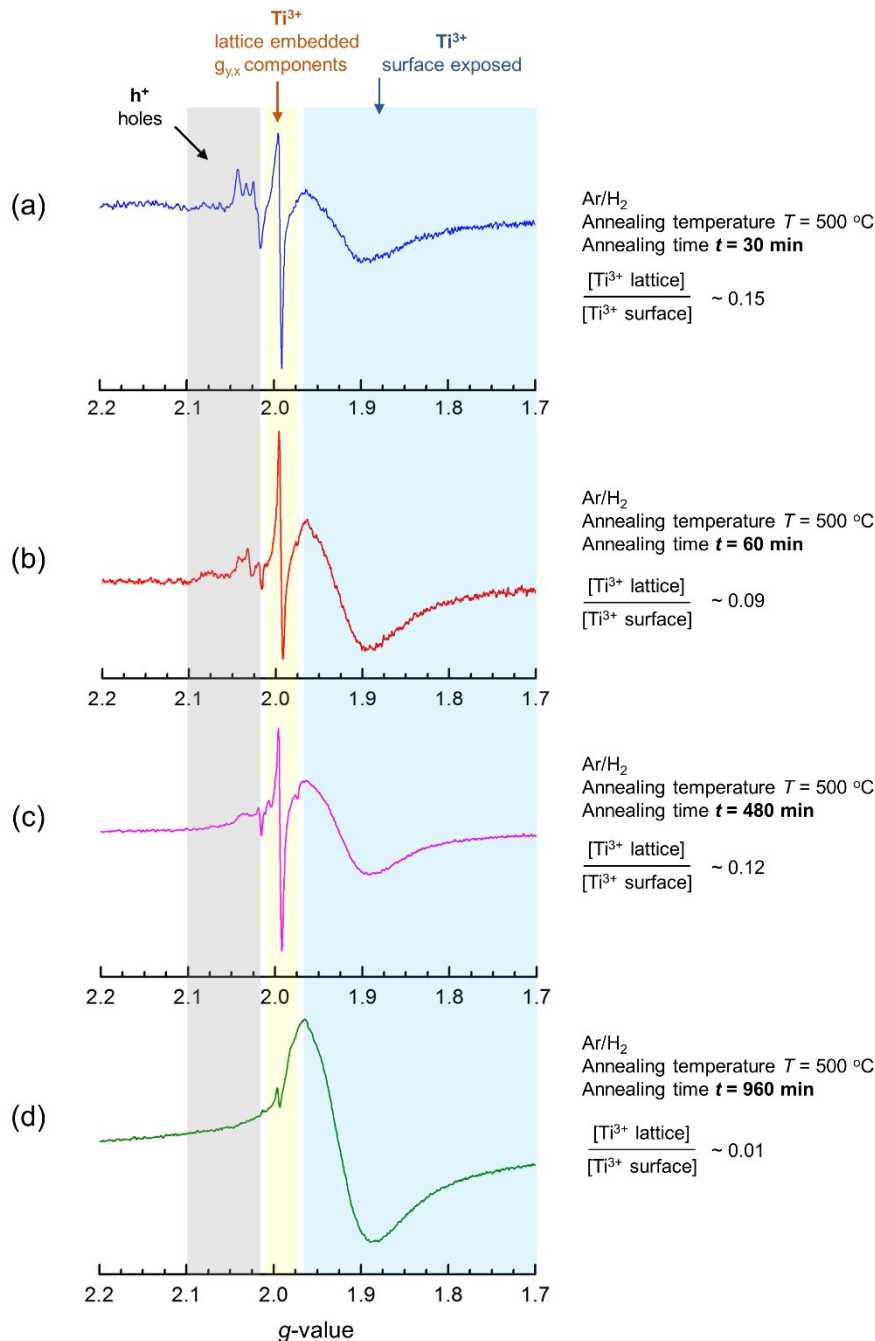


**Figure S18.** (a) X-band EPR spectrum (9153.580 MHz) of the H<sub>2</sub>-500 sample (60 min annealing time) recorded at  $T = 123$  K and 0.8 mW applied power (black line, experimental) and its computer simulation (solid red-line). The simulated EPR envelope results from the sum of three components (1, 2 and 3) as shown in panel (b). The EPR signal coming from holes ( $h^+$ ) has been neglected. Simulation parameters are shown in Table (c).

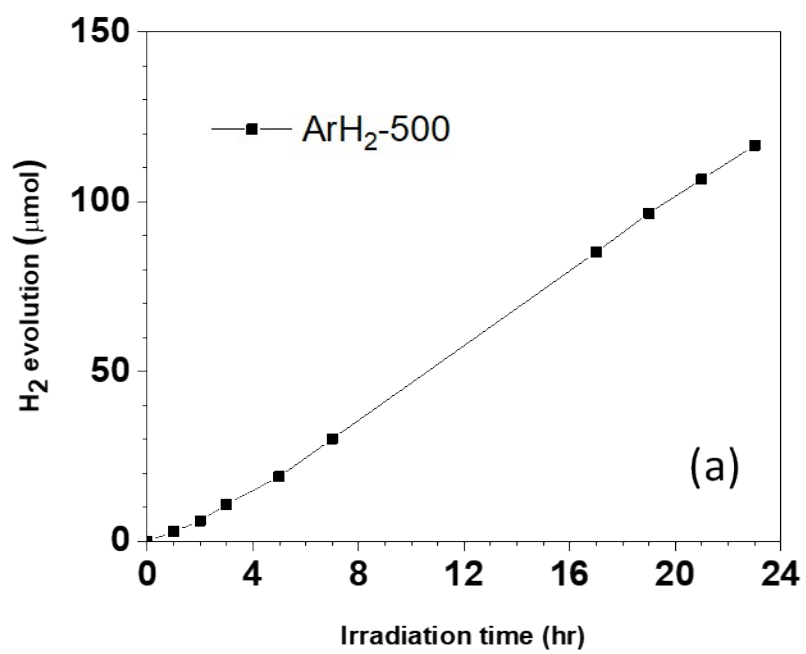


**Figure S19.** (a) X-band EPR spectrum (9162.427 MHz) of the neat anatase nanoparticles recorded at  $T = 123$  K and 0.2 mW applied power (green line, experimental) and the computer simulation (solid red-line) obtained by perturbation theory of various lattice embedded  $\text{Ti}^{3+}$  sites. (b) Computed EPR resonance signatures of the various lattice embedded  $\text{Ti}^{3+}$  sites. Signal from hole ( $\text{h}^+$ ) has been neglected.

**Comment.** The Sim-line (orange trace, Component 1) in panel (b) corresponds to the computer simulation of the EPR resonance envelope associated to the  $\text{Ti}^{3+}$  centres residing in regular lattice position characterized by a narrow distribution in tetragonal distortion ( $Oh$  or  $D_{4h}$ ). The simulated  $g$ -tensor values accounts for  $g_x = 1.998$ ,  $g_y = 1.998$  and  $g_z = 1.976$ , line width tensor  $L_w(x,y,z)$  of 0.5, 0.4, 1.0 mT, Lorentzian/Gaussian ratio of 0.80 and spherical integration of 200 (theta), 200 (phi). The blue EPR resonance-line (simulated) shown in panel (b) is also associated to lattice embedded  $\text{Ti}^{3+}$  sites but characterized by tetragonal distortions in which the energy difference ( $\delta_{l,2}$ ) among the d-orbitals spread quite significantly among the spin containing sites (see Figure S11 for definitions). Because the line-width tensor is not (usually) collinear with the  $g$ -tensor, these factors altogether prevent resolution of the  $g_{||}$  and  $g_{\perp}$  components (Component 2, Gaussian distribution,  $g_{\text{avg}} = 1.9965$ ,  $\Delta B_{\text{pp}} = 5.0$  mT). The total spin density evaluated against  $\text{CuSO}_4 \times 5 \text{ H}_2\text{O}$  ( $S=1/2$ ) standard accounts for  $0.61 \times 10^{18}$  spin /g ( $\text{TiO}_2$ ). The weighted percentage (spin concentration) of Component (1) and Component (2) is 70% for (1) and 30% for (2). The small signal at  $g = 2.014$  (marked with an asterisk in panel (a)) is attributed to the presence of trapped hole centres.



**Figure S20.** X-band EPR spectra recorded at  $T = 123$  K of  $\text{TiO}_2$  powders treated at  $500^\circ\text{C}$  in  $\text{Ar}/\text{H}_2$  atmosphere (sample coded as  $\text{Ar}/\text{H}_2$ ) showing the changes in the spin containing sites ( $\text{Ti}^{3+}$  lattice embedded,  $\text{Ti}^{3+}$  surface exposed, holes) upon varying the time ( $t$ ) of the annealing-temperature: (a)  $t = 30$  min, (b)  $t = 60$  min, (c)  $t = 480$  min, (d)  $t = 960$  min. The EPR signal intensities displayed in the plots (a) to (d) were divided by the square-root of the respective microwave powers used during signal acquisition. An estimation of the spin concentration, expressed as ratio, obtained from the experimental double integrated signal intensity between  $\text{Ti}^{3+}$  lattice embedded and  $\text{Ti}^{3+}$  surface exposed sites is given in each panel.



**Figure S21.** H<sub>2</sub> evolution stability test of TiO<sub>2</sub> powder treated at 500°C in ArH<sub>2</sub>, measured under UV irradiation for 24 hr.

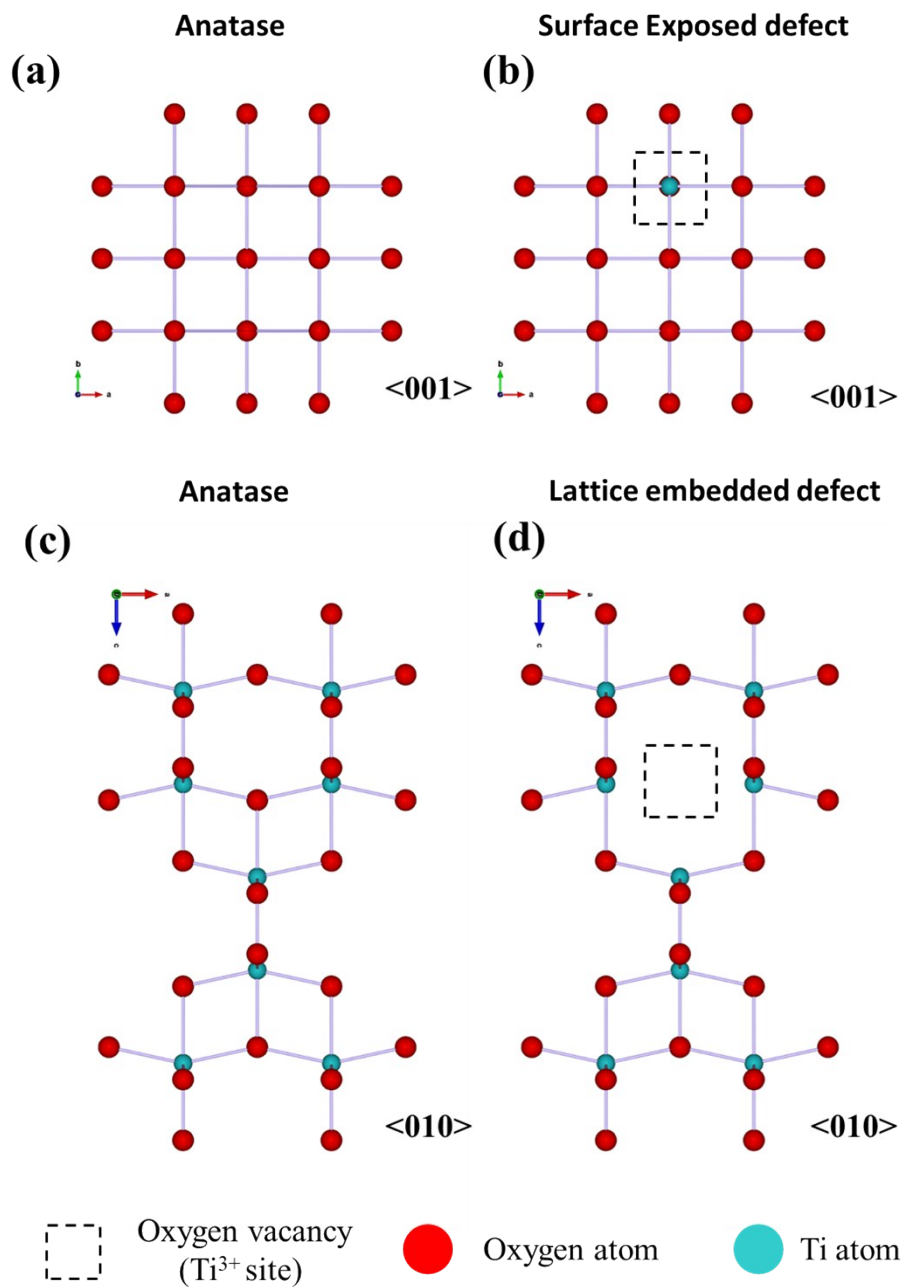


Fig. S 11

**Figure S22.** Schematic view of the surface exposed and embedded  $\text{Ti}^{3+}$  defects.

**Table S1.** Calculated phase composition from XRD diffractograms by Rietveld analysis

	Temperature (°C)	Anatase (wt. %)	Rutile (wt. %)	Ti <sub>4</sub> O <sub>7</sub> (wt. %)
air	300	100	-	-
	500	100	-	-
	700	3.4	96.6	-
	900	-	100	-
Ar	300	100	-	-
	500	100	-	-
	700	22.2	77.8	-
	900	-	100	-
Ar/H <sub>2</sub>	300	100	-	-
	500	100	-	-
	700	73.5	26.5	-
	900	-	100	-
H <sub>2</sub>	300	100	-	-
	500	100	-	-
	700	100	-	-
	900	-	-	100



**Table S2.** Experimental g-tensor parameters obtained from the X-band EPR analysis of the TiO<sub>2</sub> materials obtained at different annealing temperatures and environments, complemented by their spin-concentration ( $S_T$ , spin/g TiO<sub>2</sub>) evaluated against CuSO<sub>4</sub> × 5 H<sub>2</sub>O spin  $S = \frac{1}{2}$  standard.

Sample	Total Spin Density $S_T$ (spin/g TiO <sub>2</sub> )	Ti <sup>3+</sup> Lattice embedded	Ti <sup>3+</sup> Surface exposed	Hole Oxygen-based radical, e.g. Ti <sup>IV</sup> - O•, Ti <sup>IV</sup> -O-O•
Anatase	$0.61 \times 10^{18}$	$g_{x,y} = 1.998$ $g_z = 1.976$	-	Small signal at $g = 2.014$
Air-300	$1.02 \times 10^{18}$	$g_{x,y} = 1.999$ $g_z = 1.977$	-	$g_{avg} = 2.016$
Air-500	$1.85 \times 10^{18}$	$g_{x,y} = 1.997$	$g_{avg} \sim 1.932$	$g_z = 2.043$ , $g_{x,y} = 2.022$
Air-700	$4.20 \times 10^{18}$	$g_{x,y} = 1.994$ $g_z = 1.944$	$g_{avg} \sim 1.930$	$g_z = 2.010$ , $g_y = 1.970$ , $g_x = 1.921$
Air-900	-	-	-	-
Ar-300	$0.95 \times 10^{18}$	$g_{x,y} = 1.998$ $g_z = 1.976$	-	$g_z = 2.018$ , $g_{x,y} = 2.014$
Ar-500	$3.25 \times 10^{18}$	a) $g_{x,y} = 1.998$ , $g_z \sim 1.945$ b) $g_{x,y} = 1.993$ , $g_z \sim 1.945$	-	$g_z = 2.011$ , $g_y = 1.971$ , $g_x = 1.921$ (dominant specie)
Ar-700	$1.31 \times 10^{18}$	$g_{x,y} = 1.994$ $g_z = 1.94$ (estimated)	-	$g_z = 2.010$ , $g_y = 1.972$ , $g_x = 1.921$
Ar-900	$0.40 \times 10^{18}$	$g_{x,y} = 1.999$ , $g_z = 1.976$	-	$g_{avg} = 2.016$
Ar/H <sub>2</sub> -300	$1.16 \times 10^{18}$	$g_{x,y} = 1.997$ , $g_z = 1.976$	-	$g_{avg} = 2.015$ Broad resonance $g_{avg} = 2.190$
Ar/H <sub>2</sub> -500	$2.64 \times 10^{18}$	$g_{x,y} = 1.996$	$g_{avg} \sim 1.934$	a) $g_z = 2.041$ , $g_y = 2.030$ , $g_x = 2.016$ b) Broad resonance $g_{avg} = 2.077$
Ar/H <sub>2</sub> -700	$5.24 \times 10^{18}$	$g_{avg} = 1.948$ ( $\Delta B_{pp} = 10.1$ mT)		-
Ar/H <sub>2</sub> -900	$33.1 \times 10^{18}$	$g_{avg} = 1.949$ ( $\Delta B_{pp} = 11.0$ mT)		-
H <sub>2</sub> -300	$0.94 \times 10^{18}$	$g_{x,y,z} = 1.998$	-	$g_{avg} = 2.015$ Presence of broad resonance at $g = 2.190$
H <sub>2</sub> -500	$4.33 \times 10^{18}$	$g_{x,y} = 1.996$ $g_z \sim 1.974$	$g_{avg} \sim 1.934$	$g_z = 2.037$ , $g_{y,x} = 2.017$ Presence of weak signal at $g = 2.103$
H <sub>2</sub> -700	$6.40 \times 10^{18}$	$g_{avg} = 1.951$ ( $\Delta B_{pp} = 8.9$ mT)		-
H <sub>2</sub> -900	$34.1 \times 10^{18}$	$g_{avg} = 1.956$ ( $\Delta B_{pp} = 6.3$ mT)		-

## References

1. R. Stephen Berry and John C. Mackie. Spin—Orbit Coupling and Electron-Affinity Determinations from Radiative Capture of Electrons by Oxygen Atoms. *J. Chem. Phys.* **1965**, 43, 3067.
2. Mario Chiesa, Maria Cristina Paganini, Stefano Livraghi and Elio Giamello. Charge trapping in TiO<sub>2</sub> polymorphs as seen by Electron Paramagnetic Resonance spectroscopy. *Phys. Chem. Chem. Phys.*, **2013**, 15, 9435-9447.